

1                   **Continuous water adsorption states promoted by Ni<sup>2+</sup> confined in a smectite**

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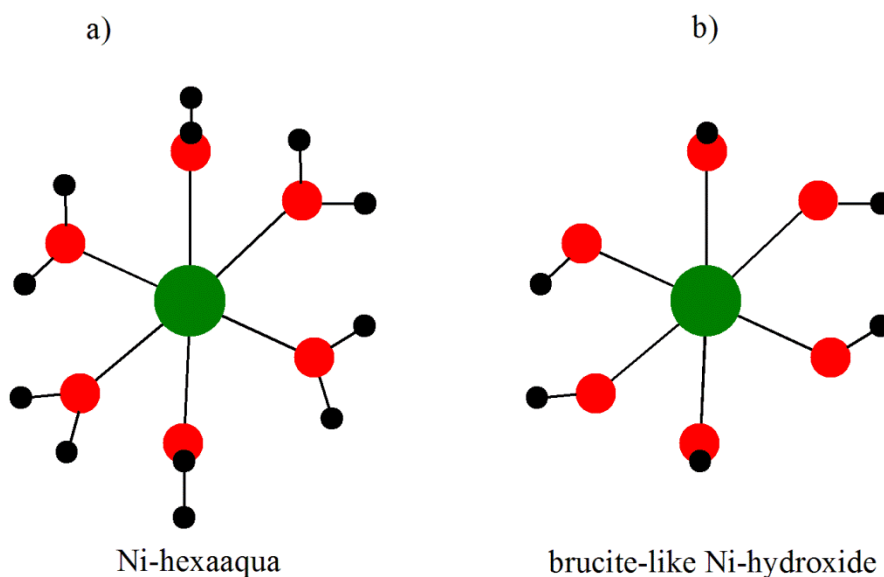
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17                   **Abstract:**

18                   The water adsorptions isotherms of the smectite Ni-fluorohectorite have been obtained from the  
19                   relative humidity dependence of X-ray diffraction intensities, and it was observed that the transitions  
20                   between the stable hydration states are continuous; a behavior which is markedly different from  
21                   previous studies of Na-fluorohectorite and Li-fluorohectorite smectite. It was also observed that the  
22                   environmental history of the clay samples is important for the detailed behavior displayed by these  
23                   isotherms. In addition, based on thermogravimetric and differential scanning calorimetry  
24                   measurements, a complete Ni-fluorohectorite dehydrated state only can be reached at long (several  
25                   hours) isotherm times at 70 °C, or at shorter times (minutes) above 150 °C. Our observations are  
26                   consistent with the existence of various forms of Ni<sup>2+</sup>-H<sub>2</sub>O complexes in Ni-fluorohectorite, and we  
27                   suggest that the present observations can be extended to other smectites with transition metals as  
28                   interlayer charge compensating cations, and consequently have significant practical consequences in  
29                   materials science and other areas where smectite clays are important.

30                   **Graphical Abstract:**



31  
 32 a) Hydration structure of Ni<sup>2+</sup> hexaaqua, which occurs for high RH conditions. b) Structure of Ni-  
 33 brucite.

34 **Introduction**

35 Smectites or clay minerals are among the most abundant minerals on Earth and are rarely  
 36 found in nature without intercalated H<sub>2</sub>O. The large scientific and technological interest in smectites is  
 37 connected to their ability to swell, most commonly caused by their interactions with H<sub>2</sub>O. (Sposito et  
 38 al., 1999) Therefore whether smectites are used for, CO<sub>2</sub>, (Giesting et al., 2012a; Michels et al., 2015;  
 39 Schaefer et al., 2012) capture, nuclear waste storage (Sellin and Leupin, 2013), heavy metal  
 40 removal (Oueslati et al., 2011) or for drug delivery (Aguzzi et al., 2014), understanding their  
 41 interactions with H<sub>2</sub>O is of paramount importance.

42 Smectite contain charge-compensating cations in their nano-sized quasi 2-dimensional  
 43 interlayers (pore width less than 2 nm). These cations balance the negative surface charge of the  
 44 smectite sheets, and as a whole this system is responsible for the well-known swelling dynamics  
 45 (Bordallo et al., 2008). Either crystalline (intercalation), or osmotic (exfoliation) swelling occurs  
 46 when clay particles interact with foreign molecules, such as H<sub>2</sub>O (Hansen et al., 2012; Tenorio et al.,  
 47 2010) or CO<sub>2</sub> (Giesting et al., 2012b; Hemmen et al., 2012) The variation of the basal spacing of  
 48 hydrated smectites is connected primarily to the number of H<sub>2</sub>O molecules intercalated within the  
 49 layers, and equilibration at different relative humidity and temperatures. The H<sub>2</sub>O molecules may be  
 50 arranged within the smectite interlayer space in structures corresponding to integral number of water  
 51 layers (WL) (Ferrage et al., 2010; Tamura et al., 2000) or more complex H<sub>2</sub>O structures (Morrow et  
 52 al., 2013; Tenorio et al., 2010). In order to study this crystalline swelling process many methods have  
 53 been employed, such as quasi-elastic neutron scattering (QENS) (Gates et al., 2012; Malikova et al.,  
 54 2007; Malikova et al., 2006; Mitra et al., 2013), Neutron Diffraction (Hawkins and Egelstaff, 1980;  
 55 Skipper et al., 1994; Zhou et al., 2015), X-ray diffraction (XRD) (da Silva et al., 2002; Dazas et al.,

56 2013; Ferrage et al., 2010; Ferrage et al., 2005; Ferrage et al., 2011; Oueslati et al., 2011; Tenorio et  
57 al., 2008; Tenorio et al., 2010), NMR (Fleury et al., 2013; Tenorio et al., 2008; Tenorio et al., 2010)  
58 and molecular dynamics simulations (Ebrahimi et al., 2014; Skipper et al., 1991a; Tambach et al.,  
59 2004; Zheng and Zaoui, 2011).

60 Synthetic clay minerals have the advantage that they possess a more homogeneous charge  
61 distribution, and also contain significantly fewer impurities (e.g. carbonates, (hydr)oxides, silica, and  
62 organic matter) than their natural counterparts (Bergaya and Lagaly, 2013). Fluorohectorites (Fh)  
63 have often been used as a representative and clean model system of natural smectite clays (Ayyappan  
64 et al., 1996; da Silva et al., 2003; Dazas et al., 2013; Hansen et al., 2012; Hemmen et al., 2010;  
65 Malikova et al., 2007; Tenorio et al., 2010). Hectorite is a 2:1 phyllosilicate, meaning that the layers  
66 are formed by two inverted silicate tetrahedral sheets, sharing their apical oxygen with one octahedral  
67 sheet sandwiched in between and it is classified as a trioctahedral smectite since  $\text{Li}^+$  substitutes for  
68  $\text{Mg}^{2+}$  in the octahedral sheet sites, which are fully occupied. The proportion  $x$  of the  $\text{Li}^+$  ions in the  
69 clay sheet structure itself determines the surface charge of the layers, which are held together in the  
70 stacked structure by sharing the interlayer cation.

