

1     Testing and modelling of stiffened aluminium panels  
2             subjected to quasi-static and low-velocity impact  
3                     loading

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9     **Abstract**

The behaviour and failure of stiffened panels made of the aluminium alloy AA6082-T6 is investigated under quasi-static and low-velocity impact loading conditions. The strain rate and inertia effects are found to be negligible suggesting that quasi-static tests might be representative for low-velocity impacts where a large mass is placed on the impactor. A simplified approach to the finite element modelling of aluminium panels under impact loading, including a regularised failure criterion, is proposed and validated against the experimental data. The effect of mesh size is investigated with shell elements of various sizes in the range from 1 to 5 times the thickness. A good correlation is obtained between experiments and simulations for fine meshes, while large shell elements have difficulties to initiate and propagate properly the observed cracks.

10    *Keywords:* Aluminium alloys, Impact loading, Design, Finite element analysis

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## 11 **1. Introduction**

12 Aluminium alloys are important in design of lightweight structures due to  
13 their good strength-to-weight ratios. This advantage combined with flexible and  
14 cost-efficient extrusion processes have enabled the application of aluminium al-  
15 loys in several business sectors, including the automotive industry [1] and the  
16 oil and gas industry. In the latter, multi-stiffened aluminium panels are used in a  
17 wide range of applications from walls and floors in offshore containers to hulls  
18 and decks in high speed ferries [2].

19 As stiffened aluminium panels are often basic building blocks of offshore  
20 structures, the research community has addressed the buckling resistance of these  
21 components over the past 15 years, e.g. Aalberg et al. [3] and more recently  
22 Paulo et al. [4]. At the same time, steel structures have been thoroughly inves-  
23 tigated in the literature, with studies including laboratory scaled experiments [5]  
24 to full-scale testing [6], analytical developments [7], and modelling and simula-  
25 tion with non-linear finite element techniques [8]. In the latter class of studies,  
26 the emphasis has often been on finite element modelling with shell elements of  
27 various sizes, as offshore structures are usually rather large and thus prevent the  
28 use of fine meshes [9]. A thorough literature review of this particular topic has  
29 been recently published by Calle and Alves [10], where the different approaches  
30 proposed in the literature for modelling of offshore steel structures subjected to  
31 impact scenarios are presented.

32 Compared to steel structures, modelling of aluminium structures may raise  
33 new challenges due to their anisotropic properties [11]. Moreover, structures

34 are usually built from several extruded parts that are welded together. Welding  
35 techniques for aluminium structures such as metal inert gas (MIG) welding and  
36 friction-stir welding (FSW) introduce heat-affected zones (HAZ) which exhibit  
37 lower strength than the base material to be joined [12, 13]. These particular  
38 features make the simulation of impact loading on aluminium structures using  
39 non-linear finite element methods challenging with regards to constitutive mod-  
40 elling.

41 Over the past decades, the numerical modelling of aluminium alloys has sig-  
42 nificantly improved with the development of advanced yield functions. An exam-  
43 ple is the yield function proposed by Barlat et al. [14] which is able to describe  
44 the complex anisotropic yielding and plastic flow of most of the aluminium al-  
45 loys in plane stress states. A drawback of these advanced models is the cost  
46 linked to the identification of parameters. Calibration of these yield functions  
47 requires at least several tensile tests in different directions with respect to the ex-  
48 trusion or rolling direction, as many parameters are involved in their mathemat-  
49 ical formulations. Even if great progress has been made in terms of calibration  
50 of these models using for instance crystal-plasticity methods [15, 16, 17], the  
51 industrial use of such approaches is still challenging and simplified methods are  
52 required.

53 Under impact loading, failure is most likely to occur and has to be accounted  
54 for in the design of an aluminium structure. Recent works [18, 19] have high-  
55 lighted that ductile failure in aluminium alloys is strongly dependent on the stress  
56 state. Moreover, failure in aluminium alloys can also be strongly anisotropic, as

57 illustrated for the AA 7075-T651 alloy by Fourmeau et al. [11]. As for the  
58 description of complex yielding and plastic flow, several models have been pro-  
59 posed to predict the observed stress state dependent failure of metals [18, 19, 20].  
60 While accurate predictions in terms of fracture initiation can be obtained with  
61 these models, their calibration requires several material tests under different  
62 stress states, thus limiting their applications in an industrial context. Moreover,  
63 the full capacity of such fracture models relies on an accurate description of  
64 the local plastic flow and strain localization using refined solid element meshes.  
65 Therefore, it is not clear that these models would provide significant improve-  
66 ments in the ductile failure prediction when applied in simulations with large  
67 shell elements.

68 This study evaluates the response of stiffened aluminium panels subjected to  
69 impact loading. The panels are subjected to quasi-static and low-velocity impact  
70 loading using a cylindrical impactor oriented either longitudinally (in parallel)  
71 or transversally to the stiffeners. Based on the obtained experimental data, a  
72 constitutive model and a failure criterion suitable for numerical simulation of  
73 large-scale offshore structures are identified and evaluated using finite element  
74 models with different mesh sizes.

## 75 **2. Material tests**

76 The stiffened aluminium panels are composed of extruded profiles of alloy  
77 AA6082 in temper T6. The nominal chemical composition of the alloy is given  
78 in Table 1. AA6082 is the most common structural aluminium alloy due its com-  
79 bination of high strength, corrosion resistance and availability as rolled plates

80 and extruded profiles of various form. Moreover, its mechanical properties are  
81 comparable in terms of yield strength to regular offshore steels. The aluminium  
82 panels are assembled by use of friction-stir welding and each panel consists of  
83 five extruded profiles, as shown in Figure 1. The extruded profile has two stiff-  
84 eners with a thickness of 3 mm, while the base plate has a thickness of 4 mm.  
85 A small increase in thickness is found at both ends of the profile, delimited by a  
86 lip (see Figure 1). The material properties of the base plate and the stiffeners are  
87 obtained from tensile testing using the specimen shown in Figure 2a. The plastic  
88 anisotropy of the extruded profile is investigated by performing tensile tests in  
89 three directions with respect to the extrusion direction. These tests are done for  
90 the base plate only. The macroscopic properties of the heat-affected zone (HAZ)  
91 are evaluated using the slightly bigger specimen depicted in Figure 2b. These  
92 tests will be referred as cross-weld tensile tests.

93 Digital Image Correlation (DIC), using a in-house software [21], and a grip  
94 extensometer are applied to measure strains. The gauge length of the extensome-  
95 ter is 35 mm in the tests of the base plate and stiffener material and 57.5 mm in  
96 the tests of the heat-affected zone around the welds, respectively. The force is  
97 measured by the load cell of the universal testing machines used to perform the  
98 tensile tests.

	Si	Fe	Cu	Mn	Mg	Cr	Zn	Ti	Others
Min (%)	0.70			0.40	0.60				0.05
Max (%)	1.30	0.50	0.10	1.00	1.20	0.25	0.20	0.10	0.15

Table 1: Nominal chemical composition of the AA6082 in temper T6.

99 The tensile tests were carried out at a speed of 1.35 mm/min for the base  
 100 plate and stiffener materials and 2.1 mm/min for the HAZ to ensure a quasi-  
 101 static strain rate. The engineering stress-strain curves are shown in Figure 3a for  
 102 the base plate material and exhibit relatively strong anisotropy of the yield stress.  
 103 The plastic strain ratios (or Lankford coefficients) are presented in Table 2 and  
 104 it is evident that also the plastic flow is anisotropic. From Figure 3b, it can be  
 105 seen that the stiffener material exhibits a somewhat lower yield stress, while the  
 106 overall shape of the engineering stress-strain curve is similar to that of the base  
 107 plate material. It is believed that a difference in cooling rate could be responsible  
 108 for the lower yield stress as this process parameter can have a large impact on  
 109 the mechanical properties of a 6xxx alloys.

110 The engineering stress-strain curve from the HAZ is shown in Figure 3b.  
 111 The yield stress is reduced and the work-hardening increased compared with the  
 112 base plate and stiffener materials. These results are in accordance with existing  
 113 experimental data for AA6082 in temper T6 [12]. It should be noted that material  
 114 behaviour within the HAZ is strongly heterogeneous and thus the engineering  
 115 stress-strain curve in Figure 3b represents the overall behaviour of the HAZ.  
 116 Figure 4 shows the strain field determined by DIC on a cross-weld tensile test  
 117 and demonstrates the strongly heterogeneous strain field even at low strain levels.

$R_0$	$R_{45}$	$R_{90}$	$R_0$ (stiffener)
0.41	1.53	0.98	0.47

Table 2: Plastic strain ratios of AA6082 in temper T6.

### 118 **3. Component tests**

#### 119 *3.1. Test set-up and procedures*

120 The component tests are carried out under both quasi-static and low-velocity  
121 impact loading. The impactor is shown in Figure 5a. It has a cylindrical shape  
122 with hemispherical end caps to avoid crack initiation at sharp edges. Two loading  
123 configurations are investigated. The impactor is placed either transversally to or  
124 longitudinally (in parallel) with the stiffeners, later referred to as transverse or  
125 longitudinal orientation of the impactor. The plates are fixed in-between two  
126 massive steel frames made of welded square hollow sections (thickness 20 mm,  
127 100 mm width) (Figure 5b). In the bottom frame, 50 mm wide cut-outs make  
128 possible the use of continuous stiffeners in the aluminium panel. To increase  
129 the support of the plates, 8 mm thick L profiles are positioned between the plate  
130 and the bottom frame, reducing the gap of 50 mm to 10 mm in the area of the  
131 cut-outs. Teflon sheets are placed at the interfaces between the panel and the top  
132 and bottom frames to reduce the effect of friction forces (Figure 5 c). A total  
133 of eight M16 bolts in property class 12.9 (i.e., two bolts per side) are used to  
134 keep the plate in position between the top and bottom frames during testing. The  
135 bolts are only loosely tightened. The specially designed rig is then fixed to a rigid  
136 foundation. A lubricant is applied on the impactor to reduce the effect of friction.  
137 For more details on the clamping system, the reader is referred to Gruben et al.  
138 [22].

