



From Nucleation to Percolation: The Effect of System Size when Disorder and Stress Localization Compete

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A phase diagram for a one-dimensional fiber bundle model is constructed with a continuous variation in two parameters guiding the dynamics of the model: strength of disorder and range of stress relaxation. When the range of stress relaxation is very low, the stress concentration plays a prominent role and the failure process is nucleating where a single crack propagates from a particular nucleus with a very high spatial correlation unless the disorder strength is high. On the other hand, a high range of stress relaxation represents the mean-field limit of the model where the failure events are random in space. At an intermediate disorder strength and stress release range, when these two parameters compete, the failure process shows avalanches and precursor activities. As the size of the bundle is increased, it favors a nucleating failure. In the thermodynamic limit, we only observe a nucleating failure unless either the disorder strength is extremely high or the stress release range is high enough so that the model is in the mean-field limit. A complex phase diagram on the plane of disorder strength, stress release range, and system size is presented showing different failure modes - 1) nucleation 2) avalanche, and 3) percolation, depending on the spatial correlation observed during the failure process.

Keywords: disordered systems, fiber bundle model, stress release range, nucleation and growth, percolation, spatial correlation

1 INTRODUCTION

It is nearly a century since Alan Arnold Griffith developed his energy criterion for the fracture propagation of cracks in near-continuous solids [1, 2]. His celebrated work has revolutionized the world of materials science. Griffith considered a single sharp crack in an otherwise homogeneous elastic medium. In Griffith's theory, the crack propagation is considered as an equilibrium problem where the balance between two energies: reduction of strain energy, and increment in surface energy is measured during the crack propagation. He found that the critical stress σ_c to cause a crack of length l, to extend is $\sigma_c = (2Yg/\pi l)^{1/2}$ [3], where Y is Young's modulus and g is the surface energy per unit area. However, this is an idealized case that requires a pre-existing crack or notch in a homogeneous medium to concentrate the applied stress. In general, the initiation of a fracture in a solid is a much more complex process. Most engineering materials are far from homogeneous, there will always be a distribution of dislocations, flaws, and other heterogeneities present. The nucleation and propagation of a crack in heterogeneous systems are not understood because of the complexities of the stress singularities at the crack tip [4]. As the applied stress is increased, micro-cracks are likely to occur randomly on the heterogeneity and are uncorrelated. As the density of micro-cracks increases, the stress fields of the micro-cracks interact and the micro-cracks become correlated. The micro-cracks eventually may coalesce to form a through-going fracture. This irreversible process is a

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part of damage mechanics and is an integral part of the nucleation and propagation of fracture in heterogeneous environments.

During the failure process of a disordered system, a complex interplay is observed between quenched heterogeneities and local stress concentration. The former one leads to non-localized damage mechanics while the latter favors the formation of localized cracks. As a consequence of this interplay, we observe system size dependence of nominal stress distribution [5-7], scalefree avalanche size statistics [8–11], self-affine crack morphology [12], etc. In the limit of infinitesimal disorder strength, the crack grows within a disordered system in a nucleating manner [13-16]. On the other hand, when disorder strength is infinitely high, the effect of local stress concentration becomes irrelevant and the failure process is random in space [17, 18] like percolation. The reason behind the damage mechanics at high disorder is the heterogeneities in a material that create energy barriers which act as a resistance on the way of crack propagation and ultimately arrest the crack motion: a phenomenon known as lattice trapping or intrinsic crack resistance [19-24]. At intermediate disorder, the situation is more interesting, where the failure process takes place through a number of avalanches showing scale-free distributions of energies emitted during the avalanches [9] and mean-field exponents [25-28].

Tuning the strength of disorder is not achievable easily in experiments. Though through heat treatment one can tune the length scale of disorder in phase-separated glasses [29]. Earlier experiments also studied the role of a varying disorder strength during pattern formation in random spring network [30-32], the study of roughness of a fracture surface [33], the transition from nucleation to damage mechanics in random fuse network [34, 35], etc. Linear elastic fracture mechanics, on the other hand, predicts the load distribution around an Inglis crack [36, 37] to be $1/r^2$ -type where *r* is the distance from the crack tip. However, this form for relaxation of local stress can be affected by many parameters like correlation among defects [38] and effect of the limited size of the sample [39]. This in turn can change the dynamics of crack propagation. We will explore here the effect all three important parameters: disorder strength, stress release range and sample size in detail.

In this article, we study the spatial correlation during failure process of a fiber bundle model (FBM) [40, 41]. Fiber bundle model is very effective yet arguably the most simplest model to understand failure process of heterogeneous materials. The effect of a variable stress release range has already been observed in the context of this model [42, 43]. With a very low stress release range, the failure process is observed to be nucleating and at the same time, the failure abruptness is affected by the size of the bundle [42, 44]. The spatial correlation, in this limit, decreases as the disorder strength is increased [45]. On the other hand, at a very high stress release range (the mean-field limit), the failure abruptness is not a function of system size and only controlled by the strength of disorder [46]. A high thermal fluctuation as well leads to a failure process, random in space, with the same universality class of a site percolation [47]. When we combine both the effects of stress release range and disorder strength, the model produces rich relaxation dynamics with different modes of failure - abrupt, non abrupt, nucleating, and random in space