71 Considering that the type of intercalated cation substantially influences the physicochemical  
72 behaviors of the system (Chatterjee et al., 2004), the present work investigates the effects of  
73 intercalating Fh with the divalent cation  $\text{Ni}^{2+}$ , hereafter termed NiFh clay. Ni-intercalated cations in  
74 smectites have been extensively studied by the scientific community (Grassi et al., 2014; Michels et  
75 al., 2014; Oueslati et al., 2011; Pascual et al., 2014; Pitteloud et al., 2001; Skipper et al., 1991b;  
76 Zheng and Zaoui, 2011; Zheng et al., 2010), partly motivated by substantial increase of the amounts  
77 of heavy metals in the environment. Our interest in NiFh is also partly based in the well-known fact  
78 that  $\text{Ni}^{2+}$  and water in basic pH conditions form a Ni-hydroxide (Ni-brucite) structure, with chemical  
79 formula  $\text{Ni}(\text{OH})_2$ , and that the Ni-brucite structure coexist with the NiFh particles (Michels et al.,  
80 2014). Therefore new features related to simultaneous NiFh-water and the Ni-brucite-water  
81 interactions could be expected, and our aim in the present work is to investigate and understand the  
82 origin of such features.

83 We here report on two different types of X-ray diffraction (XRD) experiments that were  
84 performed on NiFh powder. The first type consisted in mapping small interlayer displacements  
85 measured from the (001) Bragg peak positions as a function of relative humidity (RH) at two different  
86 fixed temperatures. In the second type of XRD experiment, the particle thickness and the lattice strain  
87 were calculated using the standard Williamson-Hall analysis method, based on the evolution of the  
88 (001) as well as higher order Bragg peaks measured at selected values of RH. In addition, thermal  
89 analysis was performed, demonstrating that NiFh clay shows a quite distinct behavior compared to  
90 that observed in other smectites upon water adsorption. We explore this in detail and suggest possible  
91 explanations for these behaviors.

## 92 1. Materials and Methods

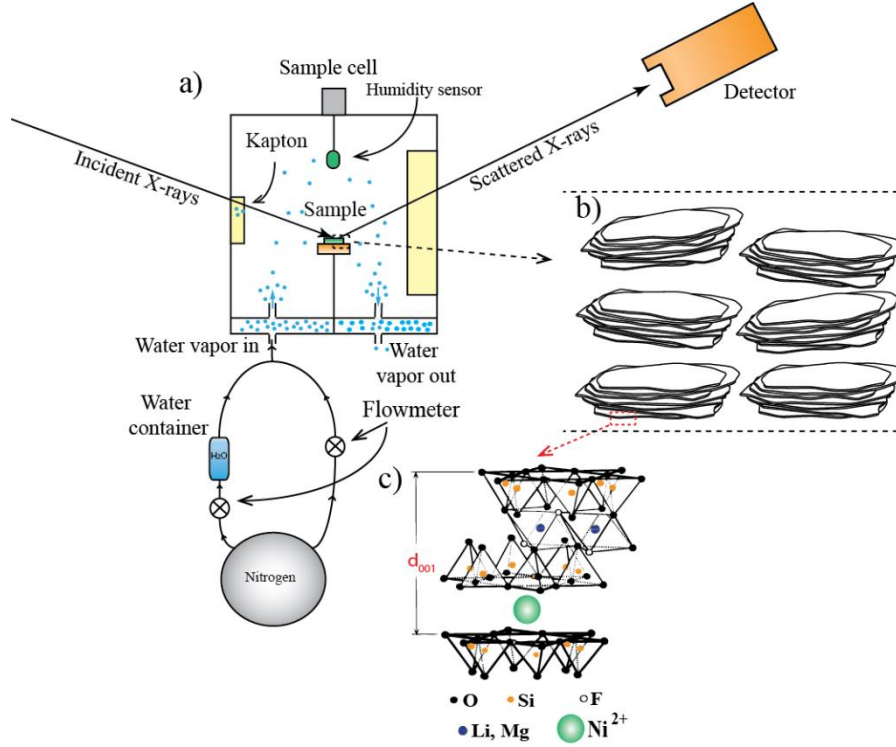
### 93 1.1 Sample Preparation

94 The Ni-fluorohectorite (NiFh) samples were obtained by a cation exchange protocol starting  
95 from Li-Fluorohectorite (LiFh), described by our group previously (Grassi et al., 2014; Michels et al.,  
96 2014). This consists in mixing LiFh and  $\text{NiCl}_2 \cdot 6\text{H}_2\text{O}$  in a water solution and collecting the powder  
97 after centrifuging. The NiFh powder is then washed in deionized water several times in order to  
98 remove residual salt. The nominal chemical formula of the product thus obtained is  
99  $\text{Ni}_{0.6} \cdot [\text{Mg}_{(4.8)}\text{Li}_{1.2}] \text{Si}_8\text{O}_{20}\text{F}_4$  per unit cell. The pH was measured to be 9.5 during the cation  
100 exchange process, as adjusted by the clay dispersion itself. The samples are left to dry and after  
101 evaporation of the water, the sedimented film is collected and placed in the sample chamber described  
102 in Figure 1. Previous to the XRD analyses, two kinds of samples were prepared from the pristine  
103 NiFh:

104 **A. Non-preheated NiFh:** Two samples of clay mineral with the same initial relative humidity  
105 (RH) were placed under  $\text{N}_2$  atmosphere and their structural change was monitored in-situ using XRD,  
106 at two different temperatures: one at 70 °C and another at 20 °C, without prior preheating of the  
107 samples.

108 **B. Pre-heated NiFh:** Two samples of clay minerals were placed on a heater plate and pre-heated  
109 at 150 °C under  $\text{N}_2$  atmosphere until their dehydrated structure was obtained, which was monitored  
110 in-situ using XRD. After dehydration, the temperature of one sample was lowered to 70 °C and a  
111 number of XRD patterns were collected for various values of RH, as detailed below. In addition, this  
112 same procedure was repeated with the other set of pre-heated NiFh at 20 °C.

113 After the sample preparation described above, the treated NiFh powder was placed inside a  
114 sample cell, in which RH and temperature could be accurately controlled. This custom made cell  
115 consists of a closed cylindrical chamber with a small kapton window for XRD transmission, a heater  
116 plate as sample holder in the center and a humidity/temperature sensor inside. The RH was controlled  
117 by regulating a pure nitrogen ( $\text{N}_2$ ) flux into the chamber. The  $\text{N}_2$  gas was separated into two branches:  
118 one going into a water container, where  $\text{N}_2$  was saturated with water vapor, and the other conducting  
119 the dry  $\text{N}_2$  gas to the chamber. Both channels had flow meters installed in their respective flow paths  
120 for reproducibility control of the applied RH. This apparatus allows for mixing humid and dry gases  
121 in a controlled manner. A sketch of this experimental setup is shown in Figure 1.



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**Figure 1:** a) Schematics of the sample cell which consists of a heater plate, kept inside a closed cylindrical sample chamber together with a humidity/temperature sensor. b) Aligned sedimented film of smectite clay mineral particles. c) The layer structure of the fluorohectorite with the charge compensating cation Ni<sup>2+</sup> confined in the unit cell interlayer space ( $d_{001}$ ).

## 127 1.2 XRD Experiments

128 XRD measurements on non-preheated samples were studied using an in-house (NTNU,  
129 Trondheim) Bruker NanoSTAR X-ray scattering instrument, attached to a Xenocs X-ray source with a  
130 copper anode. Preheated samples were measured at the Brazilian Synchrotron Light Source  
131 Laboratory (LNLS) at the XRD2 beam line, using a wavelength of  $\lambda = 0.154991$  nm.