139 The quasi-static tests are carried out using a hydraulic jack to apply the load  
140 at a rate of 10 mm/min (Figure 6a). The force level is recorded by a 1000 kN load

141 cell, while the relative displacement between the impactor and the bottom frame  
142 is measured by two laser extensometers. The two extensometers are targeting  
143 each side of an aluminium beam placed on top of the impactor (Figure 6a). The  
144 two laser extensometers placed on each side of the bottom frame are used to  
145 evaluate any misalignment of the test rig with respect to the impactor. The local  
146 deformations of the plate and stiffeners are monitored by a set of two cameras  
147 placed under the test rig.

148 The impact tests are carried out using a pendulum accelerator [23]. The  
149 impactor is installed on a trolley with a total mass of 1431 kg (Figure 6b). The  
150 tests are carried out at impact velocities in-between 2 and 3 m/s. The impact  
151 velocity of the trolley is measured using a set of photocells placed in front of  
152 the stiffened plate. The clamping system previously described is fixed to the  
153 reaction wall in a vertical position (Figure 6b). The velocity and displacement  
154 of the trolley during the impact are calculated based on the force-time curve  
155 measured in the load cell on the trolley. A set of high speed cameras is used to  
156 monitor the impact area at a frame rate of 15000 frames/s. Additionally, these  
157 cameras are used to measure the displacement of the impactor and control the  
158 measurements from the load-cell (Figure 6b). Due to limited space between the  
159 plate and the rigid wall, no cameras are monitoring the local deformation of the  
160 plates.

### 161 *3.2. Results from quasi-static tests*

162 The force-displacement curves from the quasi-static tests are shown in Fig-  
163 ure 7a and b for the transverse and longitudinal impactor orientations, respec-



164 tively. The two replicates in each configuration show little scatter in stiffness  
165 and maximum force. The replicates of the transverse and longitudinal impactor  
166 orientations are defined as  $QSTE_1$ ,  $QSTE_2$  and  $QSL_1$ ,  $QSL_2$ , respectively.

167 When loading with the transverse orientation of the impactor, three types of  
168 fracture are observed (Figure 8a). First a crack is initiated and developed in the  
169 stiffeners (designated "1" in Figure 8a). The effect of this crack is visible on the  
170 force-displacement curve in Figure 7a at a displacement of about 60 mm for the  
171 first specimen ( $QSTE_1$ ) and 65 mm for the second specimen ( $QSTE_2$ ). After  
172 the first crack has propagated through the stiffener, the plate still carries the load  
173 with a reduced stiffness until a second crack is initiated on either one or both  
174 sides of the impactor (designated "2" in Figure 8a). Specimen  $QSTE_1$  exhibits a  
175 non-symmetric crack propagation, i.e., the second crack, which is parallel to the  
176 stiffeners, is propagating on only one side of the impactor. The steep reduction  
177 in the force level for the  $QSTE_2$  specimen after 80 mm of indentation is due  
178 to a third crack initiating perpendicularly to the second crack and propagating  
179 under the impactor (designated "3" in Figure 8a). A ductile failure mechanism  
180 appeared to be dominant in the stiffener (1) and in the plate (2), while the crack  
181 propagating below the impactor (3) seems to be the results of a through-thickness  
182 shear failure mechanism.

183 For the longitudinal orientation of the impactor, fracture is only observed in  
184 the plate (Figure 8b). The sudden loss of load-carrying capacity of the plate as  
185 show in Figure 7b around 68 and 72 mm of indentation for the first and second  
186 specimen, respectively, is linked to a large crack propagating under the impactor.

187 While through-thickness shear fracture seems to be the dominant failure mech-  
188 anism, a closer inspection of the camera recordings of the plate shows a surface  
189 crack initiating perpendicularly to the impactor (Figure 7b). This crack initi-  
190 ates at about 43 mm of displacement and propagates quite slowly until the crack  
191 suddenly runs under the impactor.

192 In both configurations of the impactor, the weld lines are subjected to plastic  
193 deformations only and no signs of cracking are observed. It is believed that  
194 the small lips on each side of the weld might have acted as local stiffeners and  
195 prevented localization to occur in the HAZ.

### 196 *3.3. Results from low-velocity impact tests*

197 The force-displacement curves of the low-velocity impact tests are shown in  
198 Figure 9a and b for the transverse and longitudinal orientations of the impactor,  
199 respectively. The impact velocity was in turn 2.42 and 2.61 m/s in the two tests  
200  $DTE_1$  and  $DTE_2$  with transverse orientation of the impactor. In the two tests  
201  $DLE_1$  and  $DLE_2$  with longitudinal orientation of the impactor, the impact ve-  
202 locity was 3.48 and 3.06 m/s, respectively. The two replicates show consistent  
203 results in terms of maximum force and overall ductility, while there is some scat-  
204 ter in the stiffness.

205 Figure 10 shows the fracture modes occurring in the two loading cases. In the  
206 tests with the impactor in the transverse direction, tensile failures are observed in  
207 the stiffeners (indicated with a red line in Figure 10a). Small cracks are also ob-  
208 served on the sides of the impactor (also indicated with a red line in Figure 10a).  
209 Due to the low impact speed, the impact energy was too low to propagate these

210 cracks. In contrast, the impact energy is higher in the tests with the impactor in  
211 the longitudinal direction and full loss of load-carrying capacity is obtained by a  
212 crack propagating below the impactor, as illustrated in Figure 10b.

### 213 *3.4. Comparison of quasi-static and low-velocity impact tests*

214 A comparison of the force-displacement curves of the quasi-static and dy-  
215 namic tests is shown in Figure 11a and b for the transverse and longitudinal  
216 orientations of the impactor, respectively. No significant effects of strain rate  
217 and inertia are found. This result seems reasonable as the AA6082 alloy in tem-  
218 per T6 has been found to be almost rate insensitive [24] and the impacting mass  
219 is significantly larger than the mass of the panels. The fracture modes are also  
220 found to be similar (Figure 8 and 10). From this comparison, it seems that quasi-  
221 static tests might be good representatives for low-velocity, large-mass impactor  
222 tests on stiffened aluminium panels, provided the material exhibits low rate sen-  
223 sitivity.

## 224 **4. Material modelling**

### 225 *4.1. Constitutive model*

226 To enable industrial applications of the simulation modelling, an isotropic  
227 elastic-plastic model was adopted, even if the investigated alloy exhibits a rather  
228 strong plastic anisotropy, as shown in Figure 3a and Table 2. The experimen-  
229 tal and computational costs of employing an anisotropic plasticity model are  
230 large and not always possible within an industrial context. The same constitu-  
231 tive model is applied to the plate and stiffener materials as well as the HAZ. To

232 capture the typical shape of the yield surface of an aluminium alloy [25], the  
 233 Hershey-Hosford yield function [26, 27] is employed.

234 The yield function is expressed as

$$f = \varphi(\boldsymbol{\sigma}) - (\sigma_0 + R) \leq 0 \quad (1)$$

235 where  $\sigma_0$  is the initial yield stress and  $R$  the isotropic hardening variable. The  
 236 Hershey-Hosford equivalent stress is defined by:

$$\varphi(\boldsymbol{\sigma}) = \left[ \frac{1}{2} (|\sigma_1 - \sigma_2|^a + |\sigma_2 - \sigma_3|^a + |\sigma_3 - \sigma_1|^a) \right]^{\frac{1}{a}} \quad (2)$$

237 where  $\sigma_1, \sigma_2, \sigma_3$  are the eigenvalues of the Cauchy stress tensor  $\boldsymbol{\sigma}$  and  $a$  is a  
 238 material parameter controlling the shape of the yield surface. For FCC materials  
 239 such as aluminium alloys, it is customary to set  $a$  equal to 8 based on lower  
 240 scale analyses. Figure 12 shows the resulting Hershey-Hosford yield surface  
 241 under plane stress conditions with  $a$  equal to 8 compared with the von Mises and  
 242 Tresca yield surfaces.

243 The work hardening of the aluminium alloy is described by an extended Voce  
 244 rule in the form

$$R = \sum_{i=1}^{N_R} R_i = \sum_{i=1}^{N_R} Q_i \left( 1 - e^{-\frac{\theta_i}{Q_i} p} \right) \quad (3)$$

245 where  $p$  is the equivalent plastic strain,  $\theta_i$  and  $Q_i$  represent the initial work hard-  
 246 ening modulus and the saturation stress of hardening term  $R_i$ , and  $N_R$  is the num-  
 247 ber of terms. Two terms are used to represent the work-hardening of the plate  
 248 and stiffener materials, while only one term is used for the HAZ.

249 The associated flow rule is used in this work to describe the plastic flow. Ow-  
250 ing to the limited rate sensitivity of this alloy [24] and the negligible differences  
251 in response between quasi-static and dynamic components tests (Figure 11), a  
252 rate-independent formulation of plasticity is used.

#### 253 4.2. Failure model

254 The Cockcroft-Latham failure criterion [28] is used to model ductile failure  
255 of the aluminium alloy. The failure criterion is here formulated as a damage  
256 evolution rule

$$\dot{D} = \frac{\langle \sigma_1 \rangle}{W_C} \dot{p} \quad (4)$$

257 where  $D$  is the damage variable,  $\sigma_1$  is the maximum principal stress,  $\dot{p}$  is the  
258 equivalent plastic strain rate,  $W_C$  the Cockcroft-Latham parameter, and  $\langle x \rangle =$   
259  $\max(0, x)$  is the Maccauley bracket. Failure is assumed when the damage variable  
260  $D$  has reached a critical value  $D_C$ . Since the damage variable is not affecting the  
261 elastic-plastic behaviour of the material,  $D_C$  can be set to unity without loss of  
262 generality.

263 The Cockcroft-Latham failure criterion has the benefit of having only one  
264 parameter and therefore reduces the calibration cost. This damage evolution  
265 rule accounts for the main features of ductile failure under plane stress condi-  
266 tions such as a decrease of ductility from uniaxial tension to plane strain tension  
267 followed by an increase towards equi-biaxial tension. Failure will not be pre-  
268 dicted for uniaxial compression and lower stress triaxiality, while pure shear  
269 will produce a rather large ductility [20]. However, failure under low triaxial-  
270 ity is not considered to be important in the present study because thin-walled

271 structures (as the aluminium stiffened panels investigated here) will typically  
272 accommodate compression and shear loading by buckling leading to a locally  
273 tensile-dominated problem.

