

Dear Author,

Here are the proofs of your article.

- You can submit your corrections **online**, via **e-mail** or by **fax**.
- For **online** submission please insert your corrections in the online correction form. Always indicate the line number to which the correction refers.
- You can also insert your corrections in the proof PDF and **email** the annotated PDF.
- For fax submission, please ensure that your corrections are clearly legible. Use a fine black pen and write the correction in the margin, not too close to the edge of the page.
- Remember to note the **journal title**, **article number**, and **your name** when sending your response via e-mail or fax.
- **Check** the metadata sheet to make sure that the header information, especially author names and the corresponding affiliations are correctly shown.
- **Check** the questions that may have arisen during copy editing and insert your answers/ corrections.
- **Check** that the text is complete and that all figures, tables and their legends are included. Also check the accuracy of special characters, equations, and electronic supplementary material if applicable. If necessary refer to the *Edited manuscript*.
- The publication of inaccurate data such as dosages and units can have serious consequences. Please take particular care that all such details are correct.
- Please **do not** make changes that involve only matters of style. We have generally introduced forms that follow the journal's style. Substantial changes in content, e.g., new results, corrected values, title and authorship are not allowed without the approval of the responsible editor. In such a case, please contact the Editorial Office and return his/her consent together with the proof.
- If we do not receive your corrections **within 48 hours**, we will send you a reminder.
- Your article will be published **Online First** approximately one week after receipt of your corrected proofs. This is the **official first publication** citable with the DOI. **Further changes are, therefore, not possible.**
- The **printed version** will follow in a forthcoming issue.

Please note

After online publication, subscribers (personal/institutional) to this journal will have access to the complete article via the DOI using the URL: [http://dx.doi.org/\[DOI\]](http://dx.doi.org/[DOI]).

If you would like to know when your article has been published online, take advantage of our free alert service. For registration and further information go to: <http://www.link.springer.com>.

Due to the electronic nature of the procedure, the manuscript and the original figures will only be returned to you on special request. When you return your corrections, please inform us if you would like to have these documents returned.

Metadata of the article that will be visualized in OnlineFirst

Please note: Images will appear in color online but will be printed in black and white.

ArticleTitle Equilibrium of [Si]/(SiO₂) in Carbothermic Selective Reduction of Titanium Concentrate Ore for the Preparation of Titanium Oxycarbide

Article Sub-Title

Article CopyRight The Minerals, Metals & Materials Society and ASM International
(This will be the copyright line in the final PDF)

Journal Name Metallurgical and Materials Transactions B

Corresponding Author Family Name **Zhu**
Particle
Given Name **Hongmin**
Suffix
Division State Key Laboratory of Advanced Metallurgy
Organization University of Science and Technology Beijing
Address Beijing, 100083, China
Division
Organization Tohoku University
Address 6-6-02, Aramaki-Aza-Aoba, Aobo-ku, Sendai, 980-8579, Japan
Phone +81-22-795-7309
Fax
Email hzhu@material.tohoku.ac.jp
URL
ORCID

Author Family Name **Xiao**
Particle
Given Name **Jiusan**
Suffix
Division State Key Laboratory of Advanced Metallurgy
Organization University of Science and Technology Beijing
Address Beijing, 100083, China
Phone
Fax
Email
URL
ORCID

Author Family Name **Wang**
Particle
Given Name **Qi**
Suffix
Division State Key Laboratory of Advanced Metallurgy
Organization University of Science and Technology Beijing

Address Beijing, 100083, China
Phone
Fax
Email
URL
ORCID

Author	Family Name	Jiang
	Particle	
	Given Name	Bo
	Suffix	
	Division	Department of Materials Science and Engineering
	Organization	Norwegian University of Science and Technology
	Address	7491, Trondheim, Norway
	Phone	
	Fax	
	Email	
	URL	
	ORCID	

Schedule	Received	3 April 2017
	Revised	
	Accepted	

Abstract Carbothermic selective reduction of titanium concentrate ore for the preparation of titanium oxycarbide was presented and SiO₂ was found to be partly reduced during the process. The equilibrium relationship between Si dissolved in metallic Fe and SiO₂ existed in corresponding slag was analyzed thermodynamically, and then revealed by a series of designed equilibration experiments. The results are in accordance with the calculated values obtained from thermodynamic analysis and have well explained the phenomenon that part of SiO₂ is reduced during selective reduction processing of titanium concentrate ore.

Footnote Information	Manuscript submitted April 3, 2018.
----------------------	-------------------------------------

Equilibrium of [Si]/(SiO₂) in Carbothermic Selective Reduction of Titanium Concentrate Ore for the Preparation of Titanium Oxycarbide



JIUSAN XIAO, QI WANG, BO JIANG, and HONGMIN ZHU

Carbothermic selective reduction of titanium concentrate ore for the preparation of titanium oxycarbide was presented and SiO₂ was found to be partly reduced during the process. The equilibrium relationship between Si dissolved in metallic Fe and SiO₂ existed in corresponding slag was analyzed thermodynamically, and then revealed by a series of designed equilibration experiments. The results are in accordance with the calculated values obtained from thermodynamic analysis and have well explained the phenomenon that part of SiO₂ is reduced during selective reduction processing of titanium concentrate ore.

<https://doi.org/10.1007/s11663-018-1455-9>

© The Minerals, Metals & Materials Society and ASM International 2018

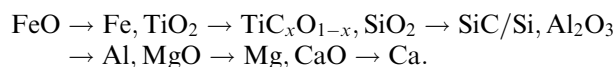
I. INTRODUCTION

TITANIUM and its alloys have received great attention from researchers worldwide due to its excellent properties such as high melting point and extraordinary corrosion resistance. However, its application has been limited by high cost arising from the Kroll Process.^[1] In the past two decades, several new methods for producing titanium have been proposed, such as FFC Cambridge process,^[2] Ono Suzuki (OS) process,^[3,4] molten oxide electrolysis (MOE) process,^[5] and the USTB electrolysis process.^[6,7] In the USTB process, high-purity titanium can be collected at the cathode through the electrolysis of titanium oxycarbide (TiC_{0.5}O_{0.5}) as consuming anode in molten salt, while the carbon and oxygen component in the oxycarbide anode evolve into the carbon oxide (CO, CO₂) gases.

The anode material for the USTB process, namely titanium oxycarbide (TiC_{0.5}O_{0.5}), can be synthesized through not only carbothermic reduction of TiO₂, but also selective carbothermic reduction of titanium raw materials such as high-grade titanium slag and titanium-iron ores,^[8] which mainly contains FeO/Fe₂O₃, TiO₂, CaO, MgO, SiO₂, and Al₂O₃. A lot of related

researches focusing on the reduction of titanium raw materials have already been proposed since the 1950s.^[9-15] Coley has studied the reduction mechanism of ilmenite into iron and titanium oxycarbide at temperatures ranging from 1587 K to 1790 K (1314 °C to 1517 °C).^[16] Dewan has studied the carbothermic reduction of ilmenite concentrates in different gas atmospheres including hydrogen, argon, and helium, demonstrating the transformation from TiO₂ to Ti₃O₅ and then Ti₂O₃, which is converted to titanium oxycarbide.^[17,18]

The equilibrium constants for relevant reactions during selective carbothermic reduction of titanium raw materials are plotted as a function of temperature in Figure 1. Since titanium sesquioxide (Ti₂O₃) will be reduced to titanium oxycarbide (TiC_xO_{1-x}) instead of titanium monoxide (TiO), the reaction sequence of carbothermic reduction of titanium raw materials is