[48]. The avalanche statistics as well as the effect of system sizes has also been discussed recently in the context of fiber bundle model [49, 50]. Moreover, the record statistics in the avalanche statistics and study of elastic energy has been observed to be a vital key to predicting an upcoming failure [51]. The occurrence of different regimes with increasing disorder strength is also observed in spin systems as well. Recent work in the random field Ising model shows many small avalanches at high disorder and depinning behavior when disorder strength is very weak. At an intermediate critical value of disorder strength, avalanches of all possible sizes are observed [52, 53]. A study similar to the present one is explored in the context of random fuse network [34, 35]. In the former study, the avalanche size distribution as well as the system size scaling of average avalanche size and average crack size. In the latter study, the authors have studied the masses of the largest cluster, the mass of the backbone (the number of burnt bonds forming the chain that effectively disconnects the bottom from the top of the system), and how those masses scale with the system size in order to compare them with percolation. Here we carry out a study, similar to what was observed in the resistor network for a variable stress release range and construct a phase diagram on the plane of disorder, stress release range and system size. We have mainly studied the behavior of crack density to observe the spatial correlation. The crack density is explored earlier by the same author [42, 48] but a systematic study with system size, especially a systematic study of maximum crack density with the size of the bundle was missing there. Such a study is carried out in the present article and can offer a nice insight into the fracture pattern.

2 DESCRIPTION OF FIBER BUNDLE MODEL

After its introduction by Pierce in 1926 [54], the fiber bundle model [40] has been proven to be very effective in understanding the failure event of a disordered system. Due to this, fiber bundle model is growing popularity among engineering, material science as well as academics. This model is very effective yet arguably the most simple model guided by threshold activated dynamics.

A conventional fiber bundle model consists of *L* parallel Hookean fibers in between two plates which are pulled apart creating a stress σ on each fiber. The fluctuation among the strength values of individual fibers is the measure of disorder within the model. In the present work, we chose such strength values (*h*) from a power law distribution with a slope -1 and span from $10^{-\beta}$ to 10^{β} .

$$p(h) \sim \begin{cases} h^{-1}, & \left(10^{-\beta} \le h \le 10^{\beta}\right) \\ 0. & (\text{otherwise}) \end{cases}$$
(1)

Here β is related to the span of the distribution and a measure of disorder strength. We have chosen such a distribution as such long-tailed power-law distribution [55] has already been observed for the distribution of material strength. A certain fiber breaks irreversibly when the stress applied on it exceeds its strength. The stress of that fiber is then redistributed among rest of the model.

Here, we have adopted a generalized rule for stress redistribution. If σ_i is the stress on the broken fiber *i*, then the stress redistribution on fiber *j* at a distance r_{ij} from fiber *i* will be as follows

$$\sigma_j \to \sigma_j + \frac{r_{ij}^{-\gamma}}{Z} \sigma_i \tag{2}$$

Where σ_j is the stress on fiber *j* and *Z* is the normalization factor given by

$$Z = \sum_{i,k} r_{ik}^{-\gamma} \tag{3}$$

Where k runs overall intact fibers. Two extreme limits of this redistribution rule are global load sharing (GLS) [54] and local load sharing (LLS) [4, 56] limit. γ has a very low value for the former case and stress of the broken fiber here is redistributed among all surviving fibers in almost same amount. This is also the mean-field limit of the model. In the other extreme limit, y has a very high value and a large amount of the redistributed stress is carried by the neighboring fibers of the broken one only. The effect of stress concentration is most prominent here. Earlier studies show a critical value γ_c of the stress release range, for both 1d [42] and 2d [43], around which the model transit from the mean-field limit to the local load sharing limit. After such redistribution, due to the elevated local stress profile, further fibers may also break starting an avalanche. With such a process, the bundle may break through a single avalanche or comes to a steady state with some fibers broken and some intact. In the latter situation, the external stress is increased to break the next weakest fiber starting a new avalanche. Such process goes on until all fibers are broken.

3 NUMERICAL RESULTS

A one dimensional fiber bundle model is studied numerically with varying disorder strength (β), stress release range (γ), and system sizes (*L*). In *Local Load Sharing Fiber Bundle Model: Variation in* β , we explore the local load sharing limit of the model which can be achieved by setting a very high γ . On the other hand, *Generalized Model: Variation in Both* β and γ , deals with a generalized version of the model where the stress of a broken fiber is redistributed depending on the exponent γ keeping β constant. β is varied between 0.4 and 2.0 while γ varies from 0 to 3. The size of the bundle varies from 10³ to 10⁵.10⁴ realizations (bundle replications) are considered for our numerical simulation. *Universality* shows numerical results for uniform and Weibull threshold distribution, discussing the universal behavior of our result. Finally, in *Discussions* we have provided the concluding remarks on the present article.

3.1 Local Load Sharing Fiber Bundle Model: Variation in β

In this section, we have studied the fiber bundle model in the local load sharing limit. In this limit, the stress of a broken fiber is redistributed among the nearest neighbors only. This scenario can be achieved by setting a very high value of γ .



values ranging from 0.4 to 2.0. The red dots denotes ρ_m , the maximum possible patch density. The dotted line is the locus of B(1 - B) and represents random failure events for a 1d bundle. (B) Variation of ρ_m as the disorder strength β is varied continuously. We see three different regions (1) ρ_m is 1/L here suggesting a single crack propagates through the bundle in a nucleating manner (2) Here number of cracks originates and the number increases with β (3) The failure process here random in space. ρ_m has a value close to 0.25 here. The system size is kept constant at 10⁵.