#### 274 *4.3. Computational considerations*

275 The constitutive model is implemented in ABAQUS [29] as a user-defined  
276 material model for 3D and plane stress states. In ABAQUS/Explicit, the cutting-  
277 plane algorithm [30] is adopted for temporal integration of the constitutive re-  
278 lations, while a semi-implicit algorithm [31] is used in ABAQUS/Implicit. To  
279 ensure an accurate stress update, a sub-stepping scheme is employed. The max-  
280 imum magnitude of the incremental deviatoric strain tensor is set to 10% and  
281 1% of the strain to yielding for explicit and implicit simulations, respectively. If  
282 the strain increment is larger, sub-stepping will reduce the strain increment to its  
283 maximum allowable value.

284 ABAQUS/Standard requires the consistent tangent operator in addition to the  
285 updated Cauchy stress tensor. The consistent tangent operator is obtained here by  
286 means of numerical derivation using a central difference scheme. By setting the  
287 exponent  $a$  of the Hershey-Hosford yield function equal to 2, comparison with  
288 the built-in  $J_2$  flow theory of ABAQUS/Implicit is possible. Similar results were  
289 obtained in simulation of a tensile test with solid elements both with regards  
290 to local response and equilibrium iterations, indicating a good accuracy of the  
291 computed consistent tangent operator.

292 Failure and crack propagation is handled by element elimination in the ex-  
293 plicit simulations. The stress tensor is set to zero in each integration point where

294 the damage variable  $D$  is equal to unity. Since this operation is carried out within  
295 one time step, elastic stress waves are released into the remaining mesh, thus  
296 creating some noise in the calculated forces in the simulations of the component  
297 tests. In ABAQUS/Explicit an element is removed when all integration points  
298 have reached failure. As will be shown below, this might be problematic when  
299 propagating a crack within the finite element mesh.

#### 300 *4.4. Parameters identification*

301 The proposed constitutive model requires the input of the initial yield stress  
302  $\sigma_0$  and the parameters  $(\theta_1, Q_1)$  and  $(\theta_2, Q_2)$  of the two hardening terms. The  
303 initial yield stress  $\sigma_0$  and the parameters  $(\theta_1, Q_1)$  of the first hardening term are  
304 identified directly from the uniaxial tensile tests in the extrusion direction (de-  
305 noted  $0^\circ$  in Figure 3a and b), using the true stress-strain curve computed based  
306 on the extensometer measurements. Here, the first hardening term is defined as  
307 the first to reach its saturation stress  $Q_1$ . The parameters  $(\theta_2, Q_2)$  of the second  
308 hardening term are initially identified using the experimental measurements, but  
309 are later refined using a numerical model.

310 Reverse engineering of the tensile test in the extrusion direction is performed  
311 with a solid element model of the specimen in ABAQUS/Standard. The param-  
312 eters  $\theta_2$  and  $Q_2$  are modified manually until a satisfactory agreement is found  
313 between the test and the numerical simulation in the post-necking regime. The  
314 numerical model of the tensile test is shown in Figure 13a. Due to the assump-  
315 tion of isotropy and the neck taking place perpendicularly to the loading axis of  
316 the specimen, only 1/8 of the geometry is modelled. Within the grip length of

317 the extensometer solid elements with characteristic size of 0.4 and 0.3 mm are  
318 used for the plate and stiffener materials, respectively, while a coarser mesh is  
319 used outside this area. Reduced integration with improved hourglass control is  
320 employed in these simulations. The specimen is loaded using a rigid analytical  
321 surface to represent the pinned connection. A frictionless interface is defined  
322 between the specimen and the pin using a surface-to-surface contact algorithm.

323 The results in terms of engineering stress-strain curves are shown in Figure  
324 13b. A rather good agreement is obtained until an engineering strains of 0.12.  
325 The tail of the engineering stress-strain curve is not captured properly by the  
326 finite element model, but the obtained set of parameters is considered to be suffi-  
327 ciently accurate. Since the material exhibits marked anisotropy with low plastic  
328 strain ratio in the extrusion direction, the simulations cannot be expected to de-  
329 scribe the experimental curves until failure occurs.

330 Prediction of ductile failure with a finite element model is a mesh-size de-  
331 pendent problem. A simple way to handle mesh-dependent parameters is to use  
332 a computational-cell approach, i.e., the element type (e.g. solid vs. shell) and  
333 mesh size is fixed during identification and application of the failure model [32].  
334 While some material tests can be modelled correctly with shell elements of var-  
335 ious sizes, this is not the case for the uniaxial tensile tests carried out in this  
336 study. A flat tensile specimen exhibits usually diffuse necking followed by lo-  
337 calised necking and subsequently failure. While the diffuse neck scales with the  
338 specimen width, the local neck scales with the thickness of the specimen, and  
339 thus a rather fine mesh is required to accurately capture local necking and fail-



340 ure. It follows that a shell element model with mesh size greater or equal to the  
341 the thickness of the specimen will generally not provide a reliable estimate of the  
342 failure parameters. Shell elements with a characteristic element length  $l_e$  below  
343 the specimen thickness could be used to simulate the uniaxial tensile tests, but  
344 this would require use of non-local regularization to prevent excessive thinning  
345 [12]. This approach of combining very small shell elements and non-local regu-  
346 larization is not always suitable for the simulation of large structures due to the  
347 increased computational time.

348 One convenient method to obtain an element-size dependent fracture param-  
349 eter for use in large-scale shell simulations, is to use the field measurements  
350 obtained by digital image correlation (DIC) on the specimen surface in the neck-  
351 ing region. As illustrated in Figure 14a, the elongation  $\Delta l$  of a vector of length  
352  $l_e$  in the initial configuration is extracted from the DIC measurements and used  
353 to define the boundary conditions for a single shell element with edge length  $l_e$ .  
354 This approach is very similar to the one proposed by Hogström et al. [33]. The  
355 shell element is then loaded under uniaxial tension until the elongation at failure  
356 in the experimental test is reached. The corresponding Cockcroft-Latham pa-  
357 rameter  $W_C$  is then found by integrating the damage evolution rule (eq. 4) with a  
358 temporary  $W_C$  equal to 1 and a critical damage  $D_C$  equal to a very large number.  
359 By repeating this operation for different element length  $l_e$ , it is possible to evalu-  
360 ate the mesh dependency of the failure parameter  $W_C$ , as shown in Figure 14b. To  
361 enable a direct comparison between the plate and stiffener materials which have  
362 different thicknesses it is chosen to represent the size of the shell elements by

363 the ratio  $l_e/t_e$ , where  $l_e$  is characteristic element length and  $t_e$  is the initial thick-  
364 ness of the shell element. This definition will also allow for extrapolation of the  
365 obtained data to larger thicknesses. It should be mentioned that boundary con-  
366 ditions extracted from a vector perpendicular to the loading axis of the uniaxial  
367 tensile test specimen could have been used if the plastic anisotropy of AA6082-  
368 T6 was incorporated into the constitutive model. By restricting the boundary  
369 conditions to uniaxial tension, a conservative estimate of the failure parameter  
370 should be obtained. This is motivated by the fact that the local stress state of a  
371 tensile test is usually drifting from uniaxial tension before necking towards plane  
372 strain at failure. Since the failure strain is decreasing between these two stress  
373 states, the failure locus is then underestimated by the proposed methodology.

#### 374 *4.5. Heat Affected Zone modelling*

375 The component tests did not show any sign of failure in the HAZ of the  
376 AA6082-T6 plates, and therefore a simplified modelling approach is applied in  
377 the simulations of the component tests. The width of the HAZ is set to 20 mm  
378 in both the simulations of the cross-weld tensile tests (Figure 15a) and the com-  
379 ponent tests and it is assigned a single set of material parameters, i.e., the spatial  
380 variation within the HAZ is homogenized. This approach accounts for a weaker  
381 zone in the component tests, while being simple enough to be useful in an en-  
382 gineering context. As shown in Figure 4, the heterogeneous strain field in the  
383 HAZ makes impossible a direct identification of the parameters of the constitu-  
384 tive model.

385 A reverse engineering approach is employed to determine the work-hardening

386 parameters  $\theta_1$  and  $Q_1$  by iteratively comparing the results from a shell element  
387 model of the cross-weld tensile test to the experimental results. The elastic con-  
388 stants and the exponent  $a$  of the Hershey-Hosford yield function are given the  
389 same values as for the plate and stiffener material. The yield stress  $\sigma_0$  of the HAZ  
390 is fixed to 150 MPa according to the cross-weld tensile tests. ABAQUS/Explicit  
391 with time scaling was employed to find the remaining parameters. This type of  
392 numerical simulations is mesh sensitive even before reaching maximum force  
393 due to the heterogeneous distribution of the material properties along the gauge  
394 length of the specimen. Accordingly, the material parameters are adjusted for  
395 different  $l_e/t_e$  ratios. Only small variations ( $\pm 5\%$  of the values of the hardening  
396 parameters  $\theta_1$  and  $Q_1$ ) are necessary to get a similar description of the engineer-  
397 ing stress-strain curve of the cross-weld tensile tests (Figure 15b) for a large  
398 range of  $l_e/t_e$  ratios.

399 As already stated, the proposed approach is valid as long as failure does not  
400 occur in the HAZ. If failure occurs within the HAZ, a better description of the  
401 spatial variation in the material properties across the HAZ should be employed  
402 to enable prediction of strain localization [12, 34].

## 403 **5. Numerical analyses**

### 404 *5.1. Finite element model*

405 The numerical model used to evaluate the proposed modelling approach is  
406 presented in Figure 16. Due to the symmetries in the geometry and boundary  
407 conditions, only 1/4 of the structure is modelled. The full test rig is represented

408 as a deformable body (apart from the impactor) to capture properly the stiffness  
409 of the aluminium panel and to ensure a correct representation of the boundary  
410 conditions. The steel frame is discretized with shell elements of 10 mm charac-  
411 teristic length and modelled as an elastic-perfectly plastic material with Young's  
412 modulus, Poisson's ratio and yield stress equal to 210000 MPa, 0.3 and 355  
413 MPa, respectively. The bolts used to fasten the top frame to the bottom frame are  
414 discretized with beam elements of 16 mm diameter and modelled as an elastic-  
415 perfectly plastic material with Young's modulus, Poisson's ratio and yield stress  
416 equal to 210000 MPa, 0.3 and 1080 MPa, respectively. The connection between  
417 the different steel cross-sections and between the bolts and the top and bottom  
418 frames is ensured through a tie-constraint formulation.