Therefore, in the presence of certain chemical potentials of carbon and oxygen at high temperature (*i.e.*, 1873 K [1600 °C]), the reactive elements, such as Si, Mg, Ca, and Al, are expected to form oxides, while titanium oxide is reduced to titanium oxycarbide (TiC_{0.5}O_{0.5}) and iron oxides are converted to liquid metallic Fe which is easily separated from the oxycarbide and slag phases, since Fe is supposed to dissolve in the molten salt before titanium oxycarbide during electrolysis, resulting in contamination. Similarly, oxides of Si, Mg, Ca, and Al should be removed due to their properties of insulation, which will bring a bad effect on the conductivity of anode. Thus, high-purity titanium oxycarbide can be synthesized from titanium raw materials and then

JIUSAN XIAO and QI WANG are with the State Key Laboratory of Advanced Metallurgy, University of Science and Technology Beijing, Beijing 100083, China. BO JIANG is with the Department of Materials Science and Engineering, Norwegian University of Science and Technology, 7491 Trondheim, Norway. HONGMIN ZHU is with the State Key Laboratory of Advanced Metallurgy, University of Science and Technology Beijing and also with the Tohoku University, 6-6-02, Aramaki-Aza-Aoba, Aobo-ku, Sendai, 980-8579, Japan. Contact e-mail: hzhu@material.tohoku.ac.jp

Manuscript submitted April 3, 2018.



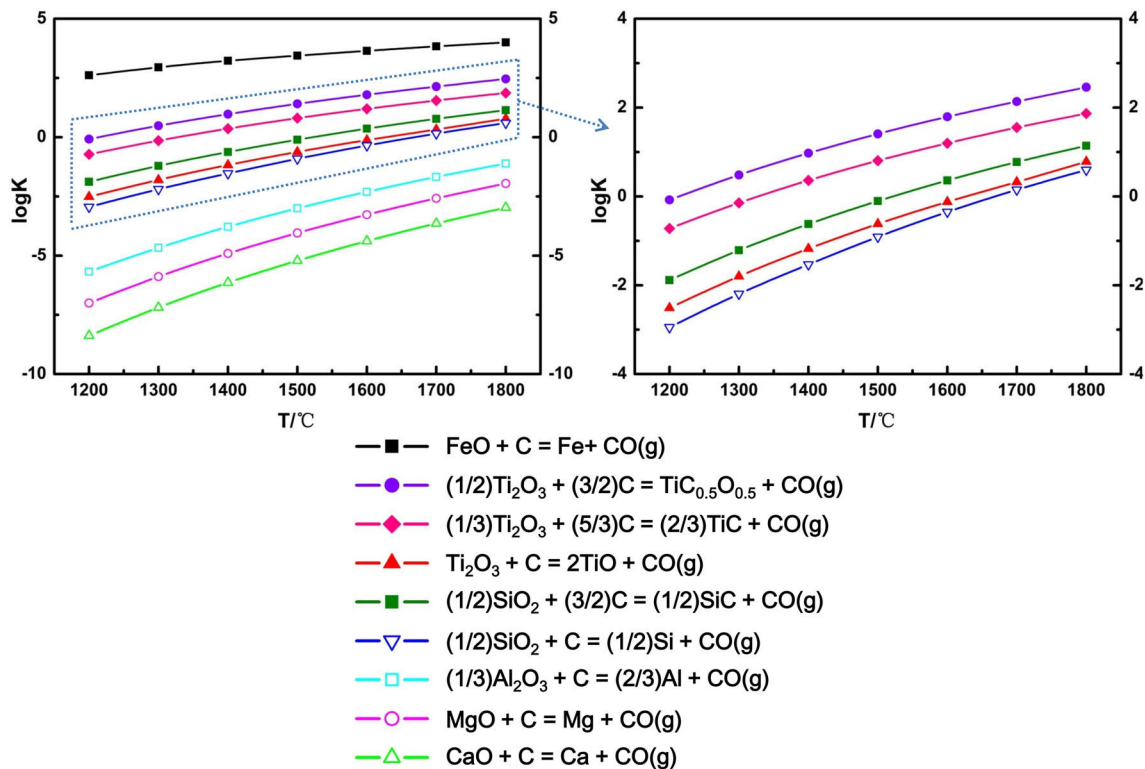


Fig. 1—Equilibrium constants for relevant reactions during selective carbothermic reduction of titanium raw materials.

78 utilized as the anode of USTB process for the preparation
 79 of metallic titanium. This process is found to be
 80 theoretically and experimentally feasible, which exhibits
 81 great potential for the utilization of low-grade titanium
 82 raw materials, a unique, cost-reducing feature to the
 83 USTB process.

84 The relationship between the reduction of TiO_2 and
 85 SiO_2 is further evaluated through thermodynamic
 86 calculation under temperature of 1873 K (1600 °C),
 87 as shown in Figure 2. The equilibrium constant of
 88 reaction from Ti_2O_3 to $\text{TiC}_x\text{O}_{1-x}$ changes with the
 89 concentration of solid solution, since the mixing Gibbs
 90 free energy of $\text{TiC}_x\text{O}_{1-x}$ exhibits quadratic relationship
 91 with the corresponding composition.^[19] SiO_2 is sup-
 92 posed to maintain in oxide form during the selective
 93 reduction process when the stoichiometric amount of
 94 carbon is provided only for the sufficient reduction of
 95 $\text{FeO}/\text{Fe}_2\text{O}_3$ to Fe and TiO_2 to $\text{TiC}_{0.5}\text{O}_{0.5}$. However,
 96 based on our previous work, a slight amount of SiO_2
 97 will be reduced into Fe-Si alloy in the selective
 98 reduction process due to the decrease of Si activity in
 99 liquid Fe,^[20] leading to the fact that part of the carbon
 100 which is designed to reduce TiO_2 to $\text{TiC}_{0.5}\text{O}_{0.5}$ will
 101 react with SiO_2 , causing the insufficient reduction of
 102 TiO_2 . Thus in this paper, we presented carbothermic
 103 selective reduction of titanium concentrate ore, and
 104 then performed a series of equilibration experiments to
 105 investigate the equilibrium relationship between Si in
 106 metallic Fe and SiO_2 in slag with the presence of
 107 titanium oxycarbide. The results were interpreted with
 108 the aid of thermodynamic analysis.

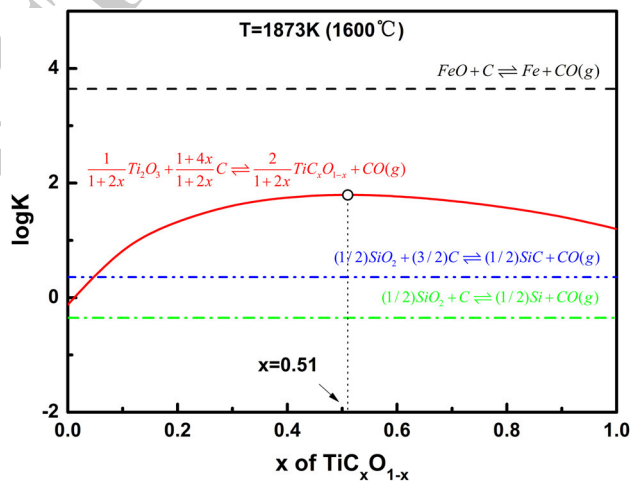


Fig. 2—Equilibrium constants for the carbothermic reduction of FeO , Ti_2O_3 , and SiO_2 related to x of $\text{TiC}_x\text{O}_{1-x}$ at 1873 K (1600 °C).

