

1 Corrosion and Degradation in MEA based post-combustion CO₂ Capture

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6 Abstract

7
8 Two of the main challenges in post-combustion CO₂ capture with ethanolamine are solvent
9 degradation and material corrosion. It has been shown that there is a correlation between
10 degradation and corrosion. The present paper examines this correlation by studying the effect
11 of 10 MEA degradation products on corrosion. Thermal degradation experiments were
12 conducted under stripper conditions for 5 weeks. 30wt% MEA solution with 1wt% of the
13 various degradation products were placed in 316 SS cylinders and stored in a thermostat
14 chamber at 135 °C. ICP-MS was used for the metal concentration analyses for all the
15 solutions, while ion chromatography was used for the quantitative determination of heat
16 stable salts anions and MEA concentrations. The solutions were also analyzed for
17 degradation products in order to study the formation and thermal stability of these
18 compounds. For corrosion monitoring, in addition to ICP-MS analyses, SEM-EDS was used
19 for examining the cylinders surface morphology and elemental composition while XRD was
20 used for corrosion product identification. In the present paper, the influence of the secondary
21 degradation products on corrosion is studied. Results show that some specific degradation
22 products, like bicine, HeGly and HEEDA enhance corrosion while others don't seem to have
23 a significant effect on corrosion of stainless steel.

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28 *Keywords: Corrosion; MEA; CO₂ capture; degradation*

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1. Introduction

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37 It is crucial to develop robust and environmentally benign technologies for CO₂ emissions
38 reduction. In this direction, CO₂ capture and storage is one potent strategy to minimize the
39 emissions of carbon dioxide from fossil fuel power plants. An effective CO₂ capture method
40 is to remove CO₂ after combustion (post-combustion CO₂ capture) by chemical absorption
41 using various aqueous amine solutions. Today, this is the most mature method for post-
42 combustion CO₂ capture and has already reached commercial stage[1]. Absorption using
43 amines as solvents has been applied successfully for several decades in areas such as natural
44 gas processing or coal gasification[2], but also in smaller for post combustion capture e.g.
45 AES Corporation CO₂ plant in Warrior Run, Maryland[3]. Various alkanolamines can be used
46 for CO₂ post-combustion capture. Monoethanolamine (MEA) is nowadays the benchmark
47 solvent, due to its good properties towards CO₂ (fast absorption rate, cheap and reasonable
48 volatility)[4].

49 Solvent degradation and equipment corrosion are two of the main problems in CO₂ capture
50 processes. By definition, degradation is the irreversible transformation of an absorbent
51 solution into other compounds. These byproducts can cause problems to the system such as
52 corrosion of the equipment, amine loss, fouling, foaming and reduction of CO₂ absorption
53 capacity[5].

54 Experience shows that amine degradation products often aggravate corrosion[5], and
55 corrosion and degradation are closely tied since the byproducts of ethanolamine (MEA) have
56 been shown to increase corrosion rates[6, 7]. Degradation of amines can be oxidative or
57 thermal, and as mentioned above, some of the degradation products are corrosive agents.
58 Oxidative degradation due to dissolved O₂ is found to be more dominant than thermal
59 degradation in pilot plants[8-10], and has been studied by various researchers [11-13]. This is
60 likely also the case in industrial plants. A detailed overview of MEA oxidative degradation
61 and the degradation compounds' mechanisms can be found elsewhere [8]. The primary
62 oxidative degradation compounds are ammonia, aldehydes and carboxylic acids. Anions of
63 strong carboxylic acids, e.g. formic acid and oxalic acid, form with MEA the so called heat-
64 stable salts (HSS). HSS reduce the CO₂ capture capacity, and induce corrosion in gas treating
65 power plants. The role of formation of heat stable salts (HSS) on corrosion has also been
66 studied in [14-16]. Among HSS formate and oxalate have a higher impact on corrosion than
67 the rest HSS. The effect of five carboxylic acids(oxalic, acetic, glycolic, propionic and
68 formic) on corrosion have been studied previously by Fytianos et al.[16]. 30wt% MEA
69 solution containing 1wt% oxalic acid and 30wt% MEA solution with 1wt% formic acid
70 showed higher impact on corrosion compared to the other tested acids.

71 Also thermal degradation of MEA has been studied by various researchers [9, 17, 18] and
72 their findings are in good agreement with regard to degradation products formation
73 mechanisms. The secondary degradation compounds are defined as products formed by
74 reaction of MEA with the primary degradation compounds[8]. The three main thermal
75 degradation products are HEIA, HEEDA and OZD with HEIA being the dominant one.
76 Degradation rates increase with temperature and CO₂ loading.

77 There are various factors that can enhance the corrosiveness in MEA based CO₂ capture.
78 Operating parameters (such as amine concentration, process temperature, CO₂ and O₂
79 content, HSS and impurities) determine the extent of corrosion within CO₂ capture plants[19,
80 20]. Generally, an increase in amine concentration, CO₂ loading, temperature and O₂ will
81 increase the corrosion rate of carbon steel in MEA based pilot plant[21].

82 Lately, stainless steel has become more popular in coal fired post-combustion CO₂ capture
83 (PCCC) as the major material of the plant [22] . From the review of J. Kittel and S. Gonzalez
84 [23], it is stated that carbon steel should not be the material of choice for PCCC, and that
85 stainless steel 316L has much higher corrosion resistance. The knowledge of corrosion
86 caused by degradation products is limited in the case of stainless steel, which has been
87 reported to suffer aggressive corrosion in the presence of formic acid in MEA plant[24].
88 According to Davis, stainless steels do not catalyze the thermal degradation of MEA[17].

89 There is a lack of data with regard to the effect of MEA degradation products on corrosion.
90 The present paper focuses on the effect of 7 oxidative and 3 thermal degradation products on
91 corrosion. The secondary oxidative degradation products of MEA, i.e. HEF, HEA, HEI,
92 HeGly, BHEOX, Bicine and HEPO are tested. In addition, the thermal degradation products
93 HEIA, OZD and HEEDA are tested in this work. In this paper the experimental results from
94 the thermal degradation and corrosion experiments are described and discussed. In addition,
95 the thermal stability and the thermal decomposition of the degradation products are
96 examined. The ultimate goal of this paper is to investigate which of the degradation products
97 of MEA could contribute to high corrosion in the CO₂ capture process.