132 The scattered X-ray intensities are plotted in terms of the scattering vector defined as  $q =$   
133  $4\pi \sin(\theta)/\lambda$ , where  $2\theta$  is the scattering angle. In order to obtain the peak positions and de-convolute  
134 the peak widths, which are intrinsic to the sample, a pseudo-Voigt peak shape function was used. For  
135 our purpose this function provides a good approximation to the convolution of Lorentzian and  
136 Gaussian functions (Wertheim et al., 1974):

$$137 \Phi(q) = \mu \frac{2}{\pi} \frac{\Gamma}{4(q - q_c)^2 + \Gamma^2} + (1 - \mu) \frac{\sqrt{4 \ln 2}}{\sqrt{\pi} \Gamma} \exp \left[ \frac{-4 \ln 2 (q - q_c)^2}{\Gamma} \right] \quad (1)$$

138  $\mu$  is the proportional constant that ranges from zero to one and it is responsible for peak shape  
139 curve being more Lorentzian or Gaussian while  $\Gamma$  denotes the experimental peak width of curves.

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$$\mu = 1.36603 \left( \frac{\omega_L}{\Gamma} \right) - 0.47719 \left( \frac{\omega_L}{\Gamma} \right)^2 + 0.11116 \left( \frac{\omega_L}{\Gamma} \right)^3$$

$$\Gamma = \left( \omega_G^5 + 2.69269 \omega_G^4 \omega_L + 2.42843 \omega_G^3 \omega_L^2 + 4.47163 \omega_G^2 \omega_L^3 + 0.07842 \omega_G \omega_L^4 + \omega_L^5 \right)^{1/5}$$

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$\Phi(q)$  is the interference function related to the measured intensity by  $I(q) = C(q)\Phi(q)$ . The  $C(q)$  parameter is related to the scattering structure factor and to the Lorentz-polarization factor, and in the region around each  $(00\ell)$  peak,  $C(q)$  can be approximated by a constant since its value fall steeply at low  $q$  and varies slowly with  $q$  in the range of the peak widths. The  $\omega_L$  and  $\omega_G$  denotes the peak widths of the Lorentzian and Gaussian components that represent the sample and instrumental contribution to the line broadening respectively. At NTNU the FWHM Gaussian instrumental resolution width, obtained from the fittings, was  $\omega_G \approx 0.196 \pm 0.005 \text{ nm}^{-1}$  (Cases of Figures 2 (b) and (d) below), and at LNLS it was  $\omega_G \approx 0.043 \pm 0.009 \text{ nm}^{-1}$  (Cases of Figures 2 (a), (c) and Fig. 3 below).

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Only for the Williamson-Hall procedure (Case of Figures 5 and 6 below) a Ge (111) single crystal analyzer was utilized at LNLS in front of the diffracted X-ray beam detector in order to sharpen the instrumental resolution, thus reducing the effects of instrumental broadening on the diffractograms but also reducing significantly the XRD intensity. This setup allows the data to be fitted with only a Lorentzian i.e. the instrumental width contribution can be neglected, and the pseudo-Voigt function is not needed.

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### 1.3 Thermogravimetric Analysis

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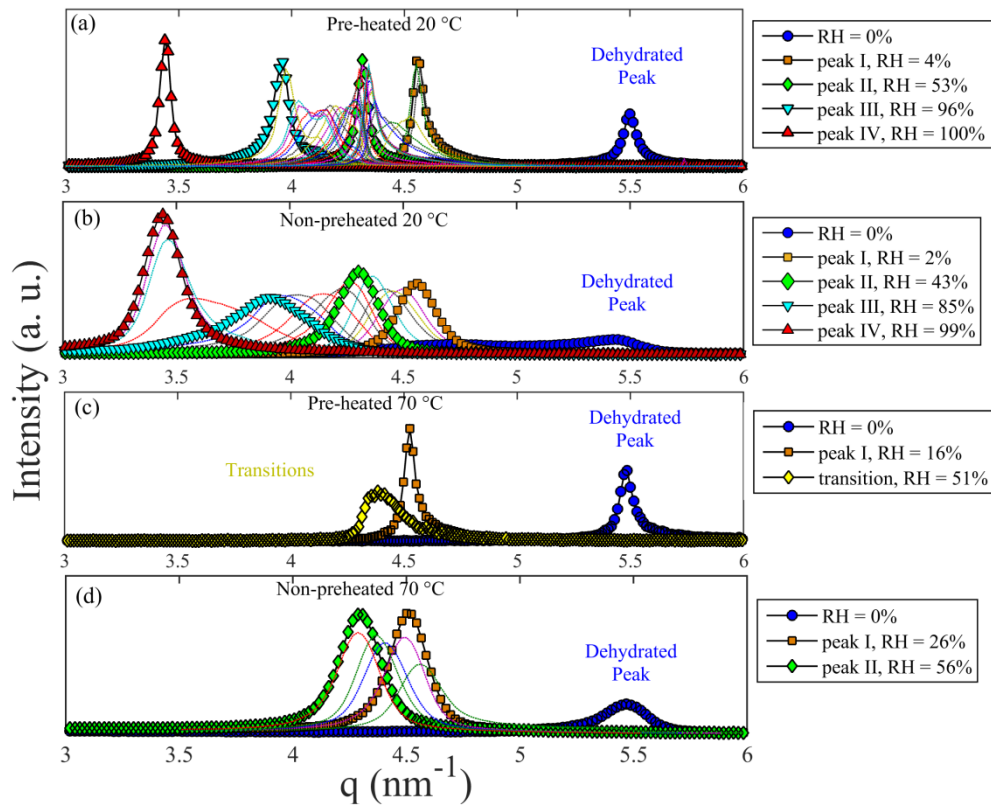
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Thermal gravimetric analysis (TGA) and Differential Scanning Calorimetry (DSC) measurements were performed using respectively a TG 209 F1 LIBRA and a DSC 214 Polyma from NETZSCH. The experimental conditions were:  $\text{N}_2$  atmosphere  $\approx (40 \text{ ml/min})$ , heating rate  $(10 \text{ }^\circ\text{C}\cdot\text{min}^{-1})$ , platinum top-opened crucible (TGA) and aluminum pierced crucible (DSC). The samples weights were approx. 20 mg. The results allowed quantifying the amount of initial  $\text{H}_2\text{O}$  in the clays layers as well as to describe the different phases as follows. Two types of measurements were performed using the TGA: one to measure the mass loss as a function of temperature carried out between  $25 \text{ }^\circ\text{C}$  to  $400 \text{ }^\circ\text{C}$  and the other to observe the evolution of the mass loss at constant temperature. The former can give the temperature where the  $\text{H}_2\text{O}$  populations de-adsorb from the sample, while in the latter (by fixing the temperature at  $70 \text{ }^\circ\text{C}$  and  $150 \text{ }^\circ\text{C}$ ) one can determine the mass loss as a function of time. Consequently, the isotherm method allows us to directly compare the relation between the initial  $\text{H}_2\text{O}$  content and the XRD Bragg peak position. On the other hand, the differential scanning calorimetry (DSC) measurements, performed between room temperature and  $400 \text{ }^\circ\text{C}$ , allows to quantify the  $\text{H}_2\text{O}$  transitions in the samples.

