

1 **Incorporation of Fe@Au nanoparticles into multiresponsive pNIPAM-AAc**
2 **colloidal gels modulates drug uptake and release.**

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1 **ABSTRACT**

2 Here, a synthetic method has been optimized for the synthesis of thermo and pH responsive
3 poly(N-isopropylacrylamide-co-acrylic acid) nanogels which are subsequently loaded with
4 Cytochrome C using a modified breathing-in mechanism. Physico-chemical properties mapped
5 using dynamic light scattering (DLS) and differential scanning calorimetry (DSC) confirm the
6 swelling/de-swelling kinetics as reversible with a volume phase transition temperature (VPTT) of
7 ~ 39 °C. Fe@Au nanoparticles were incorporated inside the nanogel networks using two different
8 methods- coating and in-situ growth. The latter bears closer resemblance to the nanogels only
9 while the former follows the trend of bare Fe@Au nanoparticles. High loading (~96%) and
10 encapsulation (500 µg/mg of nanogels) of Cytochrome C were obtained. Release experiments
11 performed using a dialysis setup and monitored using UV-vis spectroscopy show the highest
12 release at 40°C and pH 3.2 (high temperature, low pH), with maximum release from the Fe@Au
13 coated nanogels that also show a reverse swelling-collapse trend. The location of the drug,
14 incorporation and presence of Fe@Au nanoparticles and drug incorporation method are found to
15 control both the drug release mechanism and kinetics.

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21 **KEYWORDS** ((Nanogels, core-shell nanoparticles, Volume phase transition temperature,
22 programmed drug release, breathing-in))

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INTRODUCTION

Colloidal gels in the nano regime, more commonly referred to as nanogels (NGs), have attracted much attention in drug delivery applications especially for stimuli-responsive release; i.e.; release of cargo molecules under the influence of temperature, pH, ionic strength or other external parameters.[1, 2] NGs refer to swollen nanosized networks formed by physical or chemical cross-linking of hydrophilic or amphiphilic polymer chains.[3] These offer several advantages over conventional drug delivery systems such as liposomes, microspheres, cyclodextrins and so on,[4, 5] not only by providing a finer temporal control over drug release due to their large surface area but also by allowing longer circulation times and targeting properties upon suitable functionalization. Moreover, NGs have been shown to exhibit high loading and encapsulation efficiencies. Although loading efficiency gives an indication of the thermodynamic distribution of the drug, it is not sufficient to indicate delivery vehicle stability against leakage. [6] In this regard, cross-linked NGs are known to provide high encapsulation stability to the drug molecule, rendering them suitable for long term use.[7]

One of the most common methods that has been used for synthesizing thermo-sensitive NGs is precipitation polymerization, named after the fact that the polymer chain - upon reaching a critical length - collapses upon itself, producing pre-cursor particles.[8] This happens above the lower critical solution temperature (LCST) of the polymer.[9] In an aqueous solution, the polymer is in the hydrated state below the LCST, while it becomes hydrophobic above LCST. In a similar way, the cross-linked NGs obtained from this polymer swell in water under a critical temperature and collapse above it. This temperature

1 is called the volume phase transition temperature (VPTT) of the NG.[10] The transition
2 usually happens over a temperature range instead of a single temperature owing to different
3 crystalline and amorphous domains in the polymer, branching and side-chains, molecular
4 weight distributions among other factors. [11] However, this temperature driven transition
5 allows for release of cargo molecule at a desired rate by fine tuning the physico-chemical
6 properties of the NGs.

7 Multi-response is introduced in these NGs by incorporating pH-dependent co-monomers.
8 One of the frequently studied temperature and pH-dependent polymeric NGs is that of
9 poly (N-isopropylacrylamide-co-acrylic acid) (pNIPAm-AAc). [12, 13] Although
10 monomer addition, monomer-comonomer ratios can influence the final form of the NGs,
11 the core-shell morphology of the NGs is one of the frequently studied forms. It allows for
12 multiple phase transition behavior with temperature besides adding response to two
13 external stimuli, viz. temperature and pH. The cross-linked NGs also show a tendency of
14 undergoing compression owing to a cross-link gradient in the shell.[8]

15 Although there exist several synthesis methods to control the size and physico-chemical
16 properties of NGs,[7] pNIPAm-AAc NGs have rarely been investigated to understand their
17 morphological changes under various synthetic conditions and optimize the synthesis
18 process. Moreover, there exists no comprehensive study highlighting the dependence of
19 one synthetic parameter on the other with the primary aim to reduce the size of the gels in
20 the nano regime. The size reduction among other factors is an essential requirement for
21 effective drug delivery applications to avoid phagocytotic sequestration. Recent
22 investigation using poly(acrylic acid) (PAA) nanogels with AuNPs has shown that gels
23 accumulated in the liver and spleen because of their capture by phagocytic cells.[14]

1 Further, the need to optimize the synthesis route and identify the most important synthetic
2 parameters that influence the properties arises with a growing interest in these NGs towards
3 hyperthermia applications when coupled with magnetic nanoparticles.[15] If the VPTT of
4 these NGs can be maintained higher than the normal body temperature, they can be in the
5 circulation long enough to reach the site of action, where these will collapse owing to
6 hydrophobic change at temperatures above VPTT. A pH responsive release can be utilized
7 in drug delivery in intracellular compartments, such as the late endosomes (pH=5) or
8 lysosomes (pH 4.5-5), or can be tuned to respond to the slightly acidic extracellular fluid
9 surrounding tumors (pH=6.5-7.2). [16] The dual response from both temperature and pH
10 allow controlled release of a drug of interest in response to both the stimuli. In this respect,
11 reversible swelling-collapse behaviour of the NGs determines the probability of repeated
12 cycles of controlled release. However, reversibility of the NGs is a challenge to control
13 owing to different degrees of collapse of the chains and conformational architecture of the
14 polymers. Additionally, incorporation of multifunctionality can also be achieved via
15 coupling NGs with magneto-plasmonic nanoparticles (NPs), whereby introducing imaging
16 and targeting modalities along with stimuli sensitivity in one hybrid system.[17-19]
17 Here, we report a comprehensive study of the synthesis of pNIPAm-AAc NGs using free-
18 radical emulsion polymerization that show reversible swelling-collapse behaviour. The
19 influence of stabilizer and cross-linker have been investigated to understand the physico-
20 chemical properties of the NGs. In order to incorporate imaging and targeting modalities,
21 the NGs Fe@Au core@shell NPs were incorporated into the NGs using two methods –
22 coating and in-situ growth, resulting in enhancement of drug release properties. The NGs
23 and Fe@Au incorporated NGs were subsequently loaded with a heme protein- Cytochrome

1 C (Cyt C), a model protein drug- to perform an array of release studies. The effects of both
2 temperature and pH on the release of Cyt C have been studied to determine the dominating
3 factor for release. The effect of drug localization and hybrid composition on the drug
4 release kinetics and mechanisms have been studied. The NGs and their hybrids described
5 herein are promising candidates for hyperthermia applications when coupled with magnetic
6 NPs owing to stability and stimuli programmed release over sustained time periods.

7 **EXPERIMENTAL**

8 ***Materials and Methods***

9 N-isopropylacrylamide (NIPAm), acrylic acid (AAc) ($d= 1.051 \text{ g mL}^{-1}$), Cyt C from bovine
10 heart, sodium dodecyl sulphate (SDS), potassium persulphate (KPS), N, N' –
11 Methylenebis(acrylamide) (BIS), iron pentacarbonyl ($\text{Fe}(\text{CO})_5$, 99.99%), octadecene
12 (ODE, 90%), oleylamine (OAm, 70%), chloroauric acid (99.999%), sodium citrate and O-
13 [2-(3-Mercaptopropionylamino)ethyl]-O'-methylpolyethylene glycol (PEG-SH) of
14 molecular weight 5000 Da were purchased from Sigma Aldrich. n-Hexane and
15 hydrochloric acid (HCl 37% fuming) were purchased from Merck Millipore. Sodium
16 hydroxide (pellets AnalaR NORMAPUR® ACS) was purchased from VWR. NIPAm was
17 recrystallized before further use (described below) while the other chemicals were used as
18 received. All solutions were prepared using distilled de-ionized water (MQ water,
19 resistivity $\sim 18.2 \mu\Omega\text{-cm}$) purified by Simplicity® Millipore water purification system which
20 was further purified using $0.45 \mu\text{m}$ syringe filters. Cellulose dialysis tubing (Sigma Aldrich)
21 with a MWCO of 14kDa was used for performing dialysis, both for purification of the NGs
22 and the release studies.

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1 ***Recrystallization of NIPAm***

2 NIPAm was recrystallized in order to remove impurities that could inhibit the polymerization
3 reaction. The protocol has been adapted and modified from the one reported by Wu *et al.* [20] In
4 a typical recrystallization setup, 5 g of the monomer, NIPAm, was dissolved in 50 mL of n-Hexane
5 at 110°C in a one-necked glass flask equipped with a water condenser and the reaction was run for
6 2 hours. Thereafter, the flask was directly put into an ice bath for 30 minutes to allow
7 recrystallization of the purified monomer. This solution was then filtered using ϕ 90 mm filter
8 paper circles, yielding the pure monomer. After drying the purified NIPAm, it was stored at -20°C
9 to prevent absorption of moisture.

10 ***Synthesis of pNIPAm-AAc NGs***

11 pNIPAm-AAc NGs were synthesized using free radical emulsion polymerization.[21] A
12 molar composition of 85 % PNIPAm, 10 % AAac and 5 % BIS was used. [22] In essence,
13 NIPAm (1.6mMoles) and BIS (90.8 μ Moles) were put directly into the reactor under
14 nitrogen atmosphere. Thereafter, 10mL of SDS-solution with different concentrations (2
15 mM, 4 mM, 5 mM, 5.5 mM) was added and the solution left to stir under nitrogen flow for
16 30 minutes. Prior to addition of the initiator KPS (400 μ L of 103.6 mM), AAac (126 μ L of
17 1.46 M) was added into the solution. The reaction was allowed to run for 3 hours. The NG
18 solution was poured into a pre-washed dialysis tube (MWCO 14kDa) and dialysed
19 overnight to remove unreacted monomers and residual reactants.

20 ***Synthesis of Fe@Au NPs***

21 Fe@Au NPs were synthesized using a previously reported protocol developed by the
22 authors. [17] In essence, a mixture of ODE (50 mL) and OAm (740 μ L) was degassed under
23 Ar atmosphere and vigorous stirring at 120°C for 30 min. The temperature was raised to

1 180⁰C, following which, 1.8 mL of Fe(CO)₅ was injected and the reaction was continued
2 for 20 min. After cooling down to room temperature, the magnetic bar coated with Fe NPs
3 was washed with a 1:2 ratio (by volume) of hexane and acetone. Fe NPs were magnetically
4 separated, washed with acetone and dried in a stream of nitrogen. 5 mg of dried Fe NPs
5 were dissolved in 10 mL of 10 mM sodium citrate solution using sonication at 80⁰C for
6 half an hour. Citrate stabilized Fe@Au NPs were synthesized by dropwise addition of 10
7 mL of 1.5 mM chloroauric acid to the Fe seed NPs under vigorous stirring. The reaction
8 was allowed to run for 20 minutes. Thereafter, the solution was cooled down to room
9 temperature, and Fe@Au NPs were magnetically separated to remove free Au NPs.
10 PEG coating of the Fe@Au NPs was done using a method reported previously by the
11 authors.[17] Briefly, 2mg of PEG-SH was mixed with 5mg of Fe@Au NPs in a total volume
12 of 5ml MQ water and left to stir for 1 hour. PEG coated Fe@Au NPs were collected by
13 centrifugation.

14 ***Incorporation of Fe@Au NPs within NGs***

15 Two different methods were used to prepare combinations of NGs with Fe@Au NPs. The first
16 method was based on post-synthesis coating, while in case of the second method, the NGs were
17 grown atop the Fe@Au seeds via heterogeneous nucleation.

18 **Coating**

19 In order to coat Fe@Au NPs with a representative NG, 3.3 mg of the lyophilized NG was
20 added to 5 ml solution of Fe@Au NPs (concentration of 1mg/ml) and left to stir at 500 rpm
21 for 2 hours. Thereafter, the NG coated Fe@Au NPs were separated using centrifugation at
22 14,500 rpm for 20 minutes. The sample would be referred to as Fe@Au_NG_c.