419 The stiffened aluminium plate and the homogenized HAZ are modelled using  
420 reduced integration shell elements of various length to thickness ratio ( $l_e/t_e =$   
421 1, 2, 3, 4, 5). Since the stiffeners and the plate are sharing some nodes, the  $l_e/t_e$   
422 ratio of the stiffeners is slightly larger than that of the plate. Uniform meshing  
423 is applied for the stiffened plates, leading to around 40000 elements for an  $l_e/t_e$   
424 ratio of 1 and around 7000 elements for an  $l_e/t_e$  ratio of 5.

425 The impactor is modelled using a rigid body definition with a mesh size of 4  
426 mm. The test rig and the impactor have the same mesh size independently of the  
427  $l_e/t_e$  ratio of the stiffened plate to allow a one-to-one comparison of the numerical  
428 results. The Teflon sheets and lubricant are not included in the numerical model  
429 and replaced by a surface-to-surface contact definition between the aluminium  
430 stiffened plate and the steel parts (both the impactor and steel frames) with a

431 small friction coefficient of 0.05.

432 Since the quasi-static and dynamic components exhibit similar force levels  
433 and failure mechanisms, only the quasi-static tests are simulated numerically.  
434 ABAQUS/Explicit is used for this purpose with time scaling to reduce the com-  
435 putation time. The velocity of the impactor is gradually increased using a smooth  
436 function over 10% of the simulation time and then kept constant until the end of  
437 the simulation. The simulation time is chosen to obtain negligible inertia effects.

438 The material properties identified in Section 4.4 are summarised in Table  
439 3. Only the parameters for the finest shell mesh are reported here. To handle  
440 the mesh dependence, several sets of material parameters are used in accordance  
441 with the targeted mesh size. For the plate and stiffener materials, only the fracture  
442 parameter  $W_C$  varies with the mesh size, while the hardening parameters  $\theta_1$  and  
443  $Q_1$  depend weakly on the mesh size for the HAZ material.

	$E$ (MPa)	$\nu$ (-)	$\sigma_0$ (MPa)	$a$ (-)	$\theta_1$ (MPa)	$Q_1$ (MPa)	$\theta_2$ (MPa)	$Q_2$ (MPa)	$W_C$ (MPa)
Plate	66000	0.3	271.5	8	36591.2	17.8	1300.0	88.0	64.3
Stiffener	66000	0.3	238.7	8	63294.5	36.6	1440.0	90.0	64.2
HAZ	66000	0.3	150.0	8	3450.0	140.0	-	-	-

Table 3: Material parameters for  $l_e/t_e = 1$ .

## 444 5.2. Results

445 The force-displacement curves obtained for the transverse impactor orienta-  
446 tion are shown in Figure 17a. By including the test rig as a deformable body in  
447 the numerical model, the overall response is rather well reproduced in the simula-  
448 tions. A general observation is that increasing the element size produces a slight

449 increase of the overall stiffness. The effect of a crack developing in the stiffener  
450 (between 60 and 65 mm of displacement) is rather well captured for  $l_e/t_e = 1$   
451 (Figure 17a). The simulations with larger aspect ratios  $l_e/t_e$  exhibit only limited  
452 mesh dependence, indicating that the proposed mesh scaling method is working  
453 correctly. This is confirmed by the comparison between the numerical and ex-  
454 perimental displacements at which the first crack in the flange of the stiffener  
455 appears (Figure 17b). In the simulation, failure is assumed to initiate in a shell  
456 element when the integration point at mid-section reaches  $D = 1$ . It should be  
457 noted that the flange of the stiffener experiences a stress state close to uniaxial  
458 tension but with a small stress gradient through the thickness. Regarding the  
459 crack propagating into the web of the stiffener, all simulations predict that one or  
460 two elements remain intact near the plate (Figure 17d), while in the experiments  
461 complete failure of the stiffener web is observed (Figure 8b).

462 The simulations predict the force increase after the failure has propagated  
463 into the stiffener web, while the force reduction in the last part of the tests (around  
464 80 mm) is not captured, independent of the mesh size. This force reduction is  
465 caused by the crack propagating below the impactor in the tests (denoted "3" in  
466 Figure 8a), which is not predicted in the simulations. This is most likely caused  
467 by a combination of several effects. Firstly, the failure in the plate (next to the  
468 impactor, denoted "2" in Figure 8a) is not predicted properly in the simulations,  
469 as shown in Figure 17c. In this region of the plate, the shell element is subjected  
470 to biaxial tension on the back face and biaxial compression on the front face  
471 (i.e., the impactor side). According to the Cockcroft-Latham criterion, damage

472 will not evolve until the maximum principal stress becomes positive. Secondly,  
473 as stated in section 4.3, ABAQUS/Explicit will delete an element only when all  
474 integration points through the thickness have reached failure. In view of Figure  
475 17c, these two factors affect the prediction of failure initiation and propagation  
476 in the plate. Indeed, failure in the integration points on the tensile side of the  
477 plate is usually predicted before the actual crack initiates in the experiments,  
478 but the integration points next to the impactor do not reach failure for the given  
479 displacement. This prevents the elimination of the element and therefore the  
480 propagation of a crack below the impactor. Finally, the mesh discretisation is  
481 probably a key issue in the prediction of the crack on the edge of and below  
482 the impactor. The impactor used in this study has a radius of 25 mm, while the  
483 element length is ranging from 4 to 20 mm for  $l_e/t_e$  equal to 1 and 5, respectively.  
484 The mesh has then to describe the impactor with 6 to 1 element and is therefore  
485 not accurate enough.

486 Also for the transverse impactor orientation, the simulated global response is  
487 in good agreement with the experimental data (Figure 18a), but the force level is  
488 slightly overestimated for large displacements. The finest mesh used in this study  
489 ( $l_e/t_e = 1$ ) predicts the complete loss of load-carrying capacity of the structure  
490 rather accurately, while the other meshes predict structural failure either too late  
491 or not at all. The high-frequency oscillations observed in the force-displacement  
492 curves are caused by elastic stress wave generated when through-thickness in-  
493 tegration points fail and the stress tensor is abruptly set to zero. Figure 18b  
494 shows the displacement required to initiate a crack in the plate under the im-

495 pactor. While the location of the predicted cracks correlates rather well with the  
496 one observed in the tests (Figure 8b), crack initiation is delayed for  $l_e/t_e$  equal to  
497 2 and 3. In the simulations with  $l_e/t_e$  equal to 4 and 5, material failure is not pre-  
498 dicted at all. As discussed above, requiring that all through-thickness integration  
499 points must reach failure before deleting an element, prevents crack initiation and  
500 thus the complete loss of capacity of the structure. Notwithstanding, the simu-  
501 lated crack pattern (Figure 18c) for the finest mesh(es) is rather close to the one  
502 observed in the tests (Figure 8b). In particular, the FE models with  $1 \leq l_e/t_e \leq 3$   
503 are able to predict the crack developing perpendicularly to the impactor.

### 504 *5.3. Discussions*

505 From the results presented in section 5.2 it appears that, despite the regular-  
506 ized failure parameters, failure initiation can only be predicted correctly when a  
507 fine shell mesh with element size being equal to the thickness is used. In the in-  
508 vestigated aluminium panels, even if failure initiation is captured properly for the  
509 finest meshes, the propagation of the cracks is still difficult to predict accurately.

510 According to the numerical results presented in Figure 17 d and 18 c, the  
511 different mesh sizes applied for the aluminium panels are still able to predict  
512 the crack locations accurately. Taking into considerations the numerical costs  
513 summarized in Table 4 and 5 for the transverse and longitudinal impactor orien-  
514 tations additional conclusions can be drawn. The numerical cost behind the use  
515 of a fine shell element mesh ( $l_e/t_e = 1$ ) is quite important in term of normalised  
516 CPU time. Here the CPU time is normalised by the one required to solve the  
517 finest mesh. This increase in computational cost is linked to the large number



518 of elements combined with a smaller initial stable time step. It is also shown in  
519 Table 4 and 5 that the differences in the computational cost for the  $l_e/t_e = 3, 4$   
520 and 5 meshes are small due to a similar number of elements and that the initial  
521 stable time step is the same. The time step in these numerical models is actually  
522 dominated by an element located in the test rig. In design of aluminium stiffened  
523 structures against impact loading, a mesh size of 3 to 4 times the thickness could  
524 be used in preliminary simulations. In the area where the damage parameter is  
525 rather large (in the present analyses 0.5), the mesh should then be refined towards  
526 an element aspect ratio of one.

$l_e/t_e$	Number of elements	Normalised CPU time	Initial time step (s)
1	41918	1	4.437e-07
2	14345	0.134	9.415e-07
3	9105	0.046	1.209e-06
4	7403	0.031	1.209e-06
5	6683	0.025	1.209e-06

Table 4: Summary of numerical models data for the transversal impactor orientation.

$l_e/t_e$	Number of elements	Normalised CPU time	Initial time step (s)
1	41918	1	4.437e-07
2	14424	0.134	9.415e-07
3	10414	0.071	1.209e-06
4	7442	0.030	1.209e-06
5	6683	0.029	1.209e-06

Table 5: Summary of numerical models data for the longitudinal impactor orientation.

