

33 **1 INTRODUCTION**

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35 Supplementary cementitious materials (SCMs) are frequently used as a partial substitute for
36 cement clinker to reduce the emissions caused by cement manufacturing. However, some of the
37 SCMs traditionally used have limited availability when compared to the global increase of cement
38 demand [1]. Therefore, alternative SCMs need to be found that are available in sufficient quantity
39 and show similar or even improved interaction with cement clinker hydration.

40 Limestone is commonly used as an SCM and, according to the European cement standard EN 197-
41 1, it can replace up to 5%wt clinker in CEM I Portland cements and up to 35%wt in CEM II
42 Portland-limestone cements [2]. Due to the additional carbonates, the stable AFm phases which
43 form are hemi- and monocarbonate and not monosulphate. This change in phase stabilities results
44 in the so-called “ettringite stabilization”, because the sulphate-containing hydration phase
45 ettringite does not transform to monosulphate when the sulphate sources (e.g. gypsum) are
46 depleted. This stabilization of ettringite beyond sulphate depletion results in a relative increase
47 in the volume of hydrates, a corresponding decrease in porosity, and an increase in compressive
48 strength [3,4].

49 The limited aluminium content in Portland cements means this effect is somewhat limited [4], but
50 it can be amplified if additional aluminium is provided to the system by adding aluminium-
51 delivering SCMs [5–8]. In this study, we used metakaolin as a model material for industrially
52 available calcined clays supplying silicates and alumina to the system. Metakaolin is also known
53 to be a pozzolanic material. During its pozzolanic reaction, the silicates provided by the
54 metakaolin react with the portlandite formed during the hydration of the Portland cement to
55 produce additional C-S-H [9], and the aluminium provided by the metakaolin reacts with the
56 portlandite to form additional AFm phases [5–8]. Moreover, the aluminium is also partly taken up
57 by the C-S-H [10] and can lead to the formation of calcium aluminate silicate hydrate phases (C-A-
58 S-H phases), as observed in composite cements containing slag, fly ash or metakaolin [11,12].

59 Because high-grade limestone required by EN 197-1 [2] is not sufficiently available in all parts of
60 the world, various other carbonate sources are in the focus of ongoing research. One possible
61 alternative carbonate source that has been studied is dolomite [13]. This is a double salt consisting
62 of calcium, magnesium and carbonate ions with the formula $\text{CaMg}(\text{CO}_3)_2$ and can function as a
63 source of CO_2 and magnesium. In the alkaline environment of cement paste, dolomite can undergo
64 a so-called “dedolomitization reaction” [14–16]. In this reaction, dolomite reacts with portlandite
65 to form brucite and calcite. However, Zajac et al. have recently shown that the reaction of dolomite
66 in Portland cement paste results in the formation of a similar phase assemblage to that of
67 limestone addition [17,18]. They further show that the magnesium originating from the dolomite
68 reaction results in the formation of hydrotalcite [17,18] (in the present study given as
69 $\text{Mg}_6\text{Al}_2(\text{OH})_{18}\cdot 3(\text{H}_2\text{O})$, according to [19]), which can lead to an increase in the compressive
70 strength [17,20].

71 The present study focuses on the reaction of dolomite powders in combination with metakaolin
72 used as SCM in Portland cement-based pastes. Metakaolin is used as a model material for SCMs
73 that provide Al and Si. The aim was to determine whether dolomite reacts in these Al-rich systems,
74 and what the reaction products are. We also wanted to understand more about the rate-limiting
75 factors of the dolomite reaction in these systems.

76 We investigated the phase assemblage and microstructure of hydrated cement pastes, in which
77 40%wt of Portland cement was replaced by various combinations of dolomite and metakaolin.
78 The pastes were sealed-cured at 20 °C, 38 °C and 60 °C for up to 360 days. Elevated curing
79 temperatures were applied to accelerate the reaction of the dolomite and to be able to study its
80 reaction products. The following techniques were used to characterize phase assemblage and
81 microstructure: XRD, TGA, SEM-EDS and MIP. The phase assemblage was compared with those of
82 a plain Portland cement paste and systems containing limestone so that we could identify the
83 effect of dolomite on the phase assemblage. In addition, a paste with a high w/b ratio, and well-
84 hydrated samples exposed to additional portlandite and or water were prepared to study the
85 impact of increased porosity or additional portlandite on the dolomite reaction.

86

87 **2 EXPERIMENTAL**

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89 **2.1 MATERIALS**

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91 The mixes investigated in this study were prepared using a Portland cement clinker (C) supplied
92 by Norcem AS, natural dolomite (D) and limestone (L) supplied by Miljøkalk AS, and laboratory-
93 grade metakaolin (M) supplied by Imerys (Metastar501) and gypsum supplied by Merck. The
94 cement clinker was ground in a laboratory ball mill until a Blaine surface area of approx. 400
95 m²/kg was achieved. The other materials were used as received. The chemical and mineralogical
96 composition of all materials were investigated with XRF (Table 1) and QXRD (Table 2 and Table
97 3). It should be noted that the Portland cement clinker used shows a relatively high alkali content.
98 The particle size distributions of the materials used were determined from the average of 3
99 independent measurements with laser diffraction (Malvern Mastersizer 2000E) (Figure 1).

100 The experimental matrix is given in Table 4. Taking into account the sulphate content of the
101 clinker as determined by XRF, laboratory grade gypsum was added when preparing the mixes to
102 achieve a sulphate content of 2.5%wt per gram of binder in all mixes. This sulphate level was
103 determined by isothermal calorimetry at 20 °C on the sample with the highest metakaolin content.
104 At a sulphate content of 2.5%wt, the secondary aluminate peak appeared after the silicate peak
105 and thereby the system is assumed to contain sufficient amounts of sulphates. The sulphatation
106 was, however, not checked at the elevated curing temperatures (38 °C and 60 °C). It should be
107 noted that the values in Table 4 do not reflect the actual amounts in the mixes, as they do not
108 include the amount of gypsum added to the system. A pure Portland cement (100C = ground
109 Portland cement clinker + gypsum) was used as a reference. In the other mixes, 40%wt of the
110 Portland cement clinker was replaced with metakaolin and or dolomite, or limestone. Paste
111 samples were prepared in the laboratory at 20 °C with a w/b ratio of 0.45 for all mixes, and in the
112 case of 60C20D20M, an additional paste sample was also prepared with a w/b ratio of 0.93. The

113 pastes were mixed in a BRAUN MR550CA high-shear mixer (speed 6) and cast in 12 ml plastic
114 tubes (mixing procedure: mixing for 30 seconds, waiting for 5 minutes, mixing for 1 minute again).
115 The sealed tubes were cured at 100% RH in a sealed box at 20 °C, 38 °C and 60 °C for up to 360
116 days and up to 400 days in the case of the samples prepared with the various w/b ratio samples.
117 The curing at elevated temperatures (especially 60 °C) was used to accelerate the reaction of
118 dolomite and to be able to study its reaction products.
119 Additionally, two tubes of the samples with the highest metakaolin content (20%wt) and the w/b
120 ratio of 0.45 were exposed to additional portlandite and or water after a curing time of 1 year and
121 9 months (approx. 630 d) at 60 °C. We crushed the samples to a particle size between 1 mm and
122 0.25 mm and filled them in 45 ml centrifuge tubes. To both samples 30%wt of deionized water
123 and to one of them additional 30%wt of portlandite were added. These samples were stored again
124 sealed at 60 °C at 100% RH and investigated after 28 and 90days.

125

126 **2.2 METHODS**

127

128 Prior to the investigation of the phase assemblage of the cement pastes using thermogravimetric
129 analysis (TGA) and X-ray diffraction (XRD), the hydration was stopped by double solvent
130 exchange after 28, 90, 208 and 360 days of curing. The samples prepared with the different w/b
131 ratio were stopped after 400 days of curing. For this, a 6 mm thick slice (diameter: 23 mm) was
132 cut off each cured cement paste sample. This slice was crushed in a porcelain mortar until the
133 whole sample had passed through a 1 mm sieve. The coarsely crushed cement paste was then
134 immersed in 50 ml isopropanol, shaken for 30 seconds, and left to rest for 5 min before the
135 isopropanol was decanted. This isopropanol treatment was performed twice before the sample
136 was transferred to a filtration unit where the isopropanol was filtrated out and the paste was
137 immersed in 10 ml petroleum ether. After 30 seconds of stirring, the suspension was left to rest
138 for 5 minutes. The sample was then vacuum-filtrated and subsequently dried. The 28-day-old
139 samples were dried for 15 min in an aerated oven at 40 °C. All other samples were dried overnight

140 in a desiccator under a slight vacuum (-0.2 bar) applied using a water pump. After drying, the
141 samples were crushed in a porcelain mortar until the whole sample passed through a 63 μm sieve.
142 All samples were stored in a desiccator over silica gel and soda lime until measurement. The well-
143 hydrated samples exposed to additional water and or portlandite were treated similarly with the
144 difference that they have already been crushed at exposure and, due to the additional water, the
145 amounts of isopropanol and petroleum ether were increased to 100 ml and 20 ml respectively.
146 These samples were stopped after 28 and 90 days of exposure and directly prior exposure.
147 For the TGA measurements, the resulting dry powders were poured into 600 μl corundum
148 crucibles and stored in a sample changer until measurement. The weight loss was measured from
149 40 $^{\circ}\text{C}$ to 900 $^{\circ}\text{C}$ with a heating rate of 10 $^{\circ}\text{C}/\text{min}$ in Mettler Toledo TGA/SDTA851 and TGA/DSC3+
150 devices. During the measurement, the measurement cell was purged with 50 ml/min of nitrogen
151 gas. Both devices used a similar measurement principle and were operated with the same
152 parameters. We, therefore, assume no significant differences between the results obtained from
153 the two devices.

154 The derivate curves of the TG signal, the DTG curves, were used to detect phase changes. The DTG
155 curves can be divided into several sections, in which the decomposition of specific phases can be
156 detected as a weight loss. The first peak at around 100 $^{\circ}\text{C}$ is related to the ettringite decomposition
157 and the beginning of the dehydroxylation of the C-S-H phase. C-S-H decomposes gradually
158 between 40 $^{\circ}\text{C}$ and 600 $^{\circ}\text{C}$ [21] and appears as a polynomial baseline under the other peaks. The
159 region between approx. 150 $^{\circ}\text{C}$ and 400 $^{\circ}\text{C}$ represents the stepwise dehydroxylation of the AFm
160 phases (monosulphate, hemi- and monocarbonate) and other lamellar phases, including
161 hydrotalcite (Ht). Consequently, there are several mass loss peaks visible in this temperature
162 range. Hydrotalcite shows two mass loss events, the first at approx. 270 $^{\circ}\text{C}$ and the second at
163 around 400 $^{\circ}\text{C}$ [21]. The subsequent sharp peak between approx. 400 $^{\circ}\text{C}$ and 550 $^{\circ}\text{C}$ is related to
164 the decomposition of portlandite (CH). Above 550 $^{\circ}\text{C}$ carbonates decompose by emitting CO_2 .

165 TGA was used to quantify the mass losses of various hydration phases, such as portlandite and
166 hydrotalcite. The mass loss of a phase in a specific temperature interval can be determined by

167 integration of the derivative curve with subtraction of the background by a linear baseline as
168 described by Lothenbach et al. [21]. The calculated weight losses were then normalized to the
169 sample weight at approx. 550 °C, which was assumed to be the dry binder weight, and remained
170 constant during the cement hydration, or to the original percentage of clinker in the sample
171 (100%wt or 60%wt).

172 For the XRD analyses, the powder was loaded into the sample holders by means of front-loading
173 and stored in a sample changer until measurement. For the measurements, we used a D8 Focus
174 diffractometer from Bruker with a Bragg-Brentano θ - 2θ geometry and a goniometer radius of
175 200.5 mm. The samples were measured between 5 ° 2θ and 55 ° 2θ with a step size of 0.01 ° 2θ and
176 a sampling time of 0.5 seconds per step. We used Cu-K α radiation with a wavelength of approx.
177 1.54 Å as the X-ray source. The XRD plots were qualitatively evaluated with the DIFFRAC.EVA V4.0
178 software from Bruker by using the PDF4+ database from ICDD and the COD database. For the
179 quantification of the dolomite content in the samples (QXRD), the TOPAS 5 software from Bruker
180 was used combined with the G-Factor Method [22, 23] was applied by using an external quartzite
181 standard. The quartzite was calibrated against a silicon powder from NIST (Standard Reference
182 Material 6640d). For the C-S-H phase, the model described by Bergold et al. was used in the
183 refinement [24].

184 To investigate the phase assemblage and microstructure of the samples with scanning electron
185 microscopy (SEM), a 3 mm slice was cut off the cured cement paste and immersed in isopropanol
186 for min. 1 week. Then the slice was dried, cast in epoxy and polished. In case of the well-hydrated
187 samples exposed to additional water and portlandite, parts of the samples from the double solvent
188 exchange as described above before grinding were used for polishing. We analysed polished and
189 carbon-coated sections of selected paste samples with the SEM. Elemental mapping and point
190 analyses were carried out using a Hitachi S-3400N microscope equipped with an energy
191 dispersive spectrometer (EDS) from Oxford Instruments.