1 **In-situ Method**

2 A similar protocol was used as reported earlier for the NGs. A molar composition of 85 %
3 PNIPAm, 10 % AAc and 5 % BIS was used. In essence, NIPAm (0.7 mMoles) and BIS
4 (40.2 μ Moles) were put directly into the reactor under nitrogen atmosphere. Thereafter,
5 1mg of PEG coated Fe@Au NPs and 5mL of 4mM SDS-solution were added and the
6 solution left to stir under nitrogen flow for 30 minutes. Prior to addition of the initiator KPS
7 (200 μ L of 210.7 mM), AAc (54 μ L of 1.39 M) was added into the solution. The reaction
8 was allowed to run for 3 hours. The NG solution was poured into a pre-washed dialysis
9 tube (MWCO 14kDa) and dialysed overnight to remove unreacted monomers and residual
10 reactants. The sample would be referred to as Fe@Au_NG_i.

11 ***Loading of Cyt C***

12 Loading studies of Cyt C were performed primarily with breathing-in mechanism using
13 lyophilisation.[23]. For a typical loading experiment, 20 mg of the freeze dried NG was
14 imbibed with Cyt C solution (10ml, 8.1 μ M) and left to stir for 3 hours. Thereafter, the
15 solution was poured into a pre-washed dialysis tube (MWCO 14kDa) and dialysed
16 overnight to remove the unbound Cyt C.

17 For loading Fe@Au_NG_c with the protein, 5mg of NG coated Fe@Au NPs were imbibed
18 with Cyt C solution (1ml, 8.1 μ M) and left to stir for 2 hours. On the other hand, 1.7 mg of
19 Fe@Au_NG_i was imbibed with Cyt C solution (3ml, 40.6 μ M) and left to stir for 2 hours
20 Afterwards, Cyt C loaded Fe@Au_NGs were separated from the unbound drug using
21 centrifugation.

22 Loading efficiencies (L.E., %) and encapsulation efficiencies (E.E., mg drug per mg of NG)
23 of Cyt C were calculated using the following equations.

1

$$L.E. = \left(\frac{C_{cyt,0} - C_{cyt,t}}{C_{cyt,0}} \right) * 100$$

2

3

$$E.E. = \frac{C_{cyt,0} * L.E.}{100 * C_{NG}}$$

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6 Where $C_{cyt,0}$ is the concentration (mg/ml) of Cyt C at the start of loading, $C_{cyt,t}$ is the final
7 concentration (mg/ml) of Cyt C after loading, C_{NG} is the concentration (mg/ml) of the NG
8 or NG coated Fe@Au NPs, concentrations of Cyt C being determined using the calibration
9 curve (Figure S1, Supporting Information) or the absorbance method as applicable.

10 ***Release studies of Cyt C***

11 After loading NGs with Cyt C, these were subjected to different release media to understand
12 the effect of both temperature and pH on the release kinetics. Three different biologically
13 relevant release conditions were simulated – high temperature (40°C), low pH (3.2) and
14 high temperature along with low pH (40°C, pH 3.4) using MQ water and tuning the pH
15 using different molar ratios of 1M NaOH and 1M HCl. In case of the Fe@Au NPs
16 incorporated into NGs, Cyt C release was monitored only at high temperature along with
17 low pH (40°C, pH 3.4). The release medium was maintained at a temperature slightly above
18 VPTT to account for heat losses during the time study and also to ensure maximum collapse
19 of the NGs.

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21 ***Characterization techniques***

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1 **Proton Nuclear Magnetic Resonance (¹H NMR)**

2 ¹H NMR spectra was recorded in a Bruker Advance DPX400 instrument. Lyophilized NG
3 (2 mg) was suspended in D₂O (0.8 ml) and the spectra was recorded with 128 scans at 25
4 °C. The reference peak was locked in, at 4.80 for D₂O. Chemical shifts (δ) were reported
5 in ppm.

6 **Scanning (Transmission) Electron Microscopy (S(T)EM)**

7 S(T)EM images were acquired using a Hitachi S-5500 electron microscope operating at 30kV
8 accelerating voltage. TEM images were obtained in bright field mode. TEM grids were prepared
9 by placing several drops of the solution on a Formvar carbon coated copper grid (Electron
10 Microscopy Sciences) and wiping immediately with Kimberly-Clark kimwipes to prevent further
11 aggregation owing to evaporation at room temperature. For studying the temperature effect on the
12 NGs, NG or Fe@Au incorporated NG solutions were heated to 45°C, just prior to placing drops
13 on the TEM grid.

14 **Dynamic light scattering (DLS) and zeta potential measurements**

15 The size distribution and zeta potential of the NGs were measured using a Malvern
16 Zetasizer Nano-ZS instrument, and the manufacturer's own software. All measurements
17 were done in aqueous solutions and results were averaged over triplicate measurements.

18 **Ultraviolet-visible spectroscopy (UV-vis) measurements**

19 UV-vis spectra were acquired with a UV-2401PC (Shimadzu) spectrophotometer. The
20 spectra were collected over the spectral range from 200 to 800 nm.

21
22 ***Differential Scanning Calorimetry (DSC) studies***

23 DSC studies were performed using a TA Instruments Q2000 DSC. The scan rate was 5 °C
24 min⁻¹ for both heating and cooling curves and the samples were scanned in the temperature
25 range 5–45 °C. The NG solution was loaded in a Hermetic Aluminium pan while the

1 reference pan was kept empty. The data were analysed using TA Instruments Universal
2 Analysis 2000 © software.

3 **RESULTS AND DISCUSSION**

4 pNIPAm-AAc NGs with a wide range of sizes and physico-chemical properties were
5 synthesized using free radical emulsion polymerization. Polymerization occurs via
6 homogeneous nucleation, which is initiated by the sulphate radicals generated from the
7 initiator.[24] Common steps of radical propagation and chain growth continue yielding
8 critical polymer chain length. These collapse upon themselves, forming precursor particles.
9 These collapsed particles grow by a combination of the following mechanisms viz
10 aggregation of other precursor particles, by being captured by existing particles, by
11 capturing growing oligoradicals and by monomer addition.[8] A representative ¹H NMR
12 spectrum of a representative NG showing that polymerization has proceeded onwards from
13 its precursor (NIPAm) to afford pNIPAm can be found in the Supporting information
14 (Figure S2). The NGs are stabilized by Coulombic repulsion from the charge imparted by
15 the stabilizer (surfactant).

16 The results obtained for these NGs are sub-divided into the following three subsections that
17 first reveal the size optimization results and their physico-chemical properties, followed by
18 loading and encapsulation efficiencies of Cyt C and finally culminating into the release
19 studies performed under physiological conditions.

20 ***Size optimization of the NGs and physico-chemical properties***

21 Size and surface charge of the NGs play a major role in drug delivery applications – NPs
22 smaller than 5.5 nm are removed through renal clearance mechanism while particles larger

1 than 200 nm are sequestered by phagocytotic cells of the spleen.[25, 26] In the synthetic
 2 method described above, among different parameters, the influence of stabilizer (SDS)
 3 concentration, cross-linker (BIS) concentration and relative mole ratios of the monomers
 4 and cross-linker (NIPAm, AAc and BIS) have been studied. Figure 1 (a) shows the variation
 5 of size of different NGs synthesized by modifying the reaction parameters, where the error
 6 bars indicate a measure of the polydispersity of the particle distribution. Our results are in
 7 close allegiance to those observed by Lyon et al[27], although their group has controlled
 8 monodispersity as the most important parameter. Here, we report control of physico-
 9 chemical properties of the nanogels as a function of several synthetic parameters outlined
 10 in Table 1.

11

NIPAm (mol %)	AAc (mol %)	BIS (mol %)	SDS (mM)	Size ± PDI (nm)
85	10	5	2.0	593.10 ± 0.10
82	10	8	2.0	376.10 ± 0.10
82	10	8	3.0	309.10 ± 0.02
85	10	5	5.5	182.30 ± 0.10
85	5	10	1.6	378.20 ± 0.03
89	5	6	1.6	415.70 ± 0.02
85	10	5	4.0	412.60 ± 0.07
85	10	5	5.0	231.70 ± 0.06

12

13 An increase in the BIS mole percent (from 5 to 8%) causes a substantial decrease in the
 14 size of the NGs (A-B), while a coupled increase in the SDS concentration (B-C) causes a

1 larger collapse above the VPTT, although the decrease in initial size is not sharply evident.
2 This effect can be attributed to a higher cross-linking density, causing a greater collapse.
3 On the other hand, increasing the SDS concentration from 2mM to 5.5mM (while keeping
4 the BIS mole percent constant) (A-D) causes a dramatic decrease (70%) in the size of the
5 NGs. The main role of SDS is to prevent fusion of hydrophobic nuclei during
6 polymerization.[28] This refers to initial curtailing of the size of the nucleation centres,
7 which further grow by oligomer or monomer addition. A higher initial concentration of
8 SDS provides higher charge stabilization in addition to a denser packing around the
9 incipient nucleation centres, whereby limiting the growth of the NGs owing to electrostatic
10 stabilization. The former (A) shows a volume collapse of ~45% above VPTT while the
11 latter (D) shows a volume collapse of ~95%. This reflects the fact that the hydrophobic tails
12 of the SDS chains interfere constructively in increasing the hydrophobicity of the NGs
13 above VPTT, whereby causing a more efficient collapse.

14 The mole ratios of the monomer (NIPAm), co-monomer (AAc) and cross-linker (BIS) in
15 the initial reaction mixture also determine the incorporation ratios in the final NGs, whereby
16 effecting their size and collapse properties. In a typical experiment, the mole ratios were
17 varied from 85% NIPAm : 5% AAc : 10% BIS (E) to 89% NIPAm : 5% AAc : 6% BIS (F),
18 while keeping the SDS concentration constant. An increase in the initial size (~10%) is
19 explained through a decreased cross-linking density and reduced charge stabilization from
20 the acidic groups of the AAc blocks. On the other hand, there is a slight decrease (~3%) in
21 volumetric collapse efficiency on increasing the NIPAm content. This might indicate an
22 increase in hydrophobic domains when heated above the VPTT. The effect is aided by a

1 lower cross-linking density yielding a loosely structured NG thereby needing stronger force
2 to bring together the increased hydrophobic domains during collapse.

3 The SDS concentration was found to be the most dominating factor determining the size of
4 the NGs. Figure 1(b) shows the variation of the size of the NGs as a function of initial SDS
5 concentration, (error bars indicating polydispersity of particle distribution) while all other
6 parameters were kept constant. As evident from Figure 1(b), the size of the synthesized
7 NGs decreases with an increase in the SDS concentration. The volumetric collapse
8 efficiency - defined as the relative change in volumes of the swollen and collapsed NGs -
9 increases from 45% (2mM) to 95% (5.5mM) with increasing SDS concentration as
10 indicated. As all these concentrations are well below the critical micelle concentration
11 (CMC) of SDS (~8mM at 25°C), [29] we can ignore the tendency of formation of micelles.
12 Moreover, any free SDS is assumed to be completely removed during dialysis. The effect
13 of SDS is summarized as both an increase in the incipient nucleus size, onto which the
14 chains collapse during the synthesis and an increase in the hydrophobicity of the NGs due
15 to incorporation of the hydrophobic surfactant tails (C-12 chains). The initial increase is
16 seen more as an electrosteric stabilization that also explains the lowering of polydispersity
17 while increasing SDS concentrations (Figure 1(a)). On the other hand, incorporation of
18 SDS chains confirms a more efficient collapse above VPTT. This is due to the dominance
19 of hydrophobic interaction forces at increased temperature, coupled with the temperature
20 dependent collapse of the pNIPAm blocks. However, it is worthy of mention that further
21 increase in SDS concentration did not yield smaller NGs and led to temperature driven
22 aggregation when heated above VPTT. Thus, while it is important for the NGs to collapse,
23 it is equally interesting to re-swallow them with known efficiency.

1 Bright field (BF) S(T)EM images of a representative NG at 25°C and NG solution heated
2 just prior to measurement at 45°C are shown in Figures 1(c) and 1(d) respectively. Although
3 the swelling-collapse behavior is difficult to capture in this case, the images show that upon
4 heating the NGs, they tend to associate hydrophobically owing to collapse above VPTT.
5 This effect is synergistically improved due to the presence of cross-linker. Figure 2 provides
6 an overall schematic of the parameters that affect the size of the synthesized NGs.

7 NGs comprise polymer chains that are cross-linked to each other and their physico-
8 chemical properties are guided by nature of the monomers, their relative composition in the
9 NGs and the degree of cross-linking among others. The swelling-deswelling properties of
10 the NGs are an important aspect for loading and release of biologically relevant cargo
11 molecules like proteins, peptides and nucleic acids. In case of NGs, the cross-linking
12 density among other factors plays an important role in determining the VPTT, collapse rate
13 and gel properties.