II. EXPERIMENTAL DETAILS

A. Carbothermic Selective Reduction

High-purity graphite (99.9 pct, Aladdin Chemistry Co., Ltd) and titanium concentrate ore (Chongqing Iron and Steel Co., Ltd) were used as raw materials in selective reduction experiment. The chemical composition of titanium concentrate ore was tested by X-ray fluorescence (XRF, XRF-1800, Shimadzu Corporation), as listed in Table I. The stoichiometric amount of carbon is provided only for the sufficient

Table I. Chemical Composition of Titanium Concentrate Ore

	TiO ₂	FeO	Fe ₂ O ₃	CaO	MgO	SiO ₂	Al ₂ O ₃
Weight Percent	44.9	41.4	2.4	0.3	4.2	4.3	2.5

119 of FeO/Fe₂O₃ to Fe and TiO₂ to TiC_{0.5}O_{0.5} due to the
 120 aim of selective reduction process. Ingredients were
 121 mixed uniformly according to appropriate ratio by ball
 122 milling over 4 hours and then pressed into pellets of
 123 50 mm in diameter and 15 mm in height. The pellets
 124 were placed in a graphite crucible lined with molybde-
 125 num plate, sintering at 1873 K (1600 °C) over 4 hours
 126 under flowing argon atmosphere. The reductive prod-
 127 uct was analyzed by X-ray diffraction (XRD,
 128 M21XVHF22, MAC in Japan), scanning electron
 129 microscopy (SEM, ZEISS-EVO 18, Carl Zeiss AG),
 130 and energy dispersive X-ray spectroscopy (EDX) after
 131 cooling to room temperature. The concentration of Si
 132 dissolved in metallic Fe collected from obtained pro-
 133 duct was analyzed by Inductive Coupled Plasma
 134 Emission Spectrometer (ICP, OPTIMA 7000DV, Perk-
 135 inElmer Co., Ltd).

136 B. Equilibrium Experiment

137 The product obtained from selective carbothermic
 138 reduction of titanium raw material is composed of
 139 titanium oxycarbide, slag phase, and metallic Fe with a
 140 slight amount of dissolved Si.^[20] The equilibrium system
 141 consists of three parts: titanium compound for the
 142 fixation of chemical potential, Fe-Si alloy, and slag
 143 phase including CaO, SiO₂, MgO, and Al₂O₃.

144 For the preparation of Fe-Si alloy, Fe powder
 145 (99.9 pct, Aladdin Chemistry Co., Ltd) and Si powder
 146 (99.9 pct, Sinopharm Chemical Reagent Co., Ltd) were
 147 mixed uniformly by ball milling over 4 hours. The
 148 ingredients were placed into a corundum (Al₂O₃)
 149 crucible and melted at 1873 K (1600 °C) in a MoSi₂
 150 resistance furnace under flowing argon atmosphere for
 151 10 minutes. Samples were cooled to room temperature
 152 after melting and crushed into powders for the following
 153 equilibration experiments. Particularly, we have intro-
 154 duced metallic Fe powder ($x_{[Si]} = 0$ pct) and Fe-Si alloy
 155 with a certain concentrations of Si ($x_{[Si]} = 10$ pct) for
 156 comparison.

157 The constitution of slag phase (slag 1) naturally comes
 158 from the chemical composition of titanium concentrate
 159 ore, as listed in Table I. Besides, we have introduced a
 160 different kind of slag phase (slag 2) with a lower melting
 161 point (1507 K [1234 °C]^[21]) in equilibration experiments
 162 for comparison. The chemical compositions of these two
 163 kinds of slags are listed in Table II. Powders of CaO,
 164 MgO, SiO₂, and Al₂O₃ (99.9 pct, Sinopharm Chemical
 165 Reagent Co., Ltd) were mixed uniformly by ball milling
 166 over 4 hours, and then fused under flowing argon
 167 atmosphere at 1573 K (1300 °C) for 10 minutes. The
 168 melted and solidified sample was crushed into powders
 169 after cooling to room temperature.

Table II. Chemical Composition of Simulative Slags Utilized in Equilibration Experiments

	CaO	MgO	SiO ₂	Al ₂ O ₃
Slag 1				
Weight Percent	2.7	37.2	38.0	22.1
Mol Pct	2.7	50.7	34.7	11.9
Slag 2				
Weight Percent	30.5	7.3	43.7	18.5
Mol Pct	33.3	11.1	44.5	11.1

Ingredients including titanium compound, Fe-Si 170
 alloy, and pre-fused simulative slag were mixed uni- 171
 formly according to appropriate ratios by ball milling 172
 over 4 hours. Mixed powders were placed in a graphite 173
 crucible lined with molybdenum plate, and then sintered 174
 under flowing argon atmosphere at 1873 K (1600 °C) 175
 over 4 hours for the achievement of equilibrium. The 176
 concentration of Si in obtained Fe-Si alloy was analyzed 177
 by Inductive Coupled Plasma Emission Spectrometer 178
 (ICP). 179

180 III. RESULTS AND DISCUSSION

181 A. Carbothermic Selective Reduction

182 Firstly, we have performed carbothermic selective
 183 reduction of titanium concentrate ore for the prepara-
 184 tion of titanium oxycarbide. The amount of
 185 graphite utilized as the reductant was carefully controlled
 186 according to the chemical composition of
 187 titanium concentrate ore, as listed in Table I, which
 188 indicates that titanium concentrate ore primarily
 189 consists of TiO₂ and FeO, together with a slight
 190 amount of Fe₂O₃, CaO, MgO, SiO₂, and Al₂O₃. The
 191 XRD patterns of raw material and reductive product
 192 are shown in Figure 3, declaring that titanium con-
 193 centrate ore with a main phase of FeTiO₃ is substan-
 194 tially transformed into metallic Fe and titanium
 195 oxycarbide after selective reduction process, together
 196 with a slight amount of unreduced Mg₂TiO₄ and
 197 Ti₂O₃. This phenomenon is consistent with Dewan's
 198 research, which demonstrates that the completion of
 199 carbothermic reduction to titanium oxycarbide under
 200 argon atmosphere is achieved with a reaction time
 201 longer than 5 hours.^[17] The carbothermic reduction
 202 process similarly proceeds in two stages: (1) the
 203 transformation of FeTiO₃ into metallic iron and
 204 titanium dioxide; (2) the reduction of titanium dioxide
 205 to titanium oxycarbide.^[15]



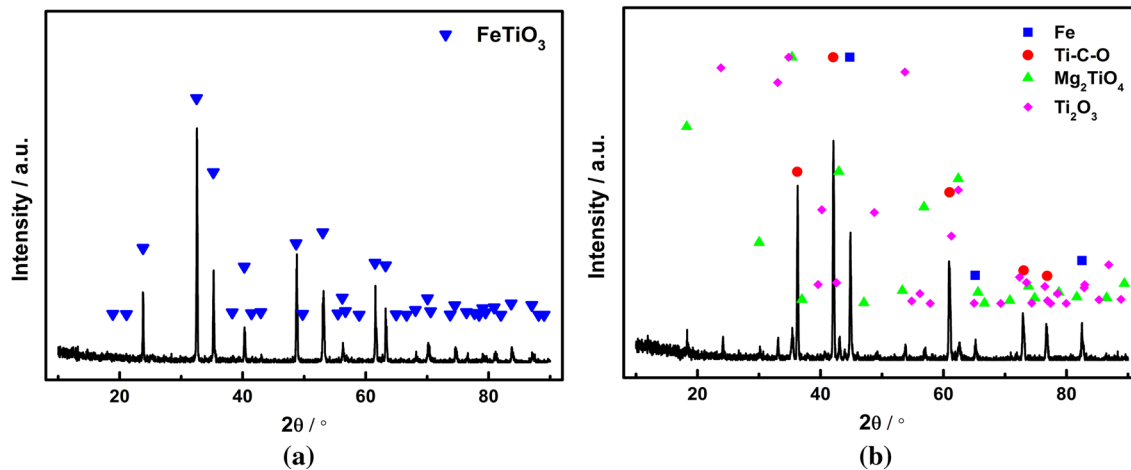
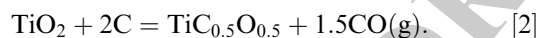


Fig. 3—XRD patterns of (a) titanium concentrate ore and (b) corresponding product of carbothermic reduction.