527 Since the present finite element models are made with a uniform mesh size it  
528 is possible to apply different failure parameters to account for the mesh sensitiv-  
529 ity of the ductile failure model. In a real engineering structure, the mesh size is

530 most likely varying along the parts of the structure and thus an automatic mesh  
 531 size regularisation rule is needed. Several mesh regularisation rules have been  
 532 proposed in the literature and are usually formulated as follows

$$A_f = A_h + (A_l - A_h) \cdot h \left( \frac{l_e}{t_e} \right) \quad (5)$$

533 where  $A_f$  is the failure parameter for a particular mesh size,  $A_h$  is the failure  
 534 parameter for a very large element size and  $A_l$  is the failure parameter for small  
 535 elements. The function  $h(l_e/t_e)$  depends on the element size; it decreases with  
 536 increasing the element length and is equal to one when  $l_e/t_e$  is equal to one.  
 537 Several failure models designed for structural steels have been proposed in the  
 538 literature [10] and are usually similar in nature to Barba's law [35]. Figure 19  
 539 shows the results of two approaches to handle mesh regularisation of failure  
 540 parameters using the data collected on the plate material (section 4.4). The first  
 541 is the RTCL regularisation scheme proposed by Törnqvist [36]:

$$\varepsilon_f = \varepsilon_n + (\varepsilon_l - \varepsilon_n) \cdot \frac{t_e}{l_e} \quad (6)$$

542 where  $\varepsilon_f$  is the failure strain for a given mesh size,  $\varepsilon_n$  is the strain at diffuse  
 543 necking and  $\varepsilon_l$  is the local strain at failure identified using an element aspect  
 544 ratio of one. By assuming that the failure strain is the equivalent plastic strain, it  
 545 is possible to determine the fracture parameter  $W_C$  using the isotropic hardening  
 546 rule. In the second approach, denoted exponential decay, the fracture parameter

547  $W_C$  is defined by

$$W_C = W_C^h + (W_C^l - W_C^h).e^{-c.(\frac{t_e}{t_e} - 1)} \quad (7)$$

548 where  $W_C$  is the Cockcroft-Latham parameter for a given mesh size,  $W_C^h$  is the  
549 fracture parameter for large shell elements,  $W_C^l$  is the fracture parameter for an  
550 element with aspect ratio equal to one, and  $c$  is a model parameter. This model  
551 requires an optimisation of the parameters  $W_C^h$  and  $c$  in contrast to the regulari-  
552 sation proposed Törnqvist [36]. This optimisation is readily performed provided  
553 DIC measurements are available from the tensile tests.

554 The resulting evolution of the fracture parameter  $W_C$  as function of the mesh  
555 size is shown in Figure 19. The exponential decay function is able to reproduce  
556 rather accurately the evolution of the fracture parameter with element size and  
557 predicts a saturation around 40 MPa. The RTCL regularisation function is giv-  
558 ing a conservative estimate of the failure parameter with a saturation at around  
559 30 MPa. The parameters obtained for the plate and stiffener materials using the  
exponential decay function are summarised in Table 6.

Material	$W_C^l$ (MPa)	$W_C^h$ (MPa)	$c$ (-)
Plate	64.4	39.9	0.62
Stiffener	64.3	30.9	0.33

Table 6: Parameters of the exponential decay function for the plate and stiffener materials.

560

## 561 **Conclusions**

562 The behaviour and failure of stiffened panels of aluminium alloy AA6082-T6  
563 subjected to quasi-static and low-velocity impact loading was investigated exper-

564 imentally and numerically. The experimental study showed that the quasi-static  
565 and dynamic tests gave similar results in terms of global behaviour and failure  
566 modes. This finding was attributed to the low rate sensitivity of the AA6082-T6  
567 alloy and the large mass of the impactor compared with the mass of the stiffened  
568 panel. Failure in the stiffened panel was initiated by a ductile fracture process,  
569 while propagation seemed to be dominated by slant (shear) fracture. The pro-  
570 posed material model gave satisfactory results for fine shell element meshes with  
571 characteristic size of the order of the plate thickness. The mesh dependence of  
572 the failure predictions was reduced by the suggested identification approach for  
573 the fracture parameter using digital image correlation, as long as the deformation  
574 mode of the stiffened panel was aptly described.

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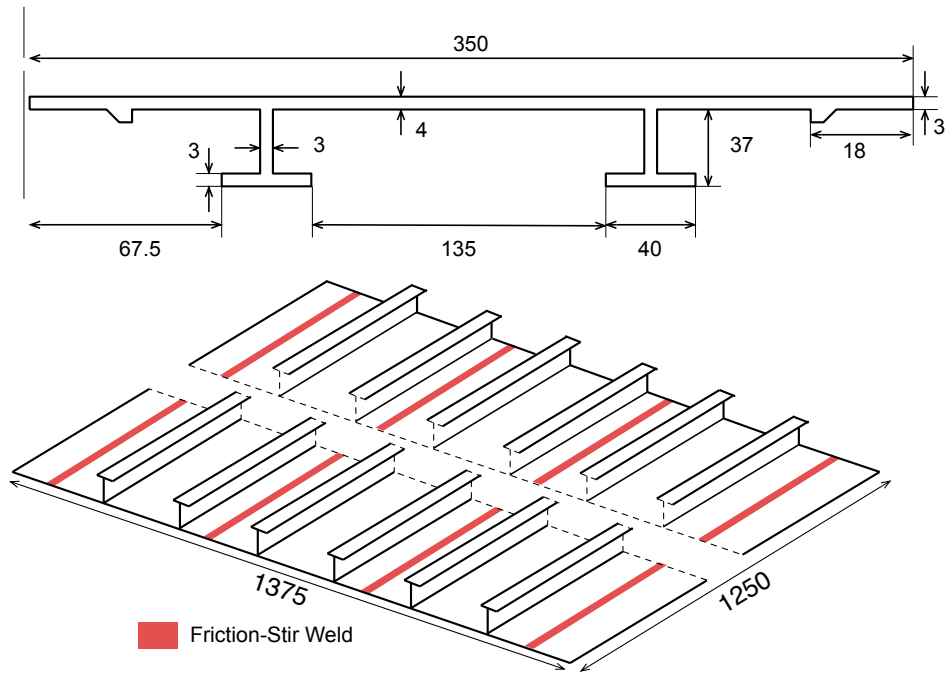
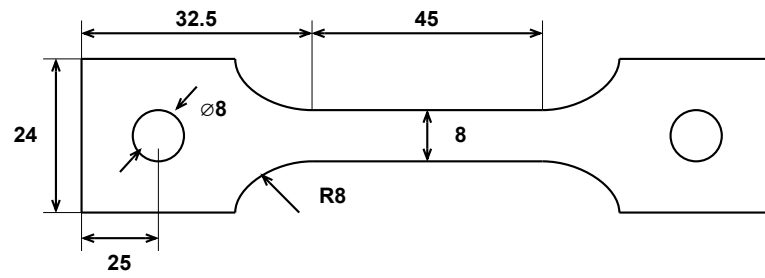
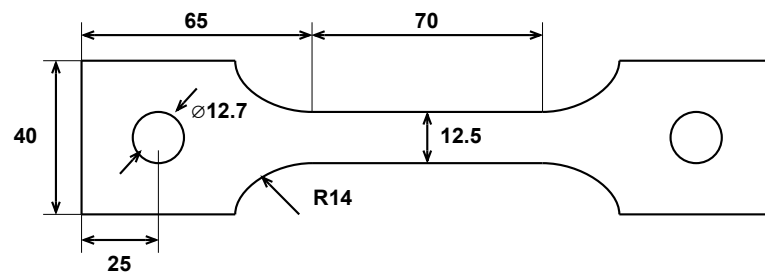


Figure 1: Illustration of the extruded profile and the assembled panels.

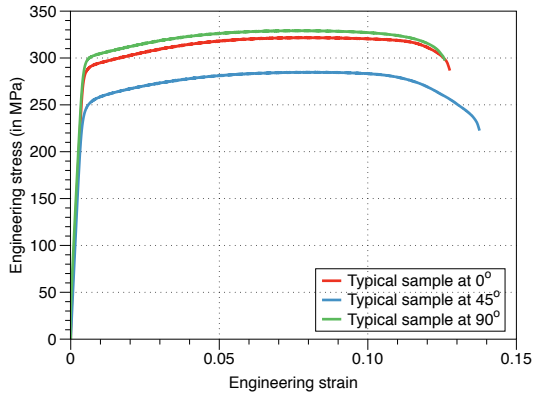


a) Plate and stiffener material

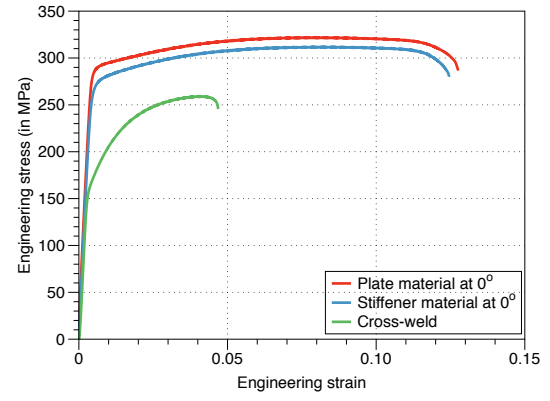


b) Specimen crossing the HAZ

Figure 2: Geometry of the tensile specimens.



a) Plastic anisotropy of plate material



b) Plate, stiffener and HAZ materials

Figure 3: Stress-strain curves of AA6082 in temper T6.

