

Impact of sample crushing on porosity characterization of hardened cement pastes by low temperature calorimetry: comparison of powder and cylinder samples

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Abstract

The impact of sample crushing on the detected porosity of hardened cement pastes by low temperature calorimetry (LTC) was studied using powder and cylinder samples. Two types of cements, CEM I and CEM III, were used to prepare the pastes. A model porous material, MCM-41, was also used in order to investigate some aspects of the measurement and the evaluation approach. The powder and cylinder samples of the cement pastes were compared in terms of the calculated ice content curves, total pore volumes and pore size distribution curves. For the two studied cement pastes, the calculated ice content curves of freezing of the powder sample differed from that of the cylinder samples, especially for the paste CEM III. The results indicate that sample crushing changed the pore connectivity as compared to non-crushed samples. One important difference between the powder sample and the cylinder samples of the paste CEM III was that the determined maximum ice content in the powder sample was much higher than that in the cylinder samples, the relatively difference being about 40-50%. However, this kind of marked difference was not found in the paste CEM I. The observed difference between the calculated pore volume of the powder and the cylinder samples of the paste CEM III is possibly due to some of the “isolated” pores which, presumably, cannot be fully filled with water in the preparation of the cylinder samples. However, sample crushing makes it possible to saturate the pores to a greater

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extent if the crushing contributes to open up the “isolated” pores. Consequently, more pores are detected in the powder samples. The argument that the “isolated” pores have a tendency to be opened up by the crushing process is supported by results using gravimetric measurements and “dynamic (water) vapor sorption” measurements on powder samples.

Keywords: Cement paste, calorimetry, freezing and thawing, pore size distribution, thermoporometry, cryoporometry

1. Introduction

The pore system in a cement based material is very complicated and the pore sizes can range from millimeter scale to nanometric level [2, 3]. The porosity at the nanometric level is of paramount importance in studying properties of cement based materials. It is the main parameter which influences, e.g., the strength, the shrinkage, the transport properties and the durability [4]. Moreover, the properties of the nanometric pores are quite essential in the modeling of some important processes of cement based materials, e.g., moisture transport [5, 6], drying shrinkage [7] and carbonation [8, 9]. Thus, the accuracy of the pore structure characterization of cement based materials is crucial. A number of methods have been used to determine the pore structure of cement based materials, e.g. mercury intrusion porosimetry (MIP) [10, 11], nitrogen adsorption/desorption (NAD) [12], scanning electron microscopy (SEM) [13], water vapor sorption [14], small-angle X-ray scattering [15, 16] and small-angle neutron scattering [17, 18]. Recently, liquid proton Nuclear Magnetic Resonance has been used and interesting results have been reported [19, 20].

Low temperature (micro-)calorimetry or LTC, also known as thermoporometry and sometimes referred to as thermoporosimetry or cryoporometry [21], is one of the used methods to investigate the porosity of materials, especially pores at the meso-level, i.e., pores with widths between 2 and 50 nm, according to the IUPAC definition [22, 23]. LTC has been used extensively to study the porosity of cement based materials, e.g., in [21, 24–26]. Compared with the traditional methods developed for porosity characterization, e.g., mercury intrusion porosimetry (MIP), nitrogen adsorption/desorption (NAD) and scanning electron microscopy (SEM), a major advantage of using LTC on cement based materials is that the measurements can be conducted on virgin samples without any drying treatment [21, 26–28], as the drying treatment in many cases results in an alteration of the pore structure of cement based materials [29, 30].

As for most of the techniques, LTC is an indirect method for porosity determination. The analysis of the measured data is not straightforward and special care should be taken in the data analysis. In LTC studies, e.g., using water as the probe liquid, the instrument records the heat flow during the freezing and melting process. Based

31 on the measured heat flow, the ice content in a sample at different temperatures can
32 be calculated. To calculate the ice content, the baseline must be determined carefully
33 and appropriate values of the heat of fusion of the water/ice confined in small pores
34 at different temperatures must be specified. The impact of the two factors, i.e., the
35 baseline determination and the values of the heat of fusion of confined water/ice, on the
36 calculated ice content has been presented in an earlier study [31]. To calculate the pore
37 size distribution, the relation between the pore size and the depressed freezing/melting
38 point must be known. Under certain basic assumptions, thermodynamic considerations
39 demonstrate that there is a unique equation between the phase transition temperature
40 of the water/ice confined in pores and the curvature of its solid-liquid interface [32, 33].
41 The quantitative relation of the freezing/melting point and the pore size can be prin-
42 cipally determined by adopting appropriate values for the thermodynamic parameters
43 of the confined water/ice. However, there is no generally consensus on which values
44 should be used for the thermodynamic parameters of water confined in pores, especially
45 at low temperatures, e.g., the surface tension, the heat of fusion and the heat capacity
46 of water and ice [22, 33, 34]. This makes the quantitative determination of the relation
47 between the freezing/melting point and the pore size difficult. Additionally, it should
48 be mentioned that the samples to be tested in LTC studies need to be fully saturated
49 by the probe liquid. If the samples are not fully saturated, the total pore volume will be
50 underestimated since the LTC method can only detect the pores filled with the probe
51 liquid. Moreover, the relation between the depressed freezing/melting temperature and
52 the pore size in this context is normally derived based on the prerequisite that the pores
53 under study are fully saturated. It has been shown that the needed thermodynamic
54 relations to be used for fully saturated pores are different from that for non-fully sat-
55 urated pores [35]. The needed thermodynamic relations for a non-fully saturated pore
56 system are very complicated, and therefore difficult to determine quantitatively, if it
57 is still possible at all. The impact of using the thermodynamic relations as derived
58 based on fully saturation for non-fully saturated system on the determined pore size
59 distribution is discussed in [35].

60 Due to the constraints as discussed above, it should be mentioned that similar
61 to many other indirect methods, LTC is a semi-empirical method for pore structure
62 characterization. This is considered especially true when working with materials like
63 hardened cement pastes, where the pore structures are very complicated as discussed
64 above. To our knowledge, perhaps at this moment there is no method, which can
65 determine the “real” or “true” pore structure of materials with complicate pore systems
66 at nanometric level. Most of the indirect methods, if not all of them, are no more than
67 semi-quantitative.

68 In LTC studies, the freezing of pore water is normally assumed to be a process initi-
69 ated by heterogeneous nucleation and then followed by progressive penetration [36, 37].

70 That is, the freezing process after the heterogeneous nucleation, i.e., the ice penetration
71 process, is controlled by the pore entry or neck sizes; while the melting process is con-
72 trolled by the pore interior sizes [26, 38]. For this reason, the freezing process indicates
73 the pore connectivity and the melting process reflects the pore interior size distribution.
74 It is noted that in some LTC studies on cement based materials, the sample size is ei-
75 ther relatively small (with the mass on the order of tens of milligrams), e.g., see [39], or
76 relatively big (with the mass on the order of several grams), e.g, see [21], depending on
77 the instrument adopted. For LTC measurements on samples with relatively small size,
78 crushing and sometimes even grinding is needed in sample preparation. The procedure
79 of sample crushing and/or grinding is also needed in sorption studies using a “Dynamic
80 water vapor sorption (DVS)” instrument due to small sample holder. In sorption stud-
81 ies, it is normally pre-assumed that the crushing and/or grinding of the cement paste
82 samples do not change the pore structure at the nanometric level, e.g., see [14, 40].
83 However, it should be mentioned that there are NMR cryoporometry studies, e.g., see
84 [41, 42], which compares the measurements on the same material but in different forms,
85 i.e., big integral and crushed samples. The results show that the pore connectivity is
86 changed by the sample crushing even though the pore interior size distribution remains
87 about the same. The impact of sample crushing, i.e., using different size of samples,
88 in LTC studies on the determined porosity of hardened cement pastes is, however, not
89 fully clear and needs to be further examined.

90 This work aims to clarify the possible impact of sample crushing on the pore struc-
91 tures of hardened cement pastes characterized by LTC. Hardened cement paste samples
92 prepared from two types of cements are included. For each hardened cement paste, two
93 types of samples, i.e., in the form of powders and cylinders, are used. One mono-sized
94 model material MCM-41 is also included in this study. MCM-41 is a silica based ma-
95 terial and the pore structure is in the form of hexagonal arrays of uniform tubular
96 channels of controlled width, which has been widely used as a model material in the
97 context of porosity characterization [43–46]. The purposes of including the model ma-
98 terial MCM-41 are two folds: firstly, it is to validate the stability of the instrument;
99 secondly, it is to validate the applicability of the LTC in the context of porosity deter-
100 mination (By comparing the analyzed results from our experiments with that provided
101 by supplier, the aim is to validate the LTC method, including the data evaluation
102 method adopted in the study, is applicable). By doing this, it is expected that the
103 possible effects due to the instrument and data evaluation method, which may lead to
104 differences in the obtained results, can be excluded. In this way, the difference noted
105 between powders and cylinders of studied hardened cement pastes can be attributed
106 to the sample crushing. The impact of sample crushing on the detected porosity of the
107 studied hardened cement pastes is demonstrated and some possible explanations are
108 proposed for the observed differences.