The product is characterized through SEM/EDX techniques, as described in Figure 4, indicating three phases of metallic Fe, titanium oxycarbide, and slag phase. According to a typical SEM/EDX image of the slag phase, as shown in Figure 5, it appears clumpy and the size is about 10 μm ; particles of titanium oxycarbide distribute in the product, with a size of 2 μm , as shown in Figure 6, while Fe is found to be spherical in shape, with a diameter of 5 μm . Particularly, the EDX image of reductive product displayed in Figure 4 exhibits an overlap of the distribution of Si and Fe element, suggesting that there is a slight dissolution of Si in the metallic Fe, with a concentration of about 2.5 pct, according to the EDX result described in Figure 6.

For the preparation of titanium oxycarbide from titanium concentrate ore, the designed reactions are



Based on the composition mentioned in Table I, there is a certain amount of mass change (33.9 pct) during the selective reduction process. The expected mass change is supposed to be lower than 33.9 pct due to unreduced Mg_2TiO_4 and Ti_2O_3 detected in XRD pattern. However, we have carefully estimated the weight of starting material and product, and the mass loss is higher than expected, as shown in Table III. This is attributed to the partly reduction of SiO_2 with the reaction of



This leads to the consumption of carbon that is originally designed to reduce TiO_2 . Due to the deviation of mass change, the concentration of Si in Fe-Si alloy obtained from the reduction of SiO_2 is calculated to be 2.5 pct, which is in accordance with not only EDX results, but also the results of ICP analysis for the collected metallic Fe as listed in Table III. All the results suggest that the main part of SiO_2 (about 77 pct) in the titanium concentrate ore stays in slag in the form of

composite oxide such as $\text{MgO}\cdot\text{SiO}_2$, as displayed in Figure 5, while the remaining SiO_2 is partly reduced to Si, as in Fe-Si alloy.

B. Thermodynamic Analysis

The reduction of pure SiO_2 thermodynamically occurs only after the transformation of Ti_2O_3 into $\text{TiC}_{0.5}\text{O}_{0.5}$ and even TiC, according to Figure 1, and thus SiO_2 is supposed to stay in the slag phase after selective reduction process, in condition of providing carbon only for the sufficient reduction of $\text{FeO}/\text{Fe}_2\text{O}_3$ to Fe and TiO_2 to $\text{TiC}_{0.5}\text{O}_{0.5}$. However, SiO_2 is found to be partly reduced into Si, as mentioned above, forming a separated phase of Fe-Si alloy at the chemical potential provided by titanium oxycarbide, due to the decreasing activity of Si,^[20] indicating that there is equilibrium relationship between Si in Fe-Si alloy phase and SiO_2 remaining in the slag phase, and the concentration of Si in metallic Fe is determined by the chemical potential of titanium oxycarbide and the activity of SiO_2 in slag phase.

It is well known that titanium oxycarbide ($\text{TiC}_x\text{O}_{1-x}$, $0 \leq x \leq 1$) is a solid solution of TiC and $\beta\text{-TiO}$, and also a type of intermediate product between Ti_2O_3 and TiC during the carbothermic reduction of TiO_2 .^[22] Therefore, we intend to control the chemical potential of the system with two kinds of redox couples: $\text{Ti}_2\text{O}_3/\text{TiC}_x\text{O}_{1-x}$ and $\text{TiC}_x\text{O}_{1-x}/\text{TiC}$. The corresponding chemical potential relationship for $\text{Ti}_2\text{O}_3/\text{TiC}_x\text{O}_{1-x}$ and $\text{TiC}_x\text{O}_{1-x}/\text{TiC}$ at 1873 K (1600 $^\circ\text{C}$) is calculated in two different ways based on previous study on thermodynamic properties of $\text{TiC}_x\text{O}_{1-x}$.^[23]

Ti_2O_3 can be easily reduced by carbon to $\text{TiC}_x\text{O}_{1-x}$ in almost the whole composition range at 1873 K (1600 $^\circ\text{C}$) and $\text{TiC}_{0.51}\text{O}_{0.49}$ exhibits the lowest chemical potential, as shown in Figure 2. However, $\text{TiC}_{0.51}\text{O}_{0.49}$ does not coexist with Ti_2O_3 at 1873 K (1600 $^\circ\text{C}$) since they will react, forming titanium oxycarbide with lower concentration of TiC, according to Figure 7, which demonstrates the reacting capacity of Ti_2O_3 and