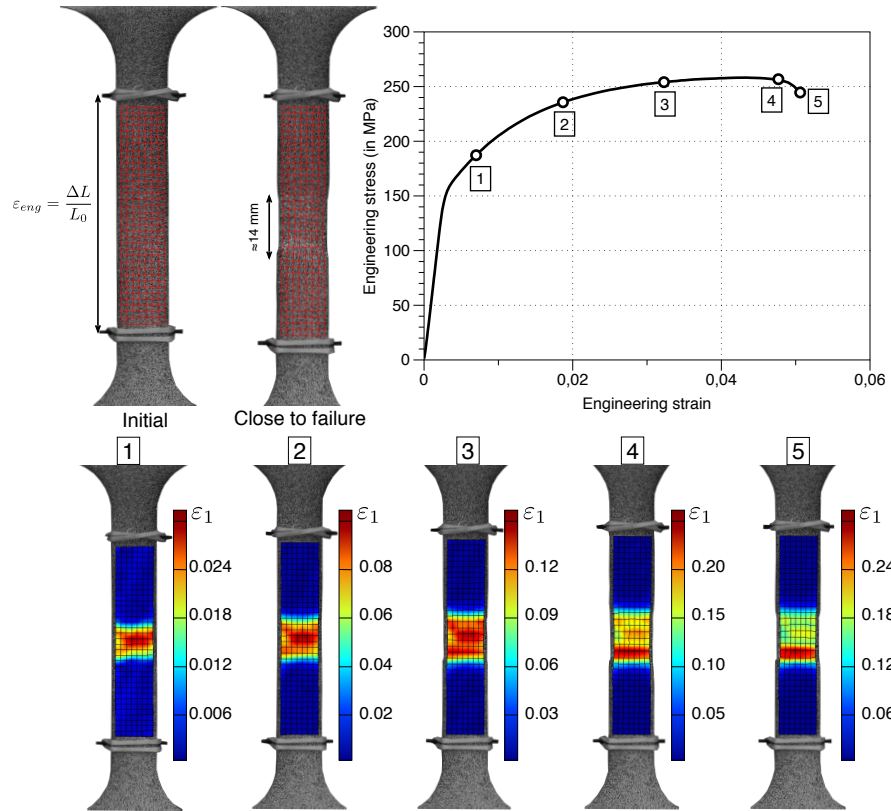
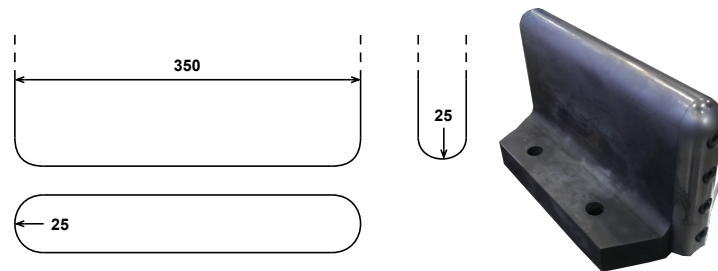
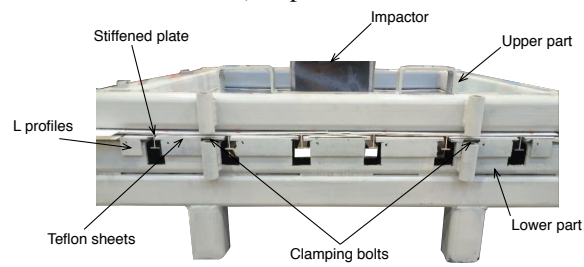


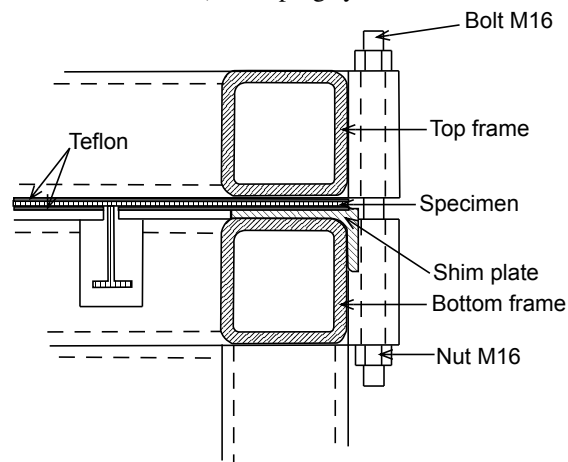
Figure 4: Heterogeneous strain field in the cross-weld tensile tests.



a) Impactor

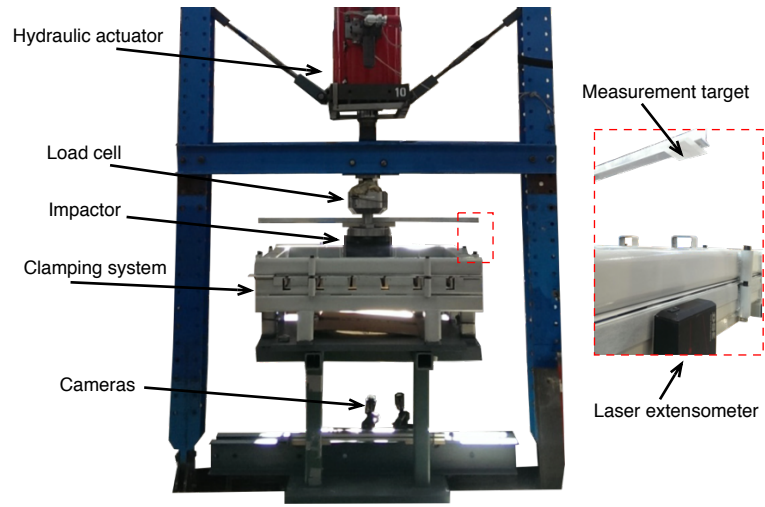


b) Clamping system

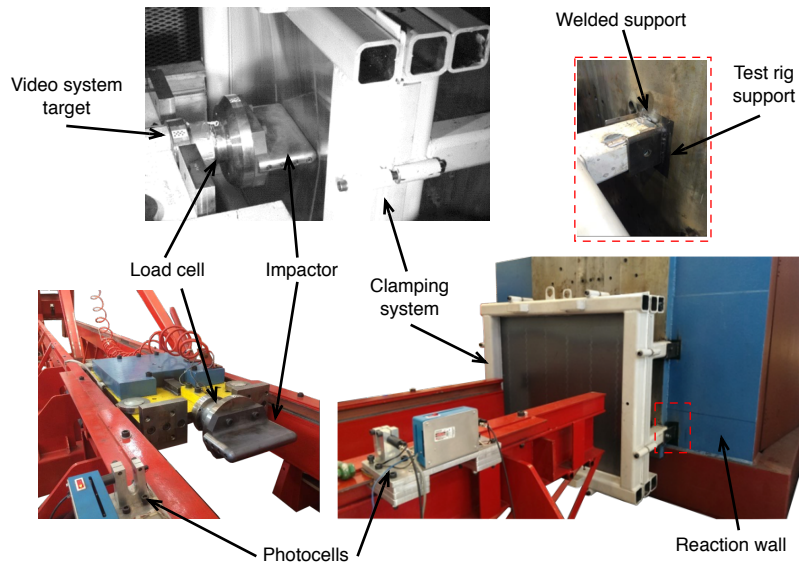


c) Close-up view

Figure 5: Shape of the impactor and clamping system for stiffened panel tests.

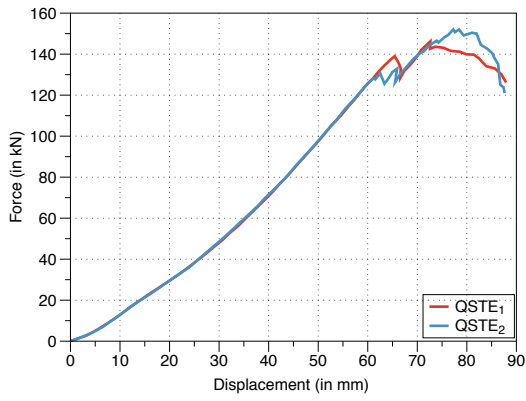


a) Quasi-static tests

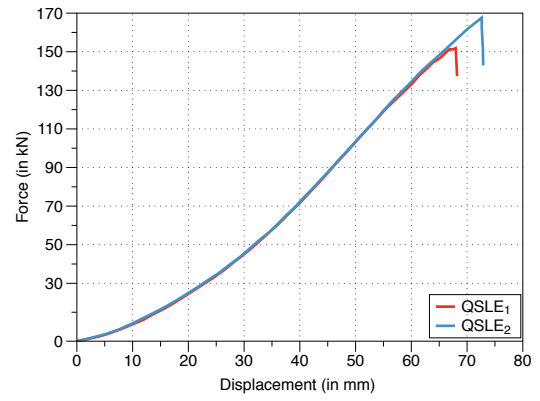


b) Low-velocity impact tests

Figure 6: Setups for component tests.



a) Transverse impactor orientation



b) Longitudinal impactor orientation

Figure 7: Force-displacement curves from the quasi-static tests.



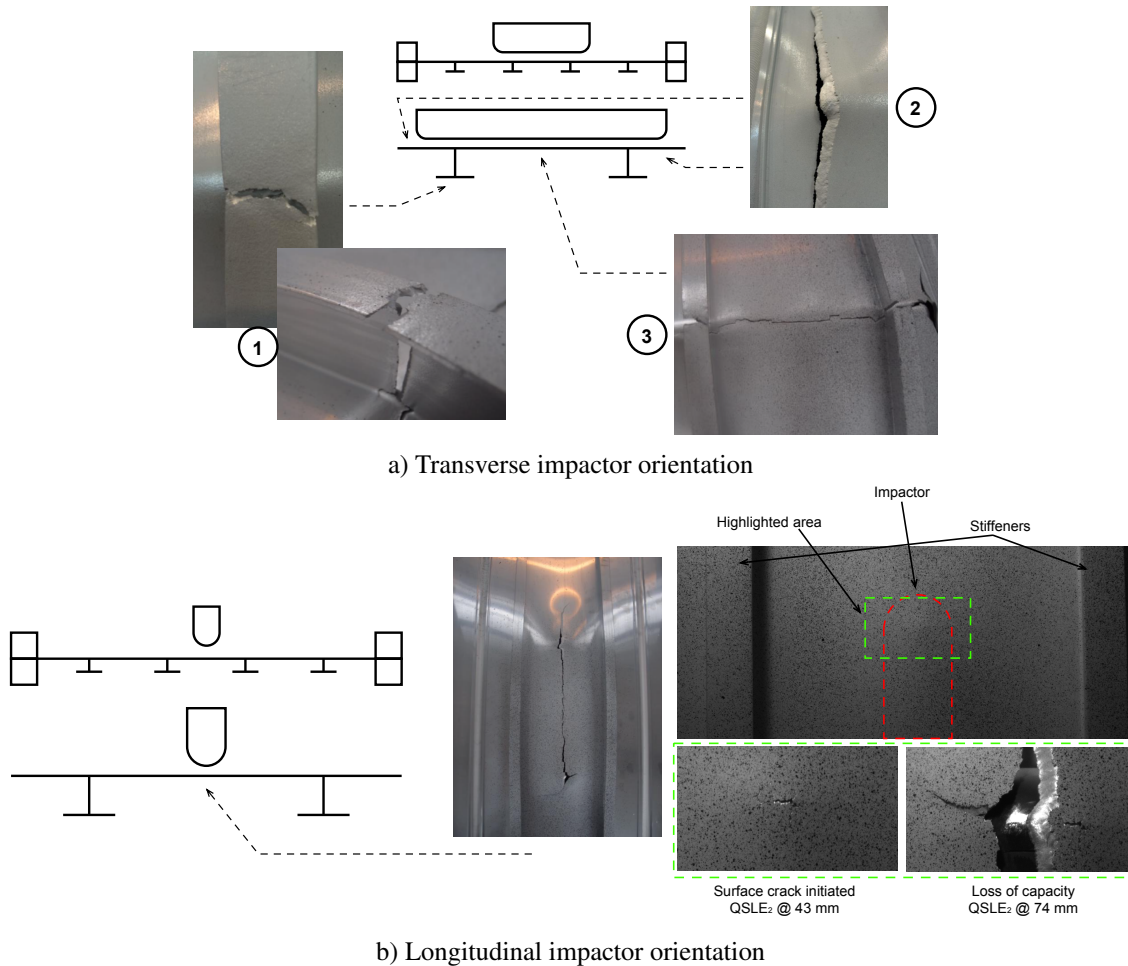
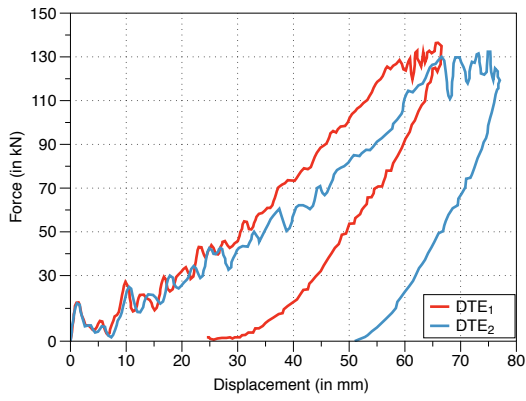
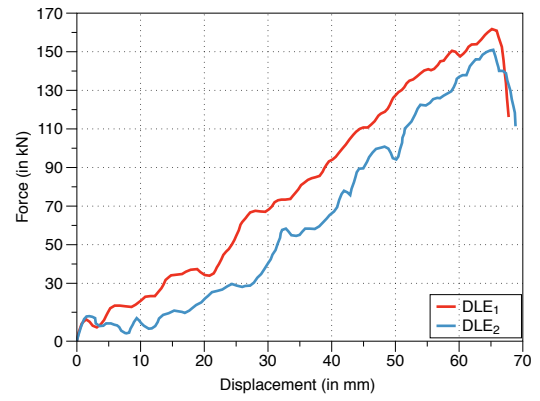