109 **2. Experimental**

110 *2.1. Materials preparation*

111 *2.1.1. Model material MCM-41*

112 The MCM-41 used in this study is in the powder form and is produced by Tianjin
113 Chemist Scientific Ltd. The nominal pore diameter and the total pore volume reported
114 by the producer are 3.0 nm and ≥ 0.70 ml/g, respectively.

115 In order to handle powders more easily in the instrument during calorimetric mea-
116 surements (Section 2.2), a cylindrical plastic vial was used as a sample holder the
117 powder sample. The size of the plastic vials ($\sim \phi 14 \times 48$ mm) was chosen to fit the
118 measuring chamber of the calorimeter. The MCM-41 powders were placed into the
119 plastic vials up to about half the volume (the dry weight was about 0.2 g) and then
120 covered with distilled water. In order to saturate the samples, the plastic vials con-
121 taining powders covered with distilled water were placed under a reduced pressure (\sim
122 40 mbar) for about 3 hours. After that, the apparent excess bulk water on top of the
123 powders was removed and the samples were stored in closed containers for two to three
124 weeks before calorimetric measurements.

125 *2.1.2. Cement pastes*

126 Two types of cements, i.e., CEM I 32.5 R and CEM III/B 42.5 N, were used to
127 prepare paste samples in this study. Refer to [Appendix A](#) for the properties and the
128 chemical composition of the cements. The water-to-cement ratio of the prepared paste
129 samples was 0.4. A paddle mixer was used to mix the fresh pastes. After mixing, the
130 fresh pastes were cast into cylindrical plastic vials followed by proper compaction. The
131 dimension of the plastic vials is about $\phi 15 \times 50$ mm. The hardened cement pastes were
132 demoulded after one day of sealed curing at room temperature (about 20 °C). Following
133 the demoulding, the paste samples were placed into slightly bigger plastic flasks (\sim
134 $\phi 25 \times 60$ mm) filled with saturated limewater for curing at room temperature. As the
135 pore solution of hardened cement pastes contains many ionic species [47], using pure
136 water for curing may lead to leaching of ions from hydration products as reported in,
137 e.g., [48]. It is a recommended practice to cure cement based materials with saturated
138 limewater, e.g. see [49].

139 The cylinder samples of the hardened cement pastes were used to obtain powder
140 samples. The cylinders were firstly vacuum saturated with saturated limewater and
141 then the crushing and grinding of the samples were conducted in a carbon dioxide free
142 chamber. Cement based materials exposed to air will carbonate, which will lead to the
143 change of the microstructure [50]. Therefore, carbon dioxide free chamber was used
144 to avoid carbonation. After that, the ground paste powders (passed through 315 μm
145 sieve) were placed into the plastic vials up to about half the volume and then covered
146 with saturated limewater. This procedure was also conducted in the carbon dioxide

147 free chamber. Following that, the plastic vials containing cement paste powders were
148 placed under reduced pressure for saturation purpose (as did for the MCM-41 powders
149 in Section 2.1.1). After the saturation, the apparent excess bulk water on top of the
150 powders was removed and the samples were stored in closed containers for two to
151 three weeks before calorimetric measurements (after calorimetric measurements, the
152 dry weight of cement paste powders and the total water content in each plastic vial
153 were determined as about 1:1).

154 For cylinder samples of the cement pastes, they were vacuum saturated and then
155 the apparent bulk water on the surface of cylinders was wiped off before calorimetric
156 measurements.

157 When calorimetric measurements were performed, the total curing time of the cylin-
158 der and powder samples of the studied cement pastes were about 1 year and 1.5 years,
159 respectively.

160 By using two types of cements, part of the effect due to cement types on the porosity
161 determination by LTC is expected to be observed. The hardened cement pastes are
162 designated as CEM I and CEM III in the following discussion.

163 2.2. Calorimetric measurements

164 A Calvet-type scanning calorimeter (SETARAM) was used in this investigation.
165 The calorimeter was calibrated and operated to work between about 20 °C down to
166 about -130 °C. The cooling and heating rate were set to be 0.1 °C per minute. A
167 freezing and melting cycle consists the temperature scanning starting from about 20
168 °C and going down to about -80 °C and then back to about 20 °C again. The cooling
169 and heating rates were adopted based on suggestions given in [51, 52], where both
170 stability of the measured results and efficiency of the measurements were considered.
171 The rates are in the same order as recommended by, e.g., [24], for the same type of
172 instrument.

173 In the measurements of the powder samples, a plastic vial same as the sample holder
174 but empty was placed in the reference chamber of the calorimeter. The purpose was
175 to counteract the effects due to some uncertainties that the plastic vial may experience
176 during freezing and melting measurements. In the measurements of the cylinder cement
177 paste samples, a totally dry paste sample (oven drying at about 105 °C until constant
178 weight) of the same dimension as the testing specimen was used as a reference sample
179 in the LTC instrument. One purpose of using the reference sample is to reduce the
180 possible uncertainties of the solid paste during the freezing and melting measurements.
181 More discussions about the influence and the benefits of using such a reference sample
182 in calorimetric measurements can be found in [31, 52].

183 Due to the energetic barrier to nucleation, bulk water can be cooled down below 0
184 °C without freezing, i.e., the supercooling behavior [34]. Because of the supercooling, it

185 is then not possible to derive any pore entry information from the freezing curves during
186 the temperature range between 0 °C and the point when the supercooling terminates
187 (the initial nucleation starts). In order to suppress the supercooling of water during
188 freezing and to derive more information from the freezing process in LTC studies, two
189 cycles of freezing and melting measurements have been suggested, e.g, see [23, 26, 53].
190 Following the same concept, a trial run of a two cycles of freezing and melting using the
191 employed instrument was conducted. However, the results showed that in this study, no
192 more useful information can be obtained compared with the case of using one freezing
193 and melting cycle, refer to [AppendixB](#) for more explanation and details. Therefore,
194 only one freezing and melting cycle was used for all the measurements.

195 The mass of the tested vacuum saturated samples before and after calorimetric
196 measurements were determined. The relative difference of the mass is less than about
197 0.15%. That is, there is almost no water loss during the calorimetric measurements.
198 After calorimetric measurements, the tested samples were oven-dried at about 105
199 °C until constant weight to obtain the dry weights. The total water content of each
200 sample was obtained by the mass difference between the dry state and the state before
201 the calorimetric measurement.

202 For the model material MCM-41, calorimetric measurements were performed on
203 two samples of the same kind. By doing this, the stability of the instrument was
204 expected to be validated (assuming the MCM-41 powders are homogenous as they
205 were collected from a rather big batch). For the cylinder samples of each cement paste,
206 three different samples were measured with the purpose to check the homogeneity of
207 the prepared cylinder samples. The powder samples of each investigated cement paste
208 were collected from two cylinders and the homogeneity was not further checked. That
209 is, only one calorimetric measurement was conducted on the powder samples of each
210 studied cement paste.

211 Additionally, using vacuum saturated samples, the total porosity of the cylinder
212 samples of the two studied cement pastes were also determined through gravimetric
213 measurements by recording the mass of the samples both in air and submerged in
214 water.

215 **3. Results and discussion**

216 *3.1. Ice content*

217 The ice content calculation based on the measured data of heat flow is central for
218 pore volume and pore size distribution determination in LTC studies. Special care
219 should be taken in determining the baseline of heat flow and choosing the appro-
220 priate values for the thermodynamic parameters of water/ice confined in pores, i.e.,
221 surface tension, heat capacity and heat of fusion. A summary of the calculation meth-
222 ods/procedures is presented in [AppendixC](#). Detailed discussions in this context and

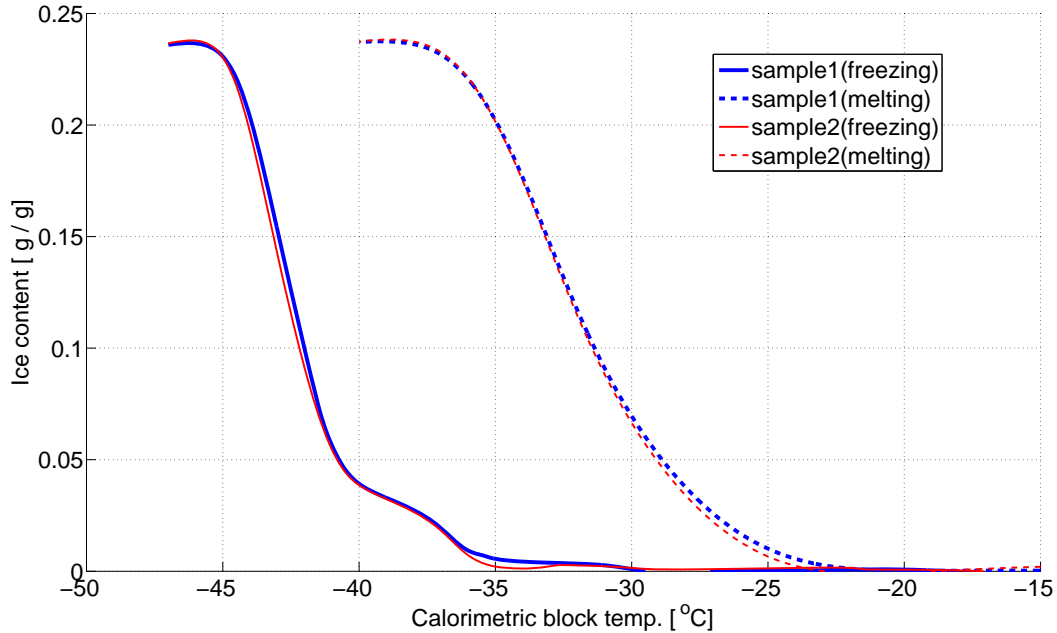


Figure 1: Calculated ice content curves of the two measured samples of the model material MCM-41 (with excess bulk ice subtracted). The ice content is expressed as gram per gram of dry material.