192 To study the threshold pore entry diameter and total porosity of the paste samples with mercury
193 intrusion porosimetry (MIP), a 7 mm slice was cut off the cured cement paste and coarsely

194 crushed in a porcelain mortar. The crushed samples were then immersed in isopropanol for at
195 least 24 h and then dried in an aerated oven overnight at 40 °C to remove the isopropanol. For the
196 MIP measurements, we used a Pascal 140/440 porosimeter from ThermoFisher Scientific and
197 defined a contact angle of 140 °. The first intrusion curve reported from the measurements was
198 used to determine the threshold pore entry diameter and the porosity of the samples. The
199 threshold pore entry diameter was defined as the intersection of two tangents on the intrusion
200 curve as described in [25]. The porosity as percentage of the sample volume equals the total
201 porosity measurable with MIP and is determined by the maximum of the intrusion curve.

202

203 **3 RESULTS**

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205 **3.1 Quantification of the dolomite reaction**

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207 **3.1.1 QXRD**

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209 QXRD was used to quantify the amount of unreacted dolomite in the hydrated cement paste
210 samples. However, the hardness of the dolomite used in this study resulted in coarse dolomite
211 particles in the powder even after grinding and therefore in the vertical exaggeration of one reflex
212 of dolomite in the XRD-patterns (commonly called spottiness-effect). This effect resulted in a
213 relatively large error in the quantification (estimated at approx. 10%wt). The results are given as
214 the amount of dolomite determined by QXRD compared to the original amount of dolomite added.
215 The theoretical amount of dolomite added equals the amounts given in Table 4. However, this is
216 not the actual amount of dolomite added, because we also added gypsum to all the samples and
217 the dolomite used contains approx. 90%wt of dolomite (Table 2).

218

219 After 360 days, the amount of dolomite was determined for all the mixes investigated. Figure 2
220 shows that the samples cured at 20 °C and 38 °C still contained a high amount of dolomite, close

221 to the actual amount of dolomite added, independently of the sample composition. However, when
222 cured at 60 °C, the samples with low metakaolin contents, especially the sample 60C40D, showed
223 a notable decrease in the amount of dolomite. For samples with a higher metakaolin content, the
224 determined amounts of dolomite were again close to the actual amount of dolomite added.
225 Figure 3 shows the amount of dolomite in sample 60C40D over a period of one year. Samples
226 cured at 20 °C and 38 °C showed a similar amount of dolomite, which was close to the actual
227 amount of dolomite added. There is only a slightly increasing trend in the degree of reaction with
228 curing time. For the samples cured at 60 °C, however, the amount of dolomite decreases
229 significantly between 90 and 208 days and stayed rather constant afterwards.
230 Based on the QXRD results, the degree of dolomite reaction, therefore, seems to depend on the
231 curing temperature, the curing time and the metakaolin addition.

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233 **3.1.2 SEM-EDS**

234

235 Figure 4 and Figure 5 show the BSE images for the 60C40D and 60C35D5M samples, which
236 showed the lowest amount of dolomite and therefore the highest degree of dolomite reaction with
237 QXRD, cured at 60 °C for 1 year. Figure 6 shows the BSE image for the 60C20C20M sample, which
238 showed limited dolomite reaction, cured under the same conditions. Alongside the BSE images,
239 the figures show the corresponding elemental maps of magnesium, aluminium, oxygen, calcium
240 and silicon for each sample.

241 The BSE images show large, uniformly grey particles, which the elemental maps show are rich in
242 Mg and Ca, but poor in Al and Si. These particles are unreacted dolomite grains (up to approx.
243 70 µm in length). In the samples where dolomite has reacted, a clear reaction rim filled with Mg-
244 rich reaction products can be observed around these grains (Figure 4 and Figure 5). The shape of
245 the original dolomite grains is still visible in the BSE images, because it is marked with a thin layer
246 of C-S-H precipitated around them, presumably at early ages, which then persisted even after the
247 dolomite started to react at later ages. In addition to the larger, partially reacted dolomite grains

248 in these samples, small fully-reacted dolomite grains can also be observed (some indicated with
249 arrows). They are completely filled with the Mg-rich reaction product and are surrounded by the
250 C-S-H rim, which indicates the original dolomite grain boundary.

251 Figure 7 gives the BSE images for the samples 60C20D20M, which showed limited dolomite
252 reaction under QXRD examination. Here we prepared two samples with the same binder, one with
253 a w/b ratio of 0.45 and the other with a higher w/b ratio, namely 0.93. We did this to investigate
254 the impact of increased porosity on the reaction of dolomite. As expected, the sample with the
255 high w/b ratio shows considerably more porous microstructure than the sample with the lower
256 w/b ratio (0.45). Moreover, significantly fewer unreacted clinker grains are visible in the high w/b
257 sample than in the low w/b sample. However, the dolomite grains do not show any significant
258 reaction in either case. So we conclude that the higher w/b ratio enhanced the clinker reaction,
259 but was not able to enhance the dolomite reaction.

260

261 **3.2 Phase assemblage**

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263 **3.2.1 SEM-EDS**

264

265 The elemental maps for the 60C40D and 60C35D5M samples cured at 60 °C for 1 year (see Figure
266 4 and Figure 5) show that the reaction rims inside the former dolomite grains are rich in
267 magnesium, aluminium and oxygen, but that they do not contain calcium or silicon. This indicates
268 that the product of the dolomite reaction contains magnesium and aluminium and its increased
269 oxygen level indicates that it is a hydrate. Outside the former grain boundaries of the dolomite, no
270 significant amount of magnesium could be detected by elemental mapping. This indicates that the
271 magnesium-containing product of the dolomite reaction formed only within the former grain
272 boundaries, indicating a low mobility of magnesium in the cement matrix, as reported in the
273 literature [1]. Point analyses were performed to further identify the reaction product of the rim
274 inside the former dolomite grains. When the results are plotted as the Mg/Si ration over the Al/Si

275 ratio, the data points describe a linear line for both samples (Figure 8). This indicates that the
276 reaction product has a fixed Mg/Al ratio (the slope of the line). The small amount of Si present in
277 the analysis originates probably from intermixing with other phases in the analysed volume. The
278 reaction product can, therefore, be identified as hydrotalcite ($\text{Mg}_6\text{Al}_2(\text{OH})_{18}\cdot 3(\text{H}_2\text{O})$) [26]. The
279 Mg/Al ratio of the hydrotalcite was approx. 3.2 in sample 60C40D and 2.4 in sample 60C35D5M.
280 In cementitious systems, Mg/Al ratios of approx. 2 are reported [17,26–30]. However, higher
281 Mg/Al ratios are possible as well, as the natural mineral hydrotalcite has a Mg/Al ratio of 3 [31].

282

283 SEM-EDS point analyses also allow us to investigate changes in the C-S-H, e.g. aluminium uptake
284 and changes in the Si/Ca ratio. The dot plots for C-S-H phase of the samples 60C40D, 60C35D5M
285 and 60C20D20M cured at 60 °C are given in Figure 9. The Al/Si ratio can be determined from the
286 slope of the lower lines, which are framing the C-S-H data clouds. While the addition of 5%wt
287 metakaolin does not seem to change the C-S-H composition significantly (60C35D5M, Al/Si: 0.04)
288 compared to the 60C40D sample (60C40D, Al/Si = 0), the addition of 20%wt metakaolin shifted
289 the C-S-H composition to a considerably higher aluminium content (60C20D20M, Al/Si: 0.32).
290 Simultaneously, the Si/Ca ratio is increased as well in this sample.

291

292 **3.2.2 XRD**

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294 The XRD-patterns in the range of 8 to 12 °2θ for samples cured for 90, 208 and 360 days at 20 °C
295 are given in Figure 10, for samples cured at 38 °C in Figure 11, and for samples cured at 60 °C in
296 Figure 12. The figures indicate the main reflections of ettringite (Et, 9.1 °2θ), monosulphate-12H
297 (Ms_{12} , 9.9 °2θ), hemihydrate (Hc, 10.8 °2θ), hydrotalcite (Ht, 11.4 °2θ) and monocarbonate (Mc,
298 11.7 °2θ). Strätlingite (7.1 °2θ) was not detected in any of the samples, irrespective of curing
299 temperature.

300 For the samples cured at 20 °C, dolomite seems to stabilize the ettringite (Figure 10), because all
301 the dolomite-containing samples show a clear ettringite peak, but the 100C sample, which

302 contains no dolomite, does not. This is in line with the findings from Zajac et al. [17]. For the 100C
303 sample, the phases observed are monosulphate and some hemicarbonates, the latter possibly
304 formed by carbonation during queuing in the diffractometer. In the 60C40D sample,
305 monocarbonate is the main AFm phase. The sample containing 5%wt metakaolin (60C35D5M)
306 has a phase composition similar to the 60C40D sample. When the metakaolin content is increased,
307 e.g. from 5 to 20% (60C35D5M to 60C20D20M), small peaks of hemicarbonates are detected as
308 well. This can be explained by the increased aluminium provided by the metakaolin, which lowers
309 the $\text{CO}_2/\text{Al}_2\text{O}_3$ ratio and leads to the formation of hemicarbonates [32]. However, the
310 hemicarbonates peaks decrease over time due to the slow reaction of the dolomite, which slowly
311 increases the $\text{CO}_2/\text{Al}_2\text{O}_3$ ratio in the sample which stabilizes monocarbonate over hemicarbonates.
312 Additionally, there is a decrease in the intensity of the AFm peaks in the XRD-pattern. This could
313 be due to increased uptake of Al by C-S-H with increased metakaolin content, as shown by SEM-
314 EDS in Figure 9, which would leave less aluminium to form AFm phases. Hydrotalcite showed a
315 shoulder on the low-angle side of the monocarbonate peaks at 20 °C but no clear peaks.
316 The samples cured at 38 °C (Figure 11) show similar phase assemblages to those of the samples
317 cured at 20 °C. In the 60C40D sample, the monocarbonate observed is partially replaced by
318 hydrotalcite over time. The sample containing 5%wt metakaolin (60C35D5M) has a similar phase
319 composition to that of the 60C40D sample, but the hydrotalcite appears not as a clear peak, but
320 rather as a shoulder on the monocarbonate peak.
321 At 60 °C (Figure 12), the stable phases differ significantly from those at 20 °C and 38 °C. The
322 ettringite peak is not observed because this phase is not stable at 60 °C [33]. The 100C sample
323 does show small peaks of hemicarbonates. The main diffraction peak observed for the 60C40D
324 sample is hydrotalcite, already present from 90 days of hydration on, and no monocarbonate is
325 detected. When 5%wt of metakaolin is added (60C35D5M), hemicarbonates are observed in
326 addition to the hydrotalcite after 90 days, but it transforms to hydrotalcite at later ages. With
327 metakaolin additions of 10%wt, monosulphate is detected together with monocarbonate, and
328 only a small hydrotalcite shoulder is observed after 1 year of curing. At higher metakaolin levels,

329 the monocarbonate is replaced by hemicarbonates. Again, this could be due to the additional
330 metakaolin, which provides more Al_2O_3 and reduces the $\text{CO}_2/\text{Al}_2\text{O}_3$ ratio [32].

331 Clear peaks of hydrotalcite can only be observed with XRD in samples containing dolomite with a
332 relatively low metakaolin content ($<10\%$ wt) cured at elevated temperatures (38°C and especially
333 at 60°C). When metakaolin additions are high, or when the samples are cured at 20°C , hydrotalcite
334 was not observable as a peak but as a small shoulder at the low-angle side of the monocarbonate
335 peak.

336

337 The hydrotalcite observed using XRD originates from the dolomite reaction because analogue
338 samples containing limestone do not show hydrotalcite-related reflections even after curing for 1
339 year at 60°C (Figure 13). The XRD plots in Figure 13 also do not show clear peaks of carbonate
340 AFm phases. This can either be explained by the fact that they are not stable, or that they are less
341 crystalline and therefore X-ray amorphous at such a high curing temperature.

342

343 Figure 14 shows the XRD-patterns for 60C20D20M samples prepared with the two different w/b
344 ratios. No hydrotalcite peaks were detected after 400 days of curing. This is in line with the
345 observations from SEM-EDS that a high w/b ratio was not able to enhance the dolomite reaction
346 of the 60C20D20M sample.

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348 **3.2.3 TGA**

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350 Figure 15 shows the DTG curves for samples 60C40D and 60C40L after curing for 1 year at 60°C .
351 Based on the XRD and SEM analysis, we would expect a considerable amount of hydrotalcite to
352 have formed in 60C40D, but not in 60C40L. Hydrotalcite is reported to decompose in two steps
353 during a TGA measurement. The first weight loss related to hydrotalcite is observed at approx.
354 220°C and the second weight loss at approx. 380°C [21]. These two steps appear as peaks in the
355 DTG curves for the sample 60C40D and are shown in Figure 15. Moreover, it should be noted that

356 small peaks of AFm phases are detected in the DTG curves of the samples 100C and 60C40L. This
357 confirms the findings with XRD, which indicated that the AFm phases in the samples cured at 60 °C
358 are probably X-ray amorphous.