14 Figure 3 (a) shows the variation of the swelling ratio (α) of a representative NG (3mM SDS,
15 8% BIS) as a function of temperature for a set of heating and cooling cycles. α is defined
16 as $(D/D_0)^3$, where D (nm) represents the hydrodynamic diameter of the NG at any
17 temperature and D_0 (nm) represents the diameter of the NG at room temperature. The NGs
18 show a reversible swelling-deswelling transition with a negligible hysteresis. The driving
19 force for aggregation is an increase in entropy from the polymer solution to a two phase
20 system of polymer and solvent. [30] The entropy of a two-phase polymer system and water
21 is greater than a polymer solution owing to a more ordered arrangement of water molecules
22 adjacent to the polymer. The positive entropy change contributes towards aggregation of
23 the polymers, thereby yielding a favourable association (free energy of association is

1 negative since positive enthalpy change is smaller than the entropy term and does not
2 influence association to a larger extent).[31]

3 Figure 3(b) shows the variation of size of a representative NG (3mM SDS, 8% BIS) as a
4 function of pH. A remarkable decrease in size (~27%) is observed under acidic conditions.
5 This is in effect due to protonation of the carboxylic acid groups of the poly AAc blocks
6 with an increasing pH. The thermodynamic model developed by Siegel can be used to
7 explain the swelling/deswelling characteristics while changing the pH of the medium.[32]

8 This model considers three sources contributing to the total free energy - (i) NG-solvent
9 system; (ii) NG-solvent mixing, (iii) deformation of polymer networks and osmotic
10 pressure of mobile ions. The NG-solvent mixing component is dominated by the poly AAc
11 segments that undergo dissociation with an increase in pH. Dissociated poly AAc segments
12 are more hydrophilic than non-dissociated segments, whereby a transition from lower to
13 higher pH causes a drastic decrease in the free energy of mixing. Hydrophobic to
14 hydrophilic transition also explains the consequent swelling of the NGs. However, this
15 dissociation is affected by the deformation degree of the polymer network, mostly affected
16 by the cross linking density and the osmotic pressure of OH^- and Na^+ ions. The sharp
17 volume transition that happens at a pH slightly below neutral conditions is reflective of the
18 fact that the pH is high enough to overcome the osmotic pressure, wherein counter-ion-
19 shielding effects occur within the poly AAc domains. [33] Once the osmotic pressure is
20 overcome, a synergistic effect of the favourable free energy of NG-solvent mixing and de-
21 cross-linking of bound poly AAc segments in the domains causes further swelling with
22 increase in pH.

1 Figure 3(c) depicts the variation of the zeta potential of a representative NG (4.2 mM SDS,
2 5% BIS) as a function of temperature. The zeta potential not only gives a measure of the
3 surface charge of the NGs but also indicates stability of NGs in solution. A higher surface
4 charge indicates higher stability owing to electrostatic forces. The zeta potential,
5 representative of the charges contributed to by the poly AAc segments, does not change
6 appreciably as a function of temperature. This is due to the fact that zeta potential is only
7 applicable to the NGs in a semi-quantitative manner as some of the charges are buried and
8 contribute partially to the calculated value. Additionally, there exists no well-defined
9 slipping plane between the NG surface and the medium.[34] Herein, the relatively constant
10 zeta potential for the NGs indicates that the poly AAc segments do not show temperature
11 dependence, and their effect in the swelling/deswelling characteristics of the NGs can be
12 widely de-coupled from the effect of the poly NIPAm chains. To further substantiate this,
13 Figures 3(d) and 3(e) depict the changes in size and zeta potential as a function of both
14 temperature and pH for a representative NG (2 mM SDS, 5% BIS). In acidic conditions
15 (pH 3), the collapse of the NGs is substantial owing to charge based interchain repulsion,
16 while in basic medium (pH 9), the NGs do not show much dependence on temperature. In
17 fact, a slight increase in size could point towards aggregation driven by hydrophobic forces.
18 On the other hand, a decrease in the zeta potential of the NGs is observed under acidic
19 conditions, while under basic conditions, the zeta potential does not change much. The
20 increase in cationic charge can be explained by a coupled effect of the collapse of the poly
21 NIPAm segments due to temperature effect and protonation of functional carboxylic groups
22 of the poly AAc segments.

1 Figure 3 (f) shows a comparison of the sizes of the Fe@Au NPs, a representative NG and
2 Fe@Au NPs incorporated into NGs (Fe@Au_NG_c and Fe@Au_NG_i) as a function of
3 temperature. An increase in the size of the Fe@Au_NG_c NPs as a result of the increase in
4 temperature is observed in contrast with only NG. Although the NG units undergo entropy
5 driven collapse above VPTT, Fe@Au NPs act as crosslinking units (Figure S3, Supporting
6 information) pulling the gelling units together, whereby increasing their effective size as
7 evidenced from their hydrodynamic sizes. On the other hand, Fe@Au_NG_i resemble the
8 NGs when it comes to temperature driven collapse, showing a volumetric collapse
9 efficiency of 94%, analogous to that of the bare NGs. Thus, the Fe@Au_NG_i behave
10 similarly to the bare NGs, while the Fe@Au_NG_c resemble the characteristics of the bare
11 Fe@Au NPs. However, the increase in size for the Fe@Au NPs (volume increase 36%)
12 may be attributed to aggregation, whereas a 67% volume increase for the Fe@Au_NG_c is
13 mostly due to the cross-linking effect of the Fe@Au NPs that pulls together the already
14 collapsed NG blocks.

15 This variation in size is further reflected in Figure 3 (g) for Fe@Au_NG_c where the
16 heating and cooling swelling ratio curves diverge from unity with increasing temperature.
17 These structures are much less reversible when compared to NGs alone, since, in addition
18 to chain length, cross-linking density and presence of Fe@Au NPs further affect their size
19 distributions as a function of temperature. For similar reasons, size dependence of NG
20 coated Fe@Au NPs with change in pH, follows a reverse trend. These NG coated Fe@Au
21 NPs are more stable than both the Fe@Au NPs and the NGs themselves owing to higher
22 magnitudes of zeta potential. (Figure S3, Supporting information) Further, the effect of
23 temperature on zeta potential follows a trend opposite to that of the NGs alone. On the other

1 hand, Figure 3 (h) shows the variation of swelling ratio for Fe@Au_PEG_i as a function of
2 temperature. Although, the heating cooling cycles resemble those of the bare NGs, the
3 hysteresis area is larger. It points to these being less reversible than the bare NGs as a result
4 of different densities of the Fe@Au cores as compared to the NGs. This density difference
5 introduces non-homogeneity in the system properties leading to less well defined swelling-
6 collapse behaviour.

7 The cross-linked NGs swell under the critical temperature and collapse above it. Thus, the
8 VPTT is an important property of these NGs that determines their biological applications.
9 Figure 4 shows DSC measurements of a representative NG (3mM SDS, 8% BIS) which
10 depicts heat flow during heating and cooling cycles (Figure 4(a),(c) respectively) and
11 specific heat capacity during heating and cooling cycles (C_P) (Figure 4 (b), (d) respectively)
12 as a function of temperature. A gradual transition is seen in all the cases over a defined
13 temperature range and the VPTT is calculated to be at the optimum peak position (heating
14 curves) as $\sim 39^\circ \text{C}$ which is in good agreement with the value calculated using the average
15 of the sigmoidal region of the size-temperature plot (Figure 3(a)). However, VPTT
16 calculated using the cooling curves gives a slightly lower value that can be attributed to a
17 slower re-organization of the polymeric chains in response to an otherwise faster
18 temperature drop rate. This is not the case for the DLS study as there is sufficient time
19 between measurements for the system to reach equilibrium. In case of the NGs coated with
20 NPs, the transitions are not smooth owing to density gradient as a result of inorganic NP
21 incorporation within the matrix. To consider this effect, the VPTT is defined as the
22 temperature where the fraction of the swelled and collapsed states is equal.[35] Using this
23 method, the VPTTs of the NG, Fe@Au_PEG_c and Fe@Au_PEG_i are found to be $37.3 \pm$

1 0.2°C, $38.3 \pm 0.5^\circ\text{C}$ and $38.9 \pm 0.8^\circ\text{C}$ respectively. Incorporation of the NPs causes an
2 increase in the VPTT as compared to the bare NGs. This is because of an inherent increase
3 of hydrophilicity of the matrix. Among the coated samples, Fe@Au_PEG_i has a higher
4 VPTT. This may be due to different total amounts of NPs incorporated within the matrix
5 or different arrangements of the Fe@Au NPs within the matrix (Figure S3, Supporting
6 Information).

7 ***Loading and Encapsulation***

8 The NGs were loaded using breathing-in mechanism where in the freeze-dried NGs or NG
9 coated Fe@Au NPs were imbibed with a concentrated solution of Cyt C. The freeze-dried
10 NGs did not show any change in physico-chemical properties when compared to the as
11 synthesized NGs (Figure S4, See Supporting Information). After the loading step, the free
12 Cyt C was removed using a dialysis tubing (MWCO 14kDa). In case of the NG coated
13 Fe@Au NPs, the free Cyt C was removed using centrifugation.

14 Cyt C is a highly water soluble heme protein with properties similar to model drug proteins.
15 The heme ligand is located in the lysine rich region of the protein, that imparts a positive
16 charge to the protein at neutral conditions ($\text{pI} = 10.1$, $M_w = 12327$ Da). [36] It is this front
17 that can interact with negatively charged molecules resulting in complex formation.
18 Coulombic forces between the negatively charged NGs and Cyt C result in the formation
19 of a polymer-protein complex. This results in a release of the counterions associated with
20 the polymer and the protein, causing an entropy gain and thus a net increase in the free
21 energy.[37] Gel swelling yields a lower polymer segment density in the network and greater
22 gel porosity,[34] yielding a higher loading capacity and thus the loading is carried out under

1 swelling conditions. High loading efficiencies of up to 95% have been obtained for these
2 NGs, with an encapsulation efficiency of ~500 μg per mg of the polymer.

3 The high loading efficiency can be attributed to the strong interaction between the NGs and
4 the Cyt C under loading conditions. While it is difficult to predict the exact location of the
5 protein inside the NG network, we hypothesize that since the initial binding events of the
6 protein are localized at the surface, a condensation of the particle periphery happens, which
7 limits deeper diffusion of the protein molecule.[34] While the location of the protein
8 influences its interaction with external factors during biological applications, it is ideal
9 when the molecule is closer to the surface and moderately bound to favour release under
10 the influence of external stimuli and/or charge exchange with electrolytes.

11 In case of the NG coated Fe@Au NPs, loading efficiencies of upto 32% have been obtained with
12 encapsulation efficiencies upto ~14 μg per mg of the Fe@Au NPs. For Fe@Au_NG_i, loading
13 efficiencies of 36.2% and encapsulation efficiencies of 109.7 μg per mg of the nanogel system
14 were observed. The encapsulation efficiency is based on Fe@Au NPs in the former case, while it
15 is based on both the Fe@Au and NGs in the latter case. This means that the efficiencies are not
16 directly comparable as it is difficult to base the calculations on one standard. However, while
17 comparing the loading efficiencies of the systems containing Fe@Au NPs with the bare NGs, there
18 is a drastic reduction in the loading capacity. This is primarily because of lower availability of
19 drug binding sites within the NG, owing to presence of Fe@Au NPs in the nanogel network.
20 (Figure S3, Supporting information)

21 ***Release Studies***

22 Cyt C loaded NGs were subjected to different release conditions to monitor their release
23 kinetics. A dialysis setup (Figure S5, Supporting Information) was employed to monitor

1 the release over time. Three different conditions were chosen to understand their behaviour
2 in response to pH, temperature and a combination of the two. These biologically relevant
3 release conditions were simulated – high temperature (40°C), low pH (3.2) and high
4 temperature along with low pH (40°C, pH 3.4) using MQ water and tuning the pH using
5 different molar ratios of 1M NaOH and 1M HCl. Further, the choice of low pH can be
6 utilized in simulating conditions where drug delivery in intracellular compartments is a
7 matter of concern like in late endosomes (pH=5) or lysosomes (pH 4.5-5). In case of the
8 NG coated Fe@Au NPs, high temperature along with low pH (40°C, pH 3.4) was the only
9 condition used to monitor the release of Cyt C.

10 Figures 5(a), (b), (c) show the release profiles of Cyt C loaded into the NGs and Fe@Au
11 incorporated NGs using the breathing-in mechanism and/or the traditional method. The
12 traditional method refers to adding a calculated amount of the drug to the NG solutions,
13 whereas the breathing-in mechanism relies on imbibing the freeze-dried NGs with a
14 concentrated drug solution. For the sake of comparison, the concentrations of both the
15 polymers and Cyt C were kept the same for both the methods.