223 the methods for ice content calculation can be found in [31]. The “C-baseline” method
 224 together with the values chosen for the relevant parameters as discussed in [31], which
 225 has been demonstrated suitable to calculate the ice content, is used in this study.

226 3.1.1. Model material MCM-41

227 The calculated ice content curves of the two measured samples of the model material
 228 MCM-41 are shown in Figure 1. It can be found that the difference between the ice
 229 content curves (both freezing and melting) of the two samples is small. That is, the
 230 stability of the instrument is concluded to be satisfactory.

231 3.1.2. Cement pastes

232 For the cement pastes CEM I and CEM III, the calculated ice content curves for the
 233 measured three cylinder samples and one powder sample of each paste are presented
 234 in Figure 2 and Figure 3, respectively. It should be mentioned that the presented ice
 235 content does not include the “bulk” ice, which is determined by calculating the ice
 236 content corresponding to the peak above 0 °C from the heat flow curve of melting.
 237 For the cylinder samples, the “bulk” ice is mainly the ice in big pores (with small
 238 temperature depression, e.g., air voids) since saturated surface dry samples were used;
 239 while for the powder samples, the “bulk” ice should contain both the ice in big pores

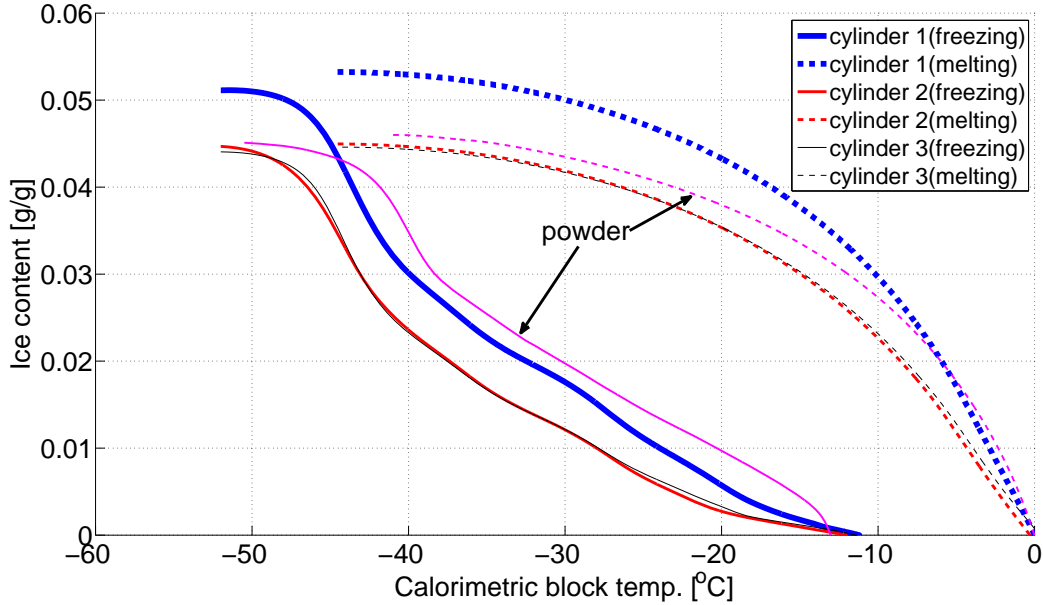


Figure 2: Calculated ice content curves of cylinder and powder samples of the cement paste CEM I (with “bulk” ice subtracted). The content of “bulk” ice in each sample is determined by calculating the ice content corresponding to the peak above 0 °C from the heat flow curve of melting. The ice content is expressed as gram per gram of dry material.

240 and the excess bulk water since there is still a portion of excess water in the powder
 241 samples as determined.

242 As can be found from Figure 2, for the paste CEM I, the calculated ice content
 243 curves of two of the measured three cylinder samples (cylinder 2 and cylinder 3) are
 244 quite comparable; while the ice content curves of one sample (cylinder 1) differ from
 245 that of the other two to a certain extent. Since the instrument is rather stable (ac-
 246 cording to the measurements on the MCM-41, see Figure 1), the difference between
 247 the ice content curves as determined for the cylinder samples is attributed to the in-
 248 homogeneity of the prepared samples. For the powder sample of the paste CEM I, the
 249 ice content curve during melting is higher than that of cylinder 2 and cylinder 3 at
 250 a same temperature while it is generally lower compared with that of the cylinder 1
 251 (except during the temperature range between -7 °C and about 0 °C). There are two
 252 main differences between the ice content curves of freezing of the powder sample and
 253 the cylinder samples. Firstly, the ice content curve of freezing of the powder sample
 254 starts at a somewhat lower temperature than that of the cylinders. That is because
 255 the starting point of the ice content curve of freezing reflects the heterogeneous nu-
 256 cleation temperature (of supercooled water) and it generally decreases as the sample
 257 size decreases [54]. Secondly, the ice content curve of freezing of the powder sample

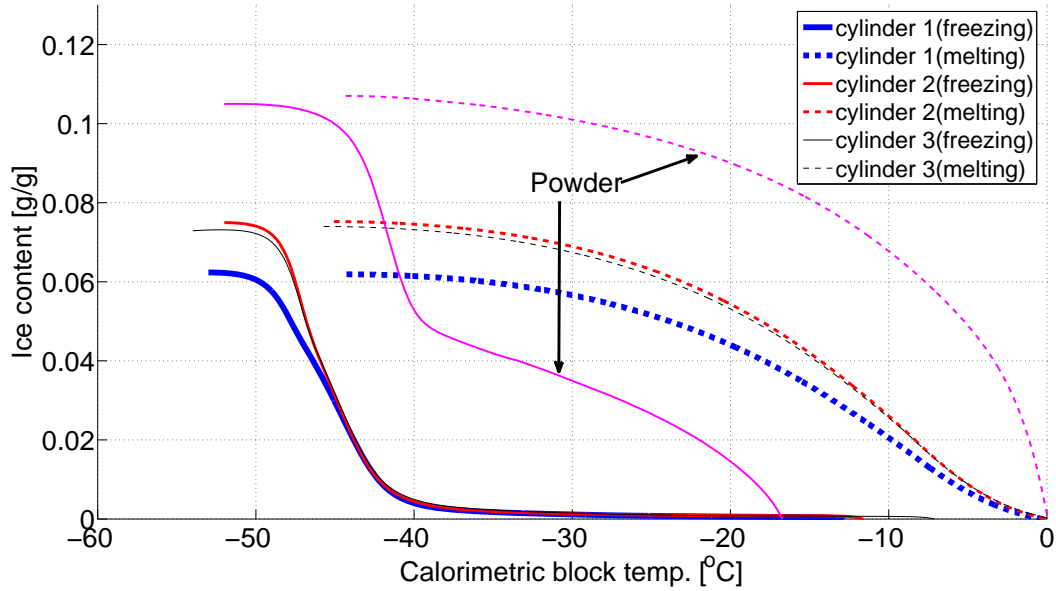


Figure 3: Calculated ice content curves of cylinder and powder samples of the cement paste CEM III (with “bulk” ice subtracted). The content of “bulk” ice in each sample is determined by calculating the ice content corresponding to the peak above 0 °C from the heat flow curve of melting. The ice content is expressed as gram per gram of dry material.

258 is higher than that of all the three cylinders at a same temperature. The more ice
 259 content detected at a same temperature during freezing indicates that the connectivity
 260 of the pores in the powder sample is increased compared with that of the cylinders, i.e.,
 261 more pores are penetrable by ice at a same temperature. The increased connectivity
 262 of the powder sample is more obvious by comparing the ice content curve of freezing
 263 of the powder sample with that of cylinder 1. Even though cylinder 1 has a higher
 264 pore volume than the powder sample (i.e., reflected by a higher maximum ice content),
 265 the ice content curve of freezing of the powder sample is still higher than that of the
 266 cylinder sample at a same temperature. The results may indicate that the crushing of
 267 cylinder samples into powders increases the pore connectivity.