359 In Figure 16 the differences between the samples containing dolomite and limestone with regard
360 to the summed hydrotalcite weight losses of both of its weight loss steps are plotted for the
361 various compositions and curing temperatures. This is done to eliminate the potential impact
362 from Portland cement. The Portland cement used contains significant amounts of magnesium
363 (Table 1) and might also cause a weight loss in the temperature range of hydrotalcite
364 decomposition, as shown in Figure 15 for the sample 60C40L. So, positive values indicate a higher
365 weight loss in the hydrotalcite temperature range for samples containing dolomite than for those
366 containing limestone. It should be noted that, due to the probability of overlapping weight losses
367 in this temperature interval, the signal recorded might not exclusively originate from the
368 decomposition of hydrotalcite. Taking into account an estimated error of 0.1%wt, it seems that
369 considerable hydrotalcite formation starts quite late. As Figure 16 shows, there are no significant
370 differences in the weight losses after 28 days and only a small difference after 90 days between
371 samples containing dolomite and limestone. Only in the 208-day-old and the 360-day-old samples
372 are significantly higher hydrotalcite weight losses detected for samples containing dolomite than
373 for those containing limestone when cured at 60 °C. The highest value is shown by the sample
374 60C40D after 360 days of hydration. With the addition of metakaolin, the hydrotalcite weight
375 losses decrease. These observed trends are in good agreement with the XRD results, where the
376 formation of hydrotalcite can be detected after 90 days and especially after 208 and 360 days in
377 samples with low metakaolin additions (< 10%wt), with the sample 60C40D showing the highest
378 hydrotalcite peaks when cured at elevated temperatures (especially 60 °C).

379 Portlandite dehydrates between approx. 400 °C and 550 °C, and a clear weight loss can be
380 observed in the DTG signal (Figure 15). Figure 17 shows the portlandite content of sample mixes
381 containing various amounts of dolomite or limestone at the temperatures investigated. A clear
382 reduction can be seen in portlandite with increasing metakaolin content. This is due to the

383 pozzolanic reaction of metakaolin. This reaction is enhanced when the curing temperature is
384 increased. It should be noted that the portlandite content for the samples containing 15%wt and
385 20%wt of metakaolin is similar. This indicates that at metakaolin contents >10%wt, its reaction
386 is limited.

387 Figure 18 plots the differences in the portlandite content between the samples containing
388 dolomite and limestone (estimated error: 0.5%wt). Therefore, negative values indicate a lower
389 portlandite content in samples containing dolomite than in those containing limestone. These
390 differences are only visible for compositions where there are also differences in the hydrotalcite
391 weight loss. The sample 60C40D shows significantly lower portlandite content than its limestone
392 equivalent (60C40L) after 208 and 360 days, when cured at 60 °C. This observation indicates that
393 portlandite is consumed during the formation of hydrotalcite from dolomite. This is in-line with
394 descriptions of the reaction of dolomite in the literature [14,15,16], where the dolomite reaction
395 requires the presence of portlandite.

396 Figure 19 shows the portlandite content of the samples 60C20D20M with a w/b ratio of 0.45 and
397 0.93, both cured at 20 °C, 38 °C and 60 °C for 400 days. In samples with the low w/b ratio, the
398 portlandite content is consumed almost completely when cured at 60 °C, but at lower curing
399 temperatures more portlandite remains. In the samples with the high w/b ratio, the portlandite
400 is completely consumed at all curing temperatures. These results indicate that in the range tested,
401 increasing the w/b ratio enhances the metakaolin reaction.

402

403 **3.3 MIP**

404

405 The threshold pore entry diameter and the porosity were estimated from the intrusion curves of
406 the first intrusion cycle of the mercury and are plotted for the various compositions in Figure 20.
407 Figure 20 a) shows the results for samples cured at 38 °C for 208 days, Figure 20 b) for samples
408 cured at 60 °C for 208 days and in Figure 20 c) for the samples cured at 60 °C for 360 days. The

409 precision of the MIP results was estimated to 20% for the threshold diameter and 1.5% for the
410 porosity according to [34].

411 Generally, the samples containing dolomite or limestone show similar results. In Figure 20 b)
412 however, there is a significant difference between the samples containing either 5%wt of dolomite
413 or limestone, which were cured at 60 °C. In order to elucidate whether this is an effect of hydration
414 or sample preparation, a set of samples cured as well at 60 °C but for 360 days was investigated
415 as well (Figure 20 c). After 360 days the samples containing dolomite or limestone show again
416 very similar results. This indicates that the differences observed in Figure 20 b) are due to sample
417 preparation effects.

418 If we compare the two curing temperatures (Figure 20 a & b), a coarsening effect can be observed
419 at the higher curing temperature. This effect is related to the transformation of ettringite to
420 monosulphate above 48 °C and a densification of the C-S-H phase at elevated temperatures, both
421 resulting in a coarsening of the pore structure [33]. Moreover, an increase in the porosity and the
422 threshold diameter can be observed if we compare the reference 100C with the sample containing
423 40%wt of dolomite or limestone at both temperatures. This is in agreement with previous findings
424 [20]. This coarsening effect of added carbonates can be counteracted by the addition of
425 metakaolin, which reduces the threshold diameter as described in the literature [9]. At high
426 metakaolin additions, the threshold pore entry diameter decreases down to values of approx.
427 0.04 µm for both curing temperatures. The addition of 5%wt of metakaolin decreases also the
428 porosity compared to the samples only containing dolomite or limestone. However, with additions
429 greater than 5%wt metakaolin, there is no further reduction in the porosity.

430

431 Figure 21 shows the threshold pore entry diameter and porosity of the samples prepared with the
432 different w/b ratio of 0.93 or 0.45 cured at 38 °C or 60 °C for 400 days. The threshold diameter
433 and the porosity are increased for the samples prepared with the high w/b ratio compared to the
434 samples with the low w/b ratio. This coarsening effect of the increased w/b ratio is expected and
435 in good agreement with the microstructural changes between samples with a high or low w/b

436 ratio as observed in the BSE images (Figure 7). While the threshold diameter is increased for the
437 samples cured at 60 °C compared to 38 °C for both w/b ratios, the porosity is rather similar for
438 the samples cured at 60 °C or 38 °C for both w/b ratios and no considerable increase in porosity
439 can be observed.

440

441

442 **3.4 Well-hydrated samples exposed to additional water and or portlandite**

443

444 Figure 22 shows the DTG curves of the sample 60C20D20M+H₂O, which was exposed to 30%wt
445 additional water after 28 and 90 days of exposure at 60 °C, compared to the same sample prior
446 exposure. The sample 60C20D20M+H₂O was used as a reference to the sample 60C20D20M+CH,
447 which was exposed to additional 30%wt water and 30%wt portlandite. As the reference samples
448 after 28 and 90 days of exposure show almost completely the same curves as the sample prior
449 exposure, we assume that they were stable during the whole exposure time. Consequently, any
450 phase changes observed in the sample 60C20D20M+CH are due to the addition of portlandite and
451 not due to the addition of water.

452 Figure 23 shows the DTG curves of the sample 60C20D20M+CH cured for 28 and 90 days at 60 °C
453 compared to the sample prior exposure. At low temperatures (40–300 °C), the samples exposed
454 to portlandite are shifted upwards compared to the sample prior exposure and show as well a
455 decreased peak in the temperature range of the dolomite decomposition. This can be explained
456 by the dilution effect of the hydrates and the dolomite in the samples where 30%wt of portlandite
457 was added. Because of this addition, these samples also show a clear peak in the temperature
458 interval of the portlandite decomposition. The two peaks related to the decomposition of
459 hydrotalcite increased with exposure time in the samples exposed to portlandite compared to the
460 sample prior exposure. This indicates the formation of hydrotalcite in these samples. The samples
461 exposed to portlandite also show an additional peak in the AFm-temperature region, probably
462 due to the formation of monosulphate. This peak seems to decrease with the exposure time. The

463 increase in the hydrotalcite peak simultaneously with the decrease in the monosulphate peak
464 indicates that due to the reaction of dolomite over time in the samples exposed to portlandite, the
465 monosulphate transforms to hydrotalcite as it is thermodynamically more stable [35].

466 Figure 24 shows the XRD-patterns of the samples 60C20D20M+CH and 60C20D20M+H₂O after 90
467 days of exposure compared to the sample prior exposure. While the sample prior exposure and
468 the reference sample show the same phase assemblage, the sample 60C20D20M+CH shows
469 additional peaks. In this sample, besides the portlandite peak due to the exposure, clear peaks of
470 monosulphate and hydrotalcite are visible. These results are in good correlation with the
471 observations made in the DTG curves of these samples.

472

473 **4 DISCUSSION**

474

475 The results of this study show that dolomite can react significantly when added to Portland
476 cement, depending on the curing age, temperature and the metakaolin content of the samples. The
477 reaction products and the limitations of this reaction are discussed in the following.

478

479 **4.1 What are the products of the dolomite reaction?**

480

481 The elemental maps obtained from SEM-EDS analysis show that when dolomite reacts magnesium
482 does not move outside the former grain boundary of the dolomite, probably due to a limited
483 mobility of magnesium in high pH environments [1]. So, the magnesium-containing reaction
484 products of dolomite form a reaction rim within the former dolomite grain boundaries. The
485 products in the reaction rim contain magnesium, aluminium and show a higher oxygen content.
486 The high oxygen content indicates that the reaction products are hydrates. No silicon can be found
487 in the rims and the calcium content is less than in the original dolomite grain. The lack of silicon
488 in the rims excludes M-S-H as a reaction product.

489 EDS point analysis of the reaction rims indicates the presence of hydrotalcite, with a Mg/Al ratio
490 of approx. 2.4–3.2 depending on the metakaolin content of the cement. The formation of
491 hydrotalcite was confirmed by XRD, where peaks of hydrotalcite could be found in samples where
492 significant amounts of dolomite have reacted. XRD did not detect clear peaks of brucite in any
493 samples. TGA analysis seems to indicate that the weight losses in the temperature range of
494 hydrotalcite correlate to the amount of dolomite reacted.

495 We conclude that hydrotalcite is the only magnesium-containing product of the dolomite reaction
496 and it forms in reaction rims within the former dolomite grains in samples cured at elevated
497 temperatures (60 °C).

498 TGA and XRD also showed that the addition of dolomite results in the formation of carbonate AFm
499 phases and therefore ettringite stabilization. This means that the dolomite delivers carbonates to
500 the system, which can form carbonate AFm phases. This is only observable in samples where
501 ettringite is stable, hence in samples cured at 20 °C or 38 °C. Calcite might also be formed due to
502 the reaction of dolomite. XRD detected calcite in the samples, but we could not confirm whether
503 this calcite was formed during the dedolomitization reaction. This is because the dolomite used
504 already contains approx. 6%wt of calcite, which might partially also participate in the formation
505 of carbonate AFm phases. SEM-EDS cannot differentiate between portlandite and calcite, so the
506 potential replacement of portlandite with calcite during the dolomite reaction cannot be
507 identified.

508 These findings are in agreement with Zajac et al., who showed that in the presence of other ions
509 dolomite reaction results in the formation of hydrotalcite and carbonate AFm phases, which lead
510 to the stabilization of ettringite [17].

511

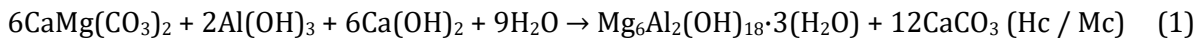
512 **4.2 What limits the hydrotalcite formation?**

513

514 As described above, hydrotalcite forms slowly over time in samples containing dolomite which
515 have been cured at 60 °C, if they contain less than 10%wt of metakaolin.

516 To elucidate the possible limitations of the hydrotalcite formation, Equation (1) gives an idealized
517 reaction based on the observations in this study. In this reaction, dolomite reacts with a source of
518 aluminium, here given as $\text{Al}(\text{OH})_3$, and portlandite to form hydrotalcite and calcite. This reaction
519 inside the reaction rims around the dolomite grains is illustrated in Figure 25. The calcite formed
520 might then precipitate in the matrix or in finely intermixed with the C-S-H in the rim, which
521 indicates the original grain boundary of the dolomite. It might also partially contribute to the
522 formation of hemi- or monocarbonate. The hydrotalcite formation could be chemically limited by
523 any of the reactants given in Equation (1). Taking into account the refinement of the pore structure
524 due to the metakaolin addition, there might be a physical limitation as well.

525



526 In the following, we discuss the various possible limitations in detail:

527

528 **4.2.1 Pore space as a physical limitation**

529

530 The MIP results for the samples cured at 60 °C (Figure 20) show that while the porosity is not
531 decreased, the threshold diameter is significantly refined with the addition of metakaolin. The
532 threshold pore diameter, which is an important parameter for transport in the pore system [36],
533 is reduced from 0.4 μm , for the samples only containing dolomite but no metakaolin, to 0.04 μm ,
534 for the samples with the highest metakaolin content. Durdzinski, 2016 reported the refinement of
535 the pore structure due to the additions of fine and reactive SCMs to be the major factor for a
536 decreased degree of reaction. This observation was explained with the restricted transport and
537 the hindered crystallization in fine pores compared to coarse pores [34]. This is in agreement with
538 the TGA results, which show a limited further reaction of metakaolin for additions of metakaolin
539 higher than 10%wt (see Figure 17). This possible hindrance of further reactions in dense systems
540 has been reported for metakaolin or other SCMs and UHPC previously [34,37,38].

541

542 The MIP results also show that curing at 60 °C considerably coarsens the pore structure. So, in
543 addition to the higher reactivity of dolomite at 60 °C than at 38 °C reported before [14–17], the
544 coarser pore structure might also facilitate the ongoing reactions due to the facilitation of
545 transport and crystallization.