We start our numerical simulation by observing the characteristics of the patches (or cracks) that are generated within the bundle during the evolution of the model. A certain patch is defined by the combination of an intact fiber and a broken fiber in its neighborhood. If we denote the intact fiber by 1 and broken fiber by 0, then any (1,0) or (0,1) combination on the one-dimensional chain of fibers will be characterized as a patch or crack. The patch density ρ at time *t* is defined by the number of patches at that time, normalized by the size *L* of the system. We also note the fraction *B* of broken fibers at time *t*. *B* is defined as the number of broken fibers divided by size *L* of the system. *Time*, in this case, is represented as the sum of stress increment and redistribution steps (see Ref. 1 for details). **Figure 1** shows the

¹At a certain point during the evolution of the model, if it goes through *m* stress increment with n_1, n_2, \ldots, n_m redistribution steps for 1*st*, 2*nd*, ..., *mth* stress increment, then the time corresponding to this scenario will be: $m + \sum n_m$.

variation of ρ with *B* for different disorder strength values ranging in between 0.4 and 2.0 for a bundle of size $L = 10^5$.

Figure 1A shows a non-monotonic behavior of ρ when B increases from 0 to 1. Earlier papers [42, 48] have already discussed that such crack density shows a non-monotonic behavior as the model evolves. B = 0 stands for the initial configuration where all fibers are intact while we obtain B = 1when all fibers are broken. The patch density ρ is zero for B = 0 as no cracks are there in the bundle. On the other hand, when the model is close to the failure point, there will be a single patch, making $\rho = 1/L$ when B approaches 1. At an intermediate B, ρ reaches a maximum value ρ_m . Before this maxima, ρ is an increasing function of B as new patches are generated within the bundle. After the maxima, the patches start to coalesce with each other and we observe a lesser and lesser number of cracks with increasing time (hence increasing B). Figure 1A shows as disorder strength is increased, ρ_m shifts to a higher value. We will discuss the variation of ρ_m with disorder next. The dotted line in the same figure is the locus of $\rho = B(1 - B)$. This dotted line represents failure events random in space for a 1d FBM. This can be understood as follows. If *B* is the fraction of fibers broken, then the probability of having a broken and an intact fiber will be B and (1 - B) respectively. A patch then will be created by placing an intact fiber beside a broken one, the probability of which will be B(1 - B) on a 1d lattice. By equating $d\rho/dB$ for this locus to zero we get the maximum $\rho_m = 0.25$ and the *B* value to be 0.5 at this maxima. The failure pattern becomes more and more random when β , the disorder strength, is high enough. On the other hand, when the disorder strength is very low ($\beta = 0.4$), we see $\rho = 1/L$ independent of the value of B. This suggests a pure nucleating failure starting from the very beginning until the global failure.

Figure 1B shows the variation of ρ_m explicitly when β is continuously varied. We observe three different regions.

- 1) Nucleation ($\beta \le 0.4$) Here ρ_m has a value close to 1/L. This suggests that only a single crack is generated within the bundle in this limit and this crack nucleates to create global failure. Due to the low strength of the disorder, the failure process here is guided by the local stress concentration at the crack tips.
- 2) Percolation ($\beta \ge 1.2$) In this limit the behavior of *B* vs ρ matches closely with $\rho = B(1 - B)$. ρ_m has a value close to 0.25. The failure events are random in space here making it reminiscent of percolation in 1d lattice. The failure process is completely guided by the disorder strength and the local stress concentration is almost non-existing.
- 3) Avalanche (0.4 < β < 1.2) in the intermediate disorder strength, there is an interplay between the disorder strength and the local stress concentration. The failure process here starts in a percolating manner but later the local stress concentration takes over making the rest of the failure events nucleating. We will be discussing this spatial correlation in more detail later in this paper.

Figure 2A shows how the maximum patch density ρ_m responds to the size of the bundle. The results are repeated for three different β values 0.3, 0.7, and 1.2, in order to cover all three regions - nucleation, avalanche, and percolation,



Variation of ξ with β . $\xi = 1$ and 0 in the region 1 and 3 respectively. In the intermediate region II, ξ decreases continuously with β .

mentioned above. When disorder strength β is low (0.3), ρ_m decreases with L in a scale-free manner with exponent -1. This suggests that the maximum number of cracks observed in the bundle decreases with increasing size and the model goes towards nucleation more and more as the model approaches the thermodynamic limit. On the other hand, for $\beta = 1.2$, ρ_m is independent of L and saturates at a value close to 0.25. As mentioned above, the failure process is percolating here and remains the same irrespective of the size of the bundle. In the intermediate disorder, where the disorder strength and local stress concentration compete with each other, we observe

$$\rho_m \sim L^{-\xi(\beta)} \tag{4}$$

Where the exponent ξ is a function of β .

Figure 2B shows the variation of exponent ξ with the strength of disorder β . ξ remains at 1 for low β where the failure is nucleating, gradually decreases in region avalanche, and becomes constant at 0 in the limit of percolation. The nature of patch density remains the same in the percolation region ($\xi = 0$) only. In both avalanche and nucleation, fewer patches grow as the size of



the bundle is increased, suggesting that the effect of the local stress concentration becomes more prominent here as the model goes towards the thermodynamic limit.