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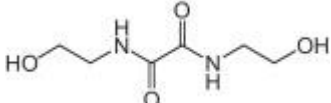
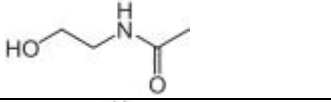
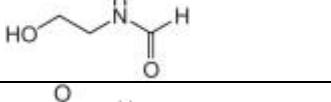
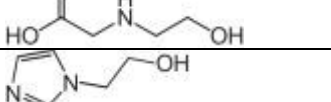
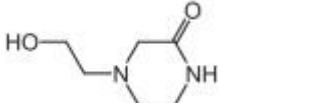


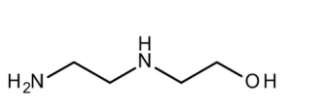

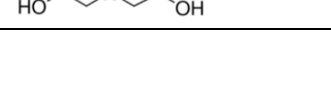
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2. Material and Methods

The chemicals used in this work were MEA, HEEDA, OZD, HEI, Bicine, HEA (Sigma-Aldrich, purity >97%), BHEOX, HEF, HEIA (Alfa Aesar, purity > 97%), HEPO (Tiger Scientific, purity >98%) and HeGly (Enamine, purity >98%). The CAS numbers, full names and structures of the tested degradation products are listed in Table 1.

Table 1: Tested MEA degradation products

Abbreviation	Compound	CAS	Structure
BHEOX	N,N'-Bis(2-hydroxyethyl)-oxamide	1871-89-2	
HEA	N-(2-hydroxyethyl)-acetamide	142-26-7	
HEF	N-(2-hydroxyethyl)-formamide	693-06-1	
HEGly	N-(2-hydroxyethyl)-glycine	5835-28-9	
HEI	N-(2-hydroxyethyl)-imidazole	1615-14-1	
HEPO	4-(2-hydroxyethyl)-2-piperazinone	23936-04-1	
OZD	2-Oxazolidinone	497-25-6	
HEIA	N-(2-hydroxyethyl)imidazolidinone	3699-54-5	
HEEDA	N-(2-hydroxyethyl)ethylenediamine	111-41-1	
Bicine	N,N-Bis(2-hydroxyethyl)glycine	150-25-4	

Solutions of 30wt% MEA containing 1wt% of degradation product were prepared gravimetrically with distilled water. The solutions were loaded with CO₂ (0.4 mol CO₂/ mol MEA). 9g of each solution was put into a 316 stainless steel cylinder equipped with Swagelok® end caps. The cylinders were heated in a thermostat chamber at 135 °C for 5 weeks. For each solution, 10 cylinders were used (two parallels for each week). Every week, the two replicates of each solution were tested for total metal concentration and for degradation product formation. A similar approach was previously used by Fytianos et

122 al.[16]. The cylinders were weighed at the start and end of the experiments for possible
 123 leakage detection. Corrosion evaluation of the liquid samples was conducted by measuring
 124 the metals concentration with ICP-MS and IC was used for the HSS anion analyses. The
 125 surface morphology of the inner part of the cylinder was examined with SEM-EDS. After the
 126 experiments, XRD was used for the identification of corrosion products. The MEA
 127 concentrations after every week were determined with IC and degradation compounds were
 128 analysed with LC-MS after week 2 and week 5. A more detailed description of the analytical
 129 methods used is given below.

130 Solutions were analysed for total Fe, Cr, Ni and Mo by a high resolution Thermo Fischer
 131 Element 2 ICP-MS, as an indication of corrosivity. Grimstvedt et al. [25] has previously used
 132 a similar approach. The approach is based on the fact that higher metal concentrations
 133 indicate more corrosion. The initial CO₂ loaded 30wt% MEA solution and the solutions after
 134 5 weeks were analysed for amine concentration and CO₂ content. Total alkalinity of the
 135 various solutions was determined by H₂SO₄ titration and the BaCl₂ method, see Ma'mun et
 136 al. [26], was used for the determination of CO₂ concentrations for the start and end samples.

137 The quantification of the degradation products was performed using an Agilent 1290
 138 Infinity LC system coupled to an Agilent 6490 Triple Quadrupole Mass Spectrometer
 139 equipped with an Agilent Jet Stream ion source (Agilent Technologies, Santa Clara, CA,
 140 USA). The components were separated by optimized reverse phase chromatographic
 141 methods, using various columns and mobile phases. Retention times were within the range of
 142 1 to 10 minutes, and the limits of quantitation (LOQ) were between 0.1 and 10 ng/ml. Details
 143 of the LC-MS methods are listed in Table 2.

144

145 Table 2. Overview of the analytical methods used to separate and quantify the degradation products

Analyte	Column	Mobile phase	LOQ (ng/ml)
HEHEAA	Supelco Discovery® HS F5 (3 µm particle size, 15 cm × 4.6 mm)	25 mM Formic acid	0.1
HEEDA	Ascentis® Express RP-Amide, 2.7 micron (2.7 µm particle size, 15 cm × 4.6 mm)	25 mM Formic acid	0.1
Bicine	Supelco Discovery® HS F5 (3 µm particle size, 15 cm × 4.6 mm)	25 mM Formic acid/methanol	1
HeGly			0.1
HEF			1
BHEOX			10
HEA	Supelco Discovery® HS F5	0.1 wt% Ammonium	0.1
HEPO	(3 µm particle size, 15 cm × 4.6 mm)	acetate/methanol	0.1
OZD			0.1
HEI			0.1
HEIA			1

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149 Ion Chromatography (IC) was used for the quantification of the HSS anions and for
150 MEA. The anions glycolate, propionate, formate, oxalate and acetate were analyzed on an
151 ICS-5000 ThermoScientific System equipped with AS15 analytical column, an ASRS300
152 suppressor and a conductivity detector and a carbonate removal device. As mobile phase, the
153 eluent generator with KOH cartridge connected to a Milliore ICW-3000 system was used.
154 MEA analysis was conducted with a Thermo Scientific Dionex IonPac™ CS19 analytical
155 column (2 x 250 mm) coupled with a Thermo Scientific Dionex IonPac™ guard column
156 CG19 (2 x 50mm) and the CSRS 300 2mm suppressor. The mobile phase was 20 mM
157 methanesulfonic acid and the cation chromatography method is described in Fytianos et al.
158 [27].

159 The surface morphology and chemical composition of cylinders were examined by
160 scanning electron microscopy (SEM) and energy dispersive spectrometry (EDS) in order to
161 examine the effects of corrosion on surface morphology and chemical composition of the
162 cylinders after 5 weeks experiment under stripper conditions. Small pieces were cut from the
163 cylinders and cleaned with ethanol to remove any deposited corrosion products. Samples
164 were placed on stubs and scanned without coating. Characterizations were carried out with a
165 Hitachi S-3400N scanning electron microscope at an acceleration voltage of 20.0 kV and a
166 working distance of 10.0 µm. EDS data were processed by using Aztec Energy Software.

167 Deposited corrosion products on the inner cylinder surfaces were collected gently by cotton
168 swabs and crushed with a mortar and pestle for qualitative characterization via powder X-Ray
169 Diffraction (XRD) (D8 Advance DaVinci, Bruker AXS GmbH). Analyses were conducted in
170 the 2θ range of 20-80° using a Cu X-ray tube, with a step size of 0.013° and a step time of
171 0.78 s. The PDF-4+ database (from the International Centre for Diffraction Data) was used
172 for the identification of species.