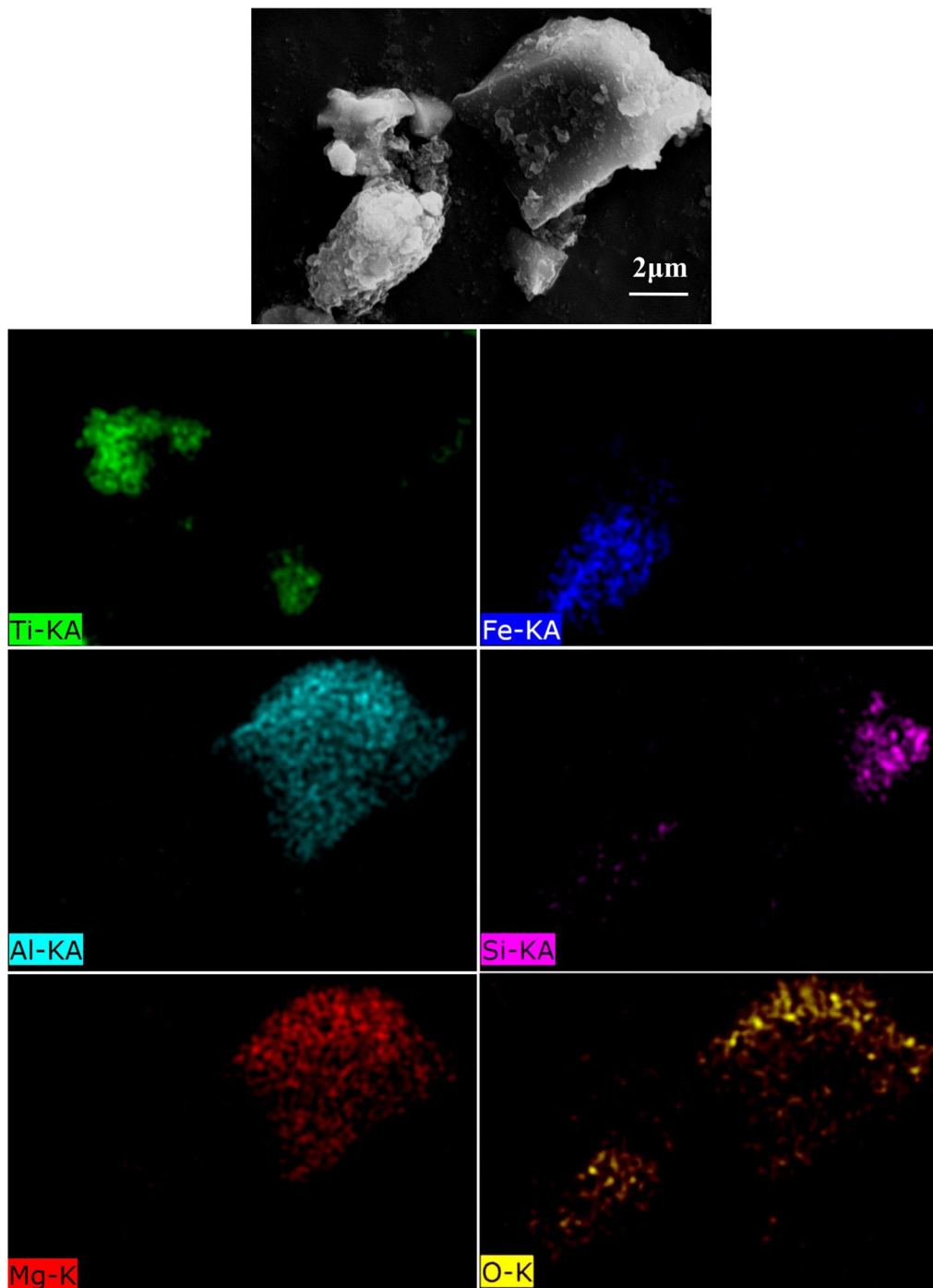


Fig. 4—SEM and EDX images of the product of carbothermic reduction from titanium concentrate ore.

288 titanium oxycarbide. The antioxidant capacity of tita-
 289 nium oxycarbide is gradually enhanced with the
 290 decrease of the concentration of TiC, reaching an
 291 equilibrium composition of $\text{TiC}_{0.417}\text{O}_{0.583}$, according to
 292 Figure 7, which suggests that $\text{TiC}_x\text{O}_{1-x}$ with a concen-
 293 tration of TiC ranging from 0 to 0.417 could coexist
 294 with Ti_2O_3 , while $\text{TiC}_x\text{O}_{1-x}$ with a concentration of TiC
 295 above 0.417 will be oxidized by Ti_2O_3 . Thus, the
 296 appropriate composition for redox couple $\text{Ti}_2\text{O}_3/\text{TiC}_x\text{O}_{1-x}$
 297 is $\text{Ti}_2\text{O}_3/\text{TiC}_{0.417}\text{O}_{0.583}$.

298 For redox couple $\text{TiC}_x\text{O}_{1-x}/\text{TiC}$, since the mixing
 299 Gibbs energy of $\text{TiC}_x\text{O}_{1-x}$ at 1873 K (1600 °C) is
 300 negative in the whole concentration range,^[24] TiC will
 301 spontaneously dissolve in $\text{TiC}_x\text{O}_{1-x}$, forming a different
 302 type of $\text{TiC}_x\text{O}_{1-x}$ with a higher concentration of TiC.
 303 Besides, titanium oxycarbide ($\text{TiC}_x\text{O}_{1-x}$) itself is a
 304 special kind of redox couple: $\text{TiO}(\text{in } \text{TiC}_x\text{O}_{1-x})/\text{TiC}(\text{in } \text{TiC}_x\text{O}_{1-x})$,
 305 while the chemical potential changes with
 306 the composition. Therefore, the appropriate

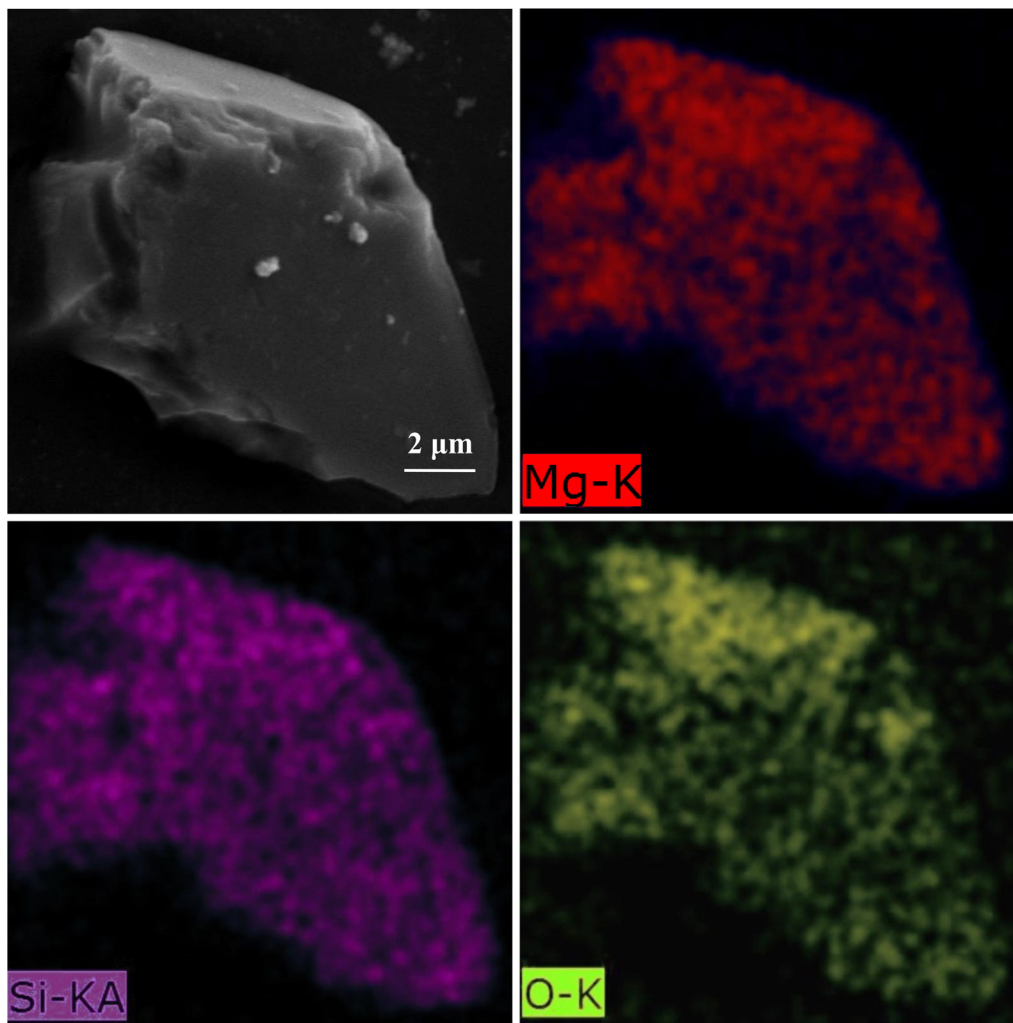


Fig. 5—SEM and EDX images of typical slag phase in product of carbothermic reduction from titanium concentrate ore.

307 composition for redox couple $\text{TiC}_x\text{O}_{1-x}/\text{TiC}$ is
 308 $\text{TiC}_{0.5}\text{O}_{0.5}$, namely $\text{TiO}(\text{in } \text{TiC}_{0.5}\text{O}_{0.5})/\text{TiC}(\text{in } \text{TiC}_{0.5}\text{O}_{0.5})$,
 309 $\text{TiC}_{0.5}\text{O}_{0.5}$, considering the aim of the selective reduction
 310 process.