546 To elucidate the effect of the pore space, we prepared samples with high amounts of metakaolin
547 (20%wt) and a high w/b ratio (0.93). The XRD plots (Figure 14) show that, despite the coarse
548 pore structure of these samples, no hydrotalcite was formed, and the BSE images also showed that
549 the dolomite did not react (Figure 7). We, therefore, conclude that the limited pore space in
550 metakaolin-rich samples does not act as a major limitation on the dolomite reaction in the
551 investigated systems.

552

553 **4.2.2 Aluminium availability**

554

555 Figure 25 shows a schematic illustration of the hydrotalcite formation inside the former dolomite
556 grain boundary including the necessary transport of ions. Aluminium is needed for hydrotalcite
557 to form, see Equation (1) and Figure 25. The addition of metakaolin to the system provides
558 additional aluminium. Besides the formation of hydrotalcite, this additional aluminium can result
559 in the increased formation of AFm phases shown in XRD (see Figure 11), and the uptake of
560 aluminium in the C-S-H shown in SEM-EDS (see Figure 9). Hydrotalcite will win the competition
561 for the aluminium over AFm phases because it is thermodynamically more stable [35], but we
562 cannot reach a conclusion on the competition with the Al-uptake in the C-S-H, due to limited
563 thermodynamic data. When comparing the Al/Si ratios of the C-S-H (Figure 9) to the Al/Si ratios
564 reported in the literature, we can see that for low metakaolin additions (5%wt), our values are
565 somewhat lower, and for high metakaolin additions, they are higher than reported in the literature
566 [12]. Our data does therefore not fit the proposed fitting equation by Dai et al. [12]. This might be
567 explained by the formation of hydrotalcite in the samples containing lower amounts of

568 metakaolin, which was shown to reduce the aluminium content of the C-S-H [17]. If aluminium
569 does act as a limiting factor in the hydrotalcite formation reaction, the addition of aluminium
570 through the addition of metakaolin should increase the amount of hydrotalcite formed. A similar
571 effect of additional aluminium has been reported on the formation of carbonate AFm phases [5–
572 8].

573

574 However, it was shown that with the addition of 5%wt of metakaolin there is no increase in the
575 amount of hydrotalcite formed. Consequently, we can conclude that any possible positive effect
576 from the additional aluminium is unable to compensate for the negative effects of metakaolin
577 addition on hydrotalcite formation. We, therefore, conclude that the availability of aluminium
578 does not act as a major limitation on the dolomite reaction in the investigated systems.

579

580 **4.2.3 Dolomite dissolution**

581

582 Hydrotalcite is a magnesium-containing hydrate. The availability of magnesium and hence the
583 dissolution of dolomite are required for hydrotalcite to form. This is illustrated in Figure 25, where
584 the formation of a reaction rims, in which the magnesium from the dolomite dissolution is
585 indicated. The dissolution of dolomite has been reported to be slow. However, it was shown to be
586 accelerated by increasing the curing temperature from 25 °C to 60 °C [14–17]. This is in good
587 agreement with our results, as we observed an increased reaction degree of dolomite when cured
588 at 60 °C compared to 38 °C or 20 °C. In samples cured at 60 °C with low or no metakaolin content,
589 QXRD showed that significant amounts of dolomite have reacted after 360 days of curing. This
590 also explains the observed hydrotalcite formation only in samples cured at elevated temperatures.
591 However, only samples with low or no metakaolin addition show significant dolomite reaction
592 and hydrotalcite formation when cured at 60 °C. This indicates that the dissolution of dolomite is
593 additionally limited by another factor besides temperature.

594

595 Equation 2 gives portlandite as one of the reactants for the hydrotalcite formation. Even though
596 hydrotalcite ($\text{Mg}_6\text{Al}_2(\text{OH})_{18}\cdot 3(\text{H}_2\text{O})$) does not contain calcium, the formation of hydrotalcite is
597 associated with the consumption of portlandite (Figure 18) to form calcite (as indicated in Figure
598 25), similarly to the reported dedolomitization reaction, where brucite is formed [14–16]. To
599 illustrate the consumption of portlandite due to the reaction of dolomite, we plotted the difference
600 in portlandite weight loss between samples containing dolomite and samples containing
601 limestone in Figure 18. Samples containing dolomite show significantly lower portlandite content
602 than samples containing limestone. This is especially visible at the elevated curing temperatures
603 and low metakaolin additions, which cause significant dolomite reaction (Figure 2) and
604 consequent hydrotalcite formation (Figure 16). The need for portlandite by the reaction of
605 dolomite can be explained by the difference between dolomite, which contains 2 mol CO_3^{2-} per
606 mol dolomite, and calcite, which contains 1 mol of CO_3^{2-} per mol of calcite. Therefore, to make the
607 magnesium available for reaction to hydrotalcite, calcium is needed to compensate for the other
608 carbonate ion from the dolomite, similar as for the dedolomitization reaction in which brucite is
609 formed. The experimental techniques in this study did not allow the differentiation between
610 carbonate-containing hydrotalcite and a carbonate-free hydrotalcite. However, as long as the
611 hydrotalcite contains less than 1 mol of CO_3^{2-} per mol of hydrotalcite [19,39–42], additional
612 calcium is needed in the reaction to compensate for the rest of the carbonate ions. This is also
613 indicated in Figure 25 by the transport of CO_3^{2-} ions out of the reaction rims.

614

615 The portlandite availability varies with the metakaolin content. A higher metakaolin content leads
616 to an increased consumption of portlandite due to its pozzolanic reaction, as TGA shows in Figure
617 17. As portlandite is consumed in the reaction of dolomite and the formation of hydrotalcite, and
618 it is, therefore, a possible limitation to this reaction in samples where most or all of the portlandite
619 has already been consumed in the fast pozzolanic reaction of the metakaolin.

620

621 It was shown from the results of the samples prepared with the various w/b ratios that the higher
622 w/b ratio enhances not only the Portland cement reaction but also the metakaolin reaction. This
623 led to the complete consumption of the portlandite in the system (Figure 19). Because metakaolin
624 addition refines the pore structure and consumes Portlandite at the same time, these two factors
625 are not completely distinguishable. Because of this, a part of the samples 60C20D20M cured at
626 60 °C for 1 year and 9 months, were exposed to 30%wt portlandite and or 30%wt of additional
627 water for up to 90 days. While the samples, which were exposed only to water, did not change
628 during exposure, the samples exposed to water and portlandite showed the formation of
629 hydrotalcite over time. It can be concluded that in samples, which contain high amounts of
630 metakaolin (20%wt) and therefore have a refined pore structure, the formation of hydrotalcite
631 can be observed as long as sufficient portlandite is available. This confirms the lack of Portlandite
632 as a limiting factor due to the correlation between portlandite consumption and hydrotalcite
633 formation, independently of the pore structure. The refined pore structure in the samples
634 containing high amounts of metakaolin might slow down the reactions but is not inhibiting the
635 formation of hydrotalcite.

636

637 **5 CONCLUSION**

638

639 Cement pastes, in which 40%wt of Portland cement clinker was replaced by either dolomite or
640 limestone in combination with 0–20%wt metakaolin, were investigated with regard to their phase
641 assemblage and microstructure. The samples were cured at 20 °C, 38 °C and 60 °C up to 360 days.
642 Additionally, samples with a high w/b ratio (0.93) were prepared and investigated after 400 days
643 of curing and parts of the samples containing 20%wt of metakaolin were exposed to additional
644 portlandite and or water for up to 90 days. The following conclusions are made based on the
645 investigations:

646 Within 360 days, significant amounts of the added dolomite react at low metakaolin additions
647 (<10%) and an elevated curing temperature (60 °C). As a result of the reaction, magnesium and

648 carbonates are supplied to the system. The carbonates originating from the dolomite stabilize
649 ettringite due to the formation of carbonate AFm phases. During the dolomite reaction,
650 portlandite is consumed and hydrotalcite is formed. Hydrotalcite ($\text{Mg}_6\text{Al}_2(\text{OH})_{18}\cdot 3(\text{H}_2\text{O})$) is the
651 only magnesium-containing reaction product from the dolomite reaction. Neither M-S-H nor
652 brucite were detected in any samples investigated.

653 To check whether additional aluminium results in additional hydrotalcite formation, up to 20%wt
654 of metakaolin was added to the system. However, instead of promoting the hydrotalcite formation,
655 the addition of more than 5%wt metakaolin inhibited the dolomite reaction and therefore also the
656 hydrotalcite formation. Thus, aluminium availability is no major limitation for hydrotalcite
657 formation.

658 The addition of metakaolin influenced the system in more ways than solely providing aluminium.
659 It also consumes portlandite during its pozzolanic reaction and simultaneously refines the pore
660 structure. From the results of the samples containing 20%wt metakaolin which were exposed to
661 additional metakaolin and or water, we can conclude that the pore structure refinement is not
662 inhibiting the formation of hydrotalcite, but might slow reactions down. By investigating the
663 samples with the high w/b ratios, the absence of portlandite in high-metakaolin samples could be
664 confirmed to inhibit the reaction of dolomite and the formation of hydrotalcite independently of
665 any impact from the pore structure.

666

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668

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673

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785 **8 TABLES**

786 Table 1: XRF results [%wt] of the clinker, dolomite, limestone, metakaolin and gypsum used.

Oxide	Clinker	Dolomite	Limestone	Metakaolin	Gypsum
SiO ₂	20.6	0.52	0.12	52.18	0.02
Al ₂ O ₃	5.6	0.01	0.06	44.92	0.09
TiO ₂	0.29	0.00	0.00	1.14	0.00
MnO	0.05	0.00	0.00	0.00	0.00
Fe ₂ O ₃	3.12	0.04	0.03	0.62	0.00
CaO	63.26	31.52	55.12	0.12	32.66
MgO	2.66	20.14	0.41	0.04	0.06
K ₂ O	1.23	0.00	0.01	0.18	0.01
Na ₂ O	0.51	0.00	0.00	0.17	0.02
SO ₃	1.37	0.00	0.02	0.14	46.47
P ₂ O ₅	0.09	0.01	0.00	0.07	0.00
LOI	-	46.79	43.57	0.29	20.39
Sum (1050 °C)	98.78	99.03	99.34	99.87	99.72

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788

789 Table 2: Mineral composition of the dolomite, limestone, metakaolin and gypsum, determined by Rietveld analysis
 790 [%wt]. Amounts given in *italics* are below the limits of quantification (1%wt). Due to the low crystallinity of mullite, its
 791 quantification is questionable.

Mineral name	Mineral formula	Dolomite	Limestone	Metakaolin	Gypsum
Calcite	CaCO ₃	5.8	100	-	-
Dolomite	CaMg(CO ₃) ₂	89.9	-	-	-
Gypsum	CaSO ₄ ·2H ₂ O	-	-	-	93.7
Bassanite	CaSO ₄ ·0.5H ₂ O	-	-	-	6.3
Anatase	TiO ₂	-	-	1.2	-
Mullite	Al ₆ Si ₂ O ₁₃	-	-	6.1	-
Muscovite	KAl ₂ Si ₃ AlO ₁₀ (OH) ₂	-	-	0.4	-
Quartz	SiO ₂	0.4	-	0.7	-
Amorphous content	-	3.9	-	91.6	-

792

793 Table 3: Mineralogical composition of the clinker used determined by Rietveld analysis [%wt]. Amounts given in *italics*
 794 are below or close to the limits of detection.

Mineral name	Formula	%wt
Alite	Ca ₃ SiO ₅	59.5
Belite	Ca ₂ SiO ₄	15.3
Aluminate	Ca ₃ Al ₂ O ₆	8.8
Ferrite	Ca ₂ (Al,Fe) ₂ O ₅	10.0
Periclase	MgO	1.5
Free Lime	CaO	0.9
Portlandite	Ca(OH) ₂	1.2
Aphthitalite	K ₃ Na(SO ₄) ₂	2.4
Arcanite	K ₂ SO ₄	0.5

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797 Table 4: Overview of the experimental matrix, including the original composition, addition of water or portlandite, w/b
 798 ratio and curing time of all samples. The times given in brackets represents the time of the samples exposed to
 799 portlandite and or water. The sulphate content per gram of binder was set to 2.5%wt for all mixes [%wt]. It should be
 800 noted that the dolomite contains approx. 90% of dolomite (see Table 2).

Name of the mix	C Portland cement clinker	D Dolomite	L Limestone	M Metakaolin	Add. Water	Add. CH	w/b ratio	Curing time
100C	100	-	-	-	-	-	0.45	28-360 d
60C40D	60	40	-	-	-	-	0.45	28-360 d
60C35D5M	60	35	-	5	-	-	0.45	28-360 d
60C30D10M	60	30	-	10	-	-	0.45	28-360 d
60C25D15M	60	25	-	15	-	-	0.45	28-360 d
60C20D20M	60	20	-	20	-	-	0.45	28-360 d
60C40L	60	-	40	-	-	-	0.45	28-360 d
60C35L5M	60	-	35	5	-	-	0.45	28-360 d
60C30L10M	60	-	30	10	-	-	0.45	28-360 d
60C25L15M	60	-	25	15	-	-	0.45	28-360 d
60C20L20M	60	-	20	20	-	-	0.45	28-360 d
60C20D20M w/b 0.45	60	20	-	20	-	-	0.45	400 d
60C20D20M w/b 0.93	60	20	-	20	-	-	0.93	400 d
60C20D20M + H ₂ O	60	20	-	20	+30%	-	0.45	630 d (+ 28/90 d)
60C20D20M + CH	60	20	-	20	+30%	+30%	0.45	630 d (+ 28/90 d)

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803 9 FIGURES

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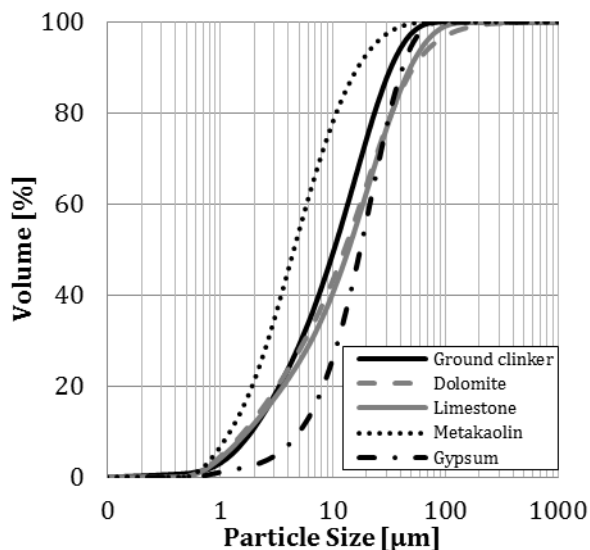


Figure 1: Particle size distributions of the materials used, determined by laser diffraction.