268 Some differences as found between the powder sample and the cylinder samples for
 269 the paste CEM I are also found for the paste CEM III (Figure 3), e.g., the inhomogeneity
 270 of the prepared cylinder samples and the increased pore connectivity in the powder
 271 sample than that in the cylinder samples. There are also some new features. The
 272 difference between the ice content curve of freezing of the powder sample and that of
 273 the cylinder samples for the paste CEM III is more significant compared with that of
 274 the paste CEM I. From the ice content curves of freezing of the paste CEM III (Figure
 275 3), it should be mentioned that the freezing behavior of the water in cylinder samples

276 is quite different from that in the powder sample. When the temperature goes down to
277 about $-40\text{ }^{\circ}\text{C}$, very limited amount of ice is formed in the cylinder samples while about
278 half of the total ice is formed in the powder sample. As we know, the freezing process
279 is an ice penetration process (after the initial heterogeneous nucleation) controlled by
280 pore entry sizes. That is, the pore entry sizes in cylinder samples of the paste CEM III
281 can be very small and ice cannot penetrate through the small pore entries. Only when
282 the temperature goes down to a very low point when homogenous nucleation becomes
283 significant (e.g., about $-40\text{ }^{\circ}\text{C}$), will the pore water then freeze irrespective of pore sizes.
284 While for the powder sample, ice has already penetrated into the sample and occupied
285 about half of the total pore volume when the temperature goes down to about $-40\text{ }^{\circ}\text{C}$.
286 The comparison of the ice content curves of freezing between the powder sample and
287 cylinder samples of the paste CEM III indicates that the crushing of cylinder samples
288 into powders may significantly change the pore connectivity.

289 Another important difference between the powder sample and the cylinder samples
290 of the paste CEM III is that the determined maximum ice content in the powder sample
291 is much higher compared with that in the cylinder samples. The relative difference is
292 up to about 40-50%. However, the marked difference is not found from the comparison
293 of the paste CEM I (Figure 2).

294 The marked difference of the determined maximum ice content between the cylinder
295 and powder samples of the paste CEM III (indicating more pores detected in the powder
296 sample) may be related to the “isolated” pores. It should be mentioned that in LTC
297 studies on cement based materials, there is normally a peak starting at around $-40\text{ }^{\circ}\text{C}$
298 (and it is extended to lower temperatures) on the measured freezing heat flow curves
299 and this peak is often associated with the so-called “isolated” pores, e.g., see [21, 38, 55].
300 Then two types of “isolated” pores can be envisaged: (1) totally isolated pores (which
301 may form due to, e.g., self-desiccation during the hydration); (2) pores connected to
302 very small pore entries in which water does not freeze above about $-40\text{ }^{\circ}\text{C}$, i.e., pores
303 “isolated” by small pore entries. It could be that some of the “isolated” pores in the
304 cylinder samples cannot be fully saturated with water during vacuum saturation due
305 to a very complicated pore structure, e.g., of which pores are less connected and the
306 whole pore system is quite tortuous; while in the powder sample, due to the increased
307 pore connectivity as a result of the sample crushing, some of the initially “isolated”
308 pores can be opened and then they are able to be filled with water during vacuum
309 saturation. Consequently, the overall saturation degree of the “isolated” pores in the
310 powder sample after vacuum saturation could be much higher compared with that in
311 the cylinder samples. Since LTC detects only the water filled pores, it is not surprising
312 that more pores are detected in the powder sample, as observed in Figure 3. The
313 impact of sample crushing on the “isolated” pores is schematically illustrated in Figure
314 4 (the small amount of water in the “isolated” pores, i.e., pore A and pore B in Figure

315 4a and pore C in Figure 4b, is schematic, which may come from, e.g., diffusion and/or
316 unreacted mixing water)¹.

317 The much more marked difference of the ice content between the powder and cylinder
318 samples as found in the paste CEM III (Figure 3) compared with that in the paste
319 CEM I (Figure 2) may indicate that there are probably more “isolated” pores in the
320 paste CEM III than that in the paste CEM I. Combined with the calculated total pore
321 volume, more discussions about the impact of the sample crushing on the detected
322 porosity will be conducted in Section 3.2.

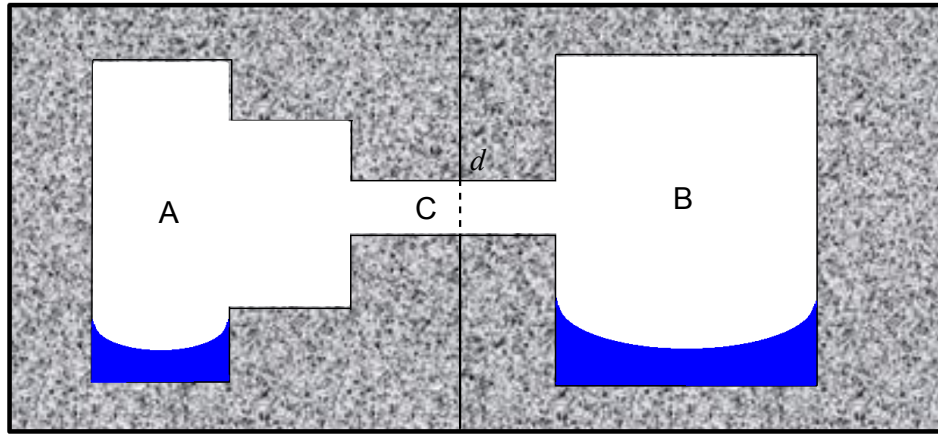
323 From the comparison of the ice content curves for the powder and cylinder samples
324 of the studied cement pastes, it can be concluded that the crushing of a sample into
325 powders increases the pore connectivity. Another important consequence of sample
326 crushing is that it could possibly result into more pores detected by LTC in the crushed
327 powders than that in a big sample, with one possible reason being that the saturation
328 degree of the “isolated” pores in a big sample can be greatly increased after the crushing
329 during saturation.

330 3.2. Total pore volume

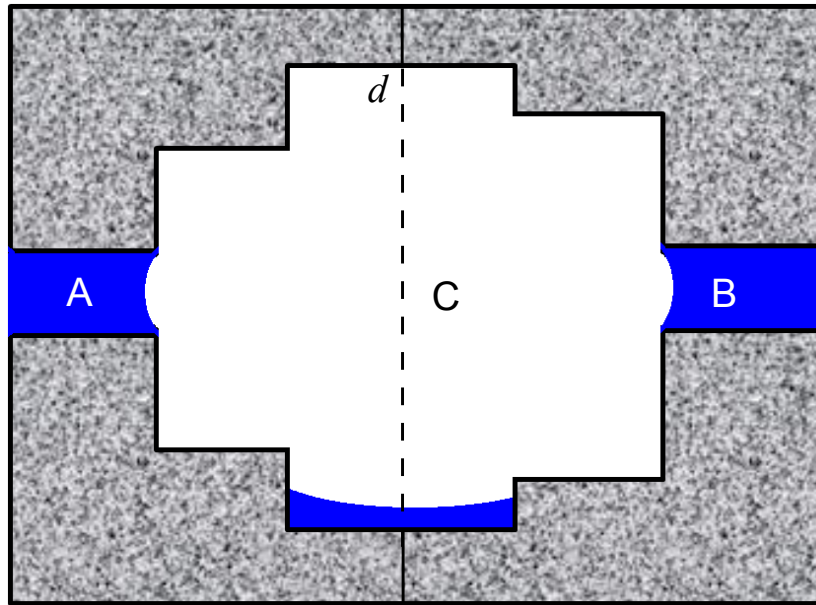
331 The total content of pore water (more accurately, it is ice and the unfreezable water
332 in pores) in each LTC studied sample is calculated by subtracting the content of “bulk”
333 ice (as explained in Section 3.1) from the determined total water content. Further
334 assuming the density of pore ice/water as 1.0 g/ml, the total pore volume of each LTC
335 studied sample is estimated². For the materials measured more than once in this study,
336 the total pore volume is obtained by averaging that of the several measurements. The
337 total pore volume of the LTC studied samples are listed in Table 1. The porosity of the
338 model material MCM-41 and the pastes CEM I and CEM III were also studied by the
339 Dynamic (water) Vapor Sorption (DVS) measurement [58]. The DVS results, together
340 with that of the gravimetric measurements, are also listed in Table 1.

¹ In Figure 4, the “isolated” pores are inside of a rather big hardened cement paste (e.g., a cylinder sample). During vacuum saturation, water may have no access to the totally isolated pores, e.g., in (a) and probably small pore entries may block or prevent water from penetrating into the “isolated” pores, e.g., in (b). However, if the sample is crushed along any line between pore A and pore B, e.g., the line d , then all the “isolated” pores are more easily to be filled with water during saturation (corresponding to powder samples).

² The density of ice is temperature dependent [56], see Eq.C.1. As the density difference in the concerned temperature range (0 °C to about -40 °C) is less than 1.0%, the temperature dependence is not considered in the estimation for the sake of simplicity. As the density of ice is less than 1.0 g/ml [56] and maybe for unfreezable pore water as well [57], this assumption probably results into underestimation of the pore volume, with the error being not greater than about 8.0% of the estimated value .



(a)



(b)

Figure 4: A schematic illustration of “isolated” pores. (a) pore A, pore B and pore C are three connected but totally isolated pores; (b) pore C is connected to but kind of “isolated” by pore A and pore B with very small sizes.

Table 1: Total pore volume of the MCM-41 and the pastes CEM I and CEM III obtained from different methods. The values are expressed in the unit of milliliter per gram dry material (ml/g).