311 C. Thermodynamic Equilibrium

312 1. Equilibrium experiment

313 Based on the thermodynamic analysis above, we have
 314 designed a series of equilibrium experiments focused on
 315 the reaction between a redox couple consisting of a
 316 titanium compound and that of silicon (SiO_2/Si), as
 317 listed in Table IV. To fix the chemical potential of the
 318 equilibrium system, we designed two equilibrium com-
 319 positions of titanium compounds, including $\text{Ti}_2\text{O}_3/$
 320 $\text{TiC}_{0.417}\text{O}_{0.583}$ (experiments 1 to 4) and $\text{TiO}(\text{in } \text{TiC}_{0.5}\text{O}_{0.5})/$
 321 $\text{TiC}(\text{in } \text{TiC}_{0.5}\text{O}_{0.5})$ (experiments 5 to 6).
 322 Metallic Fe powder ($x_{[\text{Si}]} = 0$ pct) and Fe-Si alloy
 323 ($x_{[\text{Si}]} = 10$ pct) were used, which is below and above
 324 the equilibrium concentration ($x_{[\text{Si}]} = 2.7$ pct) obtained
 325 in selective reduction process, for the distinct presenta-
 326 tion of equilibrium process from two different sides.
 327 Besides, the total amount of Fe-Si alloy was carefully

controlled to be smaller enough in all the equilibrium 328
 experiments than that of not only titanium compound 329
 but also SiO_2 existed in slag; thus, there is rarely any 330
 composition change of titanium compound and SiO_2 331
 during the equilibration. 332

Samples including titanium compounds ($\text{Ti}_2\text{O}_3/$ 333
 $\text{TiC}_{0.417}\text{O}_{0.583}$ and $\text{TiC}_{0.5}\text{O}_{0.5}$), crushed powders of Fe-Si 334
 alloy, and pre-fused slag were treated at 1873 K 335
 (1600 °C) for 4 hours to equilibrate, and the obtained 336
 Fe-Si alloys were carefully gathered and processed after 337
 cooling to room temperature for the determination of 338
 concentration of Si. The results together with corre- 339
 sponding experimental details are described in Table IV. 340

For experiments 1 and 2, the chemical potential of 341
 reacting system is controlled by $\text{Ti}_2\text{O}_3/\text{TiC}_{0.417}\text{O}_{0.583}$. 342
 The concentration of Si in Fe-Si alloy decreases from 10 343
 to 4.0 pct and 6.8 pct, after equilibrating with slag 1 and 344
 slag 2, respectively, indicating that during the equilibra- 345
 tion a certain amount of Si in Fe-Si alloy has been 346
 oxidized into SiO_2 , dissolving in slag phase, while a part 347
 of Ti_2O_3 has been transformed into TiO, dissolving into 348
 $\text{TiC}_x\text{O}_{1-x}$ solid solution. 349

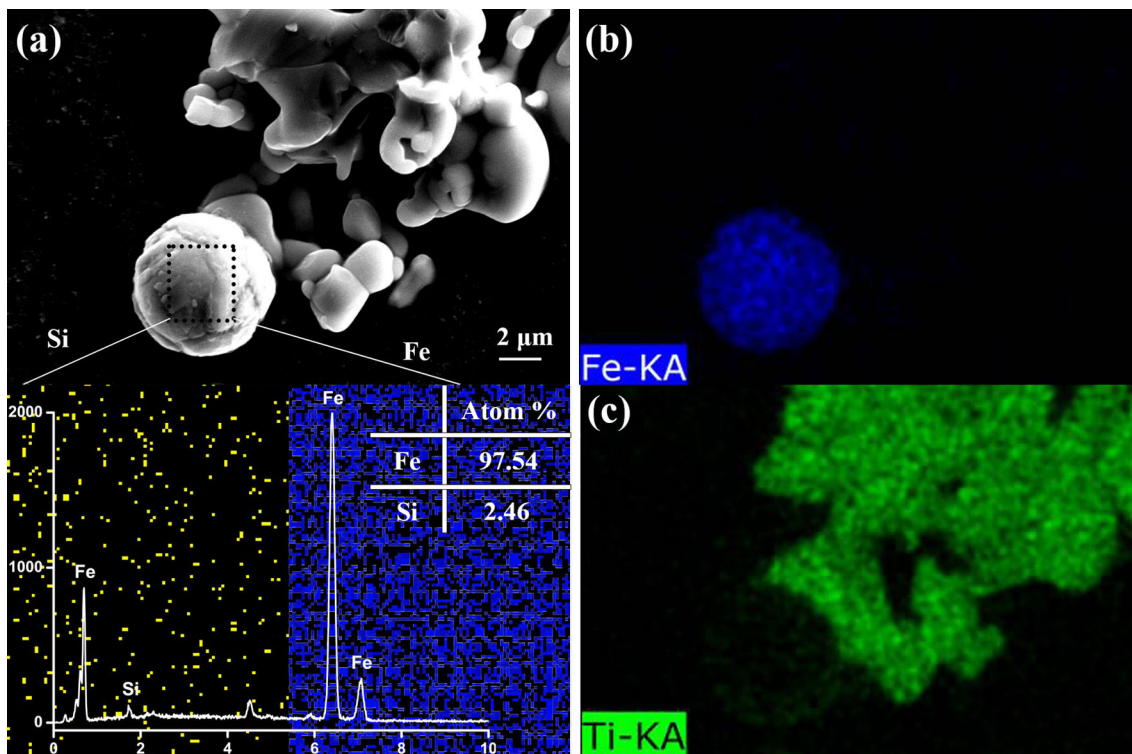


Fig. 6—SEM image of (a) product of carbothermic reduction from titanium concentrate ore and corresponding EDX images of element (b) Fe and (c) Ti, together with compositional analysis of Fe concentrate.

Table III. Theoretical and Experimental Mass Change of Carbothermic Selective Reduction, Together with Concentration of Si ($x_{[Si]}$) in Reduced Fe-Si Alloy Obtained from Calculation, EDX, and ICP Results

Mass Change (Theo.)	Mass Change (Exp.)	Reaction Ratio	$x_{[Si]}$ (Cal.)	$x_{[Si]}$ (EDX)	$x_{[Si]}$ (ICP)
33.9 pct	34.3 pct	101.2 pct	2.5 pct	2.46 pct	2.7 pct

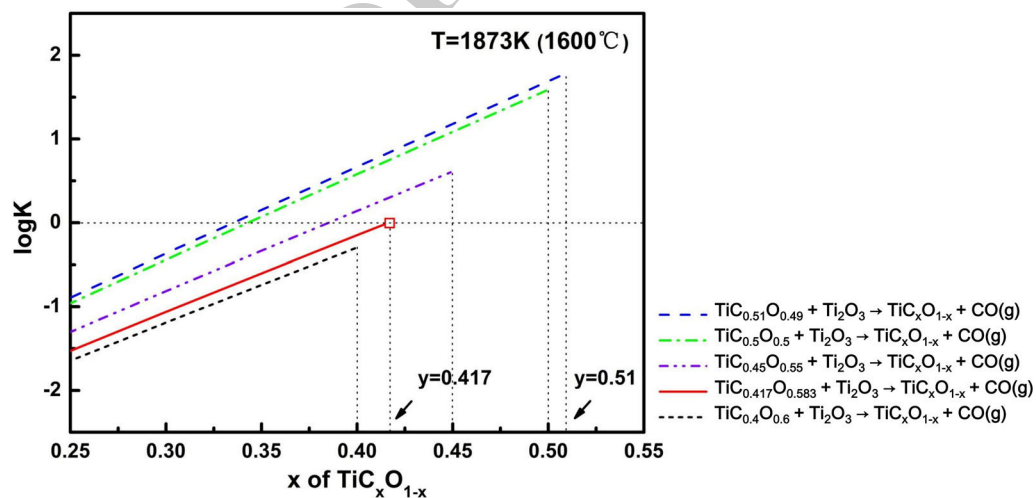


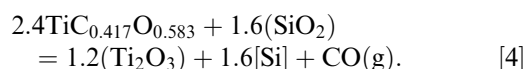
Fig. 7—Equilibrium constants of the reaction between TiC_yO_{1-y} ($y = 0.51, 0.50, 0.45, 0.417,$ and 0.40) and Ti_2O_3 turning into TiC_xO_{1-x} with a lower concentration of TiC at 1873 K (1600 °C).