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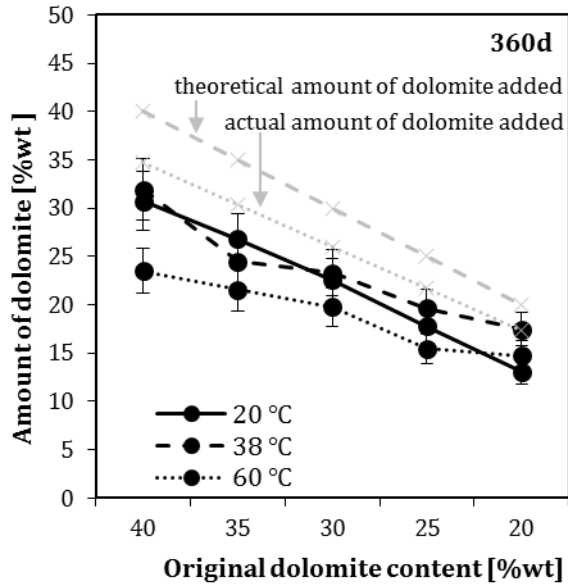


Figure 2: Amount of dolomite for the various compositions, in %wt compared to the original content of dolomite (theoretical and actual) for samples cured at 20 °C, 38 °C and 60 °C after 360 days.

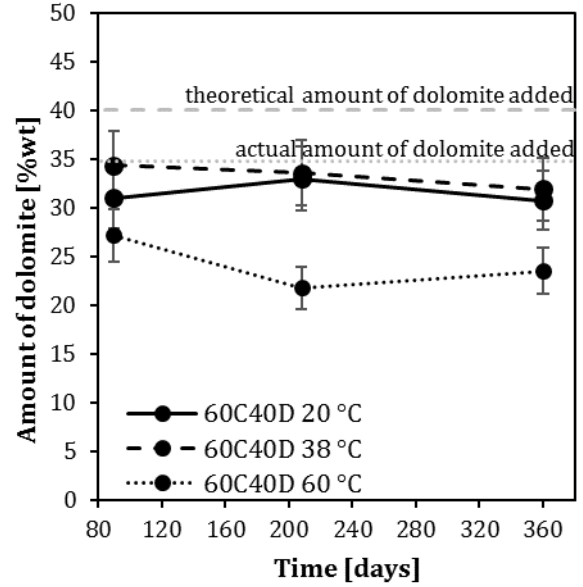


Figure 3: Amount of dolomite reacted over time, in %wt compared to the original content of dolomite (theoretical and actual) for the sample 60C40D cured at 20 °C, 38 °C and 60 °C.

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60C40D 60 °C 360d

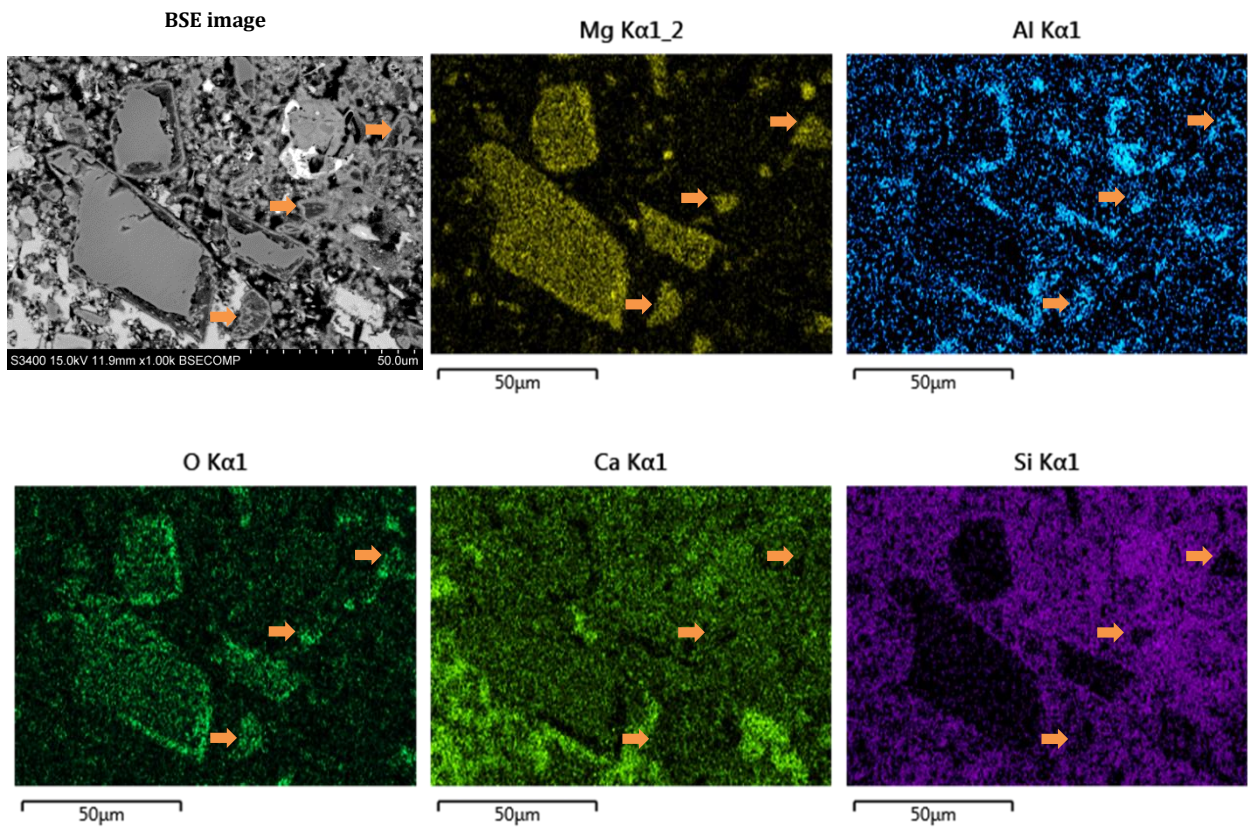


Figure 4: BSE image and elemental maps of magnesium, aluminium, oxygen, calcium and silicon for the sample 60C40D cured at 60 °C for 360 days. The arrows in the BSE image indicate small fully-reacted dolomite grains.

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60C35D5M 60 °C 360d

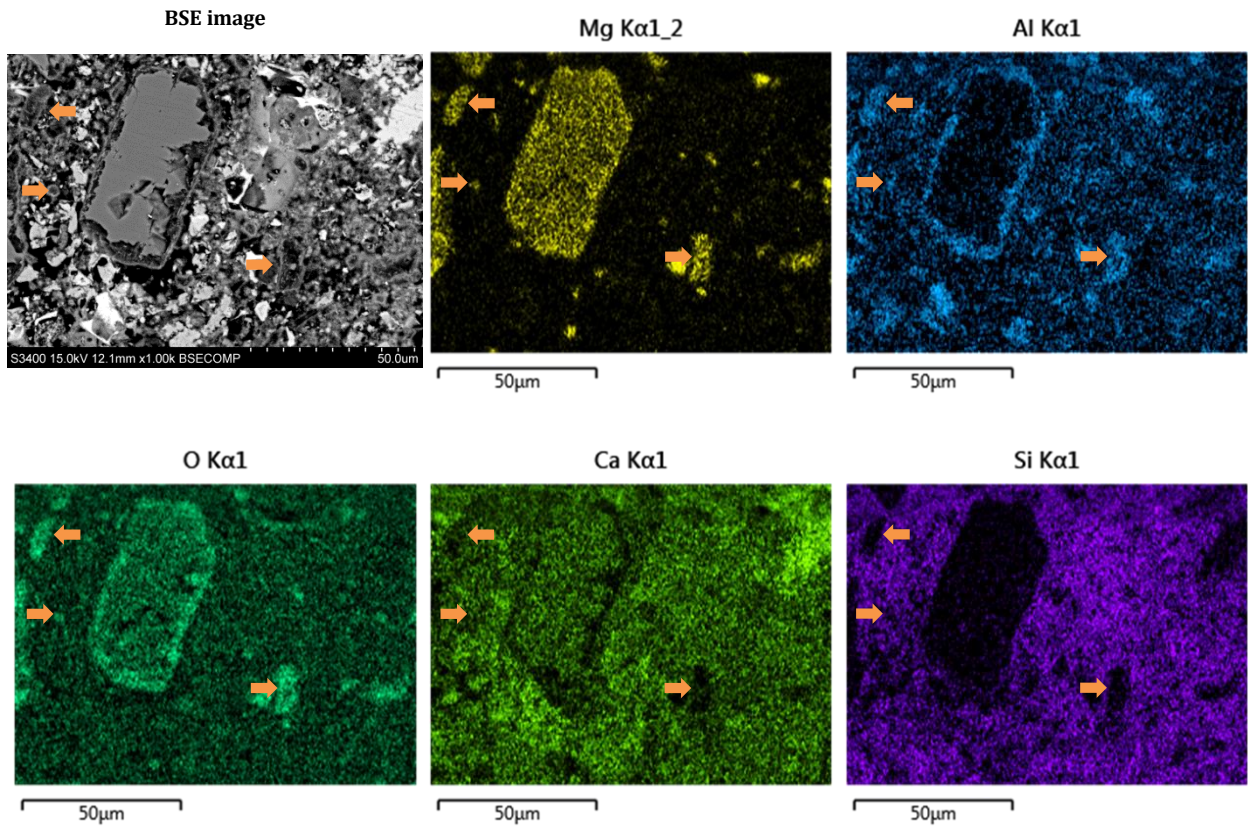


Figure 5: BSE image and elemental maps of magnesium, aluminium, oxygen, calcium and silicon for the sample 60C35D5M cured at 60 °C for 360 days. The arrows in the BSE image indicate small fully-reacted dolomite grains.

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60C20D20M 60 °C 360d

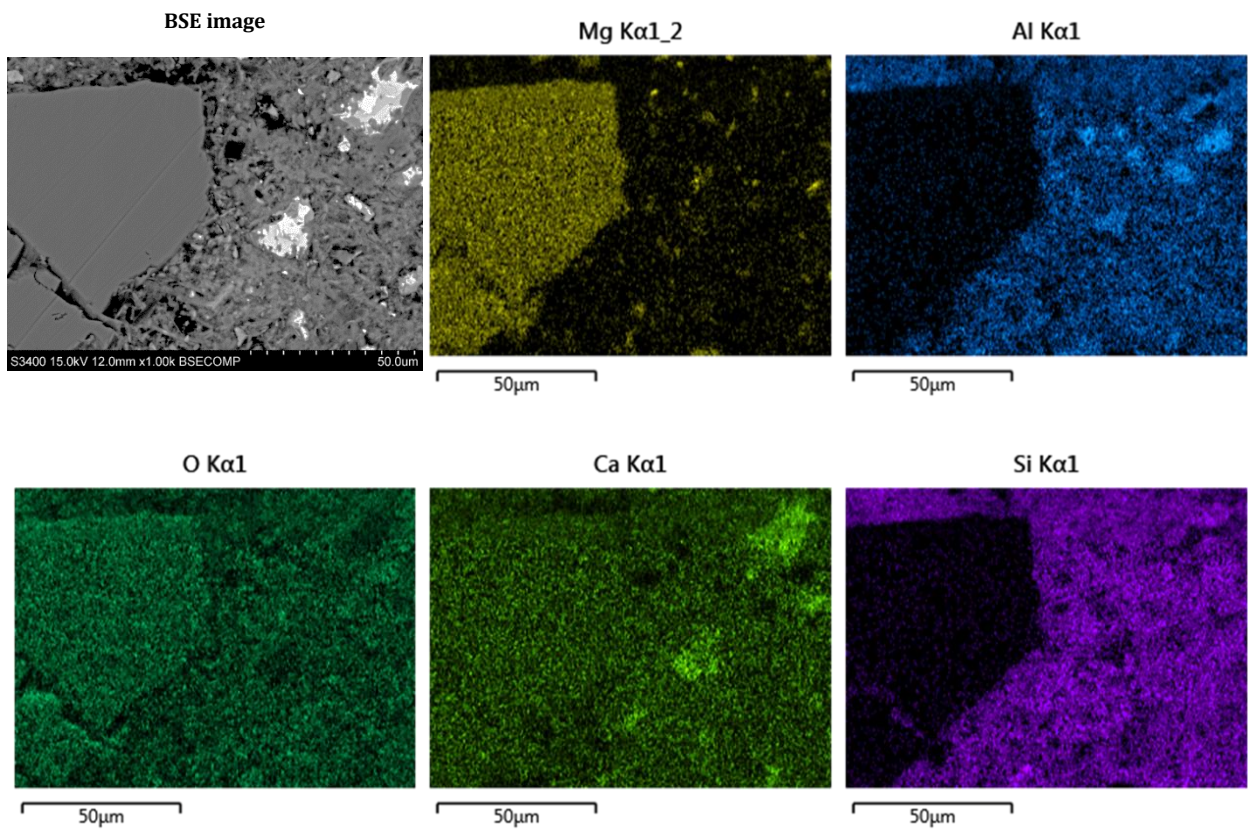


Figure 6: BSE image and elemental maps of magnesium, aluminium, oxygen, calcium and silicon for the sample 60C20D20M cured at 60 °C for 360 days.