Here, we will discuss a dynamic parameter that helps us to understand the onset of the nucleation process with time more clearly. As explained earlier, time t here is analogous to the total number of redistribution plus stress increment steps prior to the global failure. We start by breaking the weakest fiber, say *i*, at time t = 0 by the first stress increment. Let us assume further n_1 fibers break at the next time step (t = 1) upon redistributing the stress carried by the weakest fiber. We consider the distance Δr between these two consecutive events to be the minimum of distances between fiber i and other n_1 fibers that break after redistribution. Here, Δr is not the exact lattice distance as only intact fibers are considered while calculating it. The distance across a broken patch is considered to be one independent of the size of the patch. This is due to the LLS scheme that we have adopted. Whenever a fiber at a notch breaks and the redistributed stress breaks the fiber at the other notch, the failure is still nucleating, no matter how large this patch is. Next, we square this distance and average it over 10^4 realizations to get $\langle \Delta r^2 \rangle$ at time t = 0. Next, we move our reference frame to the fiber among those n_1 fibers that had the minimum distance from fiber *i*. Let's denote this new fiber as *j*. If further n_2 fibers break in the next redistribution, $\langle \Delta r^2 \rangle$ at t = 1 will be calculated by the same procedure: find Δr from the minimum of distances between fiber *j* and those n_2 fibers, square it and average over 10⁴ realizations. Such a parameter was explored earlier by Stormo et. al [57] in the context of the soft clamp model to point out the onset of localization. Figure 3 shows this variation of $\langle \Delta r^2 \rangle$ with time *t* for $\beta = 0.3$, 0.7 and 1.2. For all three disorder strength values, $\langle \Delta r^2 \rangle$ starts from a high value at low t and then decreases towards 1 when t is high. A high value of $\langle \Delta r^2 \rangle$ suggests the fibers that break consecutively are far from each other. This is a spatially uncorrelated failure. On the other hand, when $\langle \Delta r^2 \rangle = 1$, the consecutive failures take place from the neighboring fibers only. This behavior stands for pure nucleation. For $\beta = 0.3$, $\langle \Delta r^2 \rangle$ becomes 1 very fast and stays there independent of t until the bundle reaches global failure. For $\beta = 1.2$, we observe the opposite behavior where $\langle \Delta r^2 \rangle$ stays at a high value for a long time and falls to 1 just before global failure. The former behavior is nucleating (1)while the latter one is percolating (3). The visualization of the failure process for both (1) and (3) is shown below Figure 3. The x-axis of each plot is time and the y-axis is fiber index. The color gradient is over the local stress profile. The yellow color stands for the failed fibers. For (1), we see a single crack growing in a nucleating manner from the very beginning. For (3), on the other hand, there are no nucleating yellow-colored fibers and the rupture events are spatially uncorrelated. For the avalanche (2) behavior, there is a spatially uncorrelated failure in the beginning as well as nucleation close to global failure. Figure 3 shows the point for $\beta = 0.7$ where $\langle \Delta r^2 \rangle$ becomes 1 indicating onset of localized (nucleating) failure events.

Finally, we have constructed the phase diagram of disorder strength β and system size *L* to show all three failure processes. In **Figure 4**, $1/\beta$ is plotted against 1/L. This is done in this way so that the origin (0,0) of this plot corresponds to $L \to \infty$ and $\beta \to \infty$, an infinite disorder in the thermodynamic limit. As discussed earlier, if the disorder strength is increased, we start with nucleation, go through an avalanche, and finally reach







percolation behavior. The spatial correlation in rupturing events are shown in the corresponding phases. Now, if the disorder strength is kept constant and the size of the bundle is increased, a percolating behavior moves towards avalanche and an avalanche behavior moves towards nucleation. Due to weak dependence of parameters like ρ_m on *L* (see **Figure 2**), it will not be possible to see (as the system size has to be very high) this change from percolation to avalanche if we are well inside the percolation region. To see this change at relatively lower system sizes, it is required to keep the disorder strength at a value so that the model is closer to the percolation-avalanche interface. The opposite happens if the system size is decreased instead of increasing. This suggests we observe only nucleating failure in the thermodynamic limit unless the disorder is infinitely high. This effect of disorder was explored earlier in the context of random fuse network by Shekhawat et al. [34] and Moreira et al. [35]. We observe that the fiber bundle model in one dimension follows the same trend.

3.2 Generalized Model: Variation in Both β and γ

In this section, the model is studied with a continuous variation in both γ when β . We start our numerical simulation by observing the spatial correlation through the rupture events with increasing time as the bundle fails.

Figure 5 shows such correlation for β values ranging in between 0.3 and 1.2 and γ values within 0 and 3. For each small figure, the x-axis shows the time *t* and the y-axis shows the fiber index. From left to right, the figures are plotted for

increasing values of γ keeping the disorder strength β constant. On the other hand, from bottom to the top, the figures are plotted with increasing β and keeping γ constant. We observe the following behavior:

For *low* β and *high* γ , the fibers break in a nucleating manner. Due to the low stress release range, the stress concentration plays a crucial role and dominates the failure process. Moreover, due to the low value of disorder strength, the probability that the fibers break with redistribution (without any increment in external stress) from the neighborhood is high. On the other hand, for *high* β and *low* γ , the fluctuation between threshold strength as well as the stress release range is high. As a result, we observe rupture events random in space and through increment in external stress.

Now, keeping the β fixed at a low value, as we decrease γ , the model slowly goes towards the mean-field limit. In this limit, the stress of the broken fibers are redistributed among all surviving fibers. This increases the chance that whenever one fiber breaks, the next rupture event may take place from somewhere which is not the neighborhood of the broken fiber. The failure process, in this case gradually deviates from the nucleating behavior as γ decreases.

Instead, if we keep γ fixed at a high value and increase β , the fluctuation among fiber strengths increases. Here, the stress of the broken fiber is redistributed in the neighborhood (as γ is high) but due to this increase in fluctuation we will find more strong fibers in this neighborhood that will finally arrest the growth of a crack - a phenomena known as lattice trapping or intrinsic crack resistance [19–24]. This forces the growth of a different crack from a different place.