350 For experiments 3 and 4, the reaction system is similar
 351 to experiments 1 and 2 except that the resource of Fe is
 352 pure iron instead of Fe-Si alloy. After equilibrating with
 353 $Ti_2O_3/TiC_{0.417}O_{0.583}$ and slag phase, Fe is transformed
 354 into Fe-Si alloy since the concentration of Si in obtained
 355 Fe-Si alloy is found to be 3.4 and 5.6 pct, declaring that
 356 some of SiO_2 in the slag phase has been reduced into Si,
 357 forming the Fe-Si alloy during the equilibration.

358 For experiments 5 and 6, the chemical potential of
 359 reacting system is controlled by $TiO(in TiC_{0.5}O_{0.5})/$
 360 $TiC(in TiC_{0.5}O_{0.5})$. During the equilibration, the con-
 361 centration of Si in Fe increases to 2.3 and 4.5 pct,
 362 indicating that some of SiO_2 in the slag phase has been
 363 reduced into Si, dissolving in metallic Fe, while TiC in
 364 $TiC_{0.5}O_{0.5}$ was transformed into TiO.

365 2. Discussions

366 For experiments 1 to 4, the equilibrium equation of
 367 “ $Ti_2O_3/TiC_{0.417}O_{0.583}$ —Fe-Si alloy-slag-graphite cru-
 368 cible” system is



371 Substance existed in slag phase is surrounded with
 372 brackets, while substance in metallic Fe is decorated by
 373 square brackets. Since the activity of $TiC_{0.417}O_{0.583}$
 374 (solid state) is reasonable to consider as 1 in the
 375 equilibrium system, the corresponding equilibrium con-
 376 stant is as described below:

$$K = \left(\frac{a_{[Si]}}{a_{(SiO_2)}} \right)^{1.6} \times a_{(Ti_2O_3)}^{1.2} \times P_{CO}, \quad [5]$$

380 where $a_{[Si]}$, $a_{(SiO_2)}$, $a_{(Ti_2O_3)}$, and P_{CO} stand for the activi-
 381 ty of Si in Fe-Si alloy, the activity of SiO_2 in slag
 382 phase, the activity of Ti_2O_3 in slag phase, and the par-
 383 tial pressure of CO, respectively. The equilibrium con-
 384 stant can be evaluated based on previous research of
 385 Jiang *et al.*^[23] The relationship between the activity
 386 coefficient and the concentration of Si in Fe-Si alloy at
 387 1873 K (1600 °C) is described below^[25]:

$$\log \gamma_{Si} = -2.12 \times (1 - N_{[Si]})^2 - 0.22. \quad [6]$$

390

391 Since the activity of Ti_2O_3 can be calculated through
 392 Factsage 6.0 based on the composition of slag phase,^[26]
 393 the theoretical concentration of Si in Fe-Si alloy
 394 equilibrated with $Ti_2O_3/TiC_{0.417}O_{0.583}$ is plotted as a
 395 function of the activity of SiO_2 in Figure 8. It is clear
 396 that the concentration of Si in equilibrium Fe-Si alloy
 397 rises with the increase of the activity of SiO_2 . The
 398 activity of SiO_2 can also be calculated through Factsage
 399 6.0 based on the composition of slag phase (0.063 for
 400 slag 1 and 0.135 for slag 2), and thus the experimental
 401 equilibrium concentration of Si in Fe-Si alloy equili-
 402 brated with $Ti_2O_3/TiC_{0.417}O_{0.583}$ is plotted against the
 403 activity of SiO_2 in Figure 8. During the equilibration,
 404 the concentration of Si in Fe-Si alloy changes from two
 405 different sides (0 and 10 pct), approaching the values of
 406 equilibrium shown as the solid line in Figure 8.

The equilibrium equation of “ $TiC_{0.5}O_{0.5}$ —Fe-Si
 alloy-slag-graphite crucible” system is

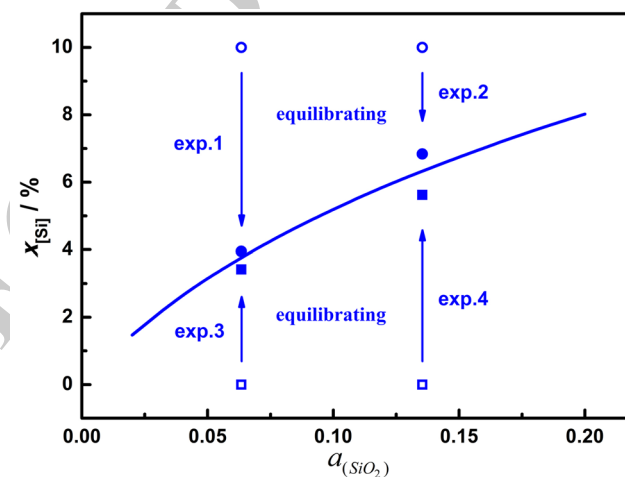
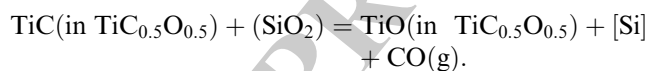


Fig. 8—Theoretical equilibrium concentration (line) and experimental concentration of Si in Fe-Si alloy before (hollow symbols) and after (solid symbols) equilibrated with $Ti_2O_3/TiC_{0.417}O_{0.583}$ and SiO_2 in slag related to the activity of SiO_2 .

Table IV. Details of Equilibrium Experiments, Including Titanium Compound Utilized, Slag Type, and Concentration of Si in Fe-Si Alloy Before and After Equilibration

	Titanium Compound	Slag (SiO_2)	$x_{[Si]}$ in Fe-Si Alloy (Pct)	
			Before	After
1	$Ti_2O_3/TiC_{0.417}O_{0.583}$	slag 1	10	4.0
2	$Ti_2O_3/TiC_{0.417}O_{0.583}$	slag 2	10	6.8
3	$Ti_2O_3/TiC_{0.417}O_{0.583}$	slag 1	0	3.4
4	$Ti_2O_3/TiC_{0.417}O_{0.583}$	slag 2	0	5.6
5	$TiC_{0.5}O_{0.5}$	slag 1	0	2.3
6	$TiC_{0.5}O_{0.5}$	slag 2	0	4.5

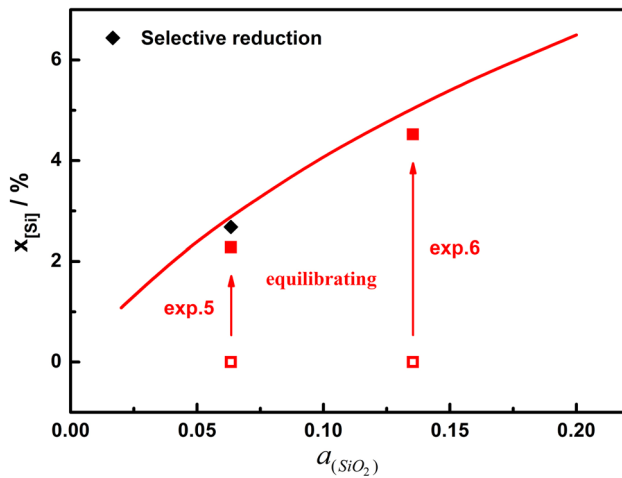


Fig. 9—Theoretical equilibrium concentration (line) and concentration of Si in Fe-Si alloy before (hollow squares) and after (solid squares) equilibrated with $\text{TiC}_{0.5}\text{O}_{0.5}$ and SiO_2 in slag related to the activity of SiO_2 , together with experimental concentration (solid diamond) of Si in reduced Fe-Si alloy obtained from selective reduction.