171 **2. Results**

172 **A. XRD:** It is well-known that smectite clays at finite values of RH expand (swell) in the z-  
 173 direction (00 $\ell$ ). Therefore, the peak positions  $q_c$ , obtained from Eq. (1), for a given RH are related to  
 174 the hydration state, which depends on the type of interlayer cation. As reported in several previous  
 175 publications(da Silva et al., 2002; Ferrage et al., 2010; Ferrage et al., 2005; Ferrage et al., 2011; Gates  
 176 et al., 2012), for Na<sup>+</sup> the stable hydration states correspond to zero water layer (0WL) (dehydrated  
 177 peak), one water layer (1WL) and two water layers (2WL). For the case of Li<sup>+</sup> as interlayer  
 178 cation(Tenorio et al., 2010), the stable hydration states are 0WL, 1WL, one-and-a-half WL (1.5 WL)  
 179 and 2WL.



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181 **Figure 2:** The diffraction peaks corresponding to the stable hydration states (see Figure 3) in NiFh with the  
 182 corresponding RH for a sample that is (a) pre-heated T = 20 °C, (b) non-pre-heated T = 20 °C, (c) pre-heated T  
 183 = 70 °C and (d) non-pre-heated 70 °C. The thin continuous lines represent transition states in between the stable  
 184 hydration states (i.e. peaks I, II, III and IV). The non-pre-heated samples measured at NTNU have broader peak  
 185 widths due to the instrumental contribution as explained in the text.

186 In this work, the expansion for NiFh was carefully followed by monitoring the position of the  
 187 (001) Bragg peak for increasing values of RH. Figure 2 shows the Bragg peak position as a function  
 188 of  $q$  for different values of RH. The value of  $d_{001}$ , i. e., the distance between the basal planes, was then  
 189 calculated using the relation:

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$$q_c = \frac{2\pi}{d_{001}} \quad (2)$$

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The peak at  $q_c \approx 5.5 \text{ nm}^{-1}$  (blue  $\circ$ ) found in all diagrams, represent the dehydrated condition corresponding to basal distance  $d_{001} = 1.14 \text{ nm}$ , implying that there is not enough space between the clay layers to accommodate  $\text{H}_2\text{O}$  molecules. The other peaks, hereafter denoted as peak I to peak IV, correspond to the conditions where the  $\text{H}_2\text{O}$  intercalates into the interlayer space

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For the pre-heated and non-pre-heated samples measured at  $T = 20 \text{ }^\circ\text{C}$  (two upper panels), the peak positions in Figure. 2 are: peak I with  $q_c \approx 4.55 \text{ nm}^{-1}$  at RH = 4%, peak II with  $q_c \approx 4.31 \text{ nm}^{-1}$  at RH = 53%, peak III with  $q_c \approx 3.96 \text{ nm}^{-1}$  at RH 96% and finally peak IV with  $q_c \approx 3.44 \text{ nm}^{-1}$  at RH = 100%. On the other hand, for samples pre-heated and non-pre-heated and measured at  $T = 70 \text{ }^\circ\text{C}$  (two lower panels) the low  $q$  peaks (III and IV) are not observed. Moreover, although the peak positions related to stable peaks remain the same when compared with the samples at  $20 \text{ }^\circ\text{C}$  they are shifted to different RH values. In addition, for the non-pre-heated sample measured at  $T = 70 \text{ }^\circ\text{C}$  a transition from peak I to peak II is observed, while the range around peak II (green  $\diamond$ ) contains multiple peaks for various RH values. This can be considered as a continuum of transitions between peak I (orange  $\square$ ) and peak III (cyan  $\nabla$ ).

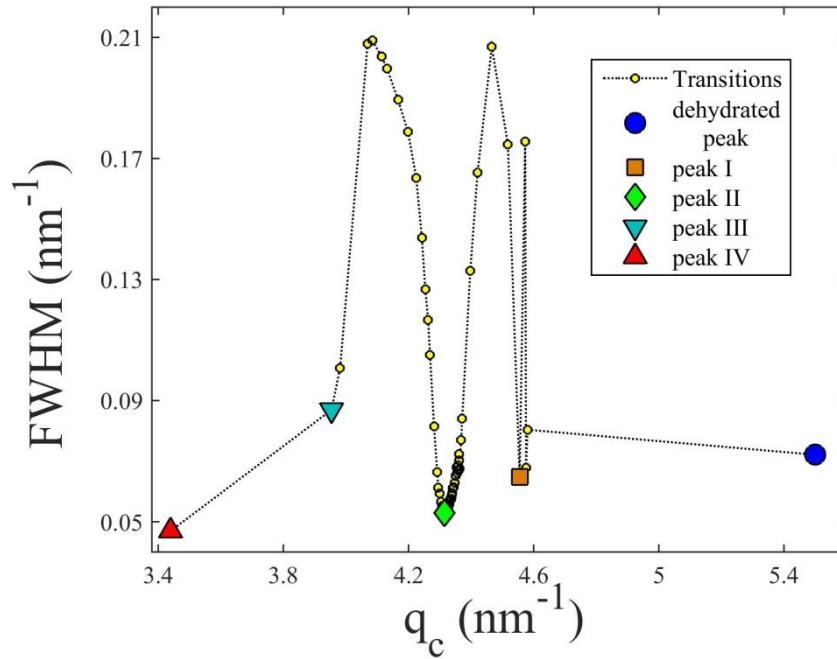
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In order to check whether these peaks are stable hydration states or not, a pseudo-Voigt function, Eq. (1), was used to fit the diffractograms depicted in Figure 2. The resulting FWHM values,  $\omega_L$ , for the pre-heated sample measured at  $20 \text{ }^\circ\text{C}$  are plotted as a function of  $q_c$  in Figure 3.





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209 **Figure 3:** FWHM ( $\omega_L$ ) as a function of the position of the Bragg peak  $q_c$  for pre-heated sample at  $T = 20^\circ\text{C}$ . The  
 210 stable hydration states (large symbols) show narrower FWHM compared to of the transient peaks.

211 It is known that the FWHM is related to the structural ordering of the crystalline material.  
 212 More ordered and/or thicker structures will generally give smaller FWHM values. Therefore the peak  
 213 positions,  $q_c$ , where the FWHM shows a minimum (large symbols in Figure 3) can be related to  
 214 hydration states where the crystalline structure has a higher degree of ordering. This distinguishes  
 215 them from the broader diffractograms of the mixed intercalated states, represented as transient states  
 216 (yellow dots  $\circ$ ) in Figure 3.

217 The average diameter of a freely rotating water molecule is approximately 0.29  
 218 nm (Israelachvili, 2011). When water is present in the interlayer space of clay minerals, the  $d_{001}$  value  
 219 is often related to the amount of  $\text{H}_2\text{O}$  “layers” present. For each hydration state there are intervals of  
 220 corresponding  $d_{001}$ . If the  $d_{001}$  value from the dehydrated state (1.146 nm) is subtracted from the  
 221 average of the stable  $d_{001}$  values, the resulting number is the effective interlayer space occupied by  
 222  $\text{H}_2\text{O}$  molecules. The results are shown in Table 1 for  $T = 20^\circ\text{C}$ .

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224 Table 1: The average value of the  $d_{001}$  corresponding to each peak type (defined as I to IV in Figure 2). The  
 225 value of 1.146 nm, which is the  $d_{001}$  of 0WL, is subtracted in order to calculate the amount of  $\text{H}_2\text{O}$  “layers” for  
 226 stable hydration state.