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3. Results

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177 In this section the MEA concentration changes and degradation products stability and
178 formation are discussed. After that, IC results for the HSS are reported and ICP-MS for total
179 metal concentration are presented. The results given in this chapter are the averages of the the
180 two cylinder parallels. Finally, surface morphology and elemental mapping were evaluated
181 by a combined use of SEM and EDS to examine the corrosion onto stainless steel.

182 As mentioned before the cylinders were weighed before and after the experiments as a
183 leakage test. The average weight loss was less than 1 % based on the total cylinder weight,
184 indicating that the cylinders were tight.

185

3.1 MEA concentration

187 The MEA concentration for all the samples after 5 weeks was between 14 to 15.5wt% and the
188 MEA concentration decreased in the different solutions in a very similar way. When looking
189 at the parallels we can see that the MEA loss differences where less than 1wt%. These results
190 are in an agreement with results from the thermal degradation study of Fytianos et al. [16].

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197 3.2 Degradation product formation and thermal stability

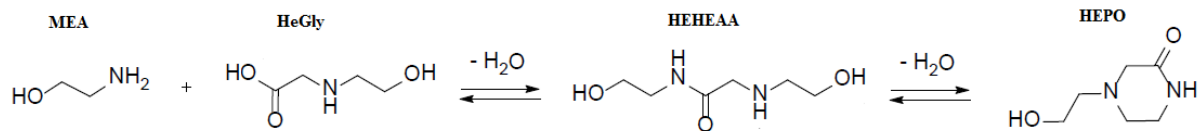
198 In table 3, the concentrations of the degradation products in mg/ L for the various solutions
 199 are shown after week 2 and 5 (the end of the experiment). For the mixtures of 30wt% MEA
 200 with 1wt% HEF and HEA only results from week 5 are given due to analytical error.

201 Table 3: LC-MS results in mg/L after week 2 and 5.

Solution	HeGly	HEF	BHEOX	HEA	HEPO	OZD	HEI	Bicine	HEIA	HEEDA
MEA										
Week 2	66	225	8	7	824	165	41		65408	14471
Week 5	15	817	8	11	238	80	29		128642	12430
MEA+HEEDA 1%										
Week 0										9130
Week 2	15	218	8	7	801	140	17		92968	15985
Week 5	6	774	8	11	282	67	12		133137	12016
MEA+OZD 1%										
Week 0						9130				
Week 2	36	261	8	7	457	158	19		99459	16380
Week 5	10	925	8	10	252	83	19		133769	11648
MEA+HeGly 1%										
Week 0	9130									
Week 2	4887	139	8	6	839	91	15		46681	15480
Week 5	663	838	8	10	920	96	14		125616	12547
MEA%+Bicine 1%										
Week 0								9130		
Week 2	72	241	8	7	820	142	16	7721	63811	15392
Week 5								5988	135626	12597
MEA+HEPO 1%										
Week 0					9130					
Week 2	86	277	8	13	10714	150	27		80142	16399
Week 5	25	862	8	13	8964	56	23		115221	12046
MEA+HEI 1%										
Week 0							9130			
Week 2	77	245	9	43	671	160	9097		67480	14255
Week 5	21	766	8	44	216	83	6832		104797	10453
MEA+HEIA 1%										
Week 0									9130	
Week 2	24	233	8	9	670	160	16		92932	14711
Week 5	8	717	8	11	258	84	11		136659	11756
MEA+BHEOX 1%										
Week 0			9130							
Week 2	56	1782	8	6	609	217	29		58972	14067
Week 5	18	2006	8	8	181	119	22		135695	14029
MEA+HEF 1%										
Week 0		9130								
Week 5	20	3072	8	10	176	105	23		135759	13607
MEA+HEA 1%										
Week 0				9130						
Week 5	15	705	8	5145	200	79	25		126884	11884

202 From Table 3, it can be observed that the dominant degradation products formed are HEIA
 203 and HEEDA in all experiments. This was expected, since the experiments were conducted at
 204 thermal degradation conditions. Furthermore, from the various solutions it can be seen that
 205 HEIA is a stable product and there is an increase from week 2 to week 5. Moreover, the
 206 MEA+OZD solution and MEA+HEEDA solutions have higher concentrations of HEIA
 207 compared to the other solutions. This is in good agreement with the work of Davis[17]. The
 208 thermal degradation reaction pathway of MEA includes the formation of OZD, then HEEDA
 209 and finally HEIA. OZD can be formed both from oxidative and thermal oxidation reactions.
 210 At 135 °C, OZD is not stable.

211 In addition to Table 3, HEHEAA was quantified for the MEA+HeGly solutions. In week 2,
 212 the concentration of HEHEAA was 755 mg/L and in week 5 HEHEAA the concentration was
 213 73 mg/L. After week 5, the HEPO concentration increased only in the MEA+HeGly solution.
 214 It appears that MEA reacts with HeGly to form HEHEAA as a first step and then, in the
 215 second stage, HEPO forms from HEHEAA. The HEPO and HEHEAA concentrations are
 216 well correlated. HEHEAA was first identified at MEA based CO₂ capture facilities by
 217 Strazisar et al. [10] and the formation reaction was verified later experimentally by Vevelstad
 218 [28]. The reactions of HEHEAA and HEPO formation are presented in Figure 1.



221 Figure 1: Formation reactions of HEHEAA and HEPO.

222 It is of great importance to know if a degradation product is thermally stable or if it reacts to
 223 form another product that might be corrosive. Originally 9130 mg/L (1wt%) of the various
 224 degradation products was added to the MEA solution. In the case of HEA it can be seen that
 225 final concentration of HEA after 5 weeks is high (5145 mg/L) compared to that in 30wt%
 226 MEA (11mg/L). This indicates that HEA is stable. HEI can be considered thermal stable also.
 227 Similarly, the final HeGly concentration in 30wt% MEA+ 1wt% HeGly is 660mg/L is more
 228 than 40 times higher than in 30wt% MEA (15mg/L). It should be noted that large amount of
 229 the added HeGly has decomposed after 5 weeks. When it comes to HEF we can see that it is
 230 more stable than HeGly, but less stable than bicine. BHEOX decomposes almost totally after
 231 2 weeks. The typical oxidative degradation products HEA, HEI and HeGly are found in low
 232 concentrations.