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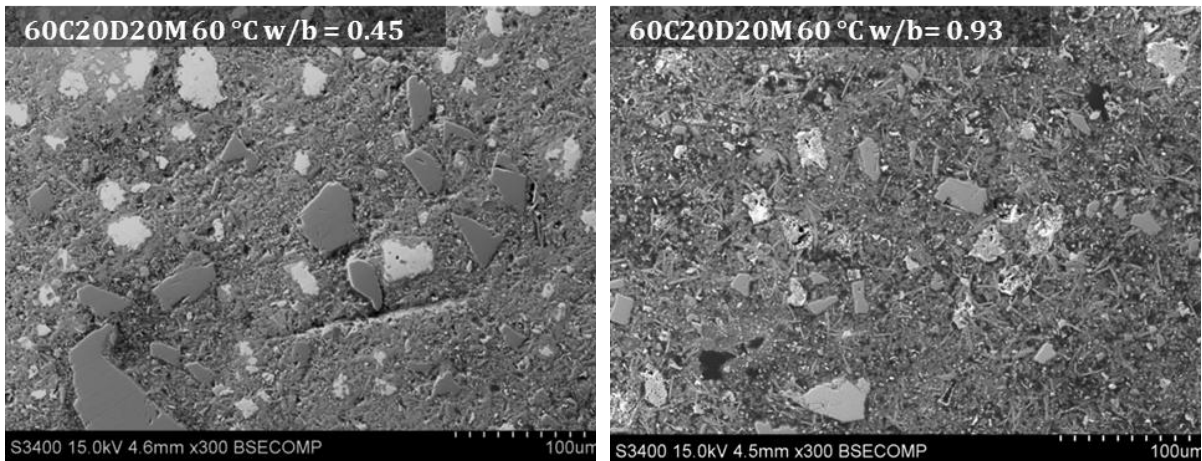


Figure 7: BSE images of the sample 60C20D20M with w/b ratios of 0.45 and 0.93 cured at 60 °C for 400 days.

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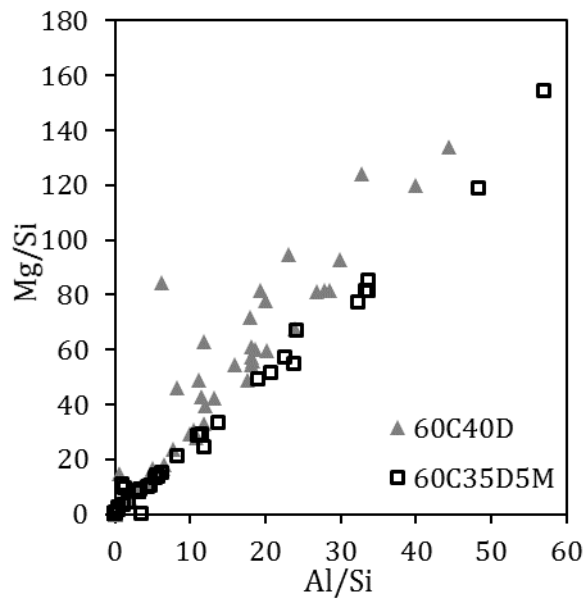


Figure 8: Mg/Si ratio over the Al/Si ratio for the point analyses of the reaction rims around the dolomite grains of samples cured at 60 °C for 360 days.

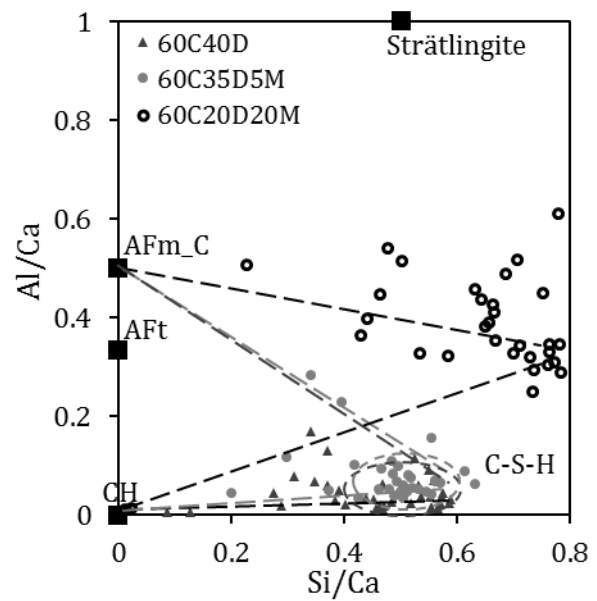


Figure 9: Al/Ca ratio over the Si/Ca ratio for the point analyses of the matrix of samples cured at 60 °C for 360 days.

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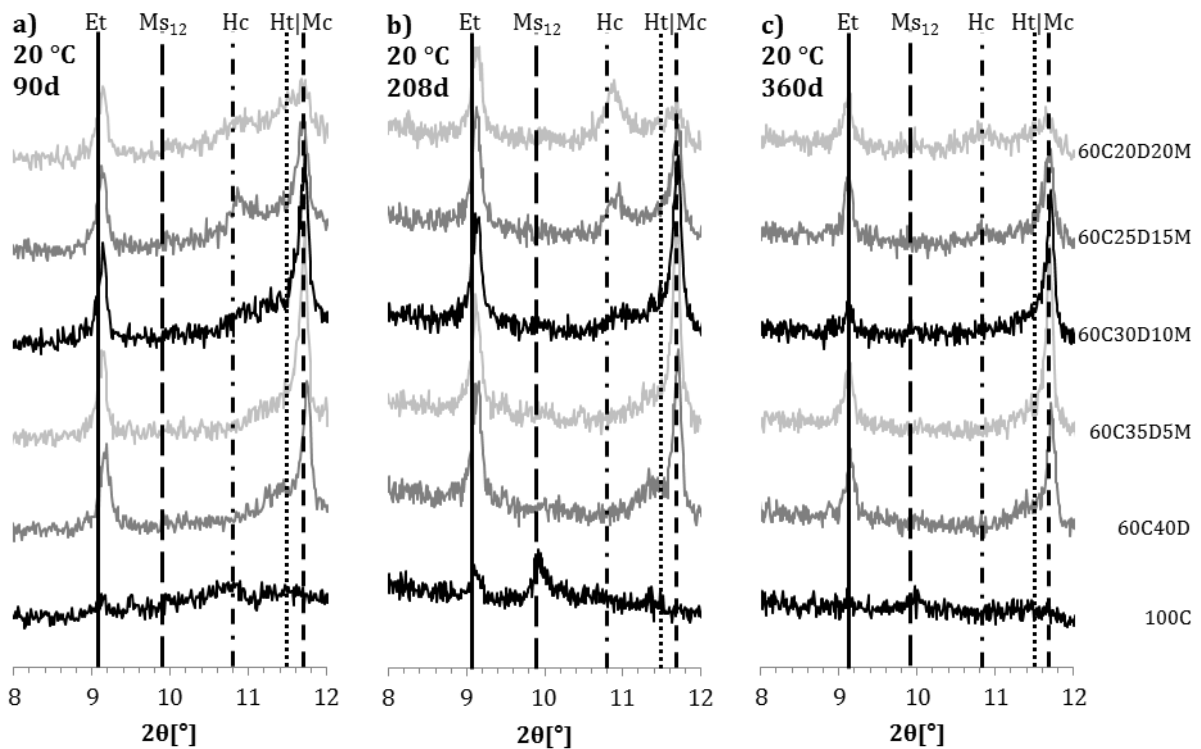


Figure 10: XRD-patterns of samples containing dolomite and the reference (100C) cured at 20 °C for a) 90 days, b) 208 days and c) 360 days.

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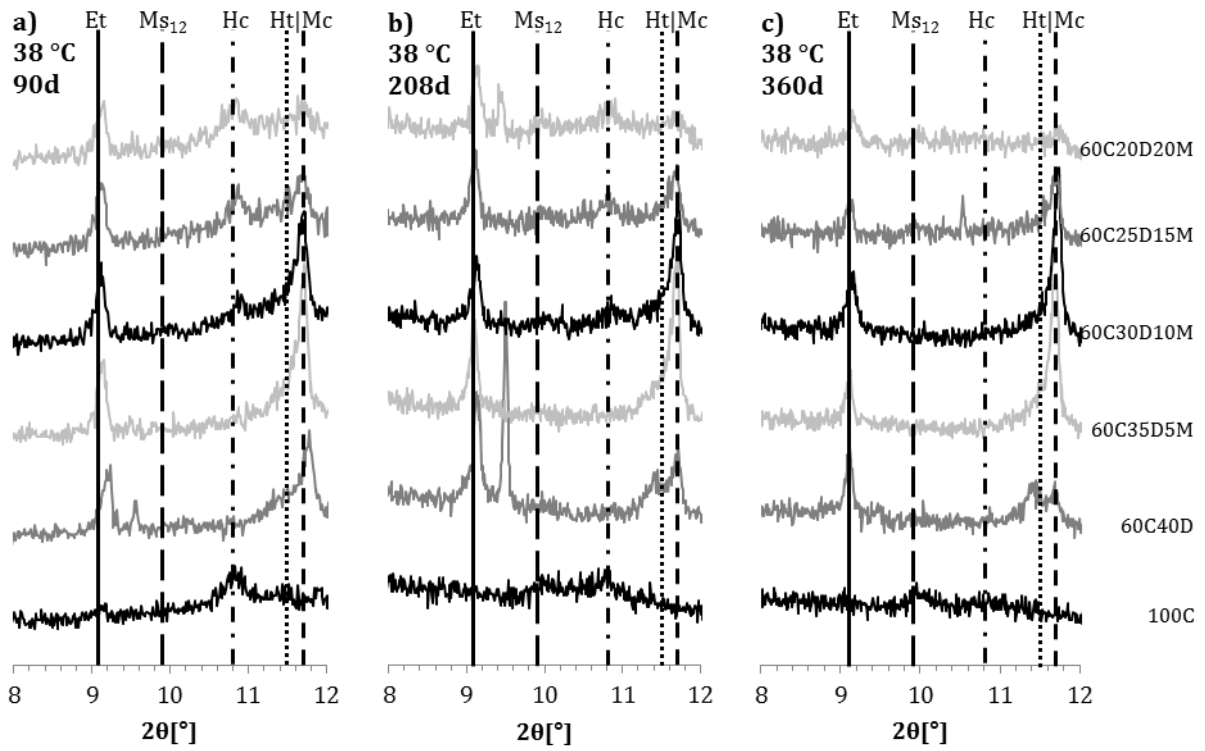


Figure 11: XRD-patterns of samples containing dolomite and the reference (100C) cured at 38 °C for a) 90 days, b) 208 days and c) 360 days.

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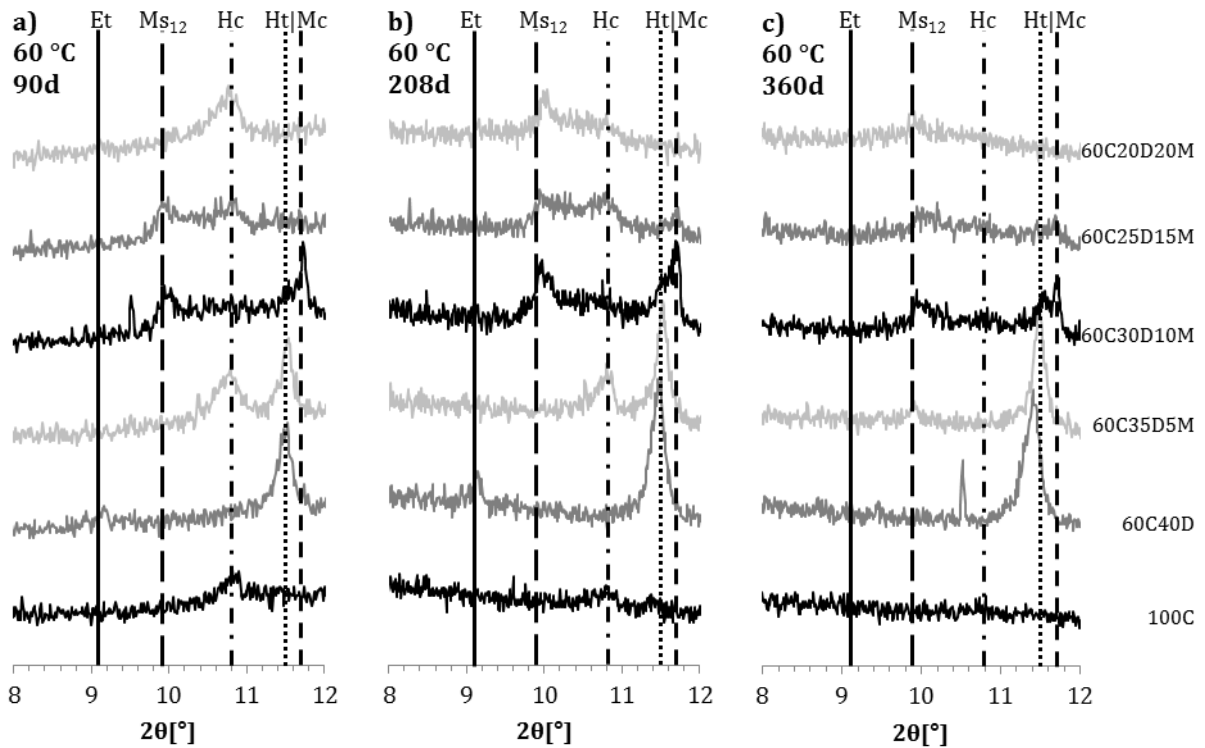


Figure 12: XRD-patterns of samples containing dolomite and the reference (100C) cured at 60 °C for a) 90 days, b) 208 days and c) 360 days.