Peak type	Mean $d_{001}$ (nm)	Subtracting the 0WL (1.146 nm)	Effective number of $\text{H}_2\text{O}$ Layers
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<b>Peak I</b>	$1.3749 \pm 0.0034$	0.2289	0.80
<b>Peak II</b>	$1.4439 \pm 0.0040$	0.2979	1
<b>Peak III</b>	$1.5685 \pm 0.0267$	0.4225	1.5
<b>Peak IV</b>	1.8269	0.6809	2.3

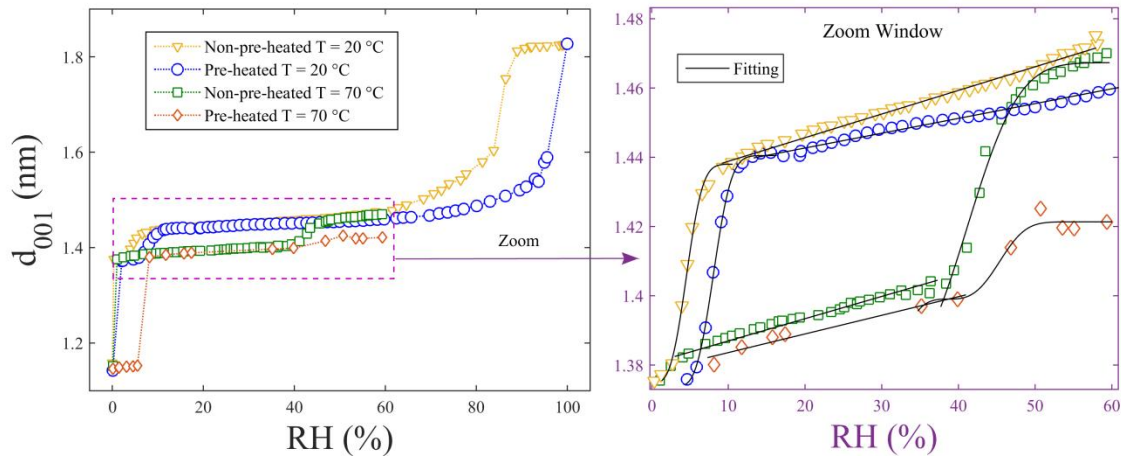
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Now we turn to the analysis of peak positions of the diffractograms in Figure 2, which can be represented in terms of  $d_{001}$  stacking distances using Eq. (2), as a function of RH. The results are shown in Figure 4.



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**Figure 4:** Stacking distance  $d_{001}$  as function of RH. The value of  $d_{001}$  is related to the amount of  $H_2O$  within the interlayer space. The inset shows the continuous transition from peak I to peak II. The transition from peak II to peak III is not abrupt as is evident since  $d_{001}$  grows continuously. The zoom window (right panel) shows the fit of an Avrami-like model (Eq. 4) to the data see description in the text. Pre-heated samples are exposed to pure dry  $N_2$  and heated to 150 °C until the dehydrated Bragg peak position is reached and then the temperature is lowered to 20 °C or 70 °C. The non-pre-heated samples are dried at 20 °C or 70 °C in pure  $N_2$  atmosphere until the dehydrated Bragg peak position is reached (see Methods).

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As RH increases,  $d_{001}$  increases as well. For low RH values, the transition from the dehydrated state to the state containing peak I is seen to be abrupt in all cases, whilst the transition from peak I to peak II displays an “S-shaped” change. The transition from peak II to peak III is not clearly visible, instead a continuous increment of  $d_{001}$  is observed from 2% to 96% of RH until  $d_{001}$  clearly changes to the peak IV position. The results indicate continuous peak shifts in the transition regions, which have not been observed in previous hydration studies of synthetic smectites (Hemmen et al., 2010; Tamura et al., 2000), and our results clearly differ from the abrupt stepwise expansion of the layer-to-layer distance with increasing water activity normally reported for smectites (Dzas et al., 2015). The change in temperature (20 or 70 °C) does not seem to affect either the continuous behavior or the presence of an abrupt transition for low RH values. However, the initial sample condition clearly influences the transition between peaks I and II. In the non-pre-heated case, this transition is observed

250 to be shifted to higher RH values at T = 70 °C (yellow and green curves in the right panel of Fig. 4) ,  
 251 whereas for the pre-heated sample at 70 °C the transition is not clearly observed.

252 For the sample pre-heated and measured at T = 20 °C (blue curve), the evolution of  $d_{001}$ , after  
 253 the transition from peak I to II, could be fitted with a linear function, with slope  $\eta$  as given by Eq. 3,  
 254 up to approximately RH = 60%.

$$255 \quad d_{001} = \eta \cdot RH \quad (3)$$

256 In the case of the measurements performed at T = 70 °C, this linear behavior is also observed for the  
 257 pre-heated sample (red curve), as well as for the non-pre-heated sample below the transition from  
 258 peak I to peak II (green curve). In addition, the evolution of  $d_{001}$ , during the transition from peak I to  
 259 peak II, was found to be well described by an exponential law:

$$260 \quad d_{I \rightarrow II} = d_{II} - (d_{II} - d_I) \exp[-K(RH - RH_I)^\alpha] \quad (4)$$

262 where,  $d_{II}$  is the interlayer spacing of the hydration state corresponding to peak II,  $d_I$  is the interlayer  
 263 spacing of the hydration state corresponding to peak I, and  $RH_I$  is the RH value at which the transition  
 264 begins for the increasing RH. Thus the whole range shown in the right panel of Fig. 4 could be fitted  
 265 with a combined equation, and the resulting parameters, represented by the continuous lines in the  
 266 right panel of Figure 4, are shown in Table 2.

267 Table 2: Fitting parameters obtained from Eq. (3) and (4).

Parameters	Pre-heated T = 20 °C	Pre-heated T = 70 °C	Non-pre- heated T = 20 °C	Non-pre- heated T = 70 °C
$\eta$ (slope)	$(4.3 \pm 0.2) \cdot 10^{-4}$	$(4.8 \pm 0.6) \cdot 10^{-4}$	$(6.8 \pm 0.2) \cdot 10^{-4}$	$(6.4 \pm 0.6) \cdot 10^{-4}$
$d_{II}$	$1.4410 \pm 0.0005$	$1.420 \pm 0.002$	$1.441 \pm 0.003$	$1.455 \pm 0.004$
$d_I$	$1.373 \pm 0.001$	$1.399 \pm 0.002$	$1.375 \pm 0.005$	$1.398 \pm 0.003$
$RH_I$	$3.5 \pm 0.6$	$38 \pm 1$	$0 \pm 1$	$36 \pm 1$
$K$	$(7 \pm 1) \cdot 10^{-3}$	$(2 \pm 1) \cdot 10^{-3}$	$(7 \pm 5) \cdot 10^{-3}$	$(2 \pm 1) \cdot 10^{-3}$
$\alpha$	3	3	3	3