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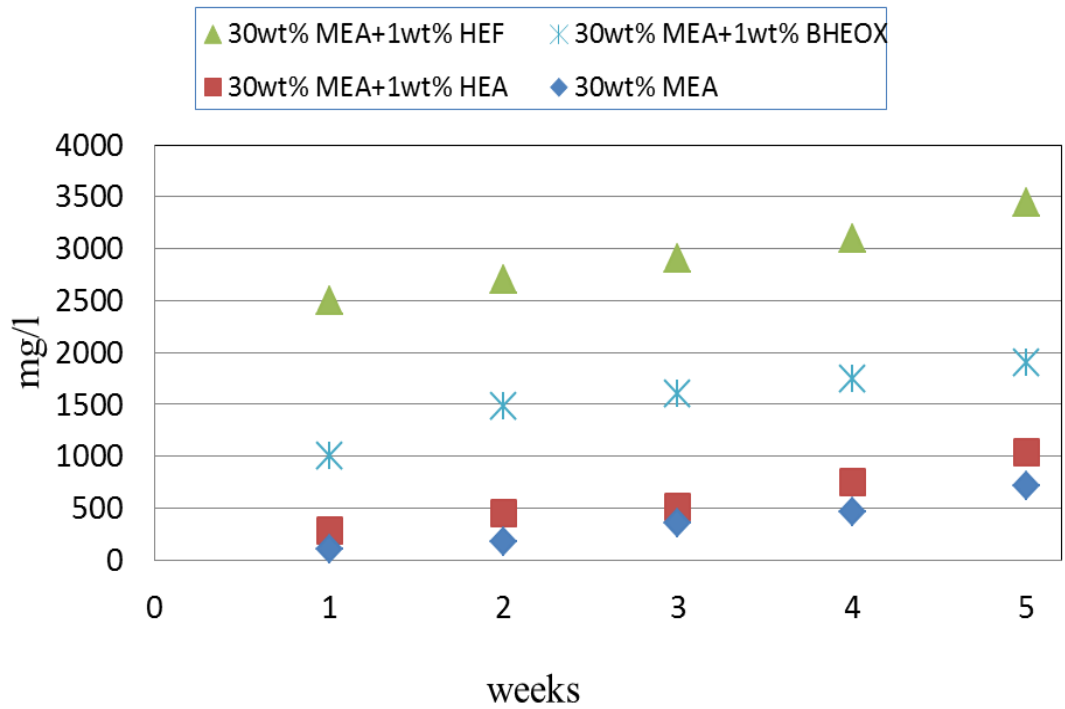
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237 3.3 IC for HSS anions

238 Formate formation is an indication of degradation, and can also be related to corrosion since
239 it can form HSS with MEA. By examining quantitatively the formate concentration with IC,
240 it can be observed that it increases each week. Most of the MEA+degradation product
241 solutions showed similar formate concentration values as the 30wt% MEA solution.
242 However, in the cases of BHEOX, HEF and HEA, significantly higher formate
243 concentrations can be observed as shown in Figure 2.



244
245 Figure 2: Concentration (mg/L) of formate for the different solutions.

246 HEF, BHEOX and HEA are formed from reactions with MEA with the corresponding
247 carboxylic acid[4, 9]. Under oxidative degradation conditions, the HEF, BHEOX and HEA
248 formations rates increase with higher oxygen concentration[13]. The HEF solution was
249 expected to give the highest formate concentrations, since HEF is formed from the reaction
250 between MEA and formic acid.

251 For BHEOX, anion IC results for the first week showed 306 mg/L oxalate which, under
252 thermal decomposition, converted to formate in the second week. From Figure 2, it can be
253 observed that the formate concentration for BHEOX solution from the first week to the
254 second does not follow the trend from the remaining weeks. This is further evidence that
255 thermal decomposition of oxalate forms formate. Higgins et al.[29] had shown that the
256 decomposition of oxalate to formate is possible. For HEA, high acetate concentrations were
257 measured, as expected.

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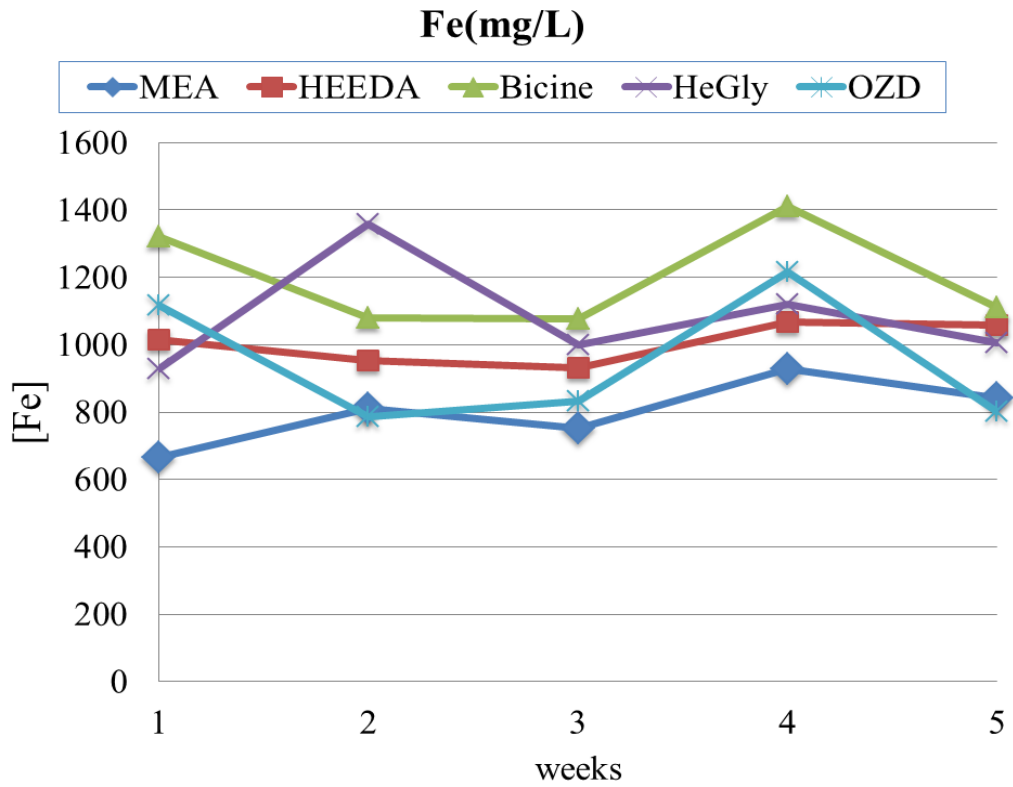
260 3.4 ICP-MS for total metal concentration