412 Since the activities of TiC and TiO are reasonable to
413 consider as equal in the $\text{TiC}_{0.5}\text{O}_{0.5}$, on the basis of
414 regular solution model introduced in previous
415 research,^[23] the corresponding equilibrium constant is
416 as described below:

$$K = \frac{a_{[\text{Si}]}}{a_{(\text{SiO}_2)}} \times P_{\text{CO}} \quad [8]$$

419 Similarly, the theoretical concentration of Si in Fe-Si
420 alloy equilibrated with $\text{TiC}_{0.5}\text{O}_{0.5}$ is plotted as a function
421 of the activity of SiO_2 in Figure 9, together with
422 experimental equilibrium concentration of Si in Fe-Si
423 alloy obtained through not only equilibrium experiment
424 but also selective carbothermic reduction of titanium
425 concentrate ore. The concentration of Si in Fe-Si alloy
426 increased from 0 pct, as displayed in Figure 9,
427 approaching the values of equilibrium shown as the
428 solid line. It is worth mentioning that the concentration
429 of Si in Fe-Si alloy ($x_{[\text{Si}]} = 2.7$ pct) obtained through
430 selective reduction of titanium concentrate ore is also
431 very close to the theoretical value of equilibrium in the
432 presence of $\text{TiC}_{0.5}\text{O}_{0.5}$. Therefore, the results of equilib-
433 rium experiments are in good agreement with the
434 theoretical value obtained from thermodynamic analysis
435 and properly explain the phenomenon that part of SiO_2
436 is reduced into Fe-Si alloy during selective reduction
437 process of titanium concentrate ore.

439 IV. CONCLUSIONS

440 Carbothermic selective reduction of titanium concen-
441 trate ore for the preparation of titanium oxycarbide was
442 demonstrated, and SiO_2 was found to be partially reduced
443 into Fe-Si alloy during the process. The equilibrium
444 relationship between Si dissolved in metallic Fe and SiO_2

in corresponding slag in the presence of titanium oxycar- 445
bide was analyzed thermodynamically, and then confir- 446
med by a series of designed equilibration experiments. 447
By controlling the chemical potential of the titanium 448
compound, the concentration of Si in the Fe-Si alloy 449
changes toward the value of equilibrium during the 450
equilibration with slag phase containing SiO_2 . The results 451
of equilibration experiments agree with the thermody- 452
namic analysis and have well explained the phenomenon 453
of partial reduction of SiO_2 during the selective carbother- 454
mic reduction of titanium concentrate ore. 455
456

ACKNOWLEDGMENTS

This work was supported by the National High 457
Technology Research and Development Program of 458
China (863 Program, No. 2012AA062302). 459
460

REFERENCES

1. W.J. Kroll: *Trans. Electrochem. Soc.*, 1940, vol. 78, pp. 35–47. 462
2. G.Z. Chen, D.J. Fray, and T.W. Farthing: *Nature*, 2000, vol. 407, 463
pp. 361–64. 464
3. K. Ono and R.O. Suzuki: *JOM*, 2002, vol. 54, pp. 59–61. 465
4. R.O. Suzuki: *J. Phys. Chem. Sol.*, 2005, vol. 66, pp. 461–65. 466
5. D.R. Sadoway: *JOM*, 1991, vol. 43, pp. 15–19. 467
6. S. Jiao and H. Zhu: *J. Mater. Res.*, 2006, vol. 21, pp. 2172–75. 468
7. S. Jiao and H. Zhu: *J. Alloys Compd.*, 2007, vol. 438, pp. 243–46. 469
8. C. Gao, B. Jiang, Z. Cao, K. Huang, and H. Zhu: *Rare Metals*, 470
2010, vol. 29, pp. 547–51. 471
9. D.G. Jones: *Trans. Inst. Min. Metall.*, 1973, vol. 82, pp. C186–92. 472
10. M. Kamal-Hussein and S.Z. El-Tawil: *Indian J. Technol.*, 1967, 473
vol. 5, pp. 97–100. 474
11. I.E. Grey and A.F. Reid: *Trans. Inst. Min. Metall.*, 1974, vol. 83, 475
pp. 39–46. 476
12. S.K. Gupta, V. Rajakumar, and P. Grieveson: *Metall. Mater.* 477
Trans. B, 1987, vol. 18B, pp. 713–18. 478
13. K. Sun, R. Takahashi, and J. Yagi: *ISIJ Int.*, 1993, vol. 33, 479
pp. 523–28. 480
14. Y. Chen, T. Hwang, M. Marsh, and J.S. Williams: *Metall. Mater.* 481
Trans. A, 1997, vol. 28B, pp. 1115–21. 482
15. M.A.R. Dewan, G. Zhang, and O. Ostrovski: *ISIJ Int.*, 2010, 483
vol. 50, pp. 647–53. 484
16. K.S. Coley, B.S. Terry, and P. Grieveson: *Metall. Mater. Trans. B*, 485
1995, vol. 26B, pp. 485–94. 486
17. M.A.R. Dewan, G. Zhang, and O. Ostrovski: *Metall. Mater.* 487
Trans. B, 2010, vol. 41B, pp. 182–92. 488
18. M.A.R. Dewan, G. Zhang, and O. Ostrovski: *Miner. Process.* 489
Extr. Metall IMM Trans., 2010, vol. 120, pp. 111–17. 490
19. B. Jiang, J. Xiao, K. Huang, J. Hou, S. Jiao, and H. Zhu: *J. Am.* 491
Ceram. Soc., 2017, vol. 100, pp. 2253–65. 492
20. J. Xiao, B. Jiang, K. Huang, S. Jiao, H. Zhu: *7th Int. Symp. High* 493
Temp. Metall. Process. 2016. 494
21. E.C. DeWys and W.R. Foster: *J. Am. Ceram. Soc.*, 1956, vol. 39, 495
pp. 372–76. 496
22. S.A. Rezan, G. Zhang, and O. Ostrovski: *Metall. Mater. Trans. B*, 497
2012, vol. 43B, pp. 73–81. 498
23. B. Jiang, K. Huang, Z. Cao, and H. Zhu: *Metall. Mater. Trans. A*, 499
2012, vol. 43A, pp. 3510–14. 500
24. B. Jiang, N. Hou, S. Huang, G. Zhou, J. Hou, Z. Cao, and H. 501
Zhu: *J. Solid State Chem.*, 2013, vol. 204, pp. 1–8. 502
25. R.J. Fruehan: *Metall. Trans.*, 1970, vol. 4, pp. 865–70. 503
26. C.W. Bale, P. Chartrand, S. Deckerov, G. Eriksson, K. Hack, R.B. 504
Mahfoud, J. Melancon, A.D. Pelton, and S. Petersen: *Calphad*, 505
2002, vol. 26, pp. 189–228. 506
507