268

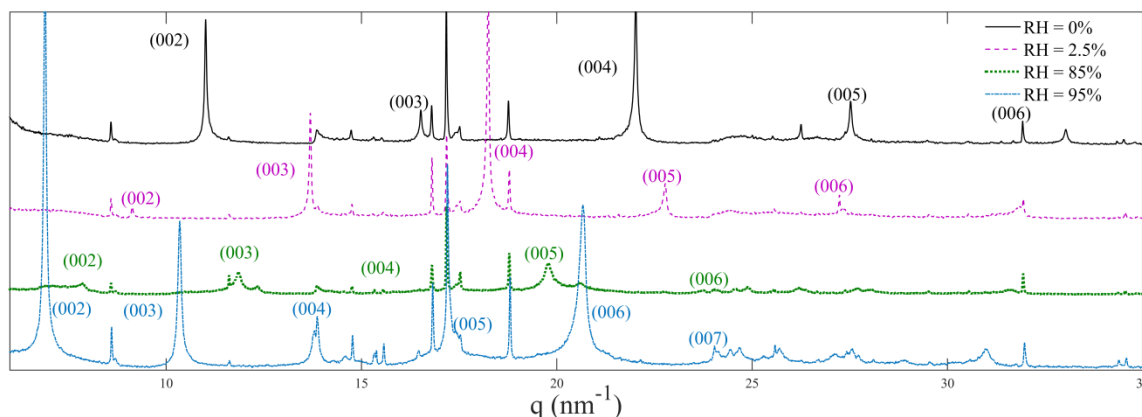
269 The slope,  $\eta$ , obtained by fitting in the linear regime, represents how H<sub>2</sub>O intercalates into the  
 270 interlayer space of NiFh in the continuous regime. Previous work performed at T = 15 °C on  
 271 NaFh(Hemmen et al., 2010) gave an estimated slope in the same range as those obtained here.  
 272 Furthermore, based on Table 2, the  $\eta$  values are related to the initial condition of the clays. The pre-  
 273 heated samples have approximately the same values of  $\eta$ , which are smaller than those observed for

274 the non-pre-heated samples. Thus our results indicate that the intercalation process in this region is  
 275 temperature independent (at least in the temperature range probed in this study), i.e. the rate of H<sub>2</sub>O  
 276 intercalation ( $\eta$ ) in the non-transition region seems to depend only on the environmental history of the  
 277 clays. It is clear however that the temperature (20 or 70 °C) shifts the RH value for which the  
 278 transition peak I  $\rightarrow$  peak II occurs. This is reflected by the fitting parameter  $RH_I$  that shows that it is  
 279 more difficult to intercalate H<sub>2</sub>O at higher temperatures. Similar observations were reported by da  
 280 Silva et al. in Na-fluorohectorite(da Silva et al., 2003).

281 The parameters  $\alpha$  and  $K$ , defined in Eq. (3) may be discussed in terms of the Avrami  
 282 equation(Gasparini et al., 2013), where  $K$  represents the transition rate constant, while  $\alpha$  is an integer  
 283 between 1 and 4, where  $\alpha = 3$  is normally interpreted as the case of 3-dimensional nucleation growth.  
 284 However, while the standard Avrami model describes time dependent processes, the present case is an  
 285 equilibrium process. The equation can however be adapted to the present case where the cation is  
 286 already solvated and the water only fills the interlayer space. The fitting shows that  $K$  is independent  
 287 of the initial condition of the sample (pre-heated or non-pre-heated) and is higher for  $T = 20$  °C than  
 288 at  $T = 70$  °C.

289 Together, these considerations, taken with appropriate care, show that smectite clay hydration  
 290 can be very sensitive to initial sample preparation and environmental history. Similar conclusions  
 291 regarding preparation and environmental history were reported in a work on the synthetic smectite  
 292 laponite by Cummins et al. (Cummins, 2007). We suggest that such behaviors also may apply to other  
 293 of the transition metals intercalated smectites.

294 **B. Layer Thickness and Lattice Strain:** The obtained XRD peak intensities are shown in  
 295 Figure 5 for the pre-heated sample at  $T = 20$  °C.



296 **Figure 5:** XRD (00 $l$ ) intensities of the NiFh samples at different RH. In this case, the use of crystal analyzer  
 297 makes the instrumental contribution to the line broadening negligible.

299 Considering that the contributions to the FWHM of the peaks are average thickness of the  
 300 clay particle stacks, plus the strain or disorder in the lattice spacing, we make use of a Williamson-  
 301 Hall (W-H)(da Silva et al., 2002) plot to analyze these data as follows:

302

$$\omega_L = \frac{2\pi}{Nd} + \xi q \quad (5)$$

303

where  $\omega_L$  is the FWHM obtained from the Lorentzian fitting of the XRD curves shown in Figure 5,

304

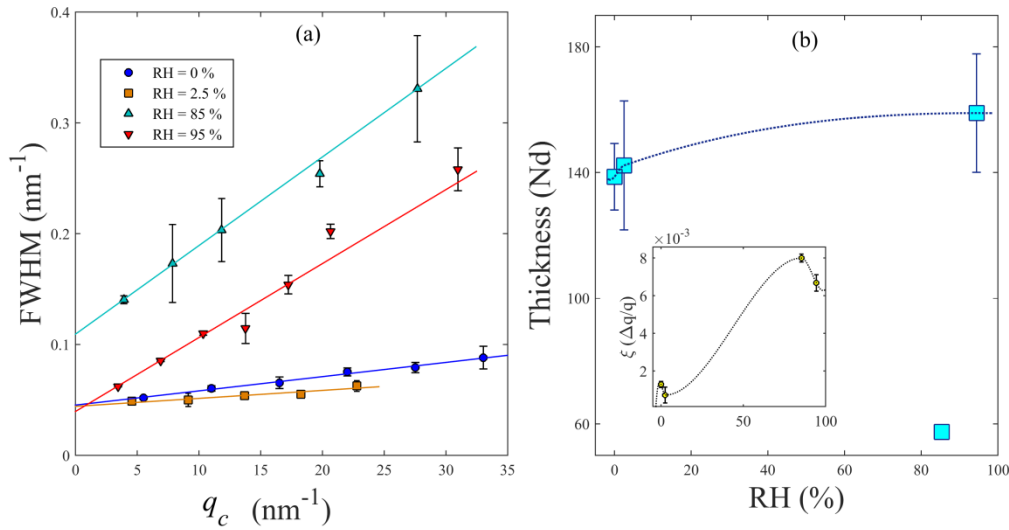
$Nd$  is the average particle stack thickness, obtained from the extrapolation of the straight line to  $q = 0$ ,

305

and  $\xi$  is the average lattice strain, defined as  $\frac{\Delta q_c}{q_c}$ , or disorder, obtained from the slope of the straight

306

line. The results of this analysis, performed for four different values of RH, are presented in Figure 6.



307

308

**Figure 6:** a) Williamson-Hall (W-H) plot, for RH values at 0%, 2.5%, 85% and 95%, with the slope and

309

intercept giving the average strain and the average particle thickness respectively. b) The estimated particle

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thickness and lattice strain (inset).