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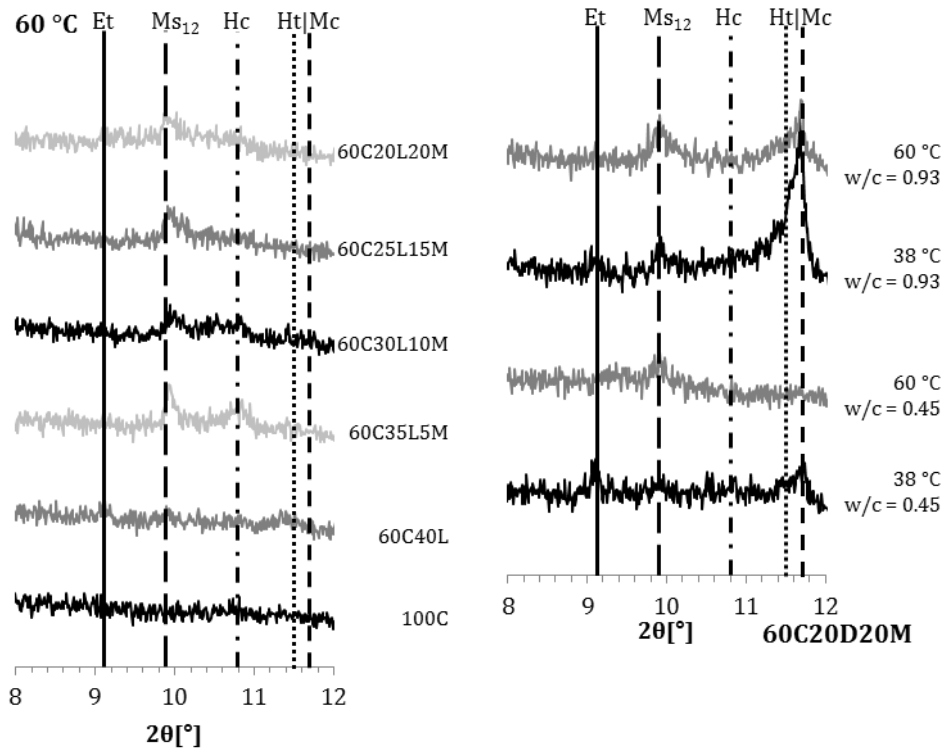


Figure 13: XRD-patterns of samples containing limestone and the reference (100C) cured at 60 °C for 360 days.

Figure 14: XRD-patterns of the 60C20D20M samples with a w/b ratio of 0.45 and 0.93, cured at 38 °C and 60 °C for 400 d.

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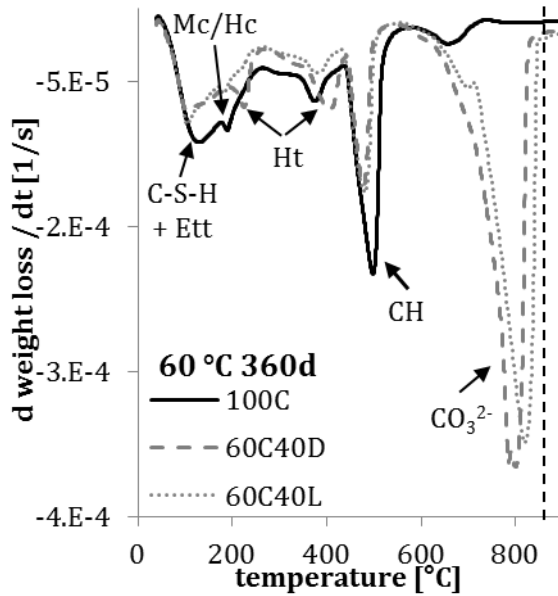


Figure 15: Differential thermogravimetric (DTG) curves for samples 60C40D, 60C40L and the reference 100C cured at 60 °C for 360 days.

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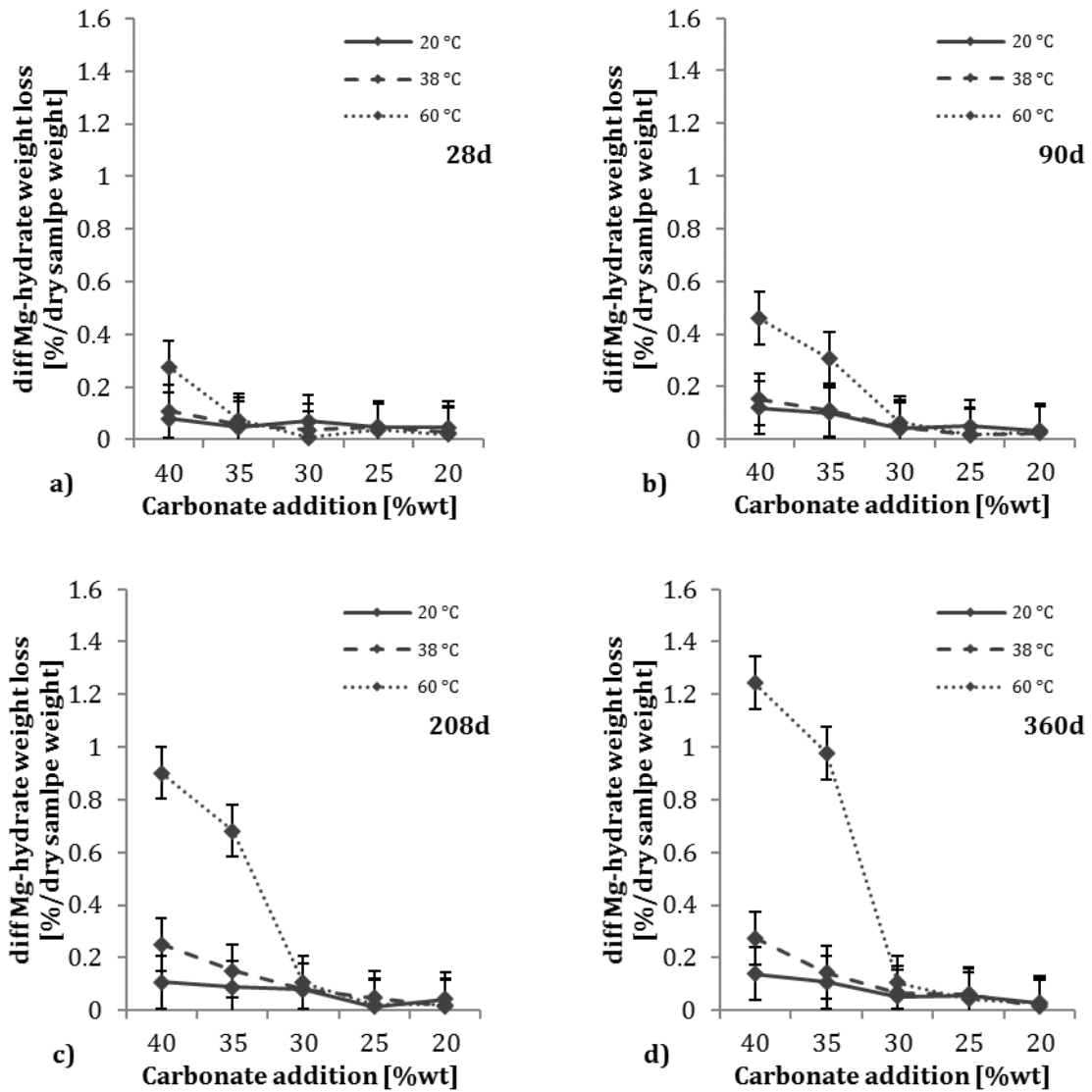


Figure 16: Development of the difference in Mg-hydrate weight losses between samples containing dolomite and samples containing limestone for the various carbonate additions, cured at 20 °C, 38 °C, and 60 °C for a) 28d, b) 90d, c) 208d, and d) 360d. It is important to note that, due to the broad shape of the peaks and the possible overlapping of weight loss peaks, these weight losses give only an indication about magnesium-containing hydrates decomposing in this temperature region, not necessarily only hydrotalcite.

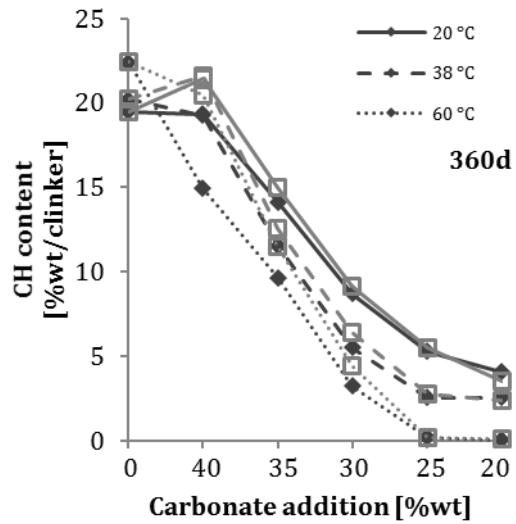


Figure 17: Development of the portlandite content for the samples containing various amounts of dolomite or limestone in combination with metakaolin and the reference 100C cured at 20 °C, 38 °C and 60 °C for 360 days. *Black diamonds* indicate the samples containing dolomite and *grey squares* indicate the samples containing limestone.

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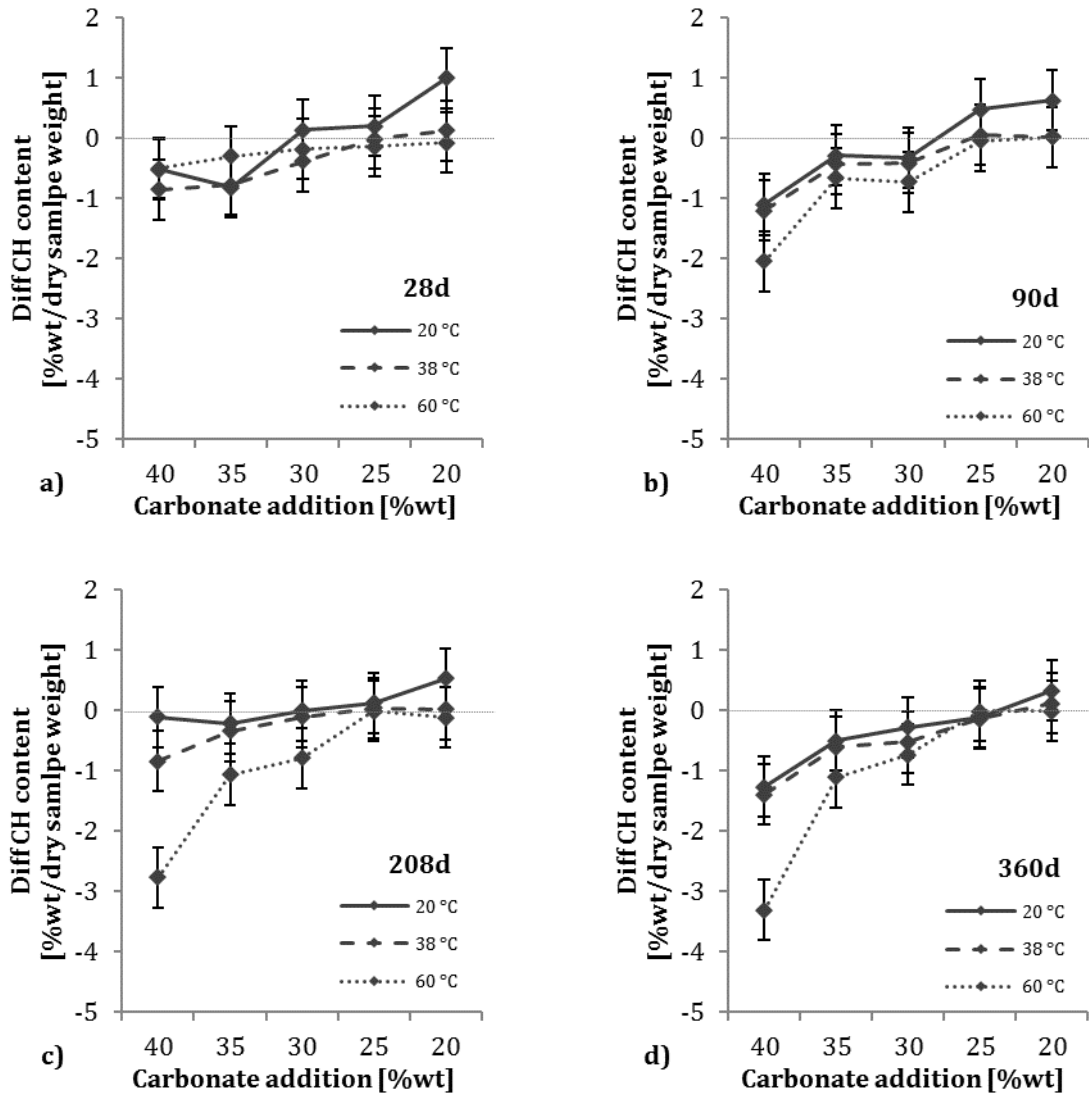


Figure 18: Development of the difference in the portlandite content between samples containing dolomite and limestone for the various carbonate additions, cured at 20 °C, 38 °C, and 60 °C for a) 28d, b) 90d, c) 208d, and d) 360d.