Figure 8: Observed fracture in the quasi-static tests.



a) Transverse impactor orientation



b) Longitudinal impactor orientation

Figure 9: Force-displacement curves from the low-velocity impact tests.

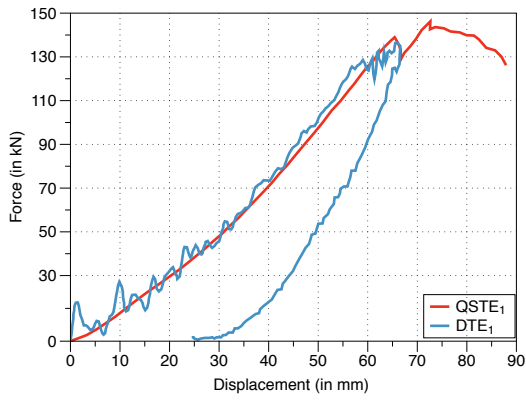


a) Transverse impactor orientation

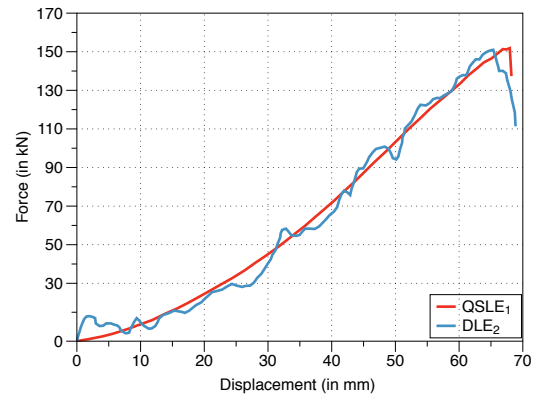


b) Longitudinal impactor orientation

Figure 10: Observed fracture in the low-velocity impact tests.



a) Transverse impactor orientation



b) Longitudinal impactor orientation

Figure 11: Comparison between quasi-static and low-velocity impact tests.

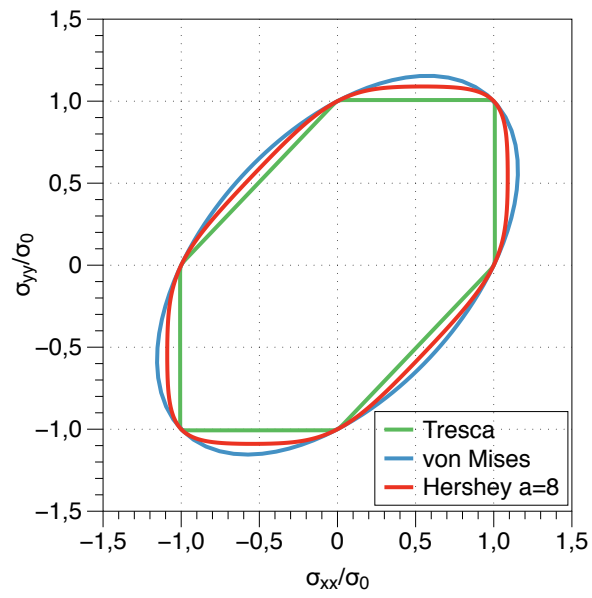
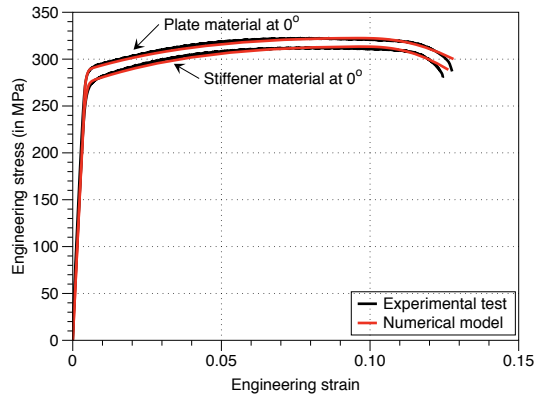
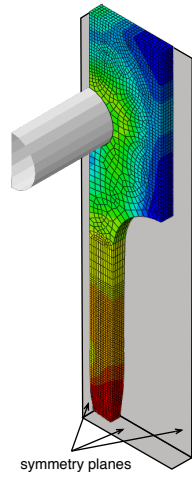


Figure 12: Hershey-Hosford yield surface under plane stress.



a) Finite element model of tensile test    b) Results from the reverse engineering procedure

Figure 13: Identification of the work-hardening parameters.

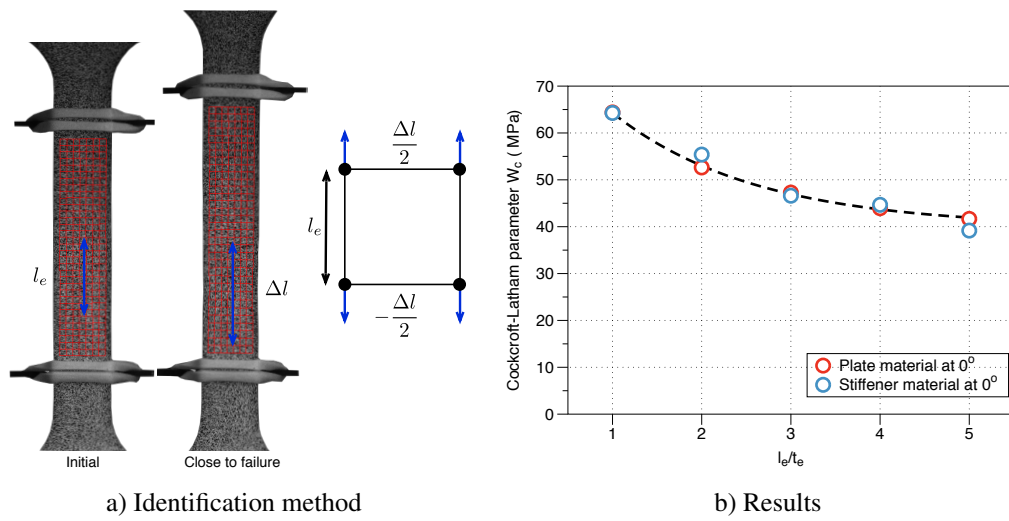


Figure 14: Identification of the failure parameter.

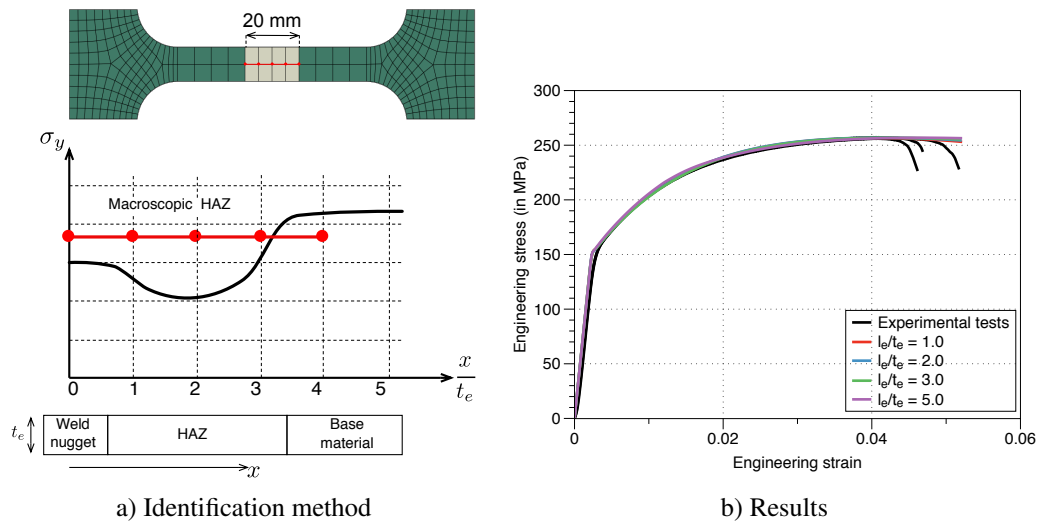


Figure 15: Identification of the macroscopic HAZ parameters.