311

Figure 6 (a) shows the W-H plots for 0%, 2.5%, 85% and 95% of RH for pre-heated samples

312

studied at  $T = 20^\circ\text{C}$ , from which both the average particle thickness and lattice strain were extracted

313

and presented in Figure 6 (b). The particle thickness,  $Nd = 146 \pm 11$ , shown Figure 6 (b), remains

314

approximately constant for 0%, 2.5%, 95% of RH. The exception is the case of RH = 85 %, where

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low particle thickness value may be attributed to the fact that at this condition the sample is in a

316

transition between the regions of the peak II to peak III (as shown in Figure 4). In this case, the

317

intercalated H<sub>2</sub>O might not occupy the clay interlayers homogeneously, resulting in an increase of the

318

lattice strain (see inset) and thus affecting the stability. The minimum strain, at low RH values, can be

319

interpreted as a condition where the clay NiFh has its basal planes more ordered in the stacking layers,

320

i.e. closer to each other due to the lack of intercalated H<sub>2</sub>O. On the other hand, an increase of H<sub>2</sub>O

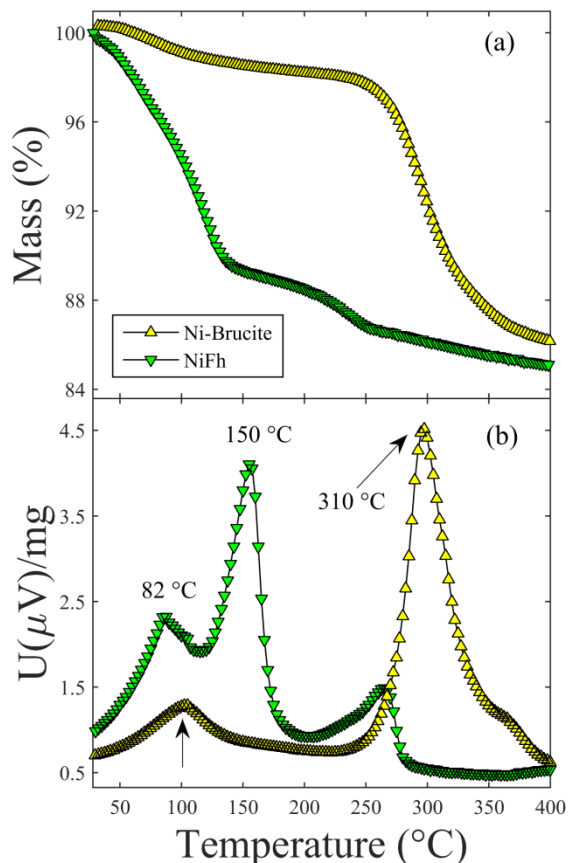
321

(higher humidity) disassociates the basal planes from each other allowing for more freedom and thus

322

possibly a more disordered (higher lattice strain) layered orientation.

323 **C. TGA and DSC:** The H<sub>2</sub>O adsorbed differentially in the sample can be identified by  
 324 measuring the mass loss as function of temperature. Figure 7 (a) shows the mass loss of both NiFh  
 325 and Ni-brucite as function of temperature and Figure 7(b) the quantifications of the H<sub>2</sub>O transitions  
 326 within the clay particles.

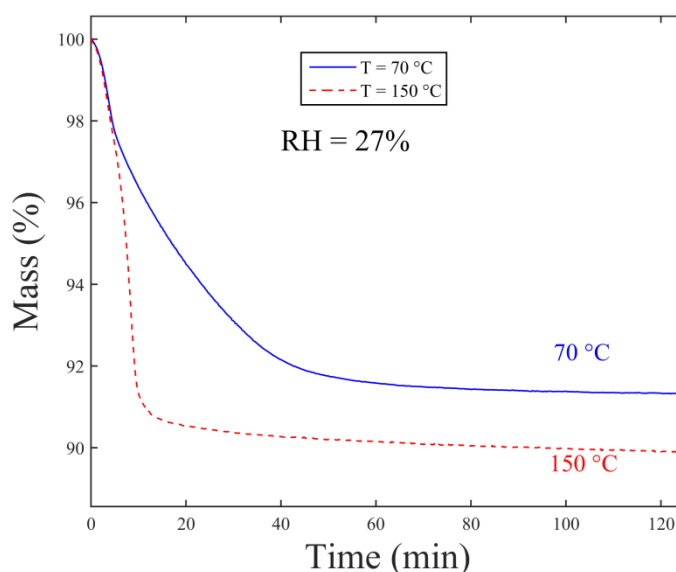


327  
 328 **Figure 7:** Mass (a) and DSC (b) of NiFh and Ni-hydroxide (Brucite) as a function of temperature. The drops in  
 329 the TGA correlates with the peaks observed in the DSC. The peak at 82 °C can be related to the H<sub>2</sub>O present in  
 330 the intraparticle meso-space, while the peaks at 150 °C and 270 °C to interlayer H<sub>2</sub>O. The Ni-brucite has a peak  
 331 at 100 °C and another at 310 °C that are related to H<sub>2</sub>O desorption and Brucite degradation, respectively.

332 It has been argued that Ni-brucite coexists with the clay mineral particles and that this is pH  
 333 dependent (Michels et al., 2014). Although the TGA curves of NiFh and Ni-brucite are different, their  
 334 DSC diagrams show some similarity. For NiFh the endothermic peaks at 82 °C and 150 °C are  
 335 associated to different H<sub>2</sub>O population that desorbs from the clay mineral, related to intraparticle and  
 336 interlayer H<sub>2</sub>O losses, respectively. The peak at 100 °C for Ni-brucite may be related to H<sub>2</sub>O  
 337 desorbing from the particle surface. These types of H<sub>2</sub>O adsorbed in the clay minerals can be linked to  
 338 previous interpretations in the literature (Bordallo et al., 2008; Gates et al., 2012). The NiFh sample  
 339 and the Ni-brucite sample show peaks at around 260 °C and 320 °C respectively that can be attributed  
 340 to the degradation of Ni-brucite, since the Ni-brucite particles are either at the surface or confined

341 within the clay mineral pores, thus likely reducing the temperature threshold for this decomposition. It  
342 is well known that when smectites submitted to high temperatures, can have their structure modified  
343 in at least two ways: reduction of the layer charge or dehydroxylation. Layer charge reduction can be  
344 due to the migration of the interlayer cation into the octahedral sites, which has been observed in  
345 smectites saturated with small cations at temperatures as low as 125 °C(Skoubris et al., 2013), and it  
346 is known as Hofmann-Klemen effect(Komadell et al., 2005). Another type of known possible structure  
347 modification is the release at temperatures above 400 °C of the structural hydroxyl groups (OH)  
348 present in smectite layers(Green et al., 1970). However, it is important to notice that for the smectite  
349 clays studied in the present work the OH groups have been replaced by fluorine. This would imply  
350 that the DSC peak at 260 °C could also mean that the fluorine groups of the layer structure are being  
351 released.

352 In order to compare the dehydration of different NiFh samples, two isotherms with  
353 temperatures fixed at 70 °C and 150 °C (temperature of dehydration of the pre-heated samples) were  
354 measured in N<sub>2</sub> atmosphere. Both samples were prepared at room temperature at RH= 27 %, which  
355 are the same conditions used in the XRD experiments on non-preheated samples. The results are  
356 presented in Figure 8. In the asymptotic limit (125 min), for the sample heated at 70 °C the amount of  
357 mass lost is 8.7% while for the one heated at 150 °C the amount is 10.1%.



358

359 **Figure 8:** The amount of mass loss based on the isotherm at 70 °C and 150 °C.