		LTC ^a	Gravimetric measurement ^b	DVS ^{b,c,d}
MCM-41		0.72	-	0.77
CEM I	cylinder	0.20	0.21	-
	powder	0.21	-	0.20
CEM III	cylinder	0.23	0.23	-
	powder	0.29	-	0.35

Note: (a). the volume of the big pores (e.g., air voids, as explained in Section 3.1) in the cement pastes is not included. From the calculation on the cylinder samples, the volume of the the big pores in pastes CEM I and CEM III is about 0.007 and 0.003 ml/g, respectively; (b). density of pore water is assumed to be 1.0 g/ml; (c). moisture content determined during desorption at the RH of 0.95 which is almost the upper limit that the instrument can work with; d). the age of the cement pastes is about 6-8 months when measured.

341 As can be seen from the results presented in Table 1, the total pore volume of the
342 paste CEM I obtained from different methods are comparable. For the paste CEM III,
343 the total pore volume determined from the gravimetric measurements is comparable to
344 that of the cylinder samples by LTC; while the total pore volume determined from DVS,
345 which is much higher than that of the cylinder samples by LTC, is more comparable
346 to that of the powder sample by LTC. The comparison between the results obtained
347 from the gravimetric and the DVS measurement for the cement paste CEM III is in
348 agreement with that of the results obtained from LTC studies on the powder and
349 cylinder samples, i.e., the sample crushing is concluded to have an impact on the
350 detectable pore volume. However, by noting that the differences between the results
351 found for the paste CEM III using different samples are not observed on the paste CEM
352 I, it should be mentioned that the effect of the sample crushing is probably dependent
353 on the (porosity) properties of the studied material.

354 3.3. Pore size distribution

355 With the obtained ice content curves of the freezing and the melting process as
356 presented in Section 3.1, it is possible to calculate both the pore entry size and the pore
357 interior size distribution of the studied material. Refer to AppendixC for explanation
358 of the calculation procedures.

359 For the measured samples in this study, it should be mentioned that the meaningful
360 temperature range of the freezing curves that can be used to calculate the pore entry
361 sizes is only from around -10 °C to about -40 °C. The pores with entry sizes bigger

362 than that corresponding to a temperature depression of about $-10\text{ }^{\circ}\text{C}$ are not detected
363 because of supercooling. That is, no ice has been formed before reaching about $-10\text{ }^{\circ}\text{C}$
364 and hence no calculation can be made. When the temperature goes down to about -40
365 $^{\circ}\text{C}$, the homogenous nucleation becomes significant [36, 37, 59] and all the freezable pore
366 water would freeze independent of the pore (entry) sizes. That is, the ice formed around
367 the homogenous nucleation temperature does not indicate any pore size information.
368 For the MCM-41, the pore (entry) size is too small to be obtained from the freezing
369 curve as the main ice content is formed below about $-40\text{ }^{\circ}\text{C}$. For the studied cement
370 pastes, the pore entry radii that can be obtained are between about 2.4 nm to about
371 5-6 nm (using Eq.C.1). As the obtained pore (entry) size range is quite limited from
372 the ice content curves of freezing of the cement pastes, the results are not presented.
373 In the following, only the ice content curves of melting are used to calculate the pore
374 interior size distribution.

375 3.3.1. Model material MCM-41

376 For the model material MCM-41, the mean ice content of the two measured samples
377 (melting curves) as presented in Figure 1 is used for the calculation of the pore size
378 distribution. The thermal lag between the calorimetric block and a tested sample is
379 assumed to be negligible. The calculated pore size distribution (PSD) curves of the
380 MCM-41 are presented in Figure 5. The pore radius corresponding to the peak in the
381 calculated PSD curves, sometimes referred to as the most frequent pore radius R_{max}
382 [60], is about 1.7 nm and 2.7 nm assuming cylindrical and spherical pores, respectively.
383 It should be mentioned that the pores of MCM-41 products are often assumed to be
384 very close to cylindrical shape [43, 44, 46, 61]. Including the spherical shape assumption
385 in the calculation is for comparison purpose only. The R_{max} of 1.7 nm based on the
386 cylindrical pore assumption is close to the value provided by the producer, i.e., the
387 diameter of 3.0 nm.

388 3.3.2. Cement pastes

389 For the cylinder samples of each cement paste, the representative ice content is
390 calculated by averaging the results of the three cylinder samples (melting curves) of
391 each paste (Figure 2 and Figure 3). One may argue that the thermal lag for the cylinder
392 samples might be different from that for the powder sample. If the thermal lag is
393 different, it should be more obvious at very low temperatures. For all the measured
394 cement paste samples, there is a characteristic peak corresponding to homogenous
395 nucleation, which starts around $-40\text{ }^{\circ}\text{C}$ and is extended to several degrees lower, on
396 the heat flow curves of freezing [55, 62]. By comparing the starting temperature of
397 the characteristic peak due to homogenous nucleation on the heating flow curves of
398 freezing, it is concluded that the thermal lag for the cylinder samples and the powder
399 sample are close, if there is any. One may also argue that the ionic concentration in

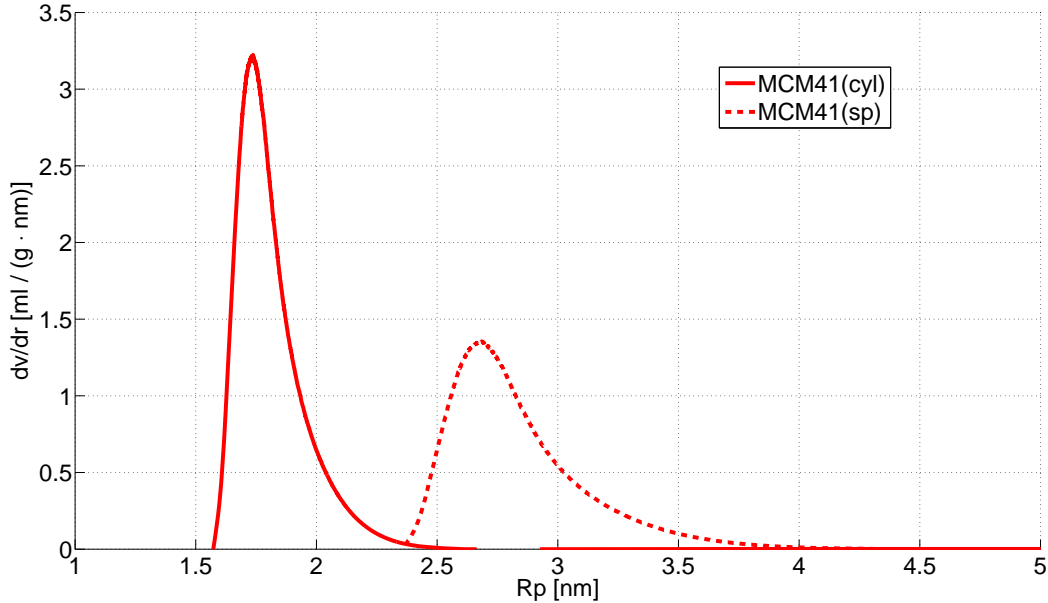
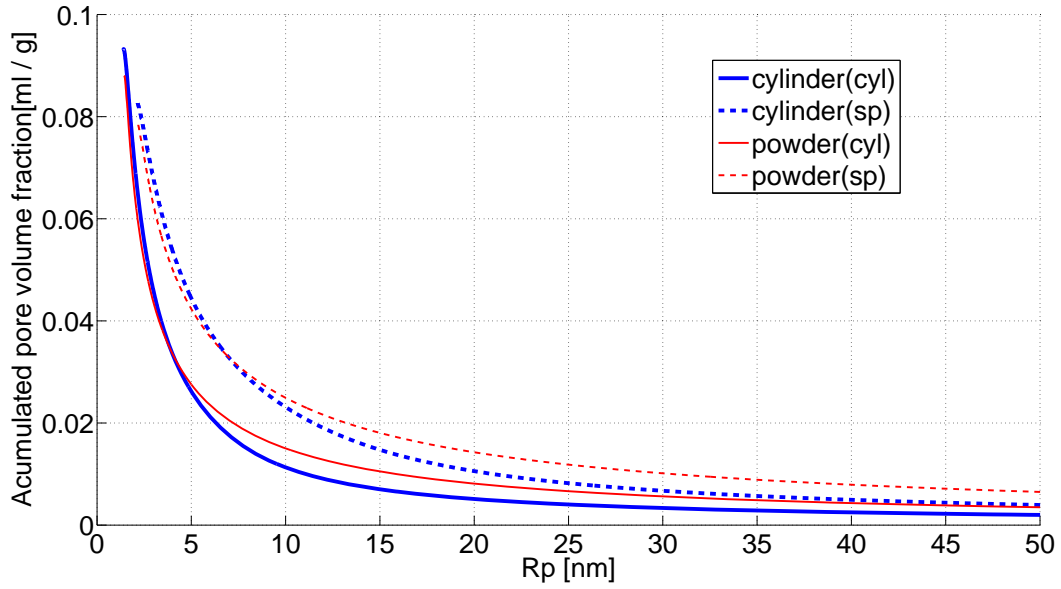


Figure 5: Calculated differential pore size distribution of the model material MCM-41 based on the cylindrical (cyl) and the spherical (sp) pore assumption.