261 Among the 4 metals (Fe, Cr, Ni and Mo) the dominant is iron followed by Ni and Cr. With
262 ICP-MS, the individual metal concentrations were measured as an indication of corrosivity,
263 see Figures 3-6. From the 10 degradation products that were tested, HEEDA, Bicine, HeGly
264 and OZD solutions with MEA showed significantly higher metal concentrations than the
265 30wt% loaded MEA solution. Thus the Figures 3-6 show results only for MEA and for the
266 previously mentioned solutions with the four specific degradation products. In Figure 3,
267 results for total iron concentration are presented. It can be seen that after one week, the bicine
268 solution has the highest total Fe concentration. Bicine solution continues to have higher Fe
269 concentration in the following weeks, however in the second week, MEA+HeGly was the
270 dominant corrosive solution. In week 5, MEA+bicine solution has the highest corrosivity.
271 OZD, in the first and fourth week, has much higher iron concentration in comparison with
272 MEA. Nevertheless, this is not the case for the rest of the weeks. There is an increasing
273 tendency in iron concentration until the fourth week, and in the last week of the experiment
274 there is a small decline. There is the possibility that some corrosion compounds are forming
275 after 4 weeks and that is why in the last week smaller Fe concentration is observed for all the
276 tested solutions. Although Fe is the dominant concentration, a significant increase of iron
277 concentration with time is not observed. That can be explained because of a protective layer
278 formation (it is discussed in chapter 3.6). Unlike iron concentration profile, total nickel
279 concentration is increasing with time as it can be observed from Figure 4.

280 For Ni concentration of the various solutions, a major difference is shown in the first and
281 fourth week. There, MEA+bicine solution followed by MEA+OZD appear to have
282 significantly higher amounts of Ni compared to MEA solution. It is still unclear if by
283 monitoring the metals Ni, Cr, and Mo, reliable results concerning corrosion can be obtained.

284 In the case of Cr, as is presented in Figure 5, Bicine, HEEDA and HeGly seem to enhance
285 solution corrosivity. From Figure 6, it can be seen that Mo concentration for MEA+bicine and
286 MEA+OZD solutions is higher compared to MEA.

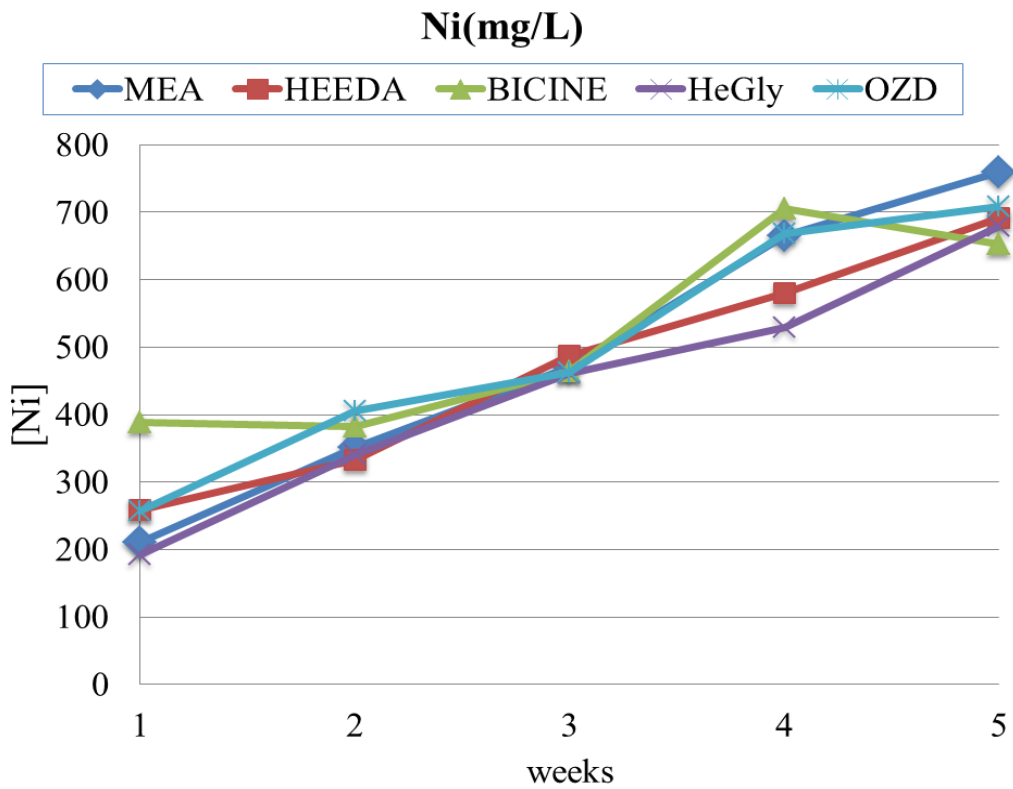
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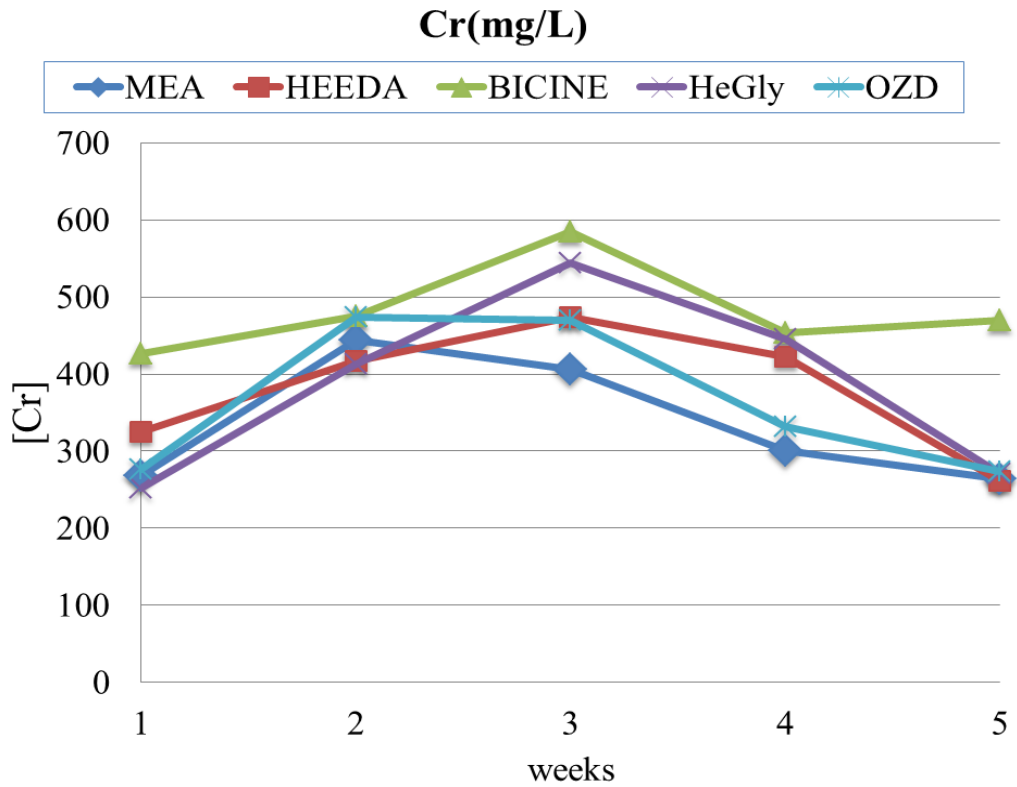
Figure 3: Total Fe concentration (mg/L) for the different solutions.