16 In case of the NGs loaded using the breathing-in protocol, temperature or pH alone is able
17 to cause a release of upto 20% of the loaded Cyt C over a day and no further increase is
18 observed beyond that. A combination of the two effects- low pH and high temperature
19 however yields a cumulative release of upto ~ 40% ranging over two days. Although, a
20 slow release kinetics is observed, this could in effect direct towards long term encapsulation
21 of the carrier molecule inside these NGs, leading to controlled release by variation of the
22 external parameters. Further, the release is assumed to occur by a “squeezing out”
23 mechanism following Fickian diffusion in the initial time period while a combination of

1 diffusion and degradation of the NGs happens at larger time points. High temperature
2 (above VPTT) and low pH (acidic) conditions enable maximum collapse of the NGs,
3 squeezing out the drug loaded in the porous network.

4 Comparing the NGs loaded using the breathing in mechanism and the traditional method,
5 a slightly higher release was observed in case of all the release scenarios for the latter
6 method. This can be explained by the fact that the drug molecule is peripherally bound to
7 the NGs while in the case of the breathing-in case, the drug molecule traverses to the inside
8 of the pores and hence requires more time or several stimuli factors to cause substantial
9 release of the cargo. However, it must be noted here that higher loading and encapsulation
10 efficiencies are obtained for the NGs loaded using the breathing-in method. An
11 optimization between the loading capacity and external stimuli affected release of the drug
12 molecule can be used to choose the potentiality of the NGs in the field of drug delivery.

13 Fe@Au_NG_c shows remarkably rapid release kinetics, releasing almost 55% of the initial
14 loaded drug over a period of ~40 hours in comparison to Fe@Au_NG_i, which exhibits
15 slower release kinetics, bearing closer resemblance to that from bare NGs. It is capable of
16 releasing around 12% over a period of ~50 hours. The higher release in case of
17 Fe@Au_NG_c is due to more peripheral localization of the drug owing to incorporation of
18 the Fe@Au NPs in the NG networks. However, this is not the case for Fe@Au_NG_i,
19 where the drug penetrates deeper into the more open nanogels networks, as also evidenced
20 from S(T)EM images (Supporting Information). A more tortuous path is thus required for
21 the loaded drug to be released, resulting in a slow kinetics. Further, the Fe@Au NPs act as
22 cross-linkers in case of Fe@Au_NG_c, pulling the gelled units closer together, and thereby
23 enhancing ‘squeezing’ out of the drug.

1 Figures 5(d), (e) and (f) show the plots of $\ln F$ as a function of $\ln t$, where F represents the
2 cumulative fraction of the drug released at time t . Majority of the drug release processes
3 from swellable polymer systems are defined by two limiting cases, that is combination of
4 Fickian and Case II transport mechanism. The latter is based on two assumptions- a
5 boundary is formed between the glassy and rubbery phase of the polymer and boundary
6 moves at constant velocity.[38, 39] The overall behavior is defined by combining diffusion-
7 controlled and visco-elastic relaxation-controlled drug release and is given by

$$8 \quad \mathbf{F = kt^n}$$

9 where, k is the rate constant and n is the diffusional exponent that determines the drug release
10 mechanism. It is observed here that drug release mechanism from the Fe@Au_NG_c and
11 Fe@Au_NG_i is that of super case transport II ($n > 1$) while for the NGs alone are representative
12 of drug release from spherical particles ($0.5 < n < 1$) and anomalous in nature. This relates to the
13 conformational arrangement of the NGs which have lesser degrees of freedom in the presence of
14 Fe@Au NPs, leading to less homogeneously defined viscous and elastic regions. On the other hand,
15 different models[40] fitted to the release data yielded a linear dependence of F with t in case of
16 traditionally loaded NGs Fe@Au_NG_c and Fe@Au_NG_i NPs, while they show square root time
17 dependence for NGs loaded with breathing-in mechanism. (Figure S9, Supporting Information)
18 These in conjunction with the diffusional exponent results show that drug incorporation method,
19 location of the drug and presence of Fe@Au NPs largely alter the drug release mechanism and the
20 kinetics. Further, the modulation of release parameters can greatly influence the release kinetics
21 as observed from the release conditions used in the study.

1 **CONCLUSIONS**

2 Synthesis of external stimuli sensitive pNIPAm-AAc NGs has been optimized by varying
3 reaction parameters. Among various parameters that included stabilizer (SDS)
4 concentration, cross-linker (BIS) concentration and mole ratios of reactants, SDS
5 concentration was observed to be the most important parameter to control the size of the
6 NGs – higher concentration of SDS led to smaller NGs. The optimized NGs were used to
7 incorporate Fe@Au NPs in order to incorporate magneto-plasmonic properties to the
8 construct using two methods- coating and in-situ growth.

9 Under the influence of temperature, the NGs show a reversible swelling/deswelling
10 kinetics, which happens due to an entropically driven expulsion of arranged water
11 molecules. The pH response occurs due to protonation/de-protonation of the AAc blocks.
12 This hydrophilic-hydrophobic transition has been confirmed to be reversible and the VPTT
13 has been determined to be ~ 39°C. The size of the NGs as a function of temperature and/or
14 pH is accounted for by a balance between Coulombic and hydrophobic forces.
15 Fe@Au_NG_i shows a similar temperature based transition. However, an opposite effect
16 is observed for Fe@Au_NG_c which happens as the Fe@Au NPs act as bridge molecules
17 pulling together the gelling units.

18 Thereafter, Cyt C was loaded into the NGs and Fe@Au incorporated NGs using a modified
19 breathing-in mechanism. This gave high loading and encapsulation efficiencies (~96% and
20 500µg/mg of NGs, respectively), showing a great capacity to retain drug solution. Using a
21 dialysis setup and three different release conditions, the release kinetics of Cyt C was
22 monitored. The release kinetics has been observed to be rather slow (over several hours),
23 which hints towards their applications in sustained retainment of the encapsulated

1 molecule. However, high release has been observed under a combination of high
2 temperature (above VPTT) and low pH (acidic) conditions, in which case maximum de-
3 swelling of the NGs is expected, leading to a squeezing release of the drug molecule. The
4 release from Fe@Au_NG_i follows similar kinetics to that of the bare NGs. However, the
5 release of Cyt C from Fe@Au_NG_c is the fastest, accounting for release of almost 55%
6 of the initially loaded drug in ~40 hours. An overall schematic is shown in Figure 6 that
7 describes the release of Cyt C from different configurations of the NG with the Fe@Au
8 NPs. The drug incorporation method, location of the drug, presence of Fe@Au NPs and NP
9 incorporation method largely alter the drug release mechanism. Thus, these temperature/pH
10 programmed NGs can be fine-tuned for both sustained entrapment of cargo molecule and
11 external stimuli directed release of the same by controlling the synthetic parameters and the
12 mode of loading the cargo molecule.

13

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16 Nanolab and the Faculty of Natural Sciences and Technology, NTNU for financial support.

17 **DISCLOSURE**

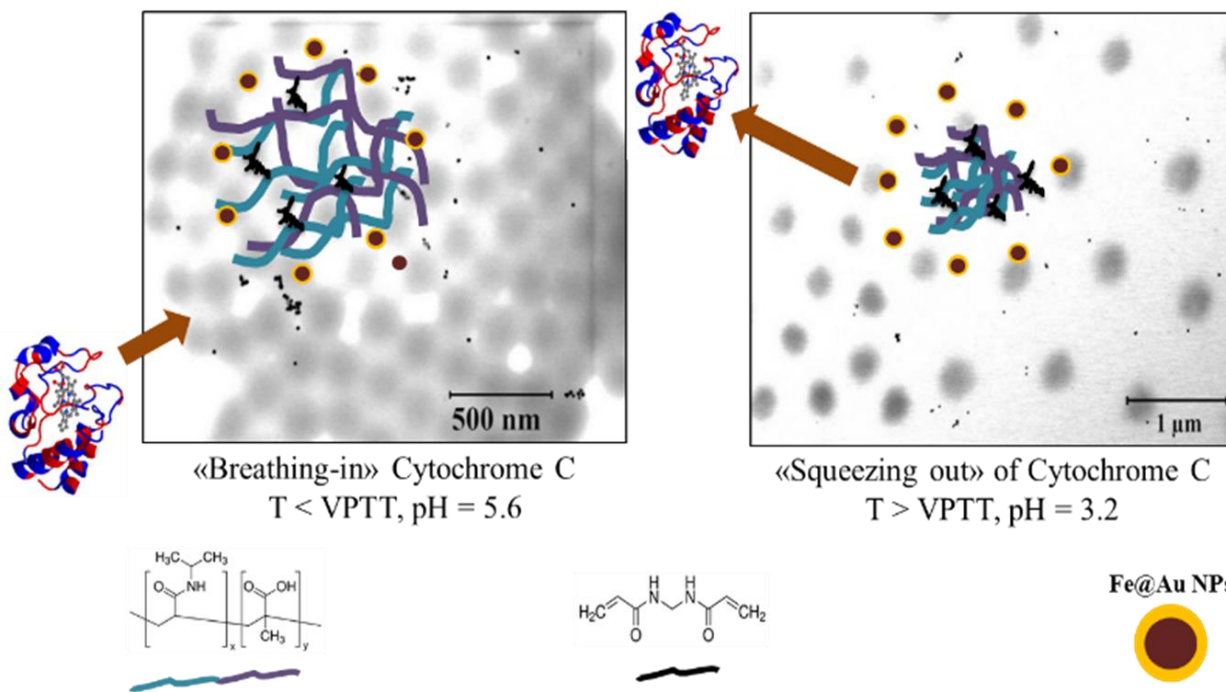
18 The authors declare that there is no conflict of interest.

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1 GRAPHICAL ABSTRACT



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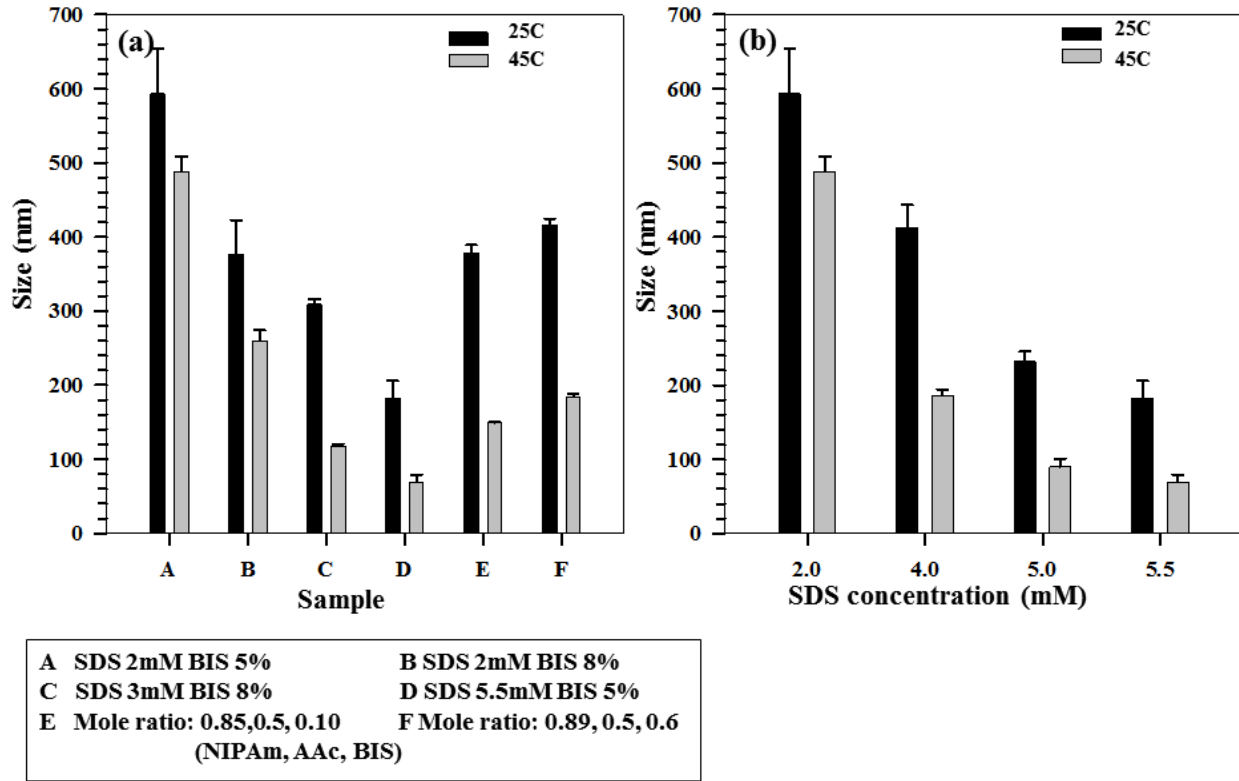
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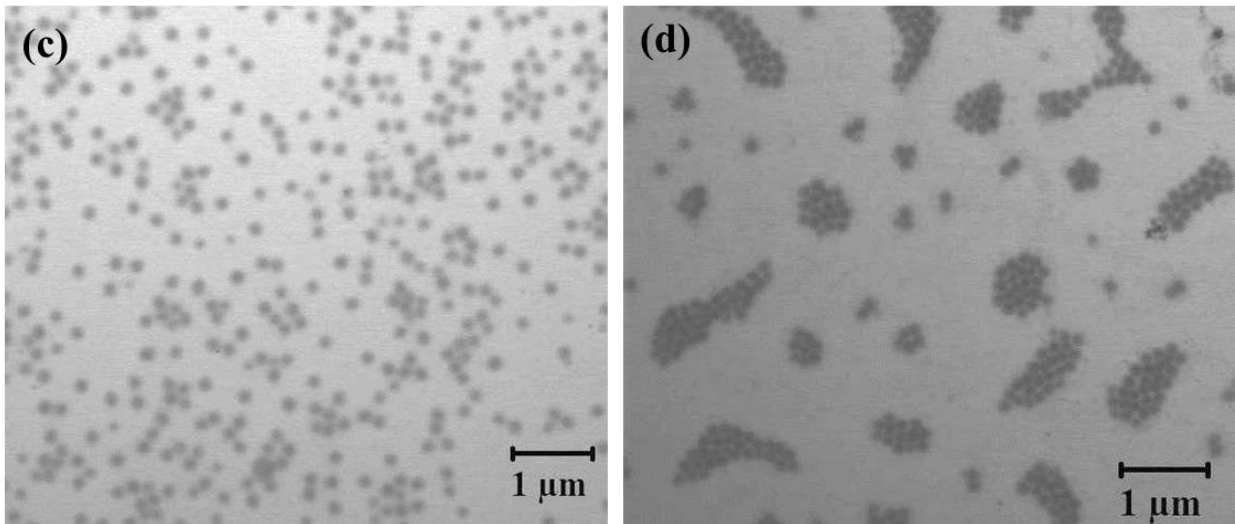
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FIGURES