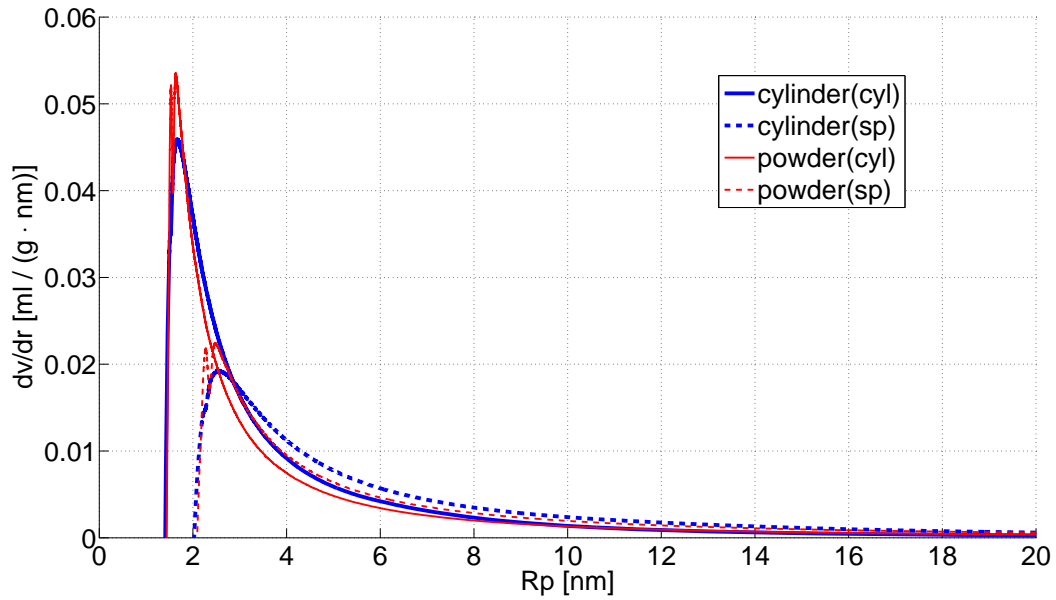
400 the pore solution might be different, since the water content in the powder sample is
 401 higher than that in the cylinder samples. A study [62] indicates that the amount of
 402 curing water has very limited effect in changing the freezing and melting behaviors of
 403 the cement pore solution. In this study, it is therefore assumed that the impact of
 404 the ions on the freezing and melting point depression for the cylinder samples and the
 405 powder sample are more or less the same. In this investigation, both the thermal lag
 406 and the effect due to ions are not considered.

407 The calculated accumulated and differential PSD curves for the pastes CEM I and
 408 CEM III are shown in Figure 6 and Figure 7, respectively. It can be found that the
 409 calculated accumulated PSD curves of the powder sample are generally higher than
 410 that of the cylinder sample, especially for the paste CEM III (Figure 7a), indicating
 411 that the volume of the relatively big pores is higher in the powder sample. However,
 412 from the calculated differential PSD curves, it is found that the portion of small pores,
 413 i.e., with the radii between about 2 to 10 nm for the paste CEM I and the radii between
 414 about 2 to 5-6 nm (depending on the pore shape assumption) for the paste CEM III,
 415 is relatively higher in the cylinder sample.

416 More big pores found in the powder sample compared to the cylindrical ones may
 417 attribute to that the sample crushing opens some “isolated” pores which may be not
 418 fully filled with water before crushing but can be fully saturated after crushing (Section
 419 3.1). Thus, more pores are detected using the powder sample. For the more small

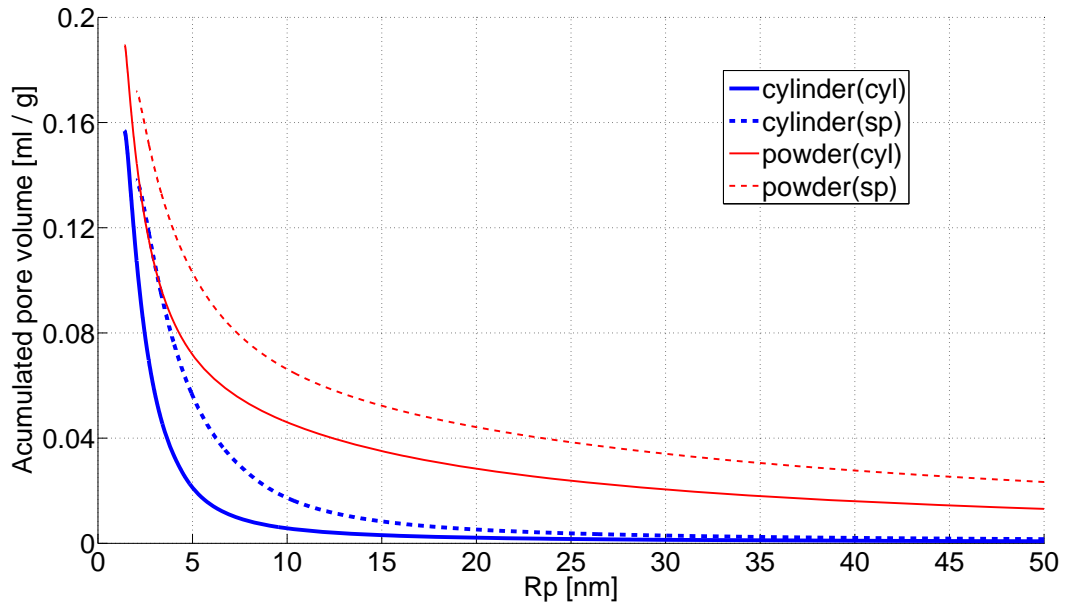


(a) Accumulated pore size distribution

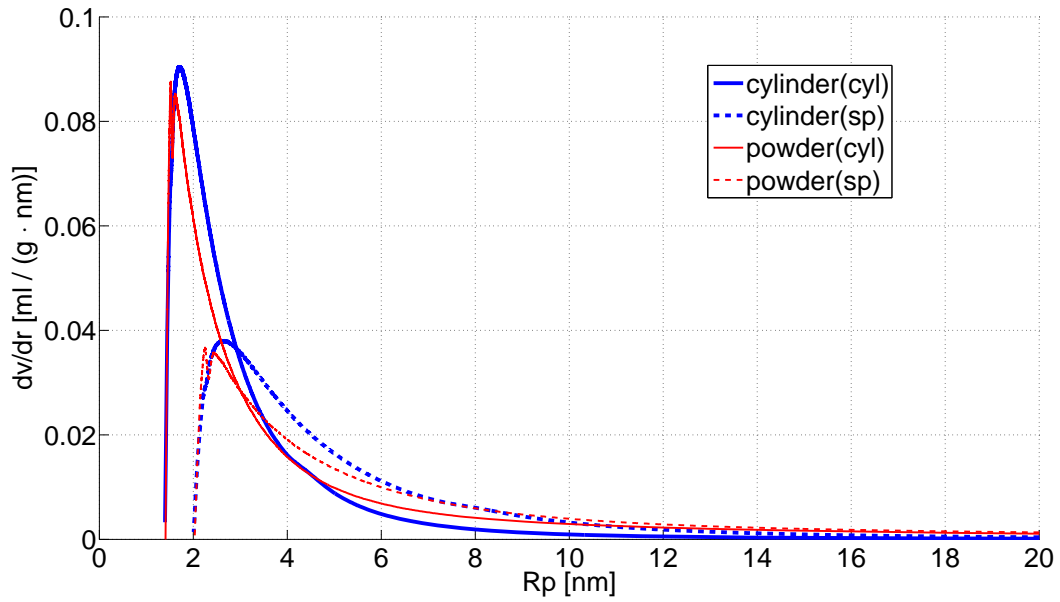


(b) Differential pore size distribution

Figure 6: Calculated accumulated and differential pore size distribution curves of cylinder and powder samples of the cement paste CEM I. The pore shape is assumed to be cylindrical (cyl) or spherical (sp).



(a) Accumulated pore size distribution



(b) Differential pore size distribution

Figure 7: Calculated accumulated and differential pore size distribution curves of cylinder and powder samples of the cement paste CEM III. The pore shape is assumed to be cylindrical (cyl) or spherical (sp).

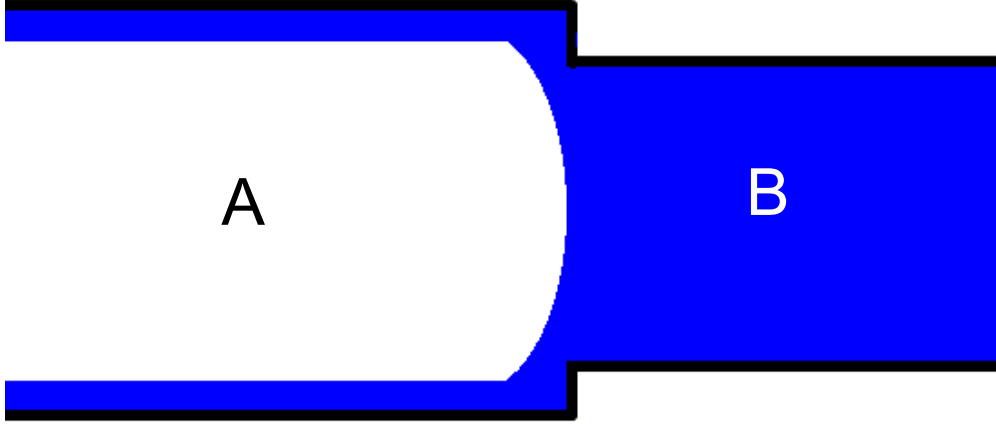


Figure 8: A schematic illustration of the concept of advanced melting, based on the description from [26, 42], refer to Footnote 3 for explanation.