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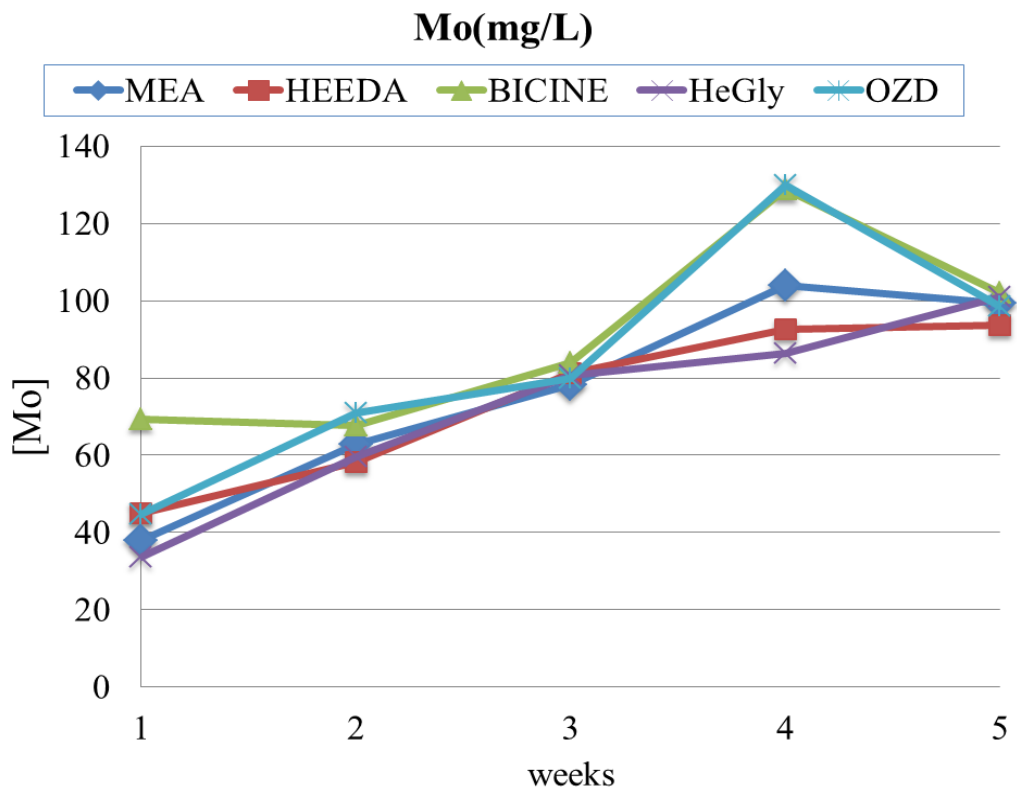
Figure 4: Total Ni concentration (mg/L) for the different solutions.



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Figure 5: Total Cr concentration (mg/L) for the different solutions.



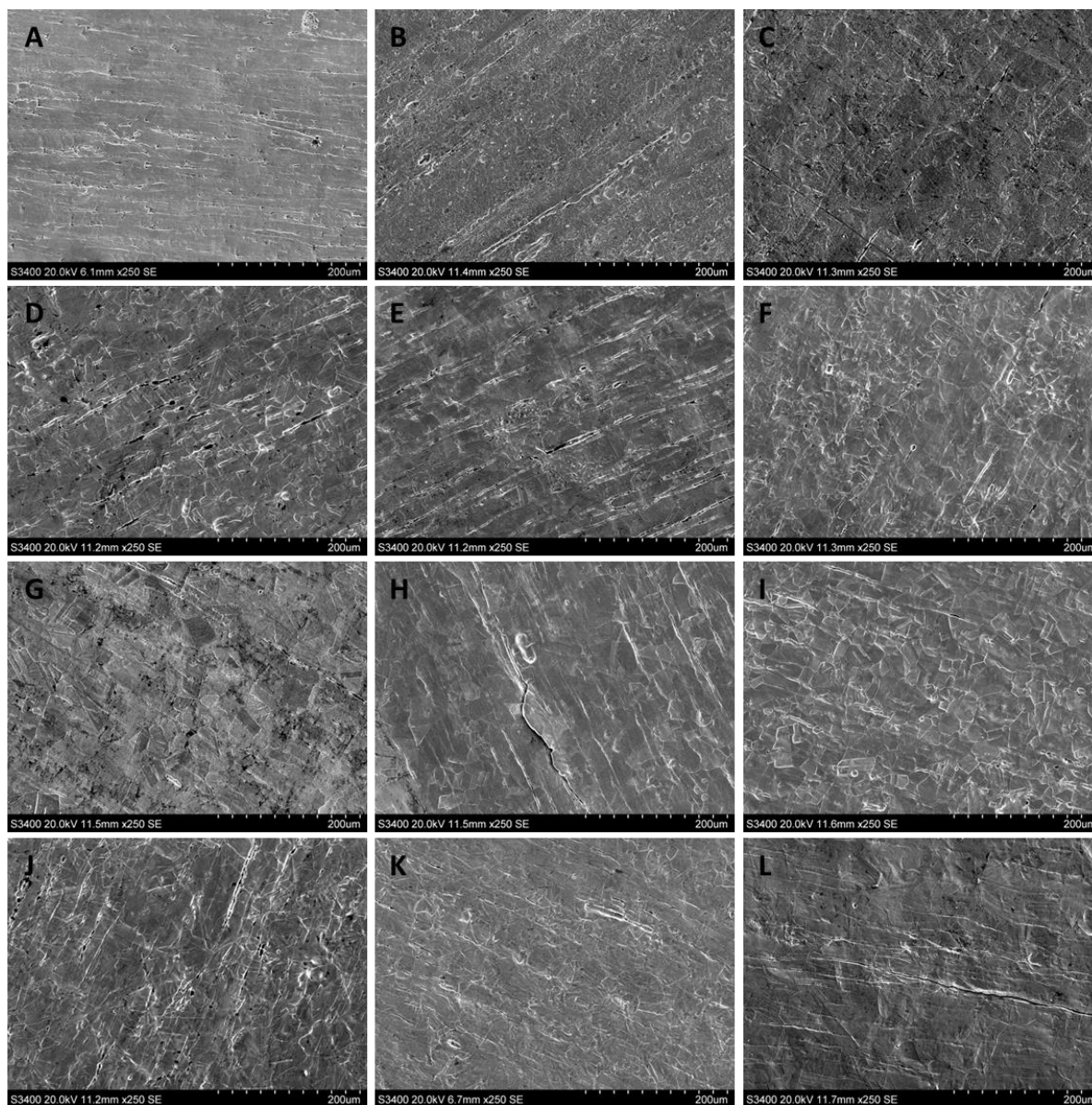
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Figure 6: Total Mo concentration (mg/L) for the different solutions.