As discussed in Figure 5, a single crack or a number of cracks are observed in the bundle depending on what the values of



disorder strength β and the stress release range γ are. In **Figure 6**, we have studied how the maximum number of cracks (ρ_m) varies as we vary both β and γ . At first, we observe the variation of ρ_m with γ for a constant value of β . The study is then repeated for β values ranging in between 0.2 and 0.8.

For low γ , ρ_m has a higher value and decreases as γ increases and crosses the critical value γ_c [42]. The results can be discussed in three parts. At an intermediate disorder ($0.3 \le \beta \le 0.5$), ρ_m saturates close to 0.25 for low γ and decreases to 1/L when γ is high. A ρ_m value close to 1/L suggests there is only one crack that propagates throughout the system. On the other hand, as already discussed in the manuscript, ρ_m close to 0.25 suggests a failure process random in space. In this limit, all three regions are accessible with a variation in γ .

At low β (<0.3), we observe that ρ_m goes to 1/*L* for high γ but does not approach 0.25 even if γ is very low. In this limit, we do not see a percolation like a random failure. This is due to very low disorder strength, where the bundle breaks very abruptly before it can reach the real maximum value of ρ_m (\approx 0.25) at low γ . On the other hand, for high β (>0.5), ρ_m reaches 0.25 easily at low γ but do not reach 1/*L* for high γ . In this case, ρ_m does not reach 1/*L* even at high γ due to the intrinsic crack resistance caused by the high fluctuation in threshold strength, which arrests an propagating crack in the process. As a result, a nucleating failure is not observed here.

Figure 7A shows the system size effect of ρ_m for $\beta = 0.6$ and for wide range of stress relaxation γ ($0 \le \gamma \le 3$). We observe ρ_m to decrease in a scale-free manner with system size *L*,

$$\rho_m \sim L^{-\zeta} \tag{5}$$

Where ζ is an increasing function of γ . At low γ , ρ_m is almost independent of *L* and saturates around 0.25. When γ is high, ρ_m responds to the change in *L* very sharply and decreases as *L* increases.



FIGURE 7 (A) System size effect of ρ_m for $\beta = 0.6$ and γ between 0 and 3. We observe $\rho_m \sim L^{-\zeta}$, where ζ is an increasing function of γ . (B) Variation of ζ with γ for $0.2 \le \beta \le 1.2$. For low γ , $\zeta \approx 0$ independent of β . For high γ , on the other hand, $\zeta = 1$ depending on whether β is low enough or not.

The variation of the exponent ζ is shown in **Figure 7B** ζ has a value close to 0 independent of disorder strength β when γ low. At such a low value of γ , the model is in the mean field limit and changing the system size does not change the dynamics of the model. As γ increases, the model slowly deviates from the mean-field limit and the effect of local stress concentration becomes more and more prominent. In this limit, ζ starts to increase slowly. When γ crosses a certain value that depends on β , ζ finally reaches 1. A Higher value of β will require a higher γ value in order to obtain $\zeta = 1$. Finally, when β is very high, ζ remains close to 0 independent of the stress release range γ . Here, the failure process is random in space, independent of both β and γ .

Figure 8 shows above mentioned three regions from the study of the maximum number of cracks (ρ_m) when both disorder strength β and the stress release range γ are varied simultaneously. The color gradient in **Figure 8** is on ρ_m , with a maximum value of 0.25 (lightest color) and minimum value of 1/L (darkest color) which is 10^{-5} as the size of our bundle is 10^5 . We observe ρ_m to have the lowest value for lowest possible β and highest possible γ .

This is due to the fact that fluctuation among local strength values are minimum here and at the same time stress release range is also minimum making the local stress concentration most prominent. At



FIGURE 8 | Histogram of ρ_m on the plane of γ vs β . The brightest color corresponds to $\rho_m \approx 0.25$ while the darkest color stands for $\rho_m \approx 1/L$. At low β , as γ increases, we go from region 3 (percolation) to 2 (avalanche) to 1 (nucleation). Same transition is observed when β decreases at high γ .



avalanche to nucleation and percolation to avalanche respectively. The three parameters constructing the plane are: disorder strength (β), stress release range (γ) and system size (L).

this situation, as we decrease γ , we go towards percolation (light color) through avalanche region. The same behavior is observed $(1 \rightarrow 2 \rightarrow 3)$ if we increase β keeping γ fix. The figure also shows the existence of γ_c [42, 43], the value of γ below which the model enters the mean-field limit. For $\gamma < \gamma_c$ we have almost an uniform gradient of light color suggesting ρ_m is close to 0.25 here independent of β . We observe a very small change in the color gradient if we increase γ at a high β .



FIGURE 10 Variation of ρ_m for uniform (upper) and Weibull (lower) distribution when both the disorder strength (δ or k) and stress release range (γ) is varied continuously. Three distinct regions (1) nucleation (2) avalanche and (3) percolation, is observed independent of the choice of threshold distribution.

Figure 9 shows different regions - nucleation, avalanche and percolation, with their unique nature of crack propagation during the failure process when the disorder strength (β), stress release range (γ) , and system size (L) are continuously varied. Figure 9A shows the plane separating the region nucleation from avalanche. The plane seems to diverge for high β . This is due to the fact that, at high beta the fluctuation among fiber strength values will be high and the γ value will also have to be very high to make the local stress concentration prominent enough to create nucleation. At the same time, since an increasing L has already been seen to favor nucleating failure, we achieve such nucleation at relatively lower γ value at higher L when β is kept fixed. Figure 9B, on the other hand, shows the plane between avalanche and percolation. We observe the same effect of L here – as L increases, the transition from percolation to avalanche takes place at a lower value of γ . At the same time, as β increases, we have to go to a higher γ value to enter the avalanche region from percolation. The sudden upward curvature of *nucleation – avalanche* plane at high β suggests that if the disorder strength is extremely high, we might not get a nucleation region. Similarly, the sudden downward curvature of avalanche – percolation plane at low β suggests that if the disorder strength is extremely low, we might not get a percolation region.