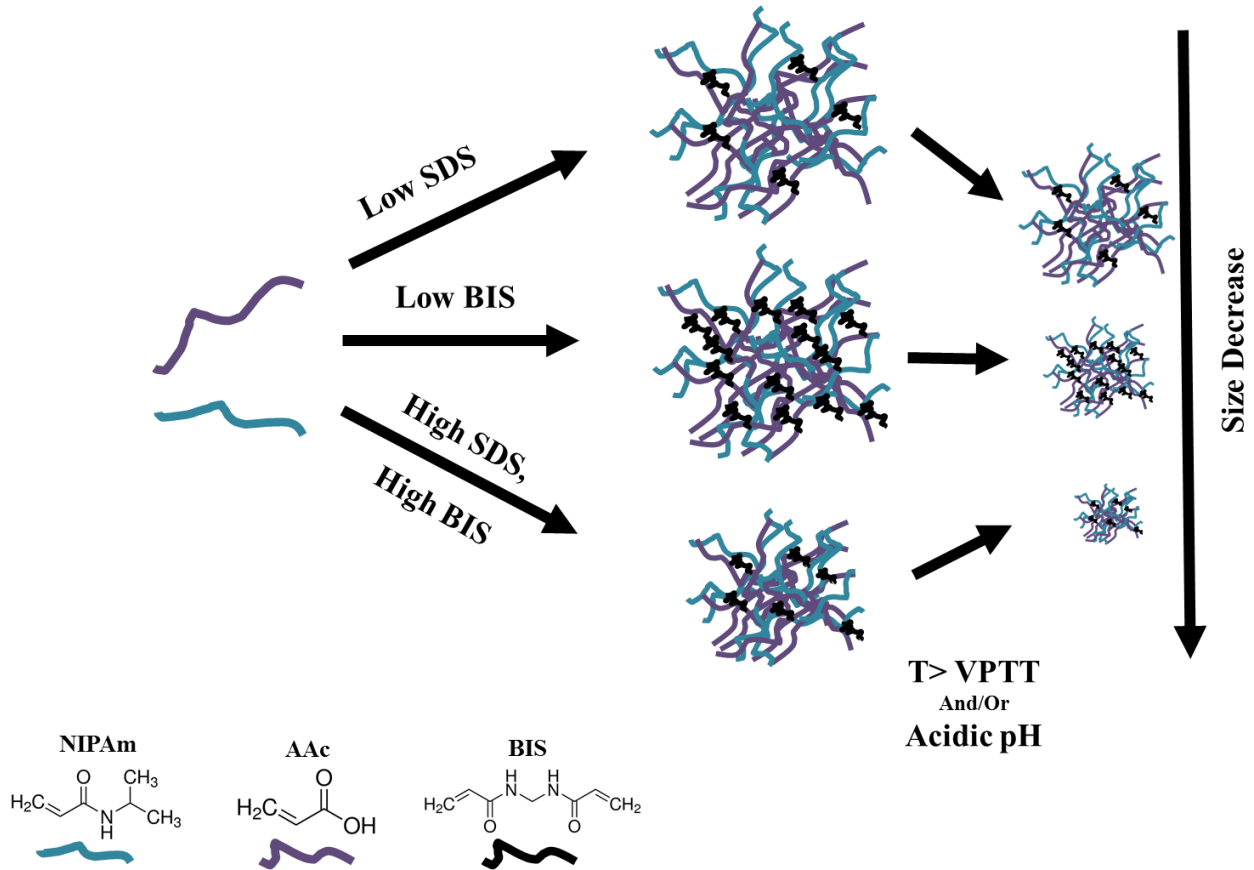


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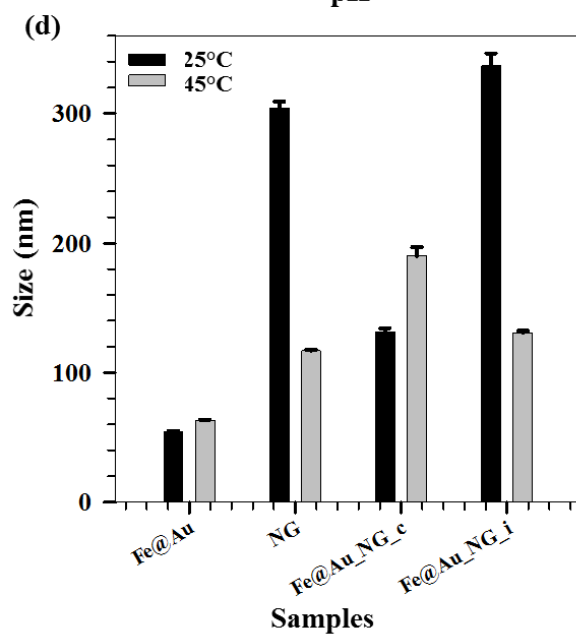
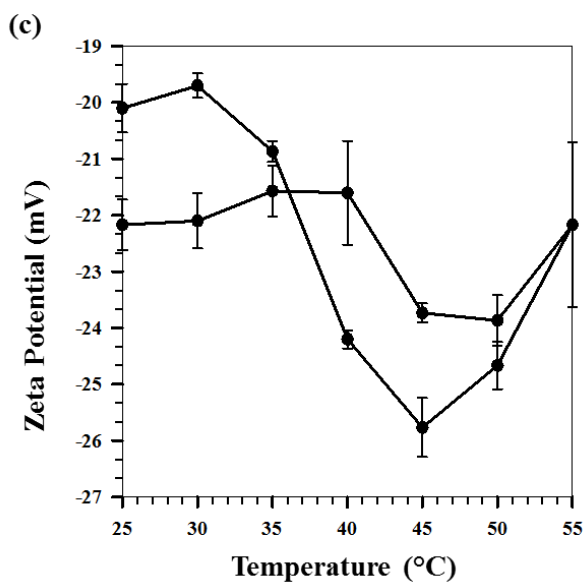
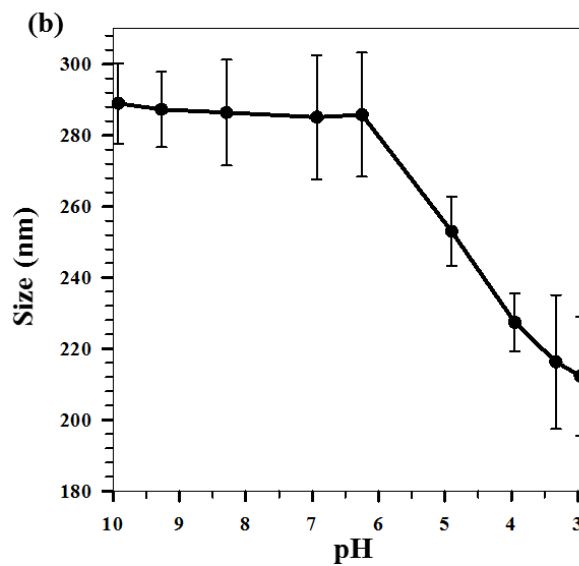
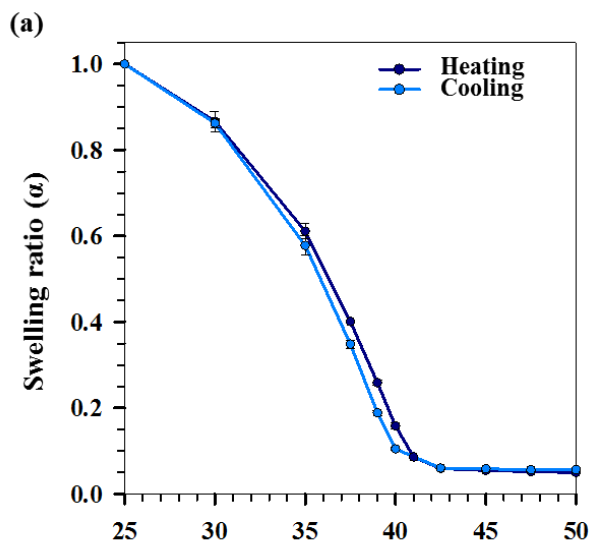
1 Figure 1 (a) Variation of size of the NGs synthesized using different parameters (b) Variation of
 2 size of the NGs as a function of SDS concentration (c) Representative BF S(T)EM image of NGs at
 3 25°C (d) Representative BF S(T)EM image of NGs at 45°C.