296 3.5 SEM-EDS

297 At the end of the experiment, surface morphology and elemental mapping were evaluated
298 by a combined use of SEM and EDS, in order to examine the corrosive effects of the
299 solutions on cylinder surfaces. Differences observed in the final surface morphology as a
300 result of corrosion can be evaluated in terms of the varying extent of corrosion and corrosion
301 mechanisms.
302



303
304 Figure 7: SEM images of cylinder surfaces (A) before immersion, and after 5 weeks of
305 immersion in (B) 30 wt% MEA solution, (C) 30 wt% MEA with 1 wt% BHEOX, (D) 1 wt%
306 HEA, (E) 1 wt% HEF, (F) 1 wt% HEGly, (G) 1 wt% HEI, (H) 1 wt% HEPO, (I) 1 wt% OZD,
307 (J) 1 wt% HEIA, (K) 1 wt% HEEDA, and (L) 1 wt% Bicine.

308 When cylinders contained 30 wt% MEA solution, without addition of corrosion products, a
309 rough surface was observed, which represents a uniform corrosion over the surface at the end

310 of 5 weeks (Figure 7B). When secondary corrosion products were added to the solutions,
311 crack formation and pitting were more prominent on cylinder surfaces. Both these features
312 were most strongly revealed for the solutions containing 1 wt% HEA, HEI and HEIA (Figure
313 7 D, G, J). Significant crack formation was also observed in the presence of HEGly, HEPO,
314 OZD and HEEDA in MEA solution (Figure 7 F, H, I, K).

315 SEM observations suggest that the presence of secondary corrosion products of MEA in
316 solutions induced more localized corrosion on the cylinder surfaces which result in the
317 emergence of crack formation and pitting. When it comes to the effect of HSS on corrosion,
318 pitting corrosion was observed.[16] In addition, it was observed that pitting on the cylinder
319 surfaces was always accompanied by corrosion cracks which suggests their promotion as a
320 result of pit formation.

321 Cylinders immersed in MEA, MEA+HEA and MEA+OZD were compared in terms of
322 elemental composition as representatives of different surface characteristics after corrosion,
323 as shown in Table 4. From the table we can see that Elemental mapping by EDS revealed that
324 the compositional homogeneity of the cylinder surfaces remained independent of the
325 corrosion mechanisms and a single phase was detected for all samples.

326 Table 4. Elemental composition of 316 SS surfaces after 5 weeks of immersion in given
327 solutions.

wt %	MEA	MEA+HEA	MEA+ OZD
Fe	58.9	59.8	60.6
Cr	16.8	16.7	17.0
Ni	11.3	11.5	11.9
Mo	2.1	2.1	2.3
C	5.9	5.1	3.9

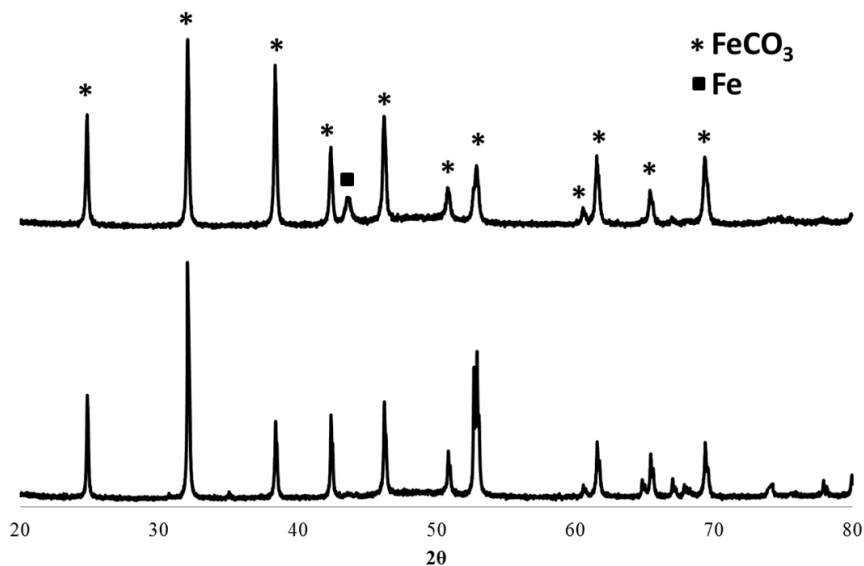
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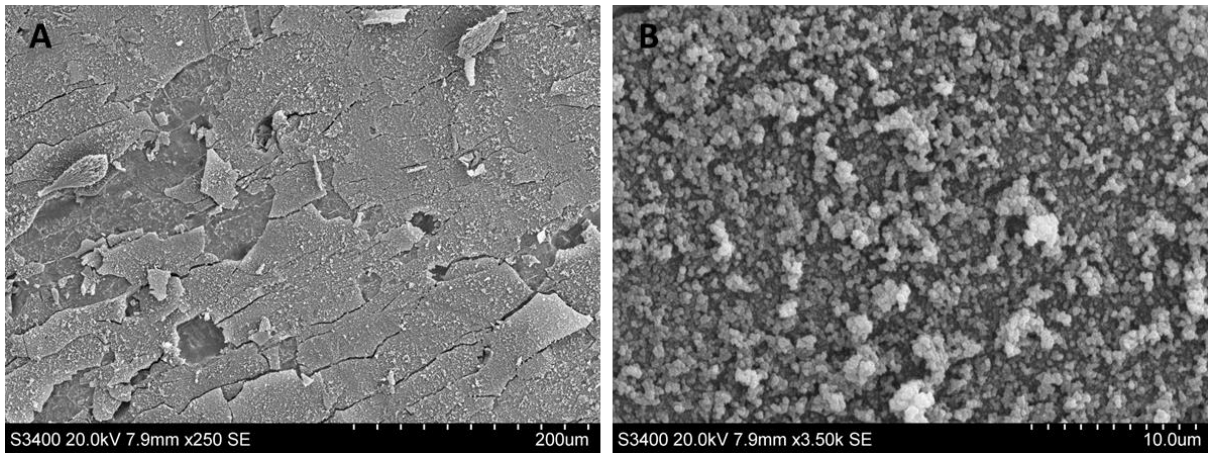
330 3.6 XRD

331 Analyses of corrosion products that deposited on the surface of 316 SS tubes were carried
332 out by powder XRD (Figure 8). Formation of highly crystalline siderite, FeCO_3 , was
333 observed on cylinder surfaces incubated in each solution. Cylinders which contained HEI,
334 HeGly, BHEOX, Bicine, and HEPO also had weak signals of iron in diffraction data of the
335 deposited products.