420 pores in the cylinder samples observed from the calculated differential PSD curves,
 421 it is possibly related to the so-called advanced melting phenomenon [42, 63]. It is
 422 normally assumed that for cylindrical pores, the melting would take place in a radical
 423 direction [42], or melting from the side (the direction perpendicular to the diameter)
 424 [26]. However, melting could also take place from the end (the direction parallel to the
 425 diameter), depending on the pore size and connectivity [26, 42]. If melting initiates
 426 from the end, it is the so-called advanced melting. The concept of advanced melting is
 427 schematically shown in Figure 8³. Since the sample crushing changes the connectivity

³ In Figure 8, pore A and B are two cylindrical pores and connected co-axially, with radii $R_A > R_B$. If the melting initiates from the side (the direction from left to right in the drawing), the ice in pore B will melt at a temperature corresponding to $1/(R_B - \delta)$ and then in pore A at a temperature corresponding to $1/(R_A - \delta)$, where δ is the thickness of the unfreezable layer close to the pore wall. That is, the ice in pore A will melt at a higher temperature compared with the ice in pore B (the smaller the curvature, the higher the melting point). But as pore A and B are connected, the melting point of the ice in pore A also depends on the size relation of the two pores. Considering the case when the ice in pore B melts, the water in pore B is in equilibrium with a ice/water interface having a curvature infinitesimally smaller than $1/(R_B - \delta)$. The ice in pore A terminates in a hemispherical cap with a curvature of $2/(R_A - \delta)$. If $2/(R_A - \delta) \geq 1/(R_B - \delta) = 2/[2(R_B - \delta)]$, i.e., approximately $R_A < 2R_B$ (assuming δ is smaller than R_A and R_B), then the ice hemispherical cap is not thermodynamically stable in pore A and the ice in pore A will melt from the end (the direction from top to bottom in the drawing). That is, due to the connectivity of the two pores and if $R_B < R_A < 2R_B$, the ice in pore A could melt from the end rather than from the side. This is the so-called advanced melting phenomenon. In such a case, the melting point of the ice in pore A from the end (corresponding to the curvature of pore B melting, $1/(R_B - \delta)$) is lower than that from the side (corresponding to the curvature of pore A melting, $1/(R_A - \delta)$). Using the same calculation (Eq.C.3, which assumes melting from the side), some big pores may be wrongly calculated as small pores due to the advanced melting.

428 of the pores, the impact of advanced melting on the powder and the cylinder samples
429 may be different. One consequence of the advanced melting on the pore size distribution
430 determination is that big pores may be wrongly calculated as small pores [42], which
431 can somewhat explain the difference between the calculated volume of small pores in
432 the powder and the cylinder samples. Additionally, some of the “isolated” pores in
433 cylinder samples may be not fully saturated as mentioned earlier (Section 3.1). It has
434 been demonstrated in [35] that the freezing/melting point of the water/ice confined in
435 the non-fully saturated pores will be further depressed compared with the condition
436 that the pores are fully saturated. The effect is that the calculated sizes of the pores
437 under non-fully saturation are underestimated. Thus, it could also explain that there
438 are more (calculated or apparent) small pores in the powder sample than what found
439 in the cylinder sample as observed for the studied pastes.

440 4. Conclusions

441 Two types of samples, i.e., in the form of powder and cylinder, were used to study
442 the impact of sample crushing on the detected porosity of hardened cement pastes
443 by low temperature calorimetry (LTC). The studied cement pastes were prepared by
444 two types of cements. The difference between the powder and cylinder samples was
445 compared in terms of the calculated ice content curves, total pore volumes and pore
446 size distribution curves.

447 For the studied cement pastes, the calculated ice content curves of freezing of the
448 powder sample differed from that of the cylinder samples, especially for the paste
449 CEM III. It indicated that sample crushing changed the pore connectivity. Another
450 important difference between the powder sample and the cylinder samples of the paste
451 CEM III was that the determined maximum ice content in the powder sample was much
452 higher compared with that in the cylinder samples, the relatively difference being about
453 40-50%. However, this kind of marked difference was not found from the comparison
454 of the powder and cylinder samples of the paste CEM I. That is, sample crushing
455 could possibly result into more pores detected by LTC depending on the (porosity)
456 characteristic properties of the studied paste.

457 About the marked difference between the calculated pore volume of the powder
458 and the cylinder samples of the paste CEM III, one possible reason could be that
459 some of the “isolated” pores which, presumably, cannot be fully filled with water in
460 the preparation of the cylinder samples. However, sample crushing makes it possible
461 to saturate the pores to a greater extent if the crushing contributes to open up the
462 “isolated” pores. Consequently, more pores can be detected in the powder samples.
463 The argument about the “isolated” pores is supported by the results of gravimetric
464 measurements on cylinder samples and the DVS measurements on powder samples.

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 470 Materials of Lund University for performing some of the tests.

471 **AppendixA. Properties and the chemical composition of cements**

472 The properties and the chemical composition of the cements used in this study are
 473 shown in Table A.1.

Table A.1: Properties and the chemical composition of the two cements used in this study.

		CEM I (CEM I 32.5 R)	CEM III (CEM III/B 42.5 N)
Density	(g/cm ³)	3.06	2.90
Fineness	(cm ² /g)	2905	4635
Water demand	(%)	26.2	32.3
Initial setting time	(min)	185	270
Loss on ignition	(%)	2.1	1.4
SiO ₂	(%)	20.6	29.2
Al ₂ O ₃	(%)	5.6	8.9
Fe ₂ O ₃	(%)	2.4	1.2
CaO	(%)	63.4	48.0
MgO	(%)	1.6	4.8
SO ₃	(%)	2.9	2.6
K ₂ O	(%)	0.7	0.6
Na ₂ O	(%)	0.2	0.2
Cl	(%)	<0.1	<0.1

474 **AppendixB. Trial run of two cycles of freezing and melting processes**

475 As mention in Section 2.2, two cycles of freezing and melting measurements have
 476 been suggested in the context of LTC studies, e.g, see [23, 26, 53]. The first freezing
 477 process is to create some ice crystals and the first melting process is run up to just below
 478 the melting point of macroscopic ice (i.e., the ice in big air voids or on the external
 479 surfaces of the testing sample), e.g., that the samples were heated to -0.05 °C in [26]
 480 and about -0.5 °C in [53]. The macroscopic ice formed in the first freezing process,

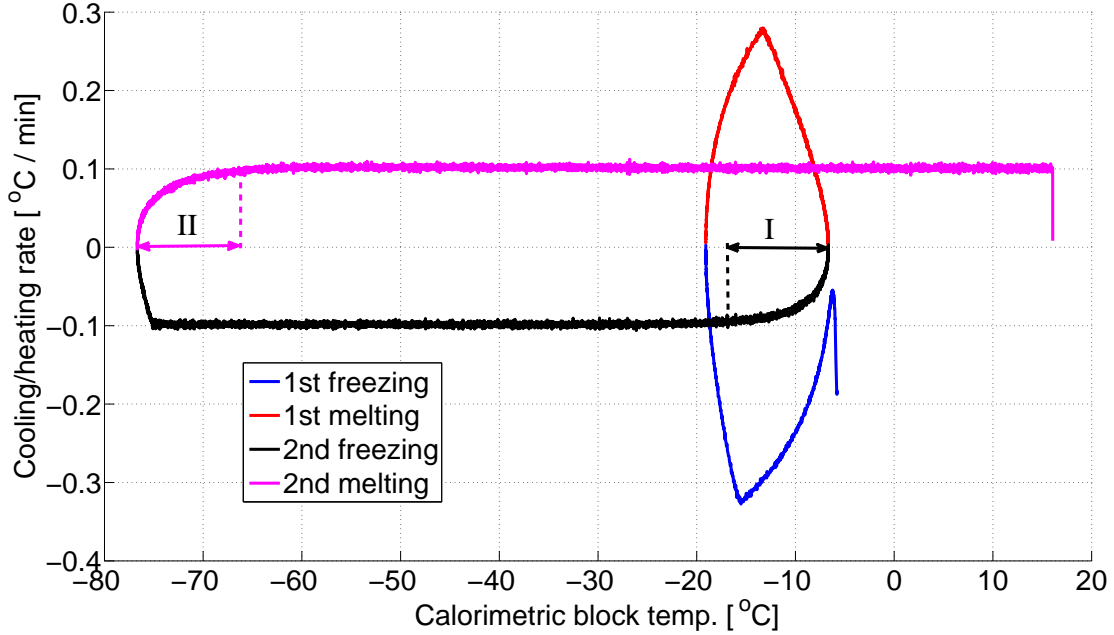


Figure B.1: The cooling/heating rates during a trial run of two cycles of freezing and melting using the employed instrument.