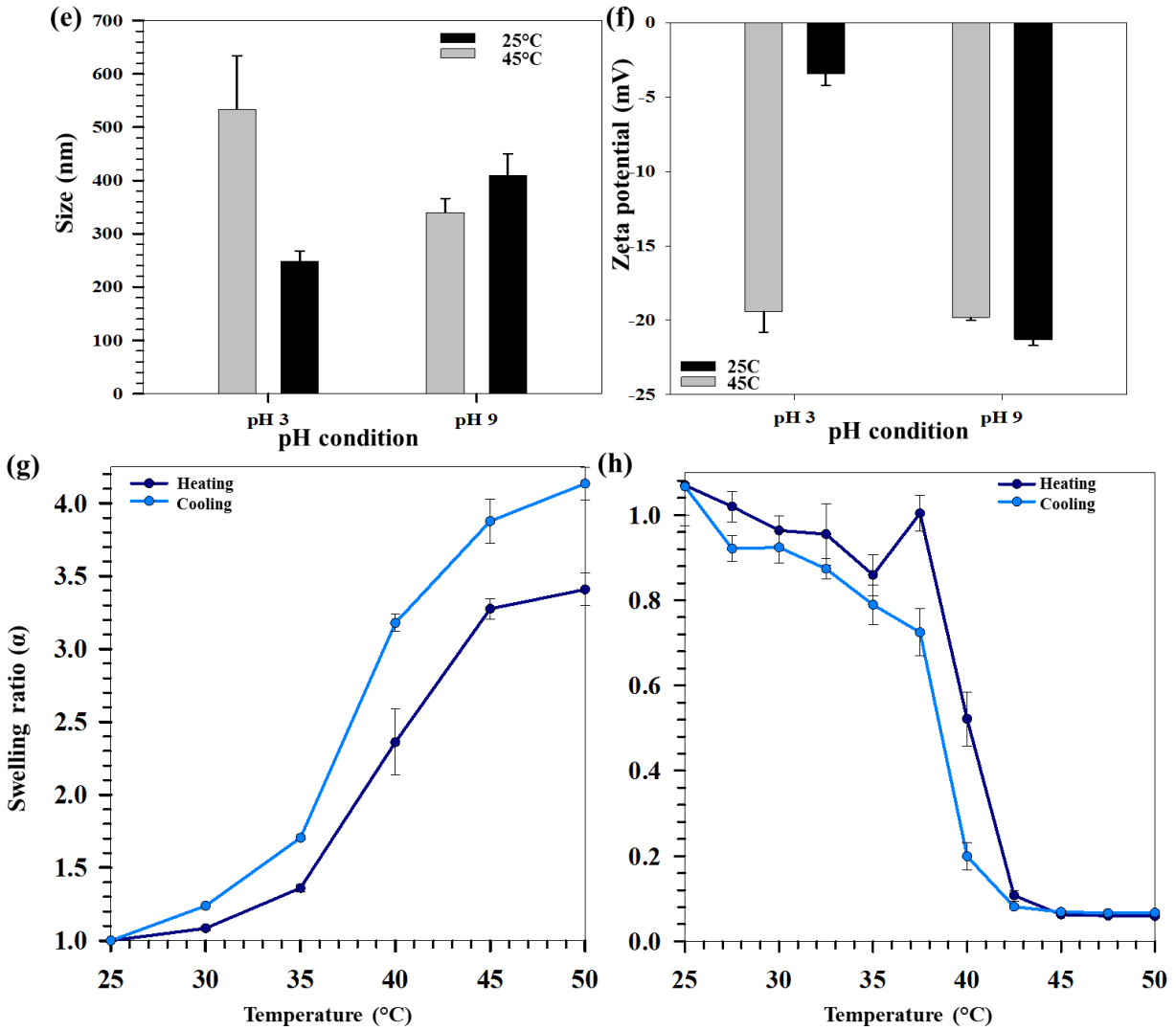
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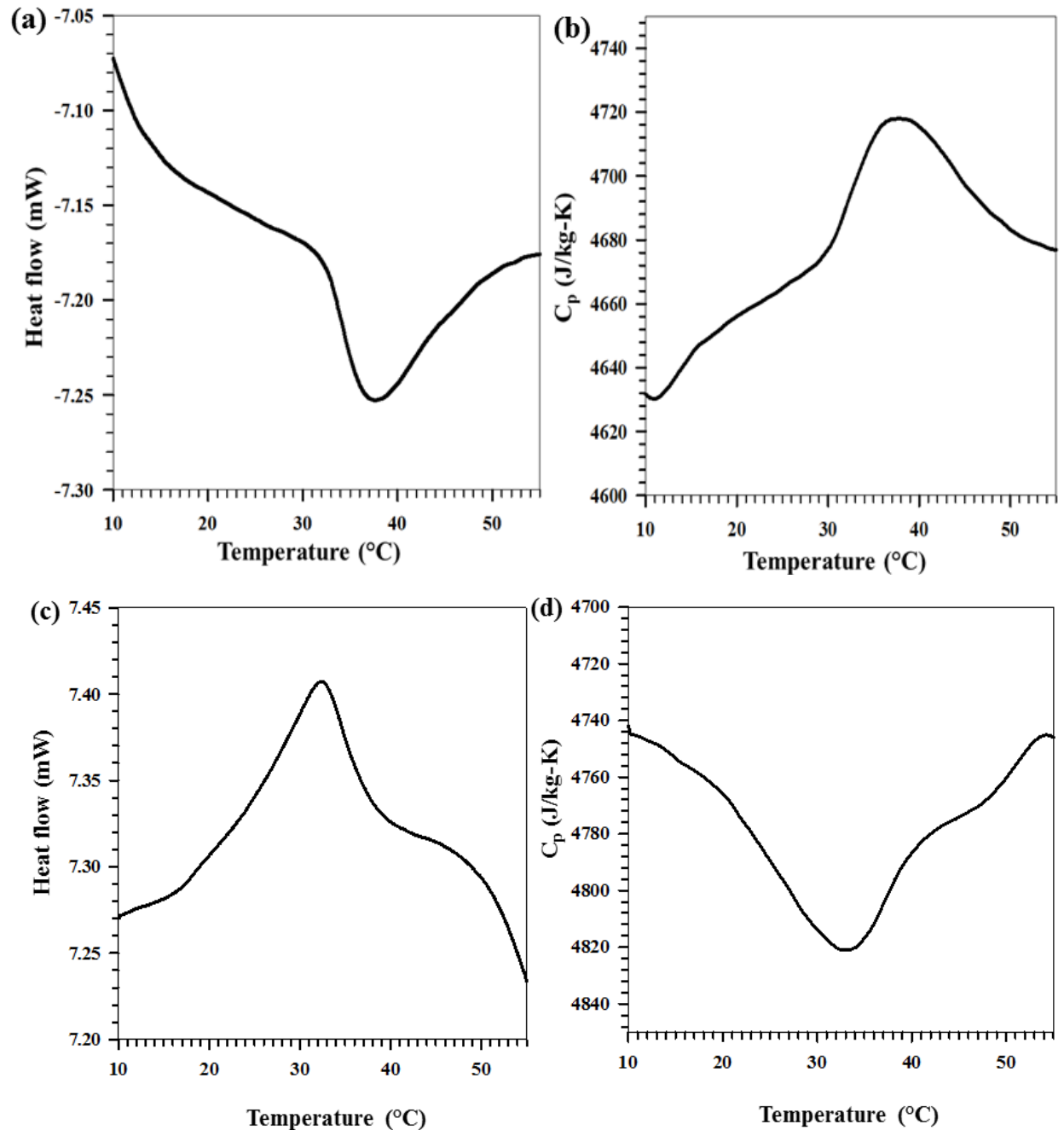
Figure 2 Schematic showing effect of various synthetic parameters on size of pNIPAm-AAc NGs.





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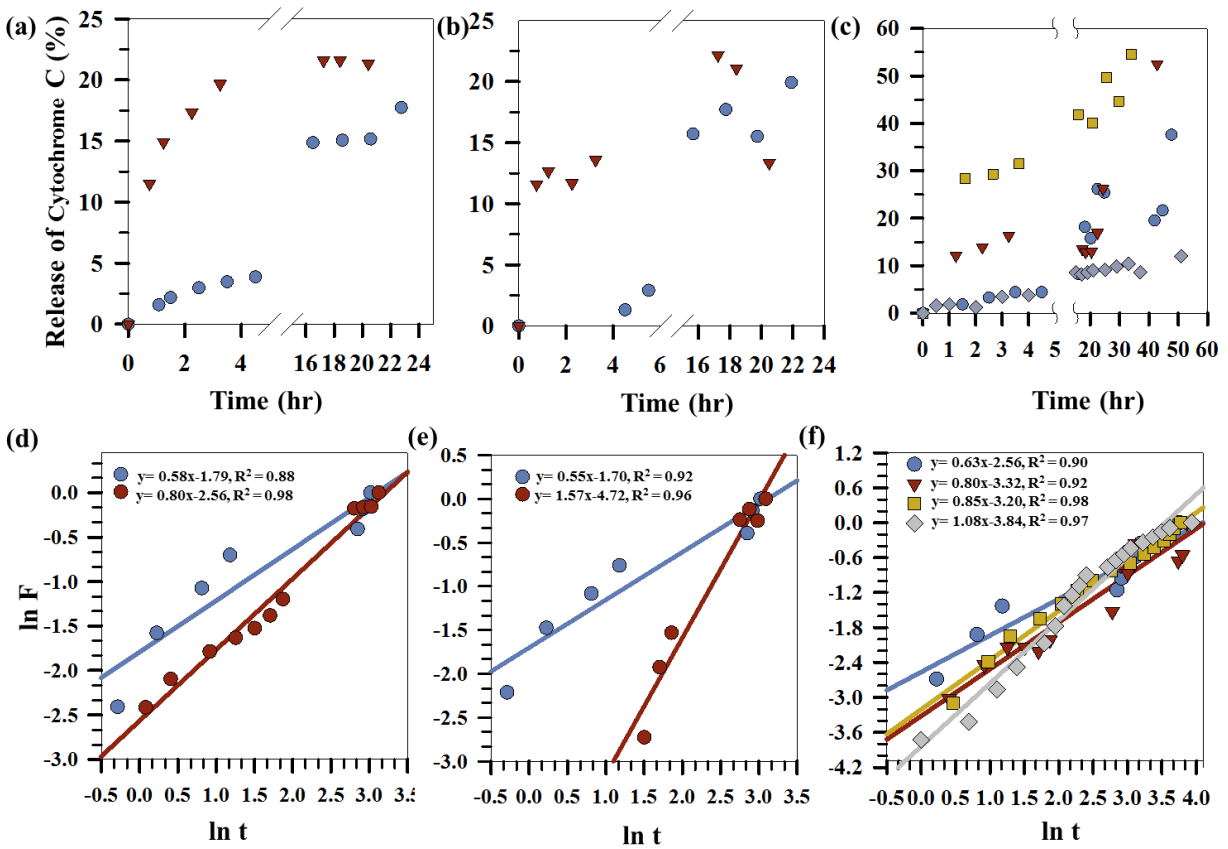
2 Figure 3 (a) Swelling ratio (α) of a representative NG as a function of temperature during heating
 3 and cooling cycles (b) Size of a representative NG as a function of pH (c) Zeta potential of a
 4 representative NG as a function of temperature during heating and cooling cycles (d) Size of
 5 Fe@Au NPs, representative NG, Fe@Au_NG_c and Fe@Au_NG_i as a function of temperature.
 6 (e) Size of a representative NG as a function of pH and temperature (f) Zeta potential of a
 7 representative NG as a function of pH and temperature. Swelling ratio of (g) Fe@Au_NG_c NPs
 8 and (h) Fe@Au_NG_i NPs as a function of temperature during heating and cooling cycles. The
 9 error bars represent standard deviation of a set of triplicate measurements.



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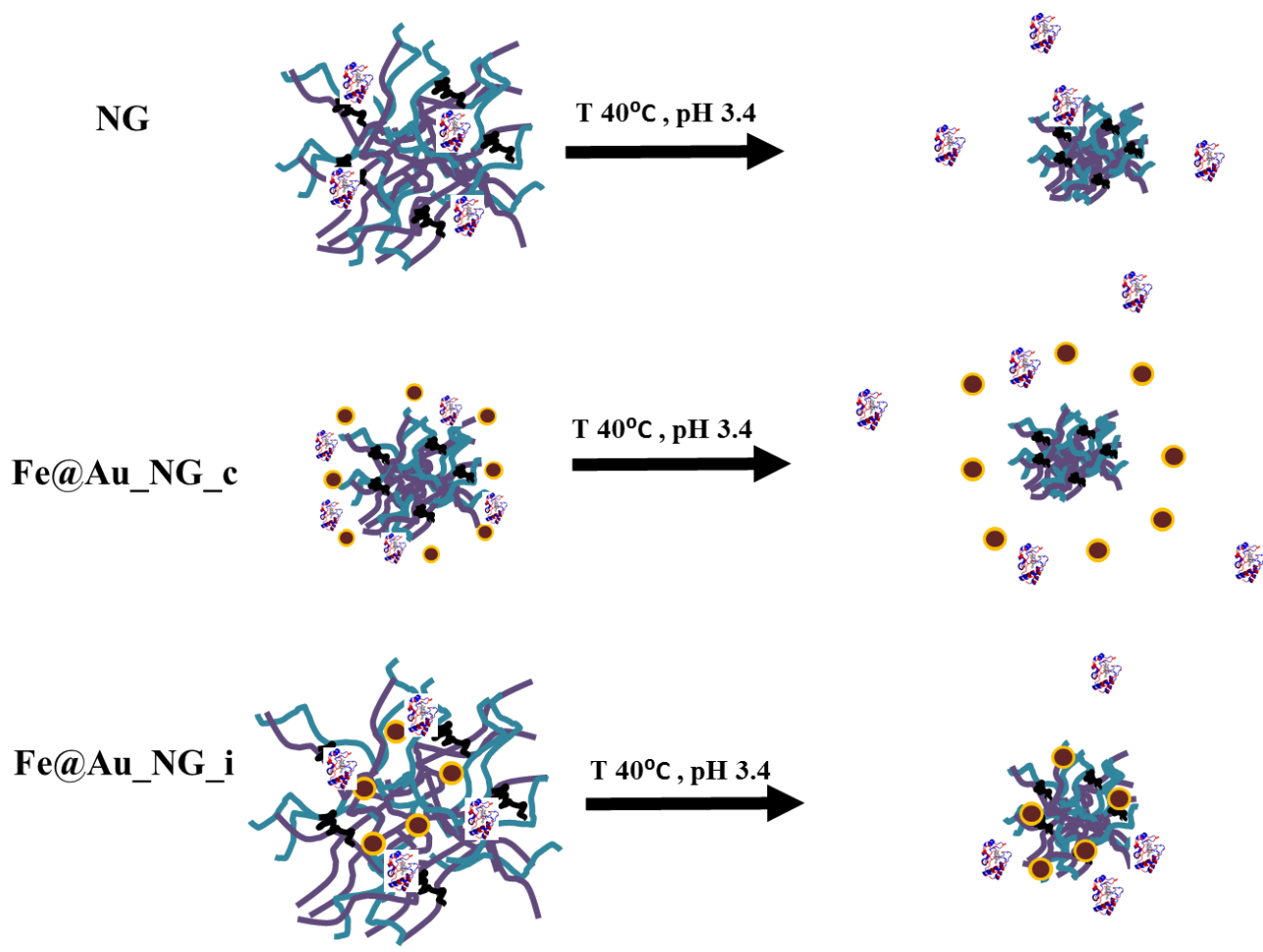
Figure 4 (a) Heat flow (during heating cycle) as a function of temperature for a representative NG (b) Specific heat capacity variation (during heating cycle) as a function of temperature for a representative NG (c) Heat flow (during cooling cycle) as a function of temperature for a representative NG (d) Specific heat capacity variation (during cooling cycle) as a function of temperature for a representative NG.