4 UNIVERSALITY

In this section we will discuss the universality of our study by using two different distributions other than the power law. For this purpose, we have adopted an uniform and a Weibull distribution as described below:

$$p(h) \sim \begin{cases} \frac{1}{2\delta}, & (0.5 - \delta \le h \le 0.5 + \delta) \\ 0, & (\text{otherwise}) \end{cases}$$
(6)

Where δ is the half-width of the distribution as well as the measure of the disorder strength.

$$p(h) \sim \left(\frac{k}{\lambda}\right) \left(\frac{h}{\lambda}\right)^{k-1} e^{-\left(\frac{h}{\lambda}\right)^k}$$
 (7)

Where k and λ are the Weibull modulus and scale parameter respectively. k controls the disorder strength in this case. We vary δ from 0 to 0.5 while k is varied between 1 and 10. The scale parameter λ is kept constant at 1. Similar to the power law distribution, the uniform distribution is also a bounded distribution while the Weibull distribution is open.

Figure 10 shows the variation of ρ_m when both disorder strength and stress release range is tuned continuously. The results for uniform distribution are shown in the left figure. The right figure shows the same for the Weibull distribution. Similar to **Figure 8**, the color gradient represents the variation in ρ_m that spans from 1/*L* to 0.25. The results suggest that all three regions (1) nucleation (2) avalanche, and (3) percolation, are observed independent of the choice of the threshold distribution. As γ increases, the model goes from the nucleating failure to a failure process random in space. On the other hand, a spatially correlated failure process is not observed by increasing δ up to 0.5 (the distribution spans from 0 to 1) as the disorder strength is not large enough. Though for Weibull distribution, both nucleating and random failure is observed at high and low *k* respectively.

5 DISCUSSIONS

As we have already discussed, two major factors governing the mode of failure in disordered solids are the strength of heterogeneities and the effective range over which the stress field is modified following a local rupture event. On the other hand, studies in random resistor network model [34, 35] claims the failure mode, in the large system size limit, to be always nucleation-driven unless the strength of disorder is extremely high. Qualitatively this is the main finding of the present paper as well as what was observed in the random fuse

REFERENCES

- 1. Dieter GE. Mechanical Metallurgy. McGraw Hill Education (1989).
- Lawn B. Fracture of Brittle Solids. Cambridge: Cambridge Solid State Science Series (1993).
- 3. Knott JF. Fundamentals of Fracture Mechanics. New York: Butterworths (1973).
- Phoenix SL. The Asymptotic Distribution for the Time to Failure of a Fiber Bundle. Adv Appl Probab (1979) 11:153–87. doi:10.1017/s0001867800031748

network model earlier [34, 35]. The simplicity of the fiber bundle model allows us to include extra parameters like stress release range compared to the random fuse network model and study its effect as well on the spatial correlation as the model evolves. The precursor events (such as scale-free size distribution of rupture events prior to global failure and scale free distribution of emitted energies during such avalanches), previously seen in the statistical models [7, 58], would imply that a nucleation-like failure would not be achievable even in the large system size limit. Such precursor events are observed experimentally [59] as well for which the extreme disorder is not necessarily the physical condition. The stress release range (analogous to fracture process zone in real experiments) comes into play here that might cause a different mode of failure, other than nucleation, even when the system size is high.

In conclusion, we present a detailed study in fiber bundle model by varying main three parameters, strength of disorder, range of stress relaxation and system size, that determines the dynamics of the model as it is acted by an external stress. An increasing disorder strength (increasing β) or stress release range (decreasing γ) favors a failure that is random in space. On the other hand, an increasing system size makes the failure more and more nucleating. The avalanche behavior is observed for all β or γ . If β is very high then it is difficult to achieve the nucleating behavior unless the value of γ is very high. On the other hand, when β is low, achieving nucleating failure is easy but it is difficult to observe pure random failure by decreasing γ . Finally, for the intermediate β value, we achieve both nucleation and percolation like failure by increasing and decreasing γ respectively.

DATA AVAILABILITY STATEMENT

The raw data supporting the conclusions of this article will be made available by the authors, without undue reservation.

AUTHOR CONTRIBUTIONS

The author confirms being the sole contributor of this work and has approved it for publication.

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- Harlow DG, and Phoenix SL. The Chain-Of-Bundles Probability Model for the Strength of Fibrous Materials I: Analysis and Conjectures. J Compos Mater (1978) 12:195–214. doi:10.1177/002199837801200207
- Bazant ZP. Scaling Theory for Quasibrittle Structural Failure. Proc Natl Acad Sci (2004) 101:13400–7. doi:10.1073/pnas.0404096101
- Alava MJ, Nukala PKVV, and Zapperi S. Statistical Models of Fracture. Adv Phys (2006) 55:349–476. doi:10.1080/00018730300741518
- Hemmer PC, and Hansen A. The Distribution of Simultaneous Fiber Failures in Fiber Bundles. J Appl Mech (1992) 59:909–14. doi:10.1115/1.2894060