336 XRD analysis showed that the deposited product forming on the cylinder surfaces was
337 siderite (FeCO_3). Siderite has been found to be the dominant product in CO_2 -containing
338 corrosion environments for carbon steel [30], but this may not be the case for stainless steel.
339 The surface images in Fig. 7 do not show any solid formation but this is due to the sample
340 preparation with removal of deposited corrosion products prior to SEM imaging. As an
341 example, in Fig. 9 SEM images for the inner part of the cylinder for the case of MEA+HEF
342 solution are shown. In this case the sample is not pre-treated with ethanol and a protective
343 layer can be observed. The SEM image in Fig. 9B resembles siderite as presented in [31].
344 This layer was collected and analysis showed it to be FeCO_3 . The formation of FeCO_3 in
345 solution and deposition on the steel surface is probably due to the very nature of the thermal
346 degradation experiment where the CO_2 pressure, and therefore liquid phase content, as well
347 as the iron concentration, are high. Previous studies have shown that siderite acts as a
348 protective layer and decreases the rate of corrosion on mild steel and carbon steel
349 surfaces[32, 33]. Since FeCO_3 was identified on the corroded surfaces of all 316 SS samples,
350 further research should be conducted on the effect of siderite in post-combustion CO_2
351 capture. Siderite can work as a protective layer and reduce the corrosion rate also in this case.
352



353
354 Figure 8: XRD spectra of corrosion products collected from cylinders immersed in 30 wt%
355 MEA solution, and 30 wt% MEA with 1 wt% bicine from bottom to top, respectively.
356 Asterisks denote peaks associated with FeCO_3 and square denotes crystalline Fe.



357

358 Fig. 9: SEM images of cylinder surfaces after 5 weeks of immersion in 30 wt% MEA
359 solution with 1 wt% HEF at 250 times magnification (A) and 2000 times magnification (B).

360

361

362 4. Discussion

363 HeGly and HEPO are the dominant oxidative degradation compounds found in pilot
364 plants[4]. HEPO, which is formed from the reaction of MEA and HeGly[4], although
365 thermally stable, does not increase the corrosion rate according to ICP-MS results.
366 Furthermore, HEIA, the dominant thermal degradation product, seems not to have a
367 deteriorating effect on stainless steel. The imidazole HEI does not cause further corrosion
368 when added to MEA solution. Bicine is a known corrosive product[34] in amine based gas
369 treating plants. The findings of the current paper, showed that bicine plays a significant role
370 in corrosion. From the ICP-MS results, it is shown that HEF and OZD can contribute to
371 higher corrosion rates compared to the MEA solution without additives. HEEDA, which is a
372 precursor for HEIA formation, is reported to cause corrosion problems in carbon steel[17].
373 ICP-MS results for Fe indicated that HEEDA increases the corrosivity of the solution. HEIA
374 is the dominant thermal degradation compound in MEA based CO₂ capture plants, but it does
375 not affect the solution's corrosivity.

376 Formate formation is directly correlated with degradation but it can also be associated with
377 increasing corrosion. The addition of formic and oxalic acid (which decomposed to formic),
378 had a higher impact on corrosion[16]. In the current work, HEF and HEA solutions, which
379 had significantly higher formate concentration compared to only MEA, had also higher metal
380 concentrations. However, formate concentration and total iron concentration cannot be
381 correlated directly.

382 ICP-MS was the main technique used in this paper for measuring the corrosivity of the
383 solutions. It is still unclear if by monitoring the metals Ni, Cr, and Mo, reliable results can be
384 obtained. Since Fe is the dominant metal in stainless steel, either Fe itself or the sum of all the
385 metals should be studied. Another important topic is the duration of the experiment. It might
386 be that a two- instead of five-week experiment could yield similar results.

387 A summary of the degradation products that are tested for their corrosivity is given in Table
388 4. The results from the previous work [16] where the effect on corrosion for five acids that
389 are found in CO₂ capture plants were studied, are also presented. In the previous work formic,
390 oxalic, propionic, acetic and glycolic acid have been tested with similar experimental
391 approach, like in the present paper. That fact that formic and oxalic acid proved to have a
392 high effect on corrosion comes with good agreement with the findings of the current paper
393 for the formamide HEF and the oxamide BHEOX.

394 SEM observations revealed that the presence of secondary corrosion products of MEA induce
395 localized corrosion on the cylinder surfaces. It is known that the alloy composition and the
396 concentration of specific corrosive species are highly effective on that type of corrosion. It
397 was observed that when pit formation due to extreme localized corrosion occurred on the
398 crystal surfaces, crack formation was also induced accordingly.

399 The main corrosion product formed on the cylinder surfaces was highly crystalline siderite,
400 FeCO₃, regardless of the solution tested.

402 Table 4: Summary of MEA Degradation Product Corrosivity

Abbreviation	Compound	CAS	Stable at 135 °C	Corrosive
BHEOX	N,N'-Bis(2-hydroxyethyl)-oxamide	1871-89-2	No. Decomposes to oxalate and then to formate	It decomposes to corrosive compounds
HEA	N-(2-hydroxyethyl)-acetamide	142-26-7	Yes	Slightly
HEF	N-(2-hydroxyethyl)-formamide	693-06-1	Less than HEA	Slightly
HEGly	N-(2-hydroxyethyl)-glycine	5835-28-9	Very little	Yes
HEI	N-(2-hydroxyethyl)-imidazole	1615-14-1	Yes	No
HEPO	4-(2-hydroxyethyl)-2-piperazinone	23936-04-1	Yes	No, but forms HeGly
OZD	2-Oxazolidinone	497-25-6	No	Slightly
HEIA	N-(2-hydroxyethyl)imidazolidinone	3699-54-5	Yes	No
HEEDA	N-(2-hydroxyethyl)ethylenediamine	111-41-1	Degrades	Yes
Bicine	N,N-Bis(2-hydroxyethyl)glycine	150-25-4	Yes	Yes
Formic Acid [16]		64-18-6	Yes	Yes
Oxalic Acid [16]		144-62-7	No. Decomposes to formate	Yes
Propionic Acid [16]		79-09-4	Yes	No
Acetic Acid [16]		64-19-7	Yes	Slightly
Glycolic Acid [16]		79-14-1	Yes	No

404 5. Conclusions

405 The effect of MEA degradation products on corrosion has been studied. From the various
406 products, HeGly, HEEDA, Bicine and BHEOX increased the corrosivity of 30wt% MEA
407 solution. BHEOX, although not thermally stable, decomposes to oxalate and then to formate
408 which form HSS. HEA and HEF enhance corrosion while HEIA, the major thermal
409 degradation product, and HEI do not seem to aggravate corrosion. HEPO, although not
410 corrosive itself, gives HeGly which is plays a major role on corrosion.

411

412

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