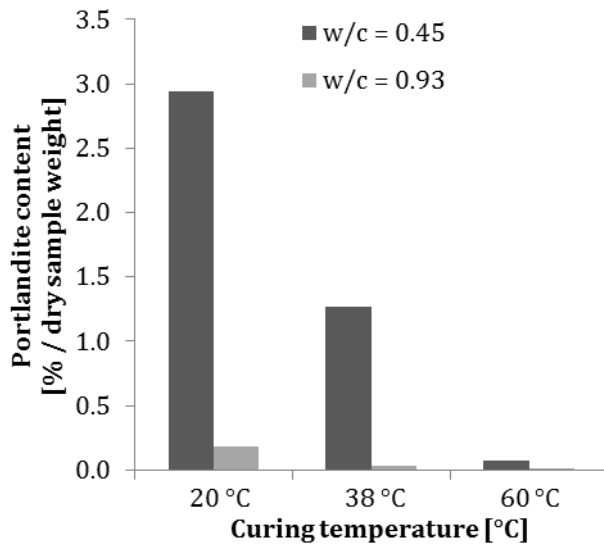


Figure 19: Portlandite content of samples with a w/b ratio of 0.45 and 0.93 cured at 20 °C, 38 °C and 60 °C for 400 days.

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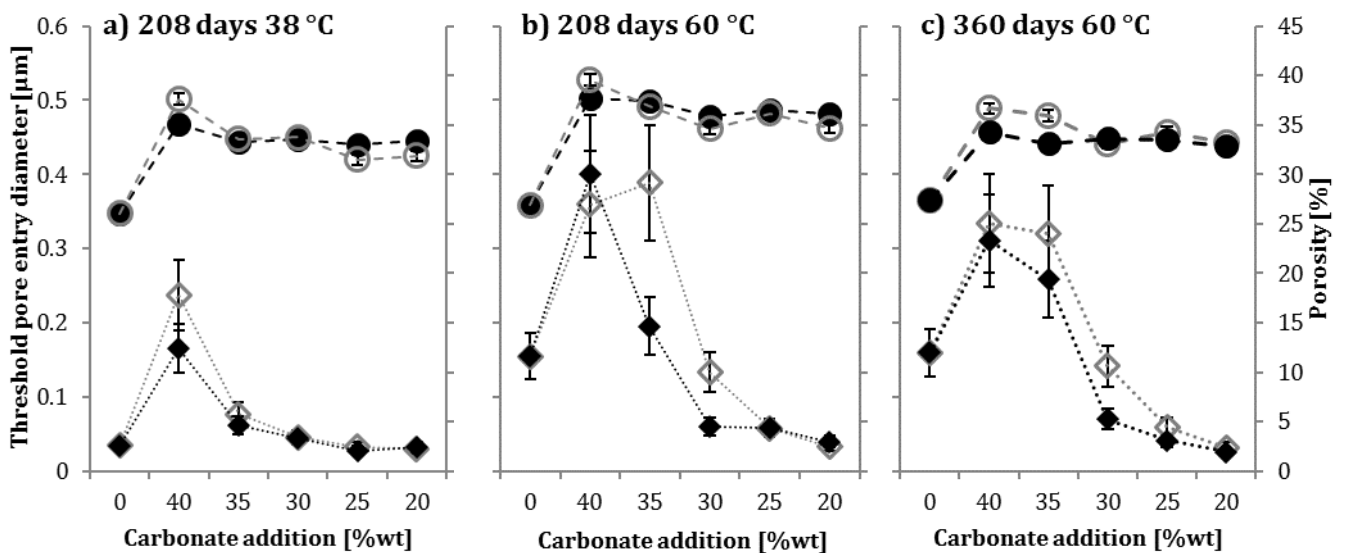


Figure 20: Development of the threshold pore entry diameter (*diamonds*) and porosity (*dots*) for samples containing dolomite (*black filled*) or limestone (*grey hollow*) cured for a) 208 days at 38 °C, b) 208 days at 60 °C and c) 360 days at 60 °C.

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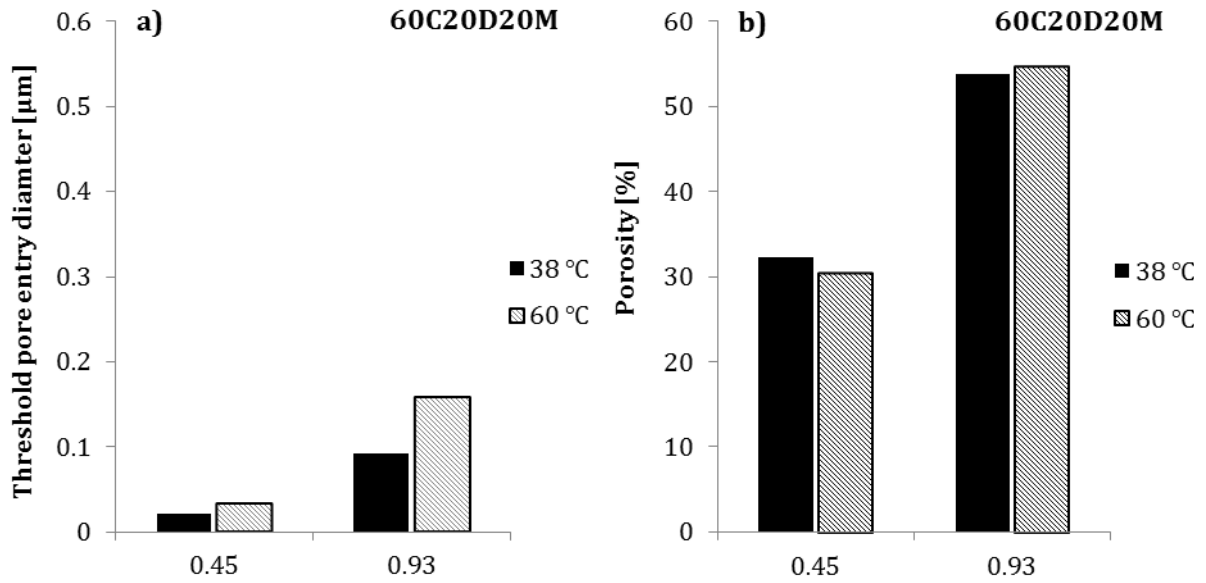


Figure 21: The threshold pore entry diameter (a) and the porosity (b) for the sample 60C20D20M prepared with a high (0.93) or low (0.45) w/b ratio cured at 38 °C or 60 °C for 400d.

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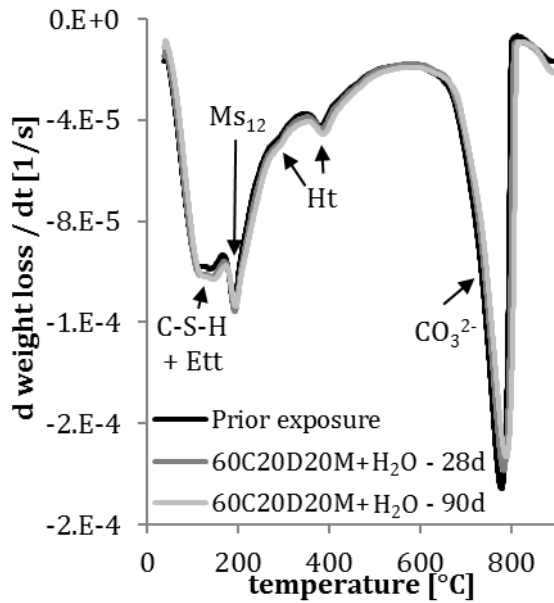


Figure 22: Differential thermogravimetric (DTG) curves for well-hydrated samples 60C20D20M exposed to 30%wt additional water for 28 or 90 days at 60 °C and the sample prior exposure.

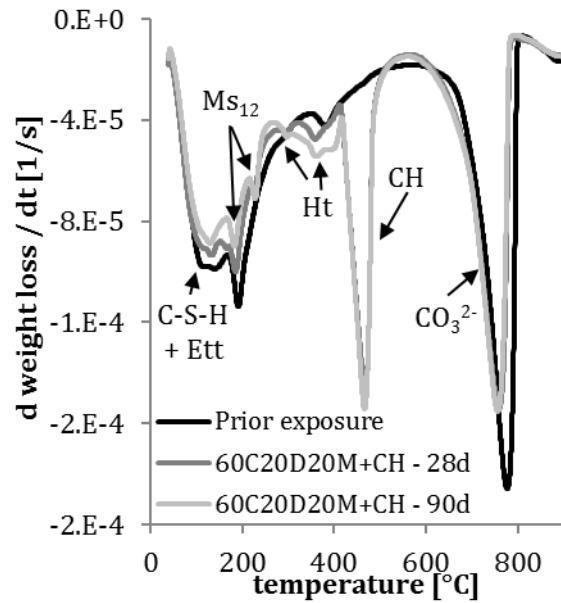


Figure 23: Differential thermogravimetric (DTG) curves for well-hydrated samples 60C20D20M exposed to 30%wt additional water and 30%wt portlandite for 28 or 90 days at 60 °C and the sample prior exposure.

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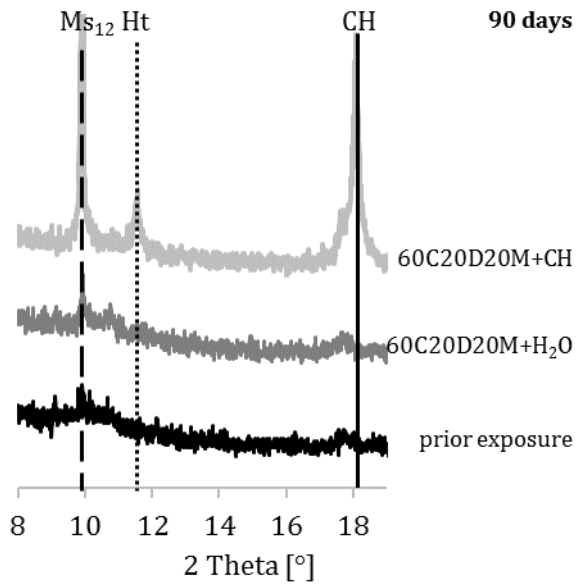


Figure 24: XRD-patterns of the well-hydrated samples 60C20D20M exposed to 30%wt of additional portlandite and or water for 90 days at 60 °C and the sample prior exposure.

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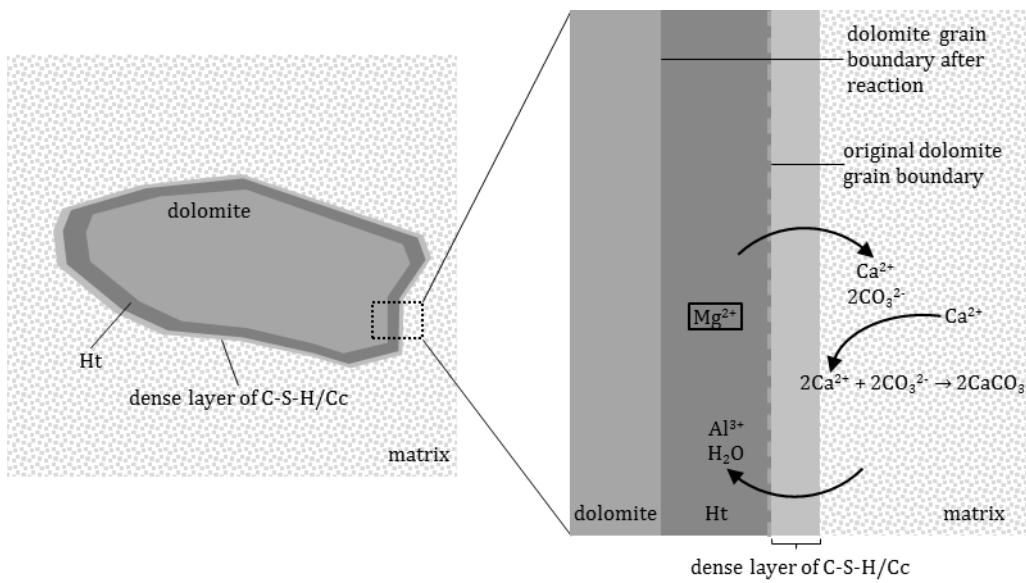


Figure 25: Schematic illustration of the hydrotalcite formation in the reaction rims around the dolomite grains.

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