- Petri A, Paparo G, Vespignani A, Alippi A, and Costantini M. Experimental Evidence for Critical Dynamics in Microfracturing Processes. *Phys Rev Lett* (1994) 73:3423–6. doi:10.1103/physrevlett.73.3423
- Garcimartín A, Guarino A, Bellon L, and Ciliberto S. Statistical Properties of Fracture Precursors. *Phys Rev Lett* (1997) 79:3202–5. doi:10.1103/physrevlett.79.3202
- Salminen LI, Tolvanen AI, and Alava MJ. Acoustic Emission from Paper Fracture. Phys Rev Lett (2002) 89:185503. doi:10.1103/physrevlett.89.185503
- Bonamy D, and Bouchaud E. Failure of Heterogeneous Materials: A Dynamic Phase Transition? *Phys Rep* (2011) 498:1–44. doi:10.1016/ j.physrep.2010.07.006
- Duxbury PM, Beale PD, and Leath PL. Size Effects of Electrical Breakdown in Quenched Random Media. *Phys Rev Lett* (1986) 57:1052–5. doi:10.1103/ physrevlett.57.1052
- Duxbury PM, Leath PL, and Beale PD. Breakdown Properties of Quenched Random Systems: The Random-Fuse Network. *Phys Rev B* (1987) 36:367–80. doi:10.1103/physrevb.36.367
- Chakrabarti BK, and Benguigui LG. Statistical Physics of Fracture and Breakdown in Disordered Systems. Oxford: Oxford Science Publications (1997).
- Manzato C, Shekhawat A, Nukala PKVV, J. Alava M, Sethna JP, and Zapperi S. Fracture Strength of Disordered Media: Universality, Interactions, and Tail Asymptotics. *Phys Rev Lett* (2012) 108:065504. doi:10.1103/ physrevlett.108.065504
- Roux Sp., Hansen A, Herrmann H, and Guyon E. Rupture of Heterogeneous media in the Limit of Infinite Disorder. J Stat Phys (1988) 52:237–44. doi:10.1007/bf01016411
- Hansen A, and Schmittbuhl J. Origin of the Universal Roughness Exponent of Brittle Fracture. Surfaces: Stress-Weighted Percolation in the Damage Zone.. *Phys Rev Lett* (2003) 90:045504. doi:10.1103/physrevlett.90.045504
- Thomson R, Hsieh C, and Rana V. Lattice Trapping of Fracture Cracks. J Appl Phys (1971) 42:3154–60. doi:10.1063/1.1660699
- Curtin WA, and Mater J. On Lattice Trapping of Cracks. J Mater Res (1990) 5: 1549–60. doi:10.1557/jmr.1990.1549
- Rice JR. Dislocation Nucleation from a Crack Tip: An Analysis Based on the Peierls Concept. J Mech Phys Sol (1992) 40:239–71. doi:10.1016/s0022-5096(05)80012-2
- Pérez R, and Gumbsch P. Directional Anisotropy in the Cleavage Fracture of Silicon. *Phys Rev Lett* (2000) 84:5347–50. doi:10.1103/physrevlett.84.5347
- Bernstein N, and Hess DW. Lattice Trapping Barriers to Brittle Fracture. *Phys Rev Lett* (2003) 91:025501. doi:10.1103/physrevlett.91.025501
- Mattoni A, Colombo L, and Cleri F. Atomic Scale Origin of Crack Resistance in Brittle Fracture. *Phys Rev Lett* (2005) 95:115501. doi:10.1103/ physrevlett.95.115501
- Hansen A, and Hemmer PC. Burst Avalanches in Bundles of Fibers: Local versus Global Load-Sharing. *Phys Lett A* (1994) 184:394–6. doi:10.1016/0375-9601(94)90511-8
- Zapperi S, Ray P, Stanley HE, and Vespignani A. First-Order Transition in the Breakdown of Disordered Media. *Phys Rev Lett* (1997) 78:1408–11. doi:10.1103/physrevlett.78.1408
- Zapperi S, Nukala PKVV, and Simunovic S. Crack Roughness and Avalanche Precursors in the Random Fuse Model. *Phys Rev E* (2005) 71:026106. doi:10.1103/physreve.71.026106
- Zapperi S, Nukala PKVV, and Šimunović S. Crack Avalanches in the Three-Dimensional Random Fuse Model. *Physica A: Stat Mech its Appl* (2005) 357: 129–33. doi:10.1016/j.physa.2005.05.071
- Dalmas D, Lelarge A, and Vandembroucq D. Crack Propagation through Phase-Separated Glasses: Effect of the Characteristic Size of Disorder. *Phys Rev Lett* (2008) 101:255501. doi:10.1103/physrevlett.101.255501
- Malakhovsky I, and Michels MAJ. Effect of Disorder Strength on the Fracture Pattern in Heterogeneous Networks. *Phys Rev B* (2007) 76:144201. doi:10.1103/physrevb.76.144201
- Ray P, and Date G. Spatial Scaling in Fracture Propagation in Dilute Systems. *Physica A: Stat Mech its Appl* (1996) 229:26–35. doi:10.1016/ 0378-4371(95)00431-9
- Parihar RPS, Mani DV, Banerjee A, and Rajesh R. Role of Spatial Patterns in Fracture of Disordered Multiphase Materials. *Phys Rev E* (2020) 102:053002. doi:10.1103/physreve.102.053002
- Silberschmidt VV. Effect of Material's Randomness on Scaling of Crack Propagation in Ceramics. Int J Fract (2006) 140:73–85. doi:10.1007/s10704-005-3994-8