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3 Figure 5 Comparison of release of Cyt C (loaded using breathing-in mechanism and traditional
 4 method) over time from (a) a representative NG at pH 3, T 25°C (b) a representative NG at pH 6,
 5 T 40°C and (c) a representative NG and NG coated Fe@Au NPs at pH 3, T 40°C. Plots of $\ln F$ as a
 6 function of $\ln t$ for (d) Case a (e) Case b and (f) Case c above respectively. F represents
 7 cumulative fraction of Cyt C.



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2 Figure 6 Overall schematic showing release of Cyt-C at pH 3.4 and temperature 40°C from three
 3 different NG combinations- NG, Fe@Au_NG_c and Fe@Au_NG_i

1 **Incorporation of Fe@Au nanoparticles into multiresponsive pNIPAM-AAc**
2 **colloidal gels modulates drug uptake and release.**

3 **Supporting Information.**

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5 Raju^b, Birgitte H. McDonagh^a, Wilhelm Robert Glomm^{a,c,†}

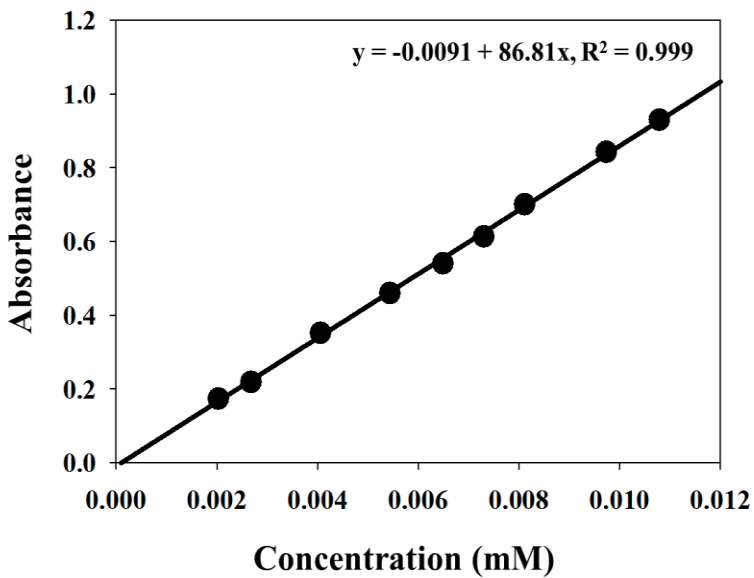
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11 Norway.

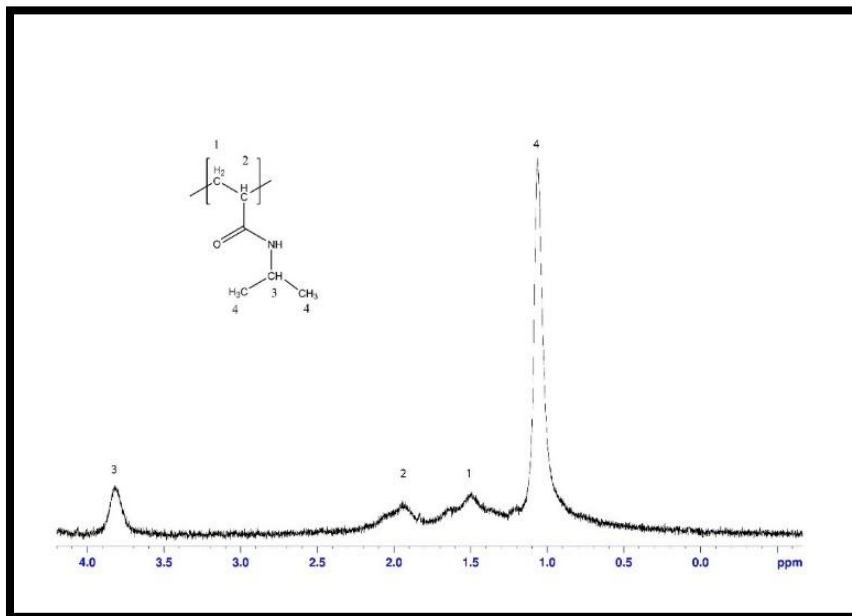
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15 (Wilhelm.Glomm@sintef.no)

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3 Figure S1 Calibration curve of Cytochrome C



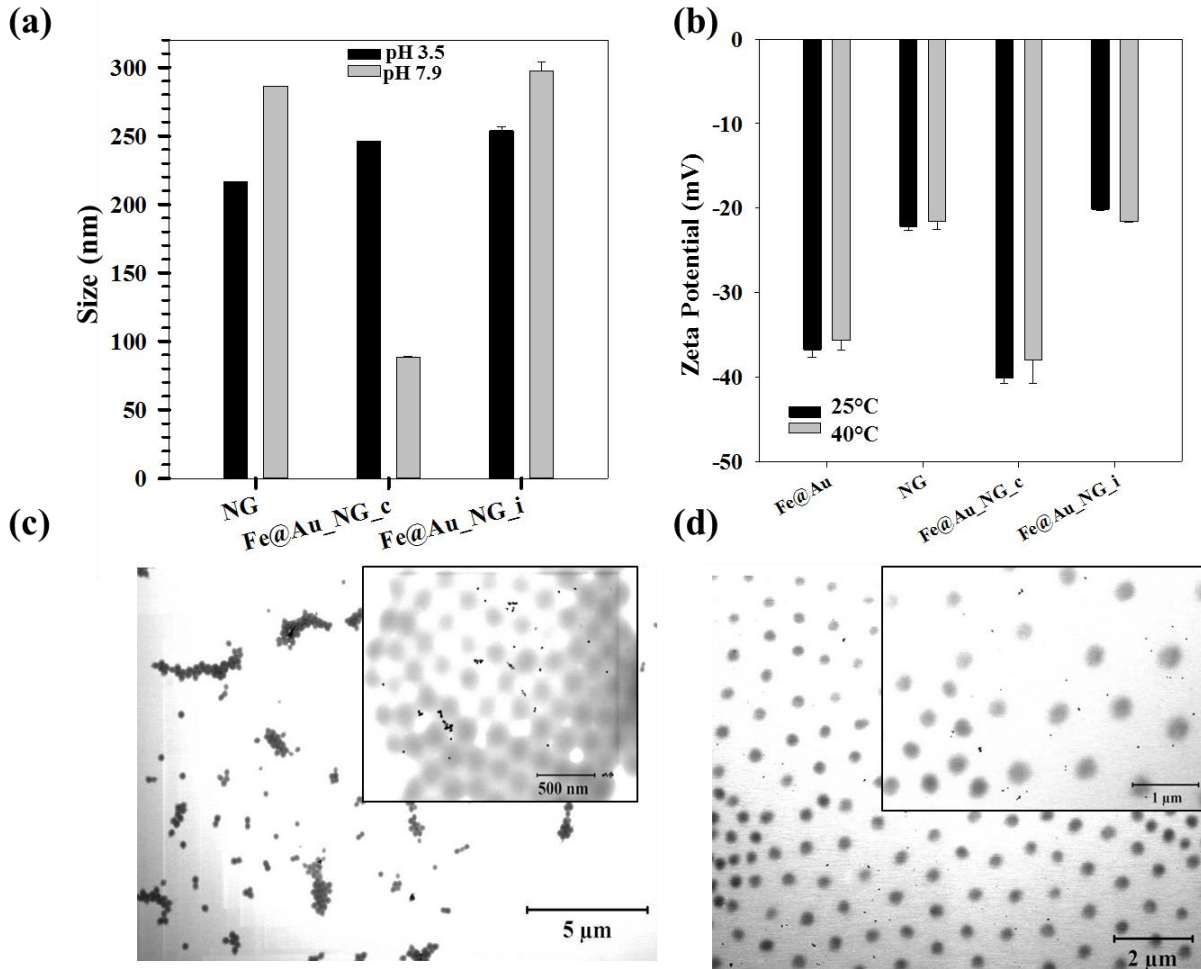
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5 Figure S2 ¹H NMR spectrum of a representative nanogel in D2O (400 MHz)

6 The presence of broad characteristic multiplets at $\delta = 1.95$ ppm (-CH₂-CH-) and $\delta = 1.51$ ppm (-

7 CH₂-CH-) along with broad singlets at $\delta = 1.05$ ppm (CH₃-CH-CH₃) and $\delta = 3.85$ ppm (CH₃-CH-

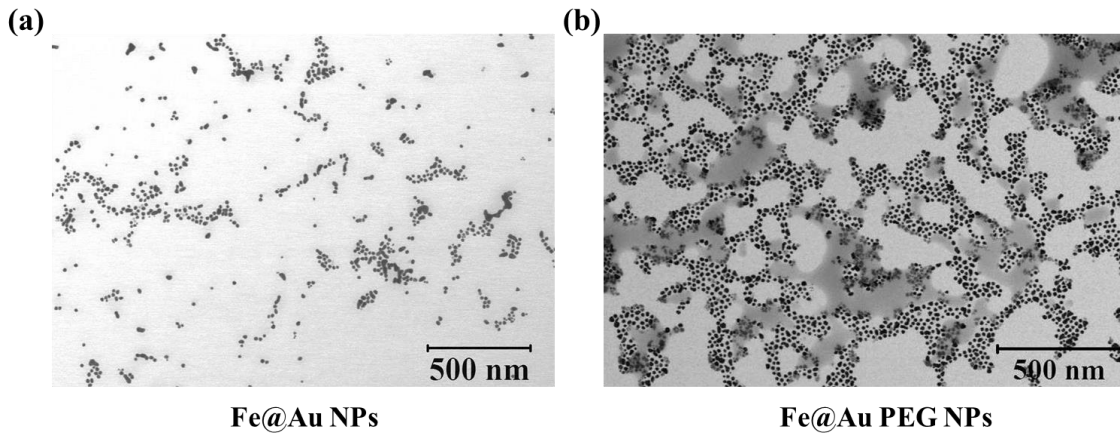
1 CH3) indicates that polymerization has proceeded onwards from its precursor (NIPAm) to afford
 2 nanogels.