### 360 **3. Discussion**

361 As described above, continuous peak shifts were found in the transition regions upon water  
362 uptake for this system. This can be related, to the existence of a Hendricks-Teller (HT) (da Silva et al.,

2002; Ferrage et al., 2007; Hendricks and Teller, 1942) type peak between two stable XRD peaks. In such a case, the positions of the HT peaks can be directly related to the amount of different states in the system. Simultaneously or alternatively, this continuous shift can be explained by different geometrical arrangements of Ni<sup>2+</sup>-H<sub>2</sub>O complexes in the interlayers, such as the presence of Ni-brucite (Ni(OH)<sub>2</sub>). Indeed Michels et al. (Michels et al., 2014) reported that Ni-brucite was formed during the cation exchange process from LiFh to NiFh due to the natural high pH of clay minerals in solution, and it was shown that such structures may remain stable, which could possibly explain the differences between dehydrated NiFh and LiFh or NaFh. Similar possibilities were also discussed by Pitteloud et al. (Pitteloud et al., 2001), who found that Ni-montmorillonite at high hydration conditions forms a complex structure of Ni<sup>2+</sup>hexaaqua, which is very close to what is found for Ni<sup>2+</sup> in aqueous solution. Based on the results in Table 1, peak II can be labeled as 1WL. This is a stable hydration state which is in agreement with Figure 3. Peak I, like the transition peaks, may be related to one type of confined Ni<sup>2+</sup>-H<sub>2</sub>O complex in which the free rotation of the H<sub>2</sub>O molecule is limited. Furthermore, peak III has an interlayer space corresponding to approximately 1.5 WL, which was also reported by Tenorio et al. for LiFh (Tenorio et al., 2010). In sum all peaks I-IV may possibly be related to different stable situations (coordination numbers and orientations of H<sub>2</sub>O molecules) of Ni<sup>2+</sup>-H<sub>2</sub>O complexes in the interlayers

The dependence of the XRD isotherms on the sample history is not clear (i.e. why there are different XRD isotherm behaviors for the non-preheated and preheated samples). One explanation might be a partial, but nevertheless minor, change in the layer structure initiated while heating the samples at 150 °C. This has indeed been previously observed in smectites saturated with small-cations at temperatures above 125 °C (Komadel et al., 2005; Skoubris et al., 2013).

From the TGA isotherm data at 150 °C the total mass loss is about 10.1%, and assuming that this is only due to H<sub>2</sub>O loss, this corresponds to approximately  $33.7 \cdot 10^{20}$  H<sub>2</sub>O molecules lost per gram of NiFh. Considering that there are approximately 0.6 Ni<sup>2+</sup> cations per unit cell of Fh (Kaviratna et al., 1996) and that the number of unit cells per gram is the Avogadro's number divided by the molar mass of the NiFh unit cell, one gets  $7.71 \cdot 10^{20}$  unit cells per gram. Thus one gets approximately  $4.6 \cdot 10^{20}$  Ni<sup>2+</sup> cations per gram of NiFh, and if one considers that all the lost H<sub>2</sub>O is associated with Ni<sup>2+</sup> cations, the coordination number would be approximately 7.3 H<sub>2</sub>O molecules per Ni<sup>2+</sup> (corresponding to 4.4 H<sub>2</sub>O molecules per unit cell). Interestingly, this value is only slightly above the Brucite structure coordination number, which is Ni<sup>2+</sup> coordinated by 6 Oxygen atoms (Michels et al., 2014). On the other hand, for the isotherm measured at 70 °C, 8.7% of the mass is lost during the experiment, which corresponds to approximately 6.3 H<sub>2</sub>O molecules per Ni<sup>2+</sup>, using the same calculation as above. This difference therefore amounts to about 1.0 H<sub>2</sub>O molecules per Ni<sup>2+</sup> cation left in the sample, which is less than 1.4% of the total mass, assuming that all and only H<sub>2</sub>O is lost at 150 °C. Consequently, the TGA measurements show that diffraction data were not collected in fully dehydrated powders, indicating that the  $d_{001}$  value obtained using XRD does not directly represent the



400 amount of H<sub>2</sub>O in the interlayer. Here it is important to point out that the detection limit of our XRD  
401 measurements is about 1.5% (signal to noise ratio) which can be estimated from the ratio of the  
402 dehydrated 0WL Bragg peak to the noise level in the 1WL peak position. Accordingly, if 1.5% of the  
403 crystallites are in the monohydrated state they are “invisible” in the XRD experiments.

404

#### 405 **4. Conclusion:**

406 We have used XRD and TGA/DSC to study the intercalation of H<sub>2</sub>O in the synthetic smectite  
407 clay fluorohectorite with Ni<sup>2+</sup> as interlayer cation using different sample preparation approaches, i.e.  
408 pre-heated and non-pre-heated clays. XRD results show that the transition upon water adsorption from  
409 the dehydrated state does not occur as an abrupt step, as commonly observed in other studies of  
410 hydration in smectites, but shows a relatively continuous change. The latter might be linked to the  
411 presence of various types of Ni<sup>2+</sup>-H<sub>2</sub>O complexes in the interlayers, such as Ni-brucite complexes or  
412 Ni<sup>2+</sup> hexaaqua and/or also to the coexistence with different hydration states within the scattering  
413 volume. As reported in our recent work (Michels et al., 2014), Ni-brucite complexes can be present in  
414 NiFh. If this behavior is common to smectite clays with transition metal cations, it will have  
415 significant practical consequences in materials science and other areas where smectite clays are used.  
416 The TGA/DSC results highlights further the effect of the initial conditions of the NiFh in hydration  
417 experiments, suggesting that the sample pre-heated at 150 °C might have a reduced layer charge due  
418 to the migration of Ni<sup>2+</sup> to the layers (Hofmann-Klemen effect). This may explain the slightly  
419 different H<sub>2</sub>O adsorption isotherms obtained using XRD for the samples pre-heated at 150 °C and the  
420 non-preheated samples.

421 Finally, via the Williamson-Hall analysis for four selected values of RH, the average particle  
422 stack thickness and the average lattice strain were obtained. The former was found to be  $Nd = 146 \pm$   
423 11 nm for all studied samples in their stable hydration states, and to decrease when the system is in  
424 transition between two hydration states. This observation may be attributed to an inhomogeneous  
425 distribution of the intercalated H<sub>2</sub>O within the clay interlayers. The lattice strain was found to be  
426 proportional to the RH. This result indicates that the intercalation of H<sub>2</sub>O in the basal planes cause  
427 disorder in the stacking layers.

428 Future work would be to perform QENS experiments in order to fully quantify the H<sub>2</sub>O  
429 mobility (rotation and translation) within the interlayer space of the clay mineral and its relation to the  
430 number of H<sub>2</sub>O layers obtained in the present work. Through the analysis of the elastic incoherent  
431 structure factor (Jacobsen et al., 2013), this technique might shed light on the identification of the  
432 different Ni<sup>2+</sup>-H<sub>2</sub>O complexes, like Ni<sup>2+</sup> hexaaqua and Ni-brucite, and their relation with the hydration  
433 stability.

434

#### 435 **Acknowledgements**

436 The authors acknowledge the Brazilian Synchrotron Light Laboratory (LNLS) technical staff  
437 at the XRD2 beam line. We also wish to thank Antônio Gasperini, Fabio Zambello and Vinícius  
438 Fonseca for their helpful assistance at LNLS, also José Pereira for assistance in the TGA  
439 measurements. Mario Altoé acknowledges CAPES for PDSE Scholarship, process number BEX  
440 4792/14-9. L. Michels, E. C. dos Santos, K. D. Knudsen, and J. O. Fossum appreciate support from  
441 the Research Council of Norway. H. N. Bordallo acknowledges support from  
442 the Carlsbergfondets (grant number Ref: 2013\_01\_0589).

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