- Shekhawat A, Zapperi S, and Sethna JP. From Damage Percolation to Crack Nucleation through Finite Size Criticality. *Phys Rev Lett* (2013) 110:185505. doi:10.1103/physrevlett.110.185505
- Moreira AA, Oliveira CLN, Hansen A, Araújo NAM, Herrmann HJ, and Andrade JS, Jr. Fracturing Highly Disordered Materials. *Phys Rev Lett* (2012) 109:255701. doi:10.1103/physrevlett.109.255701
- Schmittbuhl J, Roux S, Vilotte J-P, and Jorgen Måløy K. Interfacial Crack Pinning: Effect of Nonlocal Interactions. *Phys Rev Lett* (1995) 74:1787–90. doi:10.1103/physrevlett.74.1787
- Ramanathan S, Ertaş D, and Fisher DS. Quasistatic Crack Propagation in Heterogeneous Media. *Phys Rev Lett* (1997) 79:873-6. doi:10.1103/ physrevlett.79.873
- Peng C-K, Havlin S, Schwartz M, and Stanley HE. Directed-polymer and Ballistic-Deposition Growth with Correlated Noise. *Phys Rev A* (1991) 44: R2239–R2242. doi:10.1103/physreva.44.r2239
- Xia ZC, and Hutchinson JW. Crack Patterns in Thin Films. J Mech Phys Sol (2000) 48:1107–31. doi:10.1016/s0022-5096(99)00081-2
- 40. Hansen A, Hemmer PC, and Pradhan S. *The Fibre Bundle Model*. Germany: Wiley VCH (2015).
- Chakrabarti BK, Biswas S, and Pradhan S. Cooperative Dynamics in the Fiber Bundle Model. Front Phys (2021) 8:613392. doi:10.3389/ fphy.2020.613392
- Biswas S, Roy S, and Ray P. Nucleation Versus Percolation: Scaling Criterion for Failure in Disordered Solids. *Phys Rev E* (2015) 91:050105(R). doi:10.1103/ physreve.91.050105
- Hidalgo RC, Moreno Y, Kun F, and Herrmann HJ. Fracture Model with Variable Range of Interaction. *Phys Rev E* (2002) 65:046148. doi:10.1103/ physreve.65.046148
- 44. Roy S. Predictability and Strength of a Heterogeneous System: The Role of System Size and Disorder. *Phys Rev E* (2017) 96:042142. doi:10.1103/ PhysRevE.96.042142
- Sinha S, Roy S, and Hansen A. Phase Transitions and Correlations in Fracture Processes Where Disorder and Stress Compete. *Phys Rev Res* (2020) 2:043108. doi:10.1103/physrevresearch.2.043108
- 46. Roy S, and Ray P. Critical Behavior in Fiber Bundle Model: A Study on Brittle to Quasi-Brittle Transition. *Epl* (2015) 112:26004. doi:10.1209/ 0295-5075/112/26004
- Sinha S, Roy S, and Hansen A. Crack Localization and the Interplay between Stress Enhancement and thermal Noise. *Physica A: Stat Mech its Appl* (2021) 569:125782. doi:10.1016/j.physa.2021.125782
- Roy S, Biswas S, and Ray P. Modes of Failure in Disordered Solids. *Phys Rev E* (2017) 96:063003. doi:10.1103/physreve.96.063003
- Kadar V, Danku Z, and Kun F. Size Scaling of Failure Strength with Fat-Tailed Disorder in a Fiber Bundle Model. *Phys Rev E* (2017) 96:033001. doi:10.1103/ PhysRevE.96.033001
- Kadar V, and Kun F. System-size-dependent avalanche statistics in the limit of high disorder. *Phys Rev E* (2019) 100:053001. doi:10.1103/ PhysRevE.100.053001
- Kadar V, Pal G, and Kun F. Record Statistics of Bursts Signals the Onset of Acceleration Towards Failure. *Sci Rep* (2020) 10:2508. doi:10.1038/ s41598-020-59333-4
- Tadic B, Mijatovic S, Janicevic S, Spasojevic D, and Rodgers GJ. The Critical Barkhausen Avalanches in Thin Random-Field Ferromagnets with an Open Boundary. *Scientific Rep* (2019) 9:6340. doi:10.1038/s41598-019-42802-w
- Spasojevic D, Mijatovi S, Navas-Portella V, and Vives E. Crossover from Three-Dimensional to Two-Dimensional Systems in the Nonequilibrium Zero-Temperature Random-Field Ising Model. *Phys Rev E* (2018) 97:012109. doi:10.1103/ PhysRevE.97.012109
- Pierce FT. 32—X.—Tensile Tests for Cotton Yarns v.—"The Weakest Link" Theorems on the Strength of Long and of Composite Specimens. J Text Inst (1926) 17:355. doi:10.1080/19447027.1926.10599953
- Fisher RA, and Tippett LHC. Limiting Forms of the Frequency Distribution of the Largest or Smallest Member of a Sample. *Math Proc Camb Phil Soc* (1928) 24:180–90. doi:10.1017/s0305004100015681
- Smith RL, and Phoenix SL. Asymptotic Distributions for the Failure of Fibrous Materials under Series-Parallel Structure and Equal Load-Sharing. J Appl Mech (1981) 48:75–82. doi:10.1115/1.3157595

- Stormo A, Gjerden KS, and Hansen A. Onset of Localization in Heterogeneous Interfacial Failure. *Phys Rev E* (2012) 86:025101(R). doi:10.1103/physreve.86.025101
- Roy S, and Biswas S. Size Distribution of Emitted Energies in Local Load Sharing Fiber Bundles. *Front Phys* (2021) 9:643602. doi:10.3389/ fphy.2021.643602
- Baro J, Corral A, Illa X, Planes A, Salje EKH, Schranz W, et al. Statistical Similarity between the Compression of a Porous Material and Earthquakes. *Phys Rev Lett* (2013) 110:088702. doi:10.1103/physrevlett.110.088702

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