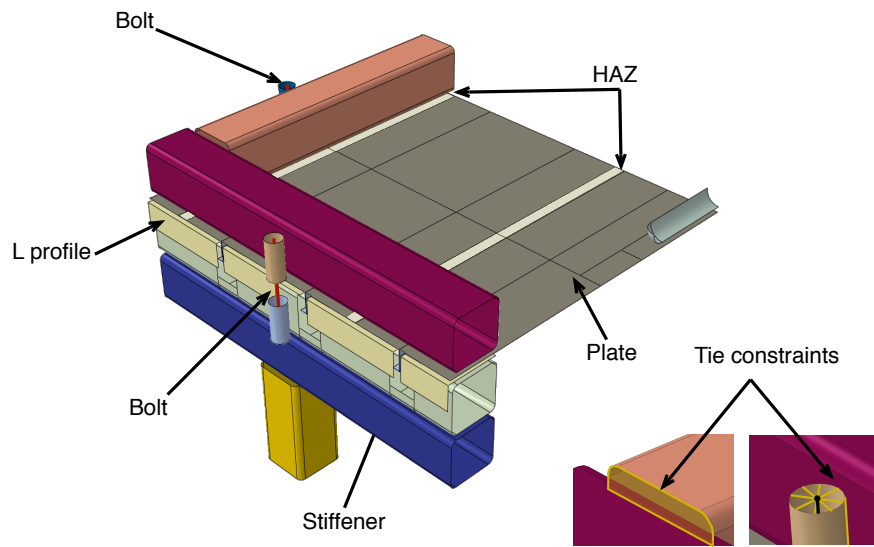
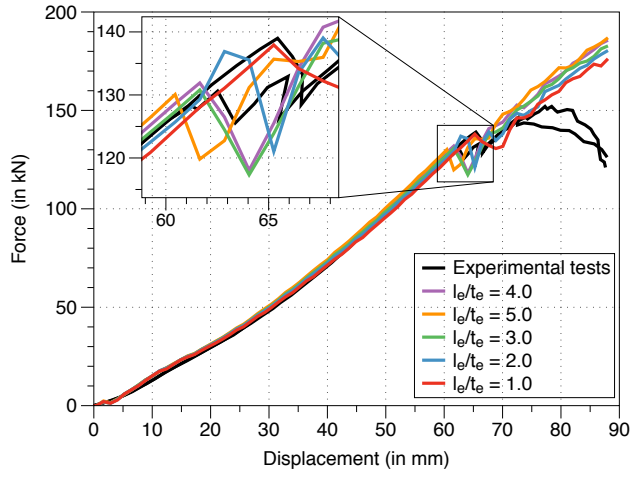
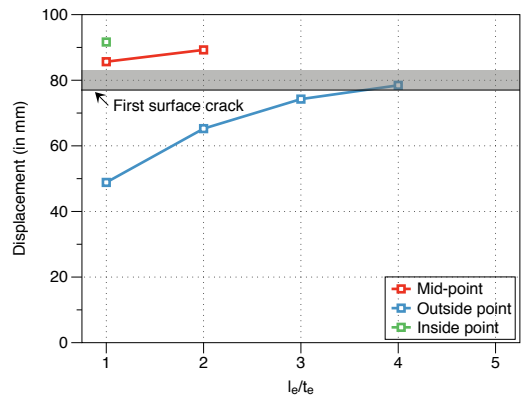
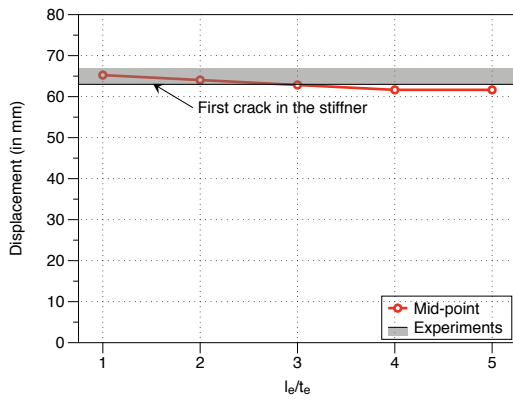


Figure 16: Numerical setup for the component tests.

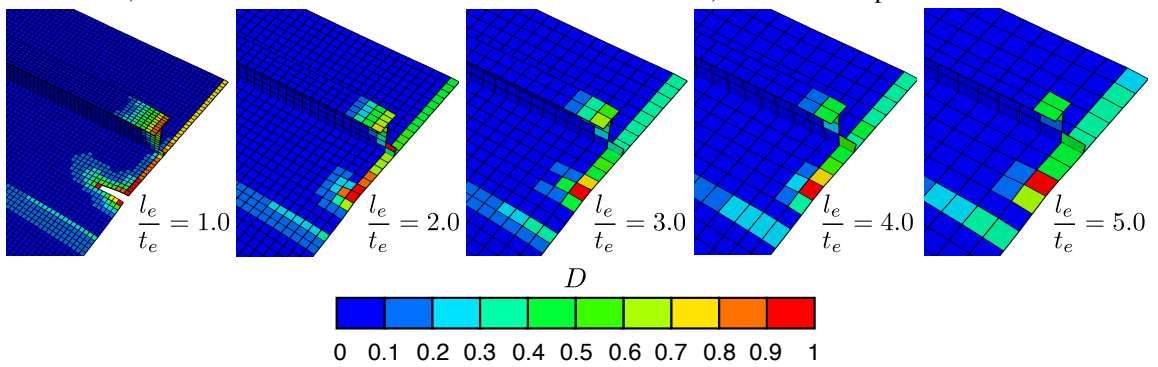


a) Force-displacement curves



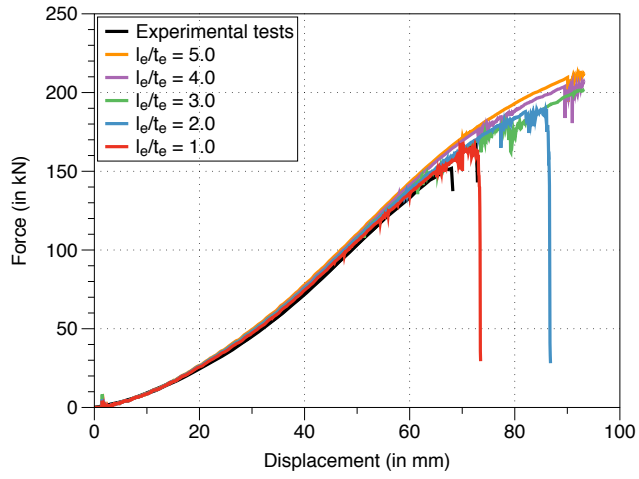
b) Failure in the stiffener material

c) Failure in the plate material

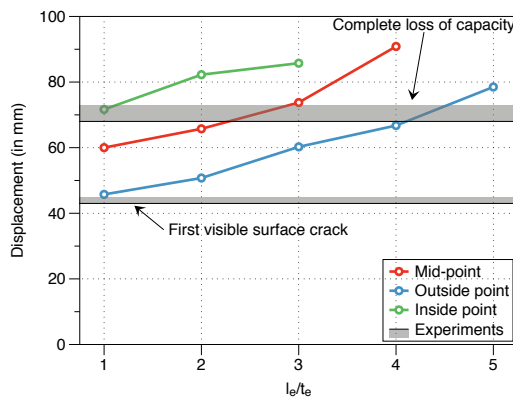


d) Damage fields on deformed specimens

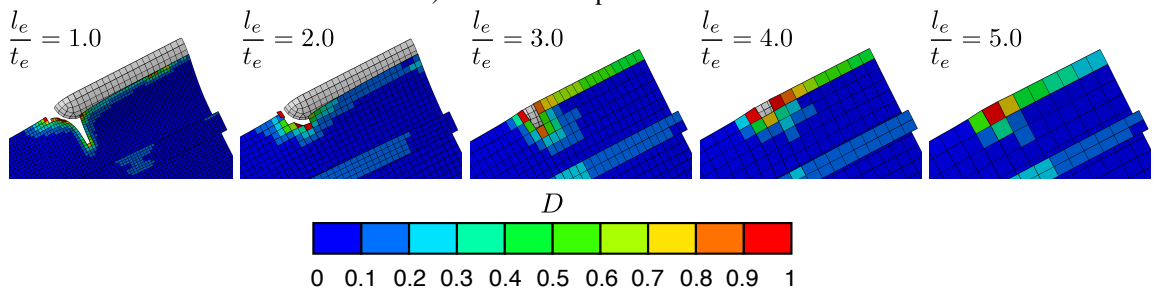
Figure 17: Results for the transverse impactor orientation.



a) Force-displacement curves



b) Failure in the plate material



c) Damage fields on deformed specimens

Figure 18: Results for the longitudinal impactor orientation.

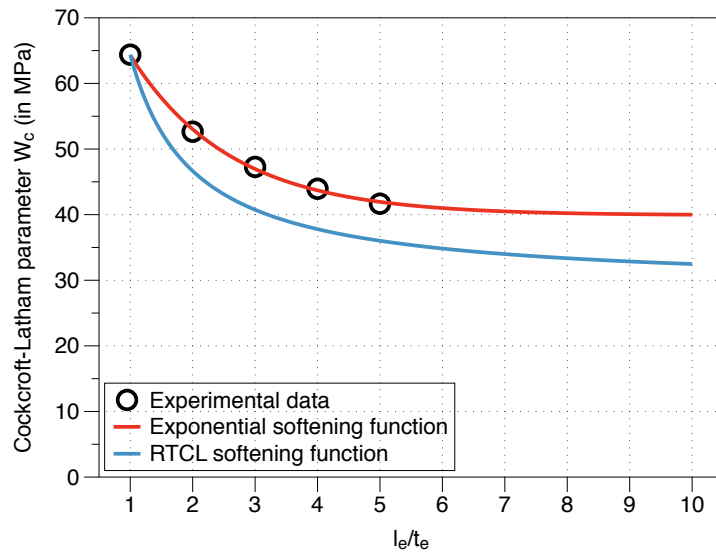


Figure 19: Illustration of mesh size regularization models.