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 4 Figure S3 (a) Variation of size of NG, Fe@Au_NG_c, Fe@Au_NG_i with pH (b) Comparison of zeta
 5 potentials as a function of temperature for Fe@Au NPs, a representative nanogel and
 6 Fe@Au_NG_c and Fe@Au_NG_i Representative BF S(TEM) image of Fe@Au_NG_c (c) at 25°C
 7 and (d) at 45°C respectively. Red circles show Fe@Au NPs

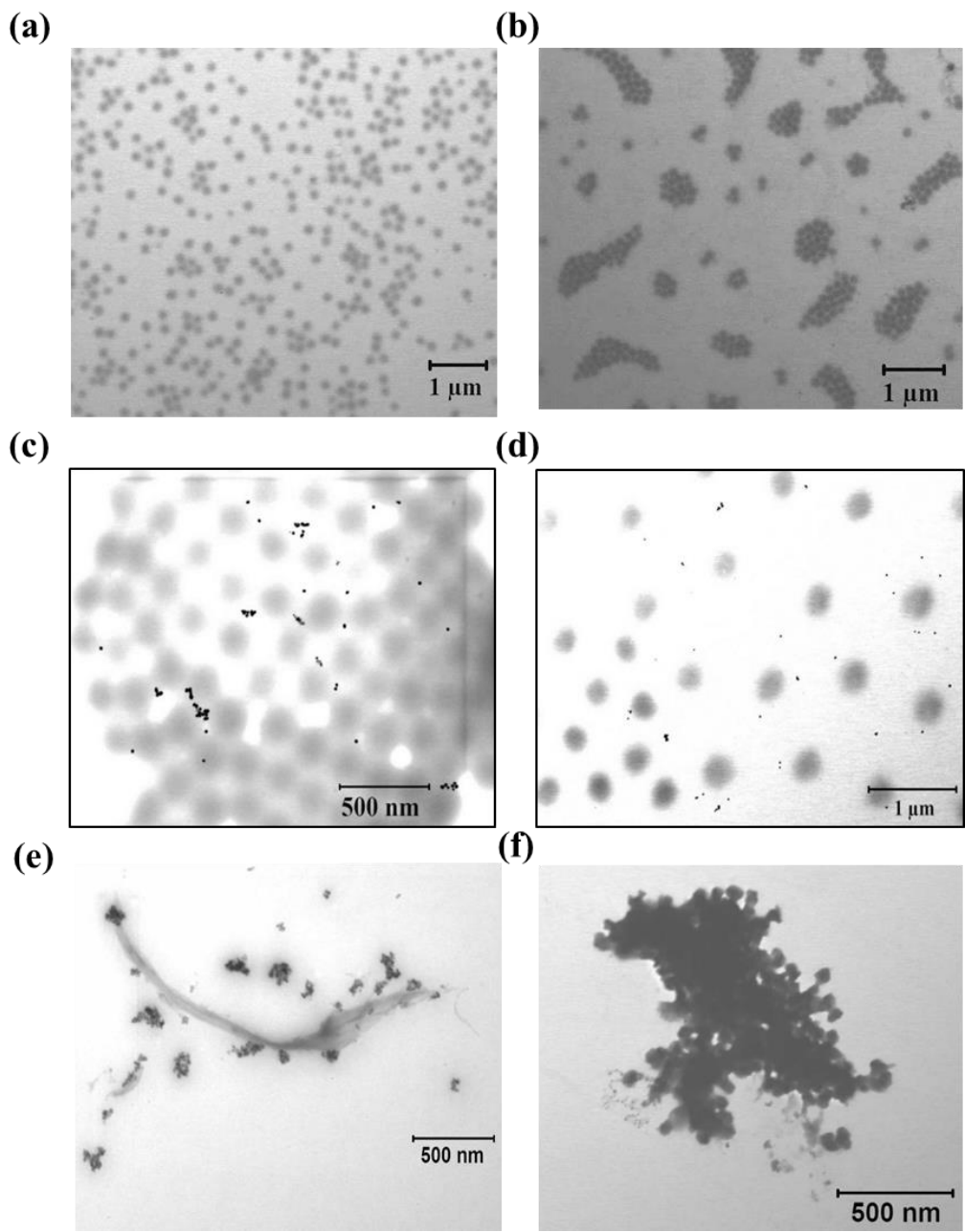
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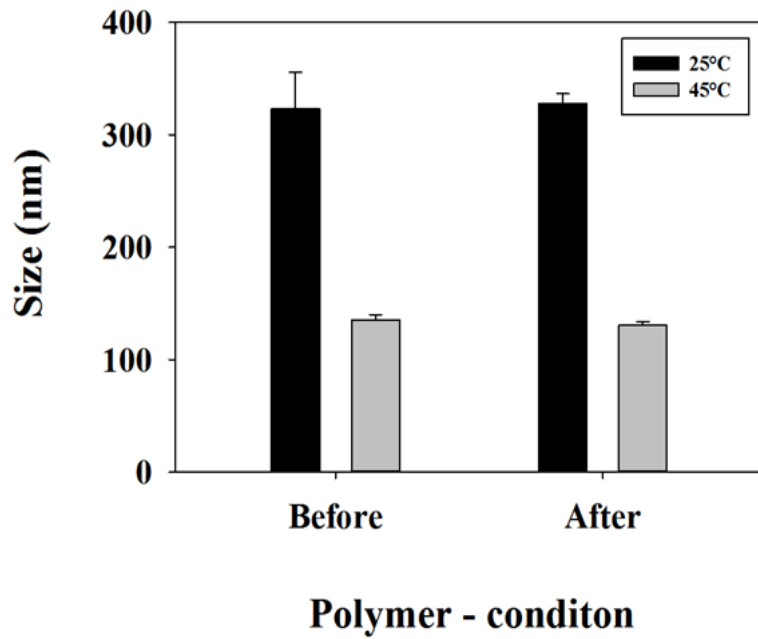
3 Figure S4 Representative S(T)EM image of (a) Fe@Au NPs and (b) PEG coated Fe@Au NPs.



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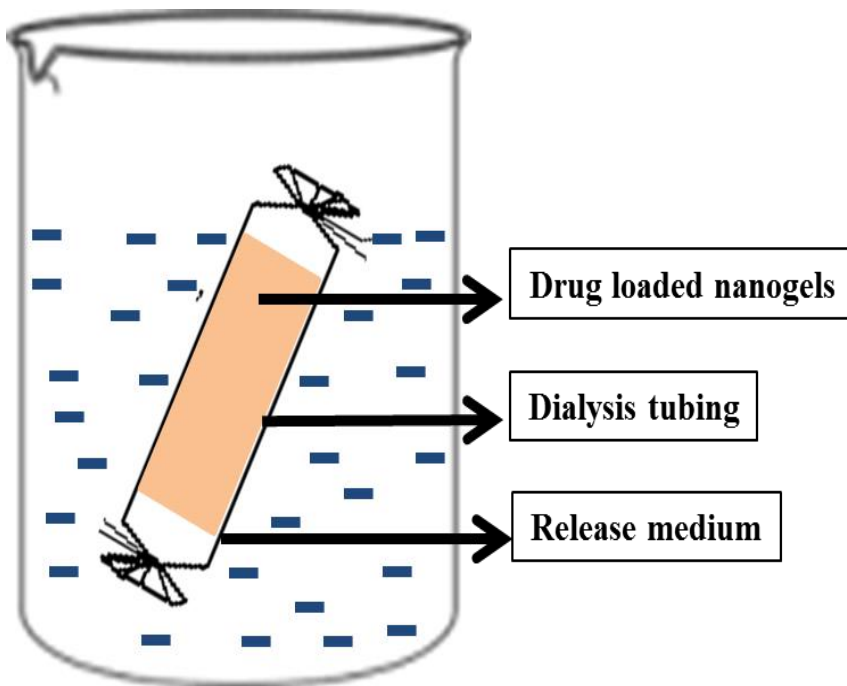
Figure S5 Representative S(T)EM images of (a) a representative nanogel at 25°C (b) at 45°C (c) Fe@AU_NG_c at 25°C (d) at 45°C (e) Fe@Au_NG_i at 25°C and (f) at 45°C respectively.

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3 Figure S6 Variation of size of a representative nanogel before and after freeze drying



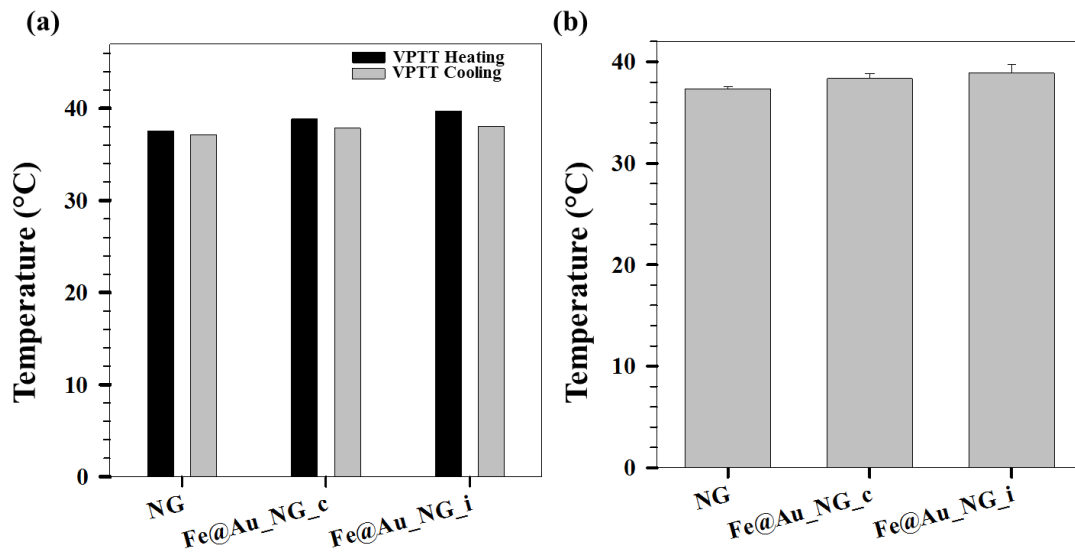
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5 Figure S7 Schematic of the dialysis setup used for studying release from the nanogels.

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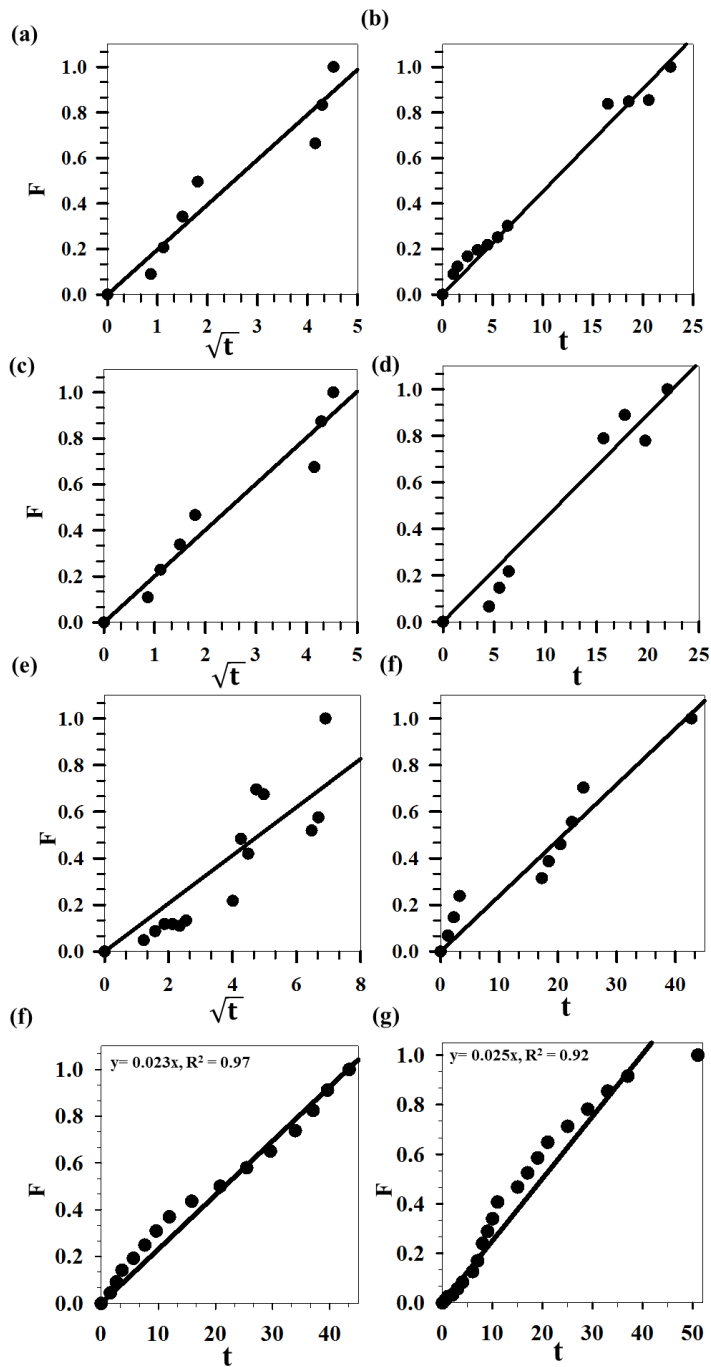
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4 Figure S8 (a) Heating and Cooling VPTTs and (b) Average VPTTs for representative nanogel,
5 Fe@Au_NG_c and Fe@Au_NG_i respectively.



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2 Figure S9 Fitted model for nanogel loaded with Cytochrome C (a) using traditional method and
 3 release monitored at 25C and pH 3 (b) using breathing in method and release monitored at 25C
 4 and pH 3 (c) using traditional method and release monitored at 40C and pH 6 (d) using traditional
 5 method and release monitored at 40C and pH 3 (e) using breathing in method and release
 6 monitored at 40C and pH 6 (f) using breathing in method and release monitored at 40C and pH 3

- 1 (g) Fitted model for Fe@Au_NG_i loaded with Cytochrome C using breathing in method and
- 2 release monitored at 40C and pH 3