481 which serves as the ice front, then penetrates through the pore network of the sample
 482 during the second freezing process. By adopting the two cycles procedure, the pore
 483 entry and interior sizes can be obtained from the second freezing and melting process,
 484 respectively.

485 A trial run of a two cycles of freezing and melting using the employed instrument
 486 was conducted. The first cycle was run on a relatively high cooling/heating rate, as
 487 suggested in [23, 26, 53], since the main purpose is just to create some macroscopic ice
 488 to be used in the second cycle. The second cycle was run on the set cooling/heating
 489 rate for measurements, i.e., 0.1 °C per minute. The cooling/heating rates during the
 490 trial run are calculated and shown in Figure B.1. It is found that it would take about
 491 10 °C for the cooling/heating rate to be stabilized when the process is changed from
 492 cooling to heating and *vice versa* (as indicated by notation I and II in Figure B.1). The
 493 importance of the stability of the cooling/heating rate during measurements in the
 494 ice content determination has been discussed in [31]. That is, the ice content during
 495 the temperatures with unstable cooling/heating rate may not be calculated properly.
 496 Assuming the testing samples are successfully heated up to just slightly lower than 0 °C
 497 in the first melting process and then the samples are cooled down immediately, the ice
 498 content of the second freezing during the temperature from about 0 °C to about -10 °C
 499 may not be obtained properly, due to the unstable cooling rate during the temperature

500 range. As will be shown later, measurements on the cylinder samples of cement pastes
501 using one freezing and melting cycle show that the supercooling also terminates at
502 around $-10\text{ }^{\circ}\text{C}$ (see Figure 2 and Figure 3). Similar to the two cycles of measurement,
503 the ice content of the freezing process during the temperature from about $0\text{ }^{\circ}\text{C}$ to about
504 $-10\text{ }^{\circ}\text{C}$ also cannot be obtained. That is, using two cycles cannot provide more pore
505 entry information (derived from the ice content curve of the freezing process) compared
506 with that using one cycle. Thus, only one freezing and melting cycle was used for all
507 the measurements.

508 In summary, the benefit of using a two cycles of freezing and melting in the context of
509 LTC studies can only be guaranteed if the cooling/heating rate of the instrument, when
510 changing the process from cooling to heating, and *vice versa*, can be stabilized rather
511 quickly, i.e., notation I and II as indicated in Figure B.1 covering a short temperature
512 range. In this study, limited added value can be obtained by using the two cycles since
513 it will take about $10\text{ }^{\circ}\text{C}$ for the cooling/heating rate to be stabilized when changing
514 the process. Consequently, only one freezing and melting cycle was adopted to improve
515 the efficiency of measurements.

516 **AppendixC. LTC data interpretation procedures**

517 The typical output measured by LTC experiments is the relation between heat flow
518 and the temperature, e.g., see Figure C.1. To derive pore size information, several steps
519 would be needed. The essential parts of the data interpretation procedures including
520 necessary assumptions have been explained in a detailed manner in previous studies [31,
521 35]. The main steps, together with important assumptions, are listed below. Interested
522 readers are advised to find more details in [31, 35].

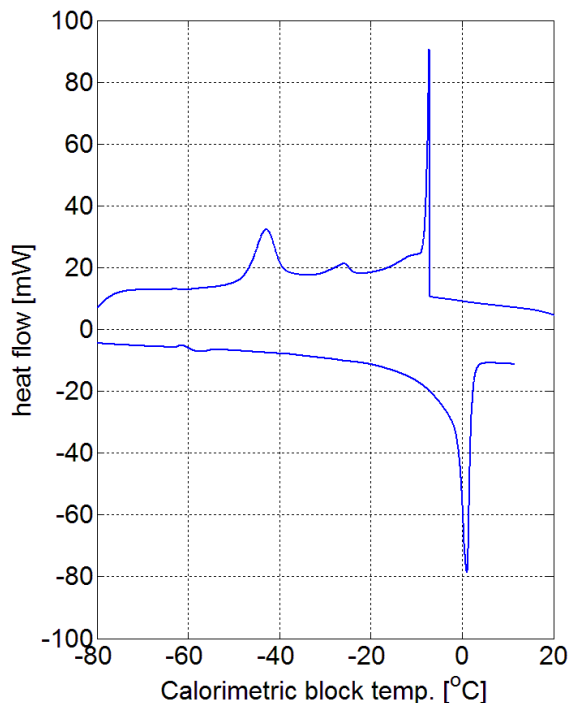


Figure C.1: Typical output from LTC measurements, i.e., the relation between heat flow and the temperature.

523 1. Calculate ice content from the heat flow curve

524 (a) Baseline calculation

525 The heat flow measured by a LTC instrument consists of two parts: (1) the
 526 contribution from the heat capacity of the system (including water and ice in
 527 the concrete sample, and also the concrete skeleton matrix), which is referred
 528 to as the “heat flow baseline” or simply “baseline” in this discussion; and (2)
 529 the contribution due to the phase transition. Hence, in order to calculate
 530 the ice content, one needs to know how much energy goes to form or melt
 531 ice at each temperature level, which basically is obtained by subtracting the
 532 baseline from the measured total heat flow. The determination of the base-
 533 line is complicated by the facts that the heat flow is continuously changing
 534 due to the change in the proportion of ice and liquid water confined in the
 535 pores and the temperature dependent heat capacities of water, ice and the
 536 solid matrix.

537 We propose two different methods, i.e., the “J-baseline” method and the “C-
 538 baseline” method. The “J-baseline” method is a recently proposed method
 539 based on extrapolation by using the accumulated heat curves measured in
 540 the freezing and the melting process. The “C-baseline” method considers

541 the heat capacity of both water and ice and the phase changing behavior
542 under the different testing temperatures. Refer to [31] for the advantages
543 and constraints of each method.

544 (b) Ice content calculation

545 With the obtained baseline, we know how much heat is related to phase
546 change of the water/ice. By dividing the heat related to phase change by
547 the heat of fusion of water, the ice content can be calculated.

548 It should be noted that the water/ice under study in this case is the water/ice
549 confined in small pores. Although bulk water/ice is relatively well studied,
550 there is no general agreement on how to obtain the important temperature
551 dependence of the heat of fusion of water/ice confined in small pores. In
552 this study, the values of the heat of fusion of water as proposed in, e.g.,
553 Ishikiriya et al. [22, 53, 57, 64], are used. Refer to [31] for reasons and
554 discussion.

555 2. Derive pore size information from the calculated ice content

556 (a) Pore volume calculation

557 From Step 1, we know the ice content corresponding to each temperature
558 level. Dividing the calculated ice contents (in mass) by the ice density, we
559 get the ice volume. The density of ice adopted in the pore size distribution
560 calculation is a temperature dependent function, which is [56]

$$\rho_{ice} \approx 0.9167 - 2.053 \cdot 10^{-4}\theta - 1.357 \cdot 10^{-6}\theta^2 \quad \text{g/ml} \quad (\text{C.1})$$

561 where θ is the temperature in Celsius degree.

562 (b) It should be noted that it is normally assumed that there is a portion of water
563 which is very close to the pore walls and it will not undergo phase transition
564 even if the temperature is very low. This unfreezable layer, referred to as
565 δ -layer, is normally treated as consisting of 2-3 layers of water molecules,
566 which corresponds to about 0.8 nm in thickness, e.g., in [33, 53, 65]. The
567 calculated ice volume plus the volume of this unfreezable layer of water is
568 the pore volume.

569 (c) Pore size calculation

570 To calculate the pore sizes, thermodynamic relation between the depressed
571 freezing point of water confined in a pore and the pore size should be used,
572 e.g., see [23, 32, 33, 36].. In addition, assumptions have to be made with
573 respect to pore shape. In this study, the classical assumptions of cylindrical
574 pores and spherical pores are followed. It is well acknowledged that the
575 real pore size/shape of cement paste materials is rather complicated and
576 the actual pore shape may deviate significantly from the ideal pore shape

577 assumed. However, with no better generally accepted model, the cylindrical
578 and spherical pore shape assumptions are used.
579 Based on different pore shape assumption and the process (either freezing
580 or melting), Brun et al. [33] proposed two equations accounting for the
581 relation between the pore size and the freezing/melting depression. The two
582 equations are used in this study, i.e.,

$$R_p = -\frac{64.67}{T - T_0} + 0.57 \quad (\text{C.2})$$

$$R_p = -\frac{32.33}{T - T_0} + 0.68 \quad (\text{C.3})$$

583 where R_p (nm) is the pore size; T and T_0 is the freezing/melting point of
584 pore water/ice and bulk water/ice, respectively. For the cylindrical pore
585 assumption, Eq.C.2 is used for the freezing process and Eq.C.3 is used for
586 the melting process; while for spherical pore assumption, Eq.C.2 is used for
587 both the freezing and the melting process.

588 (d) With the obtained pore volume and pore size at each corresponding temper-
589 ature level, the pore size distributions (e.g., differential and accumulated)
590 can